



# A separation MOF with O/N active sites in nonpolar pore for One-step C<sub>2</sub>H<sub>4</sub> purification from C<sub>2</sub>H<sub>6</sub> or C<sub>3</sub>H<sub>6</sub> mixtures

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## ABSTRACT

Here, a 2-fold interpenetrated metal-organic framework (MOF) was synthesized by imidazole derivative and carboxyl co-ligands, which features nonpolar pore environment with accessible N/O sorption sites, resulting in the suitable pore sizes (8.78 Å × 11.45 Å) to optimally interact with C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>6</sub> compared to C<sub>2</sub>H<sub>4</sub>. The C<sub>3</sub>H<sub>6</sub>- and C<sub>2</sub>H<sub>6</sub>-selective phenomena were rigorously studied by combining pure-component sorption isotherms, ideal adsorbed solution theory (IAST) calculations, molecular simulations and breakthrough curves. The MOF exhibits high C<sub>3</sub>H<sub>6</sub> uptake (53.6 cm<sup>3</sup> g<sup>-1</sup> at 10 kPa and 298 K) and benchmark C<sub>3</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> selectivity (17.1), surpassing all of the reported porous materials for C<sub>3</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation. Molecular simulations revealed both nonpolar pore environment and accessible O/N sites synergistically “match” better with C<sub>3</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>6</sub> to provide stronger multiple interactions than C<sub>2</sub>H<sub>4</sub>. Experimental breakthrough curves and transient breakthrough simulations revealed that the MOF not only can realize one-step C<sub>2</sub>H<sub>4</sub> purification (purity ≥ 99.9%) from various ratios of C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> and C<sub>3</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures with great recyclability, but also greatly increase the purity of C<sub>3</sub>H<sub>6</sub> (such as from 28.5% to 87.7% for 2/5 C<sub>3</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixture).

## 1. Introduction

Light olefins, including ethylene (C<sub>2</sub>H<sub>4</sub>) and propylene (C<sub>3</sub>H<sub>6</sub>) are the major chemical feedstock for the synthesis of polymers (e.g., polystyrene, polyvinyl chloride, polystyrene, polypropylene, polyethylene) and other bulk chemical commodities [1–3], with the predicted worldwide production were more than 200 and 130 million tons in 2023, respectively, exceeding any other organic chemical [4–5]. Within the petrochemical industry, C<sub>3</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>4</sub> are primarily obtained by methanol-to-propylene (MTO) technology and thermal decomposition of ethane (C<sub>2</sub>H<sub>6</sub>), respectively: the MTO process mainly produces C<sub>3</sub>H<sub>6</sub> of 20.9 wt% and C<sub>2</sub>H<sub>4</sub> of 51.1 wt%, and the trace unreacted C<sub>2</sub>H<sub>6</sub> (5%–9%) as the impurity exists in the ethane dehydrogenation product [6–9]. The high purity of C<sub>3</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>4</sub> are essential for the manufacture of advanced fine chemicals, thus separation of C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures and MTO products (C<sub>3</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub>) to obtain the pure C<sub>2</sub>H<sub>4</sub> and C<sub>3</sub>H<sub>6</sub> are urgently needed but remain a huge challenge owing to the similar physical and chemical properties of light hydrocarbons (Scheme S1). Currently,

the light olefin/paraffin separation heavily relies on the high energy footprints heat-driven cryogenic distillation process, which accounts for ca. 0.3% of the global energy demand [10–12]. To this end, developing non-thermal driven separation technique with low carbon emissions is the promising solutions to the global energy use and emissions crisis. It is well known that adsorptive separation technology based on porous adsorbents presents a viable solution to this dilemma and displays more attractive promise with the continued advancement of customized adsorbents [13–16].

Metal-organic frameworks (MOFs) as novel porous adsorbents, have shown tremendous advantages in powerful predictability and tunability on pore size and functionality, that enable them to meet the task-specific requirements of various applications [17–23], including olefin/paraffin separation. Currently, the MOFs for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation can be classified into C<sub>2</sub>H<sub>6</sub>-selective or C<sub>2</sub>H<sub>4</sub>-selective MOFs, in which C<sub>2</sub>H<sub>4</sub>-selective MOFs are capable of preferentially adsorbing C<sub>2</sub>H<sub>4</sub> over C<sub>2</sub>H<sub>6</sub> while C<sub>2</sub>H<sub>6</sub>-selective MOFs adsorb C<sub>2</sub>H<sub>6</sub> over C<sub>2</sub>H<sub>4</sub> [24,25]. Nevertheless, C<sub>2</sub>H<sub>4</sub>-selective MOFs are not desirable because the C<sub>2</sub>H<sub>4</sub> product is

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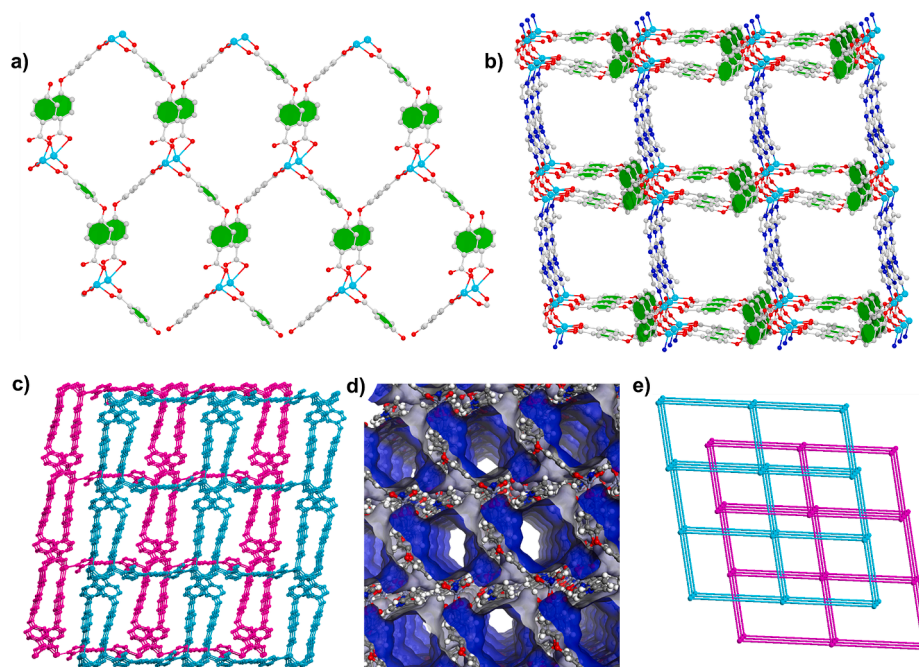
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**Fig. 1.** a) The 2D layer; b) 3D individual framework of MOF **1** with 1D channels; c) the 2-fold interpenetrated 3D networks individually coloured; d) accessible surface representation of pores in **1** viewed along the *c* axis; e) the 2-fold interpenetrated *pcu* net.

produced by additional desorption steps, and the co-adsorption of  $C_2H_6$  in adsorbents [26,27]. By comparison, the separation efficiency of  $C_2H_6$ -selective MOFs is remarkably increased since pure  $C_2H_4$  can be obtained directly at the outlet by one operation process [28]. Unfortunately, only a few cases have achieved the construction of  $C_2H_6$ -selective MOFs as installing  $C_2H_4$ -binding sites such as open metal sites (OMSs) in MOFs is easier than  $C_2H_6$ -binding sites [29,30]. Furthermore, the effect of moisture to MOFs must be considered as water vapor is ubiquitous in practical applications and might result in the structure collapse or reduced gas capacity of adsorbents. Hence, the exploitation of  $C_2H_6$ -selective MOFs with high and long-term humidity stability is the desired pursuit for  $C_2H_4$  purity goal.

Previous reports indicated that introducing inert or nonpolar functional groups in the pore walls of MOFs can prefer capturing  $C_2H_6$ , since  $C_2H_6$  has a higher polarizability ( $44.7 \times 10^{-25} \text{ cm}^3$ ) than  $C_2H_4$  ( $42.5 \times 10^{-25} \text{ cm}^3$ ), making the stronger interactions with  $C_2H_6$  through induction and dispersion interactions [3,31,32]. Furthermore, there are also some experimental and simulation results demonstrated that implantation of hydrogen bonding acceptors such as accessible oxygen or nitrogen atoms within the pores of MOFs can form multiple hydrogen bonds between framework and  $C_2H_6$  with more C-H bonds, enabling the construction of  $C_2H_6$ -selective MOFs [33,34]. Therefore, we envisage that the utilization of linkers containing nonpolar substituents (e.g., methyl) and metal nodes without OMSs can achieve the construction of  $C_2H_6$ -selective MOFs, while the introduction of accessible nitrogen and oxygen atoms on the linkers can improve the interaction strength for  $C_2H_6$ , and thus enhance the  $C_2H_6/C_2H_4$  separation performance.

Inspired by these endeavors, we presented a strategy to rationally construct MOF  $[Zn_2(\text{oba})_2(\text{dmimpym})] \cdot \text{DMA} \cdot 2\text{H}_2\text{O} \cdot 2\text{CH}_3\text{OH}$  (**1**) with nonpolar pore environment and accessible O/N adsorption sites that favour  $C_3H_6$  and  $C_2H_6$  over  $C_2H_4$ , through employing aromatic imidazole derivative, 4,6-di(2-methyl-imidazol-1-yl)-pyrimidine (dmimpym) and geometrically bent 4',4'-oxybisbenzoic linkers ( $H_2\text{oba}$ ). The methyl group functionalized pores with accessible O/N sites in the MOF **1** that on one hand preferentially adsorb  $C_2H_6$  and  $C_3H_6$  over  $C_2H_4$  at ambient pressure and temperature, on the other hand improve the humid stability of the MOF because of the hydrophobic surface of the pores. The static gas adsorption isotherms, isosteric enthalpies of

adsorption ( $Q_{st}$ ) and IAST calculations have confirmed  $C_3H_6$ - and  $C_2H_6$ -selective behavior for MOF **1**. So the MOF **1** can separate the  $C_2H_6/C_2H_4$  mixtures or  $C_3H_6/C_2H_4$  mixtures to produce the pure  $C_2H_4$  ( $\geq 99.9\%$ ) by one step, as demonstrated by simulated and experimental transient breakthrough results. Thus, together with humidity tolerance and excellent separation performance, the MOF is among the best adsorbents for  $C_2H_4$  purification application.

## 2. Experimental

### 2.1. Materials and methods

These contents were provided in [Supporting Information](#).

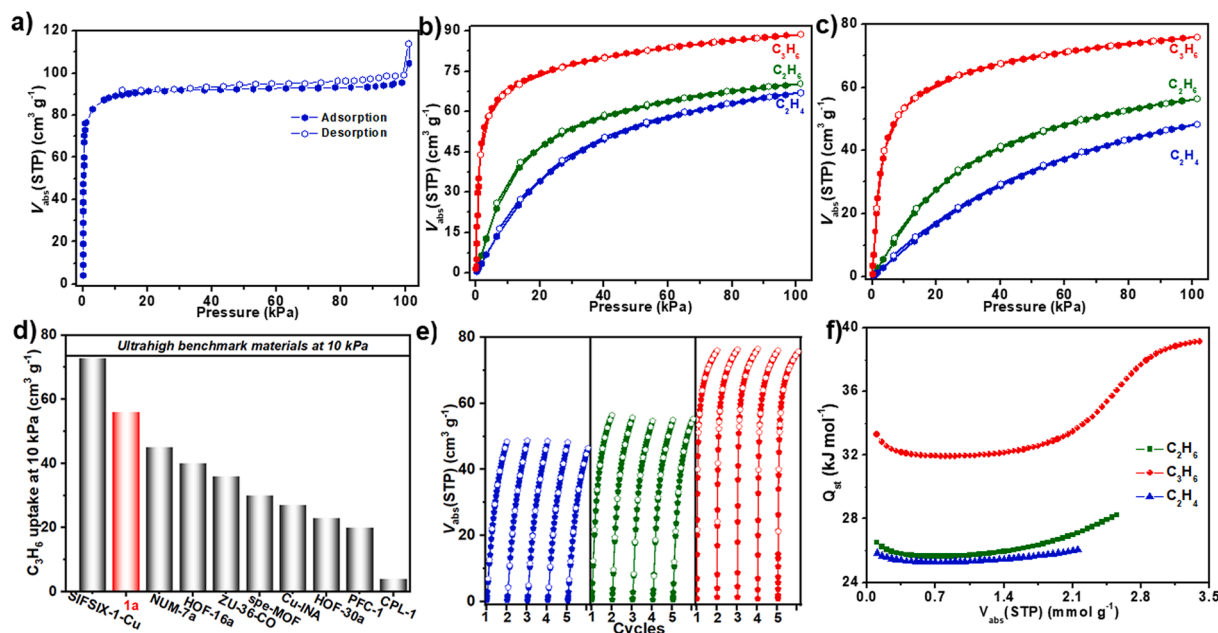
### 2.2. Synthesis of $[Zn_2(\text{oba})_2(\text{dmimpym})] \cdot \text{DMA} \cdot 2\text{H}_2\text{O} \cdot 2\text{CH}_3\text{OH}$ (**1**)

A mixture of 4,6-di(2-methyl-imidazol-1-yl)-pyrimidine (0.0241 g, 0.1 mmol), 4,4'-oxybis(benzoic acid) (0.0258 g, 0.1 mmol) and Zn ( $\text{NO}_3$ ) $_2 \cdot 6\text{H}_2\text{O}$  (0.0297 g, 0.1 mmol) was dissolved in the mixed solvent methanol (MeOH) (2 mL) and N,N'-dimethylacetamide (DMA) (4 mL) sealed in 15 mL glass vial, and heated at 95 °C for 48 h to obtain light yellow massive crystals (yield: 34%, based on Zn( $\text{NO}_3$ ) $_2 \cdot 6\text{H}_2\text{O}$ ). The sample morphology and distribution of elements were characterized by SEM-EDX ([Figure S1](#)). Anal. Calcd for  $C_{46}H_{49}N_7O_{15}Zn_2$ : C, 51.59; H, 4.61; N, 9.15%. Found: C, 51.54; H, 4.52; N, 9.11%.

## 3. Results and discussion

### 3.1. Crystal structure

Single-crystal X-ray diffraction reveals that the asymmetric unit of **1** consists of two  $Zn^{2+}$  isolated cations, one dmimpym linker, and two deprotonated  $\text{oba}^{2-}$  linkers with the framework formula  $[Zn_2(\text{oba})_2(\text{dmimpym})]$ , as shown in [Figure S2](#). Zn1 is tetrahedrally coordinated by three O atoms of three  $\text{oba}^{2-}$  linkers and one N atom of dmimpym. Zn2 coordinates with one N atom belonging to dmimpym linker in the axial position and four O atoms of three  $\text{oba}^{2-}$  linkers in the equatorial plane, resulting into a square-pyramidal geometry. The



**Fig. 2.** a)  $N_2$  sorption isotherm of **1a** at 77 K; b) and c)  $C_2H_4$ ,  $C_2H_6$  and  $C_3H_6$  sorption isotherms at 273 and 298 K; d) comparisons of  $C_3H_6$  uptakes in **1a** and top-performing adsorbents at 298 K under 10 kPa; e) repetitive  $C_2H_4$  (blue),  $C_2H_6$  (olive) and  $C_3H_6$  (red) adsorption measurements at 298 K; f) the  $Q_{st}$  curves for  $C_2H_4$ ,  $C_2H_6$  and  $C_3H_6$ . (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

adjacent Zn2 and Zn1 are joined by two carboxylic acid groups of two  $oba^{2-}$  linkers to form  $[Zn_2(COO)_2]$  binuclear secondary building units, and which are connected by  $oba^{2-}$  linkers to form the two-dimensional (2D) layer (Fig. 1a). As illustrated in Fig. 1b, dmimpym linkers connect the adjacent 2D layers to yield a three-dimensional (3D) pillared-layer framework. The frameworks are mutually interpenetrated to generate a 2-fold interpenetrated network (Fig. 1c), resulting in the reduction of channel void with the open sizes of ca.  $8.78 \text{ \AA} \times 11.45 \text{ \AA}$  along the  $c$  axis and the void volume of  $\sim 38.5\%$  calculated by a PLATON program (Fig. 1d). Topologically, this 3D framework can be simplified as a 6-connected **pcu** topology (Fig. 1e).

### 3.2. Characterization

The powder X-ray diffraction (PXRD) of synthesized bulk sample matches with the simulated one obtained from the crystal structure, suggesting its great crystallinity and high phase uniformity (Figure S3). Thermo gravimetric analysis (TGA) results reveal that the MOF underwent two steps of weight loss and then was stable up to about  $300 \text{ }^\circ\text{C}$  (Figure S4). The first weight loss about  $9.6\%$  up to  $110 \text{ }^\circ\text{C}$  is ascribed to free MeOH and  $H_2O$  molecules (calcd.  $9.4\%$ ), the subsequent weight loss from  $110$  to  $250 \text{ }^\circ\text{C}$  corroborates to the expulsion of DMA solvent molecules (found  $8.3\%$ , calcd  $8.1\%$ ).

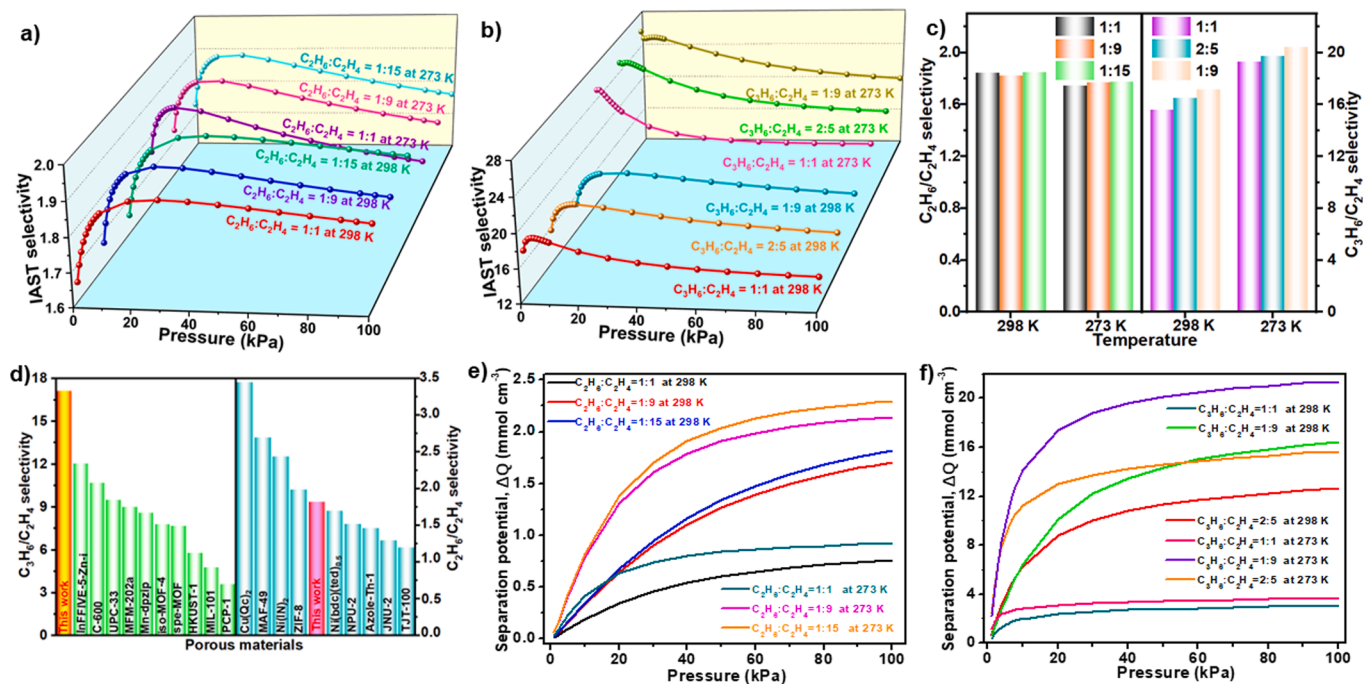
### 3.3. Gas adsorption properties

The pore characteristics of the guest-free framework **1a** were determined by  $N_2$  sorption experiment at 77 K. **1a** exhibited a distinct type-I  $N_2$  sorption isotherm with the saturated adsorption amount of  $113.9 \text{ cm}^3 \text{ g}^{-1}$  (Fig. 2a), and the Brunauer-Emmett-Teller (BET) surface area was fitted to be  $353.8 \text{ m}^2 \text{ g}^{-1}$ . The pore aperture distribution on basis of the Density Functional Theory model revealed a micropore distribution of around  $10\text{--}13 \text{ \AA}$ , which is consistent with the value of crystal structure. The accessible pore surface in **1a** is mainly modified by abundant methyl groups and benzene rings, resulting in a hydrophobic and nonpolar pore environment. The water vapor adsorption isotherm at 298 K in Figure S5 shows that only  $40.5 \text{ cm}^3 \text{ g}^{-1}$  water vapor was absorbed within **1a**, confirming the hydrophobicity [25]. Furthermore,

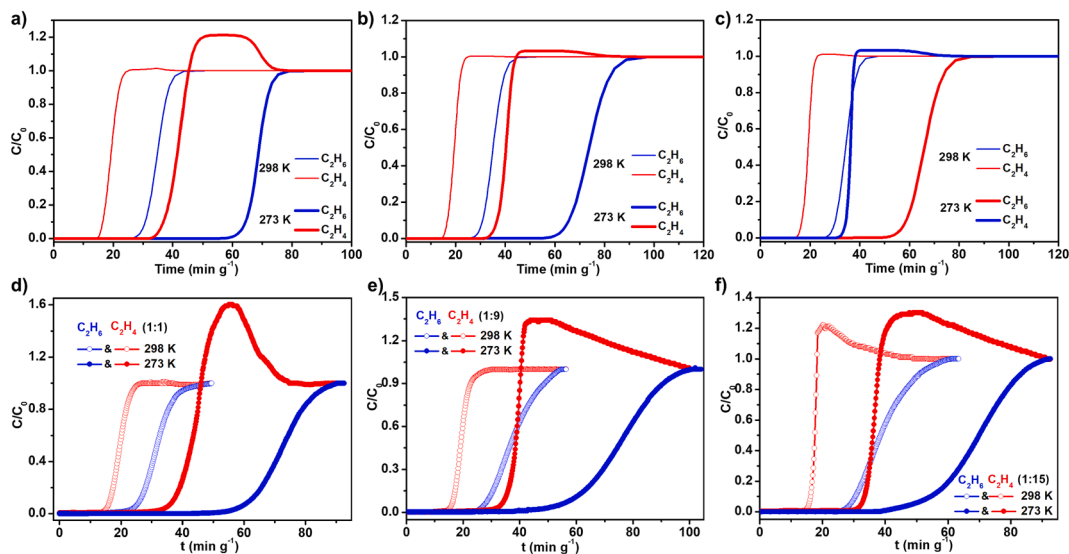
**1a** can maintain the framework crystallinity and integrity after the water vapor adsorption experiment, as evidenced by PXRD results in Figure S2.

We further recorded the sorption isotherms of  $C_2H_4$ ,  $C_3H_6$  and  $C_2H_6$  at 298 and 273 K, respectively. The adsorption capacity of **1a** for  $C_2H_6$  and  $C_3H_6$  are higher than that of  $C_2H_4$  at the same temperature, show the  $C_3H_6$ - and  $C_2H_6$ -selective behavior (Fig. 2b and 2c). Under 100 kPa, the  $C_2H_4$ ,  $C_2H_6$  and  $C_3H_6$  maximum uptakes are  $48.3 \text{ cm}^3 \text{ g}^{-1}$  ( $2.16 \text{ mmol g}^{-1}$ ),  $56.4 \text{ cm}^3 \text{ g}^{-1}$  ( $2.52 \text{ mmol g}^{-1}$ ) and  $76.0 \text{ cm}^3 \text{ g}^{-1}$  ( $3.39 \text{ mmol g}^{-1}$ ) at 298 K and  $67.1 \text{ cm}^3 \text{ g}^{-1}$  ( $2.99 \text{ mmol g}^{-1}$ ),  $70.4 \text{ cm}^3 \text{ g}^{-1}$  ( $3.14 \text{ mmol g}^{-1}$ ) and  $88.8 \text{ cm}^3 \text{ g}^{-1}$  ( $3.96 \text{ mmol g}^{-1}$ ) at 273 K, respectively. So  $C_2H_6$  and  $C_3H_6$  molecules are preferentially adsorbed into the framework through a strong adsorption affinity. Although the  $C_2H_6$  uptake of **1a** is below that some reported  $C_2H_6$ -selective MOFs, like ScBPDC ( $3.42 \text{ mmol g}^{-1}$ ) [35], CPM-736 ( $4.02 \text{ mmol g}^{-1}$ ) [36], and Co-9-ina ( $3.75 \text{ mmol g}^{-1}$ ) [37], but higher than many benchmark  $C_2H_6$ -selective MOFs such as MAF-49 ( $1.73 \text{ mmol g}^{-1}$ ) [38] and  $Cu(Qc)_2$  ( $1.85 \text{ mmol g}^{-1}$ ) [34]. It is worth noting that the  $C_3H_6$  uptake ( $53.6 \text{ cm}^3 \text{ g}^{-1}$ ) of **1a** at 298 K and 10 kPa is superior to some promising porous materials, such as NUM-7a,<sup>1</sup> spe-MOF [39], HOF-30a [40], and CPL-1 [41] (Fig. 2d). Furthermore, the continuous  $C_2H_6$ ,  $C_2H_4$  and  $C_3H_6$  adsorption measurements without reactivation at 298 K showed the uptake of  $C_2H_6$ ,  $C_2H_4$  and  $C_3H_6$  no obvious decrease at least five adsorption-desorption cycles (Fig. 2e), demonstrating excellent recyclability and regeneration capability of the material. The adsorption enthalpies ( $Q_{st}$ ) were determined from sorption isotherms at 298 and 273 K to appraise the interaction strengths of framework for  $C_2H_4$ ,  $C_2H_6$  and  $C_3H_6$  (Table S3). The  $Q_{st}$  values at zero-coverage of  $C_3H_6$ ,  $C_2H_6$  and  $C_2H_4$  were computed to be  $33.3$ ,  $26.5$  and  $25.8 \text{ kJ mol}^{-1}$ , respectively (Fig. 2f), corroborating that  $C_2H_6$  and  $C_3H_6$  have stronger affinity with the framework than  $C_2H_4$ . The  $Q_{st}$  values gradually increase at higher coverage, implying the adsorption benefits from the intermolecular interactions among the adsorbates [42–44]. It is noted the  $Q_{st}$  value for three gases are below those of MAF-49 ( $60.0 \text{ kJ mol}^{-1}$ ) [38],  $Fe_2(O_2)(dobdc)$  ( $66.8 \text{ kJ mol}^{-1}$ ) [14], IRMOF-8 ( $52.5 \text{ kJ mol}^{-1}$ ) [45], and FJI-H11-Me(des) ( $38.9 \text{ kJ mol}^{-1}$ ) [3], suggesting the low regeneration energy footprint.

To estimate the potential of **1a** for separating the mixed gases, the theoretical calculations of ideal adsorption solutions theory (IAST) for



**Fig. 3.** IAST selectivity curves of **1a**: a)  $C_2H_6/C_2H_4$ ; b)  $C_3H_6/C_2H_4$ ; c) IAST selectivity values of **1a** for  $C_2H_6/C_2H_4$  and  $C_3H_6/C_2H_4$ ; d) comparison of  $C_3H_6/C_2H_4$  (left) and  $C_2H_6/C_2H_4$  (right) selectivities in **1a** and some benchmark materials at 298 K; e) and f) separation potential of **1a** for  $C_2H_6/C_2H_4$  and  $C_3H_6/C_2H_4$  mixtures at 273 and 298 K.

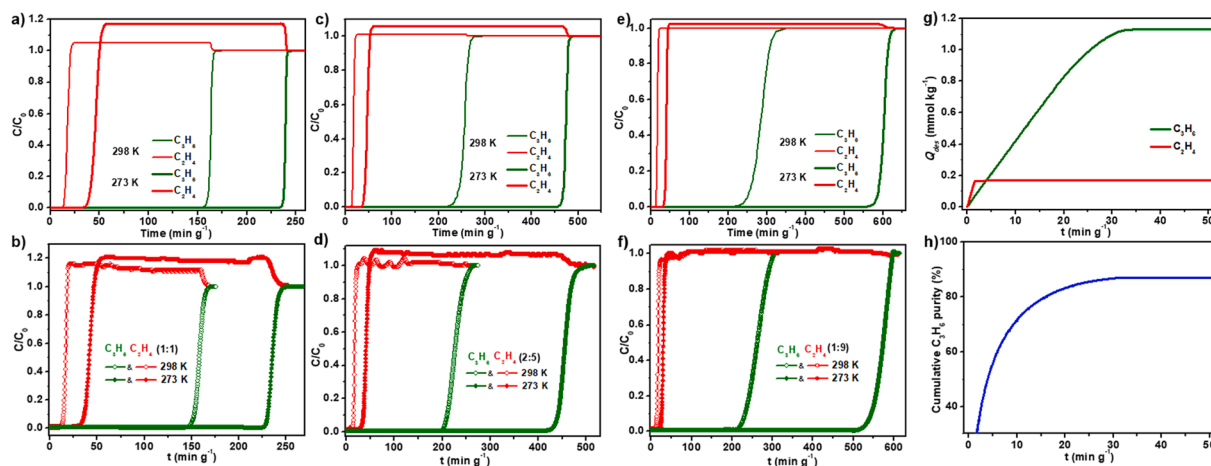


**Fig. 4.** a-c) Transient breakthrough simulation curves of **1a** for  $C_2H_6/C_2H_4$  mixtures at 298 and 273 K: a) v/v, 1/1; b) v/v, 1/9; c) v/v, 1/15; d-f) experimental breakthrough curves of **1a** for  $C_2H_6/C_2H_4$  mixtures at 298 and 273 K: d) v/v, 1/1; e) v/v, 1/9; f) v/v, 1/15.

$C_2H_6/C_2H_4$  (v/v; 1/15, 1/9 and 1/1) and  $C_3H_6/C_2H_4$  (v/v; 1/1, 1/9 and 2/5) were performed to predict the adsorption selectivity at 298 and 273 K (Figure S6-S8). As displayed in Fig. 3a and 3c, the adsorption selectivities for  $C_2H_6/C_2H_4$  mixtures of 1/1, 1/9 and 1/15 were calculated to be about 1.8, which is higher than or comparable to many  $C_2H_6$ -selective MOFs, like TJT-100 (1.2) [46], Azole-Th-1 (1.46) [47], and NPU-2 (1.52) [4], Ni(bdc)(ted)<sub>0.5</sub> (1.7) [48], and ZIF-8 (1.99) [49], but inferior to some benchmark MOFs, such as Cu(Qc)<sub>2</sub> (3.45) [34], MAF-49 (2.7) [38], ZJU-120a (2.74) [50], and Ni(IN)<sub>2</sub> (2.44) [51]. In addition, the selectivities for 1/1, 1/9 and 2/5  $C_3H_6/C_2H_4$  mixtures were 15.6/19.3, 17.1/20.4 and 16.5/19.7 at 298/273 K, respectively (Fig. 3b and 3c), which are considerably high compared to the benchmark materials

for  $C_3H_6/C_2H_4$  separation, such as C-600 (10.6) [52], UPC-33 (9.5) [53], Mn-dpzip (8.6) [54] spe-MOF (7.7) [35], and HKUST-1 (5.8) (Fig. 3d) [55].

The index to evaluate separation performance is separation potential ( $\Delta Q$ ), which is a combined selectivity-capacity metric introduced by Krishna [56–58]. According to the calculated  $\Delta Q$  shown in Fig. 3e, the amounts of pure  $C_2H_4$  can be recovered by **1a** reached up to 0.76/0.92, 1.70/2.14 and 1.82/2.30 mmol cm<sup>-3</sup> for the 1/1, 1/9, and 1/15  $C_2H_6/C_2H_4$  mixtures at 298/273 K, respectively. The corresponding pure  $C_2H_4$  values are 3.04/3.67, 12.63/15.68 and 16.43/21.39 mmol cm<sup>-3</sup> for the 1/1, 2/5, and 1/9  $C_3H_6/C_2H_4$  mixtures at 298/273 K (Fig. 3f). These results demonstrate the potential of **1a** for one-step purification of  $C_2H_4$



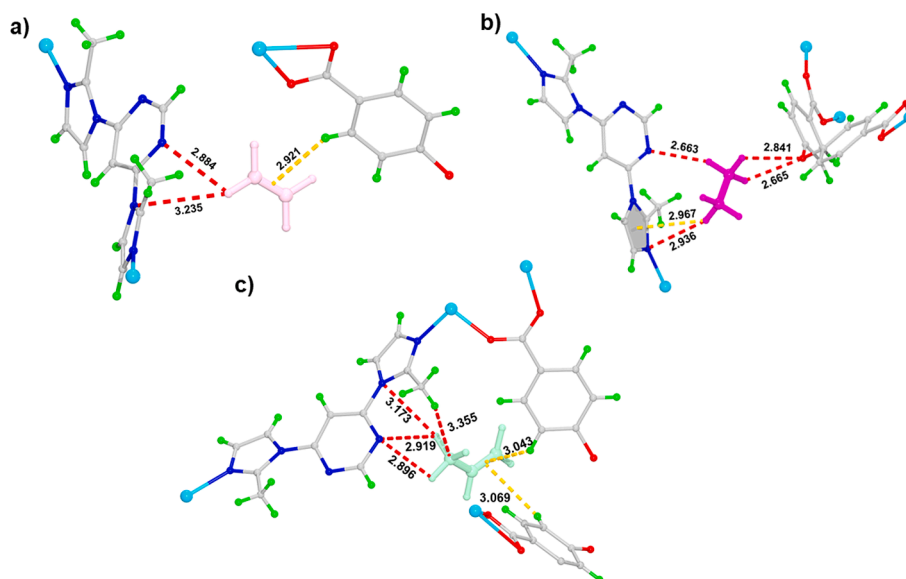
**Fig. 5.** a), c) and e) Transient breakthrough simulation curves of **1a** for  $\text{C}_3\text{H}_6/\text{C}_2\text{H}_4$  mixtures at 298 and 273 K: a) v/v, 1/1; b) v/v, 2/5; c) v/v, 1/9; b), d) and f) experimental breakthrough curves of **1a** for  $\text{C}_3\text{H}_6/\text{C}_2\text{H}_4$  mixtures at 298 and 273 K: b) v/v, 1/1; d) v/v, 2/5; f) v/v, 1/9; g) cumulative moles of  $\text{C}_3\text{H}_6$  and  $\text{C}_2\text{H}_4$  recovered during blowdown; h) cumulative purity of  $\text{C}_3\text{H}_6$  recovered from **1a** during simulated counter-current blowdown operations.

from binary  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  or  $\text{C}_3\text{H}_6/\text{C}_2\text{H}_4$  mixtures.

### 3.4. $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$ separation

To verify the feasibility of  $\text{C}_2\text{H}_4$  purification from  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixtures, transient breakthrough simulations were conducted for **1a** in a fixed bed at 273 and 298 K and 100 kPa by using the methodology described by Krishna (see also [Supporting Information](#)) [56,58]. The simulated breakthrough results in [Fig. 4a-c](#) show that efficient separations can be achieved by **1a** for 1/1, 1/9, and 1/15  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixtures, wherein  $\text{C}_2\text{H}_4$  pass through occurs first to yield the high-purity  $\text{C}_2\text{H}_4$  and then  $\text{C}_2\text{H}_6$  breakthrough the bed after a certain time. Next, the dynamic breakthrough experiments of  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  (v/v; 1/1, 1/9, and 1/15) mixtures with Ar as the carrier gas (90%, 90% and 84%, vol %) were purged into fixed bed with total gas mixture flow rate of  $5.0 \text{ mL min}^{-1}$  at 273 and 298 K. As revealed in [Fig. 4d-f](#),  $\text{C}_2\text{H}_4$  can be efficiently separated from  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixtures at two temperatures. When these three  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixtures were passed over the fixed bed of **1a**,  $\text{C}_2\text{H}_4$  first eluted through the fixed bed to yield a desirable pure  $\text{C}_2\text{H}_4$  ( $\geq 99.9\%$ ) without detectable  $\text{C}_2\text{H}_6$  signal was found (below the mass spectrometry detection limit), whereas  $\text{C}_2\text{H}_6$  was more efficiently adsorbed in adsorbent for

a certain period of time and then reached its breakthrough point. During the breakthrough duration,  $\text{C}_2\text{H}_4$  productivities from the outlet effluent which were determined to be  $1.32/4.14$ ,  $3.29/8.47$ , and  $5.65/13.42 \text{ L kg}^{-1}$  for 5/5, 1/9, and 1/15  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixtures at 298/273 K, respectively. It is worth noting that the  $\text{C}_2\text{H}_4$  productivity values for **1a** would be higher than our experimental results because of industrial practice without containing inert carrier gas. The value of  $5.65 \text{ L kg}^{-1}$   $\text{C}_2\text{H}_4$  ( $\geq 99.9\%$ ) from 1/15  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixtures (partial cracked gas mixtures) is lower than some excellent materials, like [Zn(BDC) ( $\text{H}_2\text{BPZ}$ )] [8],  $\text{Fe}_2(\text{O}_2)(\text{dobdc})$  [14], and TJT-100 [46], but is comparable or higher than some reported benchmark porous material (e.g., NKCOF-21 [28], HOF-76a [59],  $\text{Cu}(\text{Qc})_2$  [34], MAF-49 [38], and HIAM-102 [9]). The separation factor based on breakthrough experiments for 1/1, 1/9, and 1/15  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixtures were calculated to be  $1.56/2.22$ ,  $1.96/2.85$ , and  $2.67/2.65$  at 298/273 K, respectively. The experimental separation factor for the  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  were higher than the results predicted by IAST, indicating that **1a** more preferentially adsorb  $\text{C}_2\text{H}_6$  than  $\text{C}_2\text{H}_4$  in the actual  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixtures separation process. The high  $\text{C}_2\text{H}_4$  productivity indicates that **1a** has efficient separation performance of various ratios  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  mixtures around room temperature, suggesting promising  $\text{C}_2\text{H}_6/\text{C}_2\text{H}_4$  separation in practical



**Fig. 6.** Preferential binding sites for a)  $\text{C}_2\text{H}_4$ ; b)  $\text{C}_2\text{H}_6$  and c)  $\text{C}_3\text{H}_6$ .

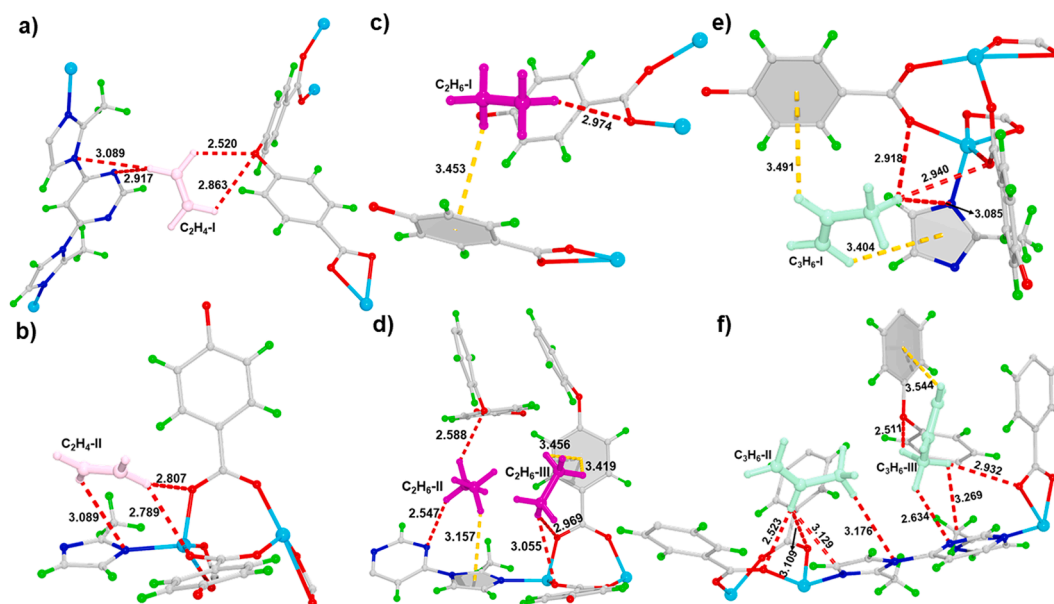


Fig. 7. Adsorption sites for a) and b)  $C_2H_4$ ; c) and d)  $C_2H_6$ ; e) and f)  $C_3H_6$  in **1a** simulated at 298 K under 100 kPa.

applications. Subsequently, multiple breakthrough experiments result for  $C_2H_6/C_2H_4$  mixtures (v/v; 1/1) exhibits that **1a** could be remained for at least five cycles without any loss in separation performance, confirming excellent separation and recovery capacity (Figure S9).

### 3.5. $C_3H_6/C_2H_4$ separation

In addition, the similar transient breakthrough simulations and experimental breakthrough curves of  $C_3H_6/C_2H_4$  (1/1, 1/9 and 2/5; v/v) mixtures with Ar as the carrier gas (90%, 90% and 93%, vol %) were also tested to evaluate the MTO products separation performance of **1a**. As presented in Fig. 5a-f, there are excellent match between the experiments and simulations breakthrough curves, which display complete separation of these  $C_3H_6/C_2H_4$  mixtures by **1a** at both 273 and 298 K, wherein  $C_2H_4$  breakthrough the bed first to produce an outflow of pure  $C_2H_4$  gas with 99.9% purity, and then  $C_3H_6$  breaks through the fixed-bed after a substantial time lapse since it's more preferentially adsorbed in **1a**. During the breakthrough duration, the collected high-purity  $C_2H_4$  gas are 35.62/50.01, 49.68/98.71, and 94.91/243.42  $L\ kg^{-1}$  for 1/1, 2/5, and 1/9  $C_3H_6/C_2H_4$  mixtures at 298/273 K, respectively. And these productivity values would be higher in industrial practice without containing inert carrier gas. The  $C_2H_4$  productivity values of **1a** are significantly higher than that of Mn-dtzip under the same conditions [54]. To estimate the possibility of **1a** for  $C_3H_6$  purification from  $C_3H_6/C_2H_4$  mixture, we also simulated counter-current blowdown operations

for 2/5/93  $C_3H_6/C_2H_4/Ar$  mixture at 298 K (Fig. 5g). The results showed that 87.1%  $C_3H_6$  purity can be recovered with productivity of 1.133  $mol\ kg^{-1}$  in a counter-current blowdown process (Fig. 5h). Thus, **1a** not only can produce 99.9% purity  $C_2H_4$  by one step, but also improves the purity of  $C_3H_6$ , which implies a promising application for obtaining pure  $C_2H_4$  and concentrate  $C_3H_6$  simultaneously from MTO products. Furthermore, this MOF also shows great recyclability and regeneration capability for multiple  $C_3H_6/C_2H_4$  mixtures breakthrough tests (Figure S10).

### 3.6. GCMC simulation

To further explore the interactions between the framework and adsorbates, the preferential binding sites for  $C_3H_6$ ,  $C_2H_4$  and  $C_2H_6$  were firstly determined by Grand Canonical Monte Carlo (GCMC) simulations. It found that the preferential adsorption sites for three gas molecules at 298 K and 100 kPa are located at the in the similar region near the dmimpym and oba coordination fragments. As depicted in the Fig. 6a, there are only two C-H...N interactions between the H atoms of  $C_2H_4$  and N atoms of dmimpym ligand with the H...N distances from 2.884 to 3.235 Å and one C-H... $\pi$  interaction between the H atoms of phenyl ring and  $\pi$  center of  $C_2H_4$  with the H... $\pi$  distances of 2.921 Å. For  $C_2H_6$  molecule, there are not only form stronger C-H...N contacts (H...N, 2.663 and 2.936 Å), C-H... $\pi$  interaction (H... $\pi$ , 2.967 Å), but also two C-H...O hydrogen bonds with O atoms of carbonyl group in oba ligand (H...O, 2.665 and 2.841 Å) were found (Fig. 6b). Compared to  $C_2H_4$  and

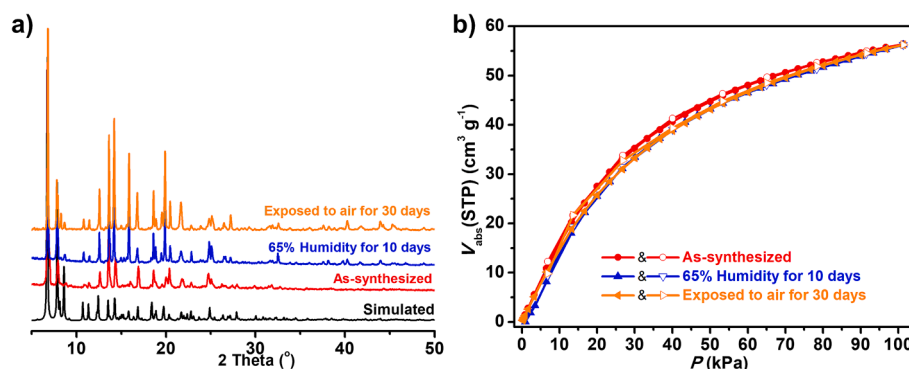


Fig. 8. a) PXRD patterns of **1** treated under different conditions; b)  $C_2H_6$  sorption isotherms at 298 K of **1a** treated under different conditions.

C<sub>2</sub>H<sub>6</sub>, C<sub>3</sub>H<sub>6</sub> with a larger molecular size enhances the interactions with the framework. As a consequence, one C<sub>3</sub>H<sub>6</sub> molecule interacts with two H atoms of phenyl ring, one H atoms of methyl group, one imidazole N atom, one pyrimidine N atom of ligands to form more strong contacts including C-H... $\pi$ /N/C interactions with the distances of 2.896–3.355 Å (Fig. 6c). The more and stronger interactions between C<sub>2</sub>H<sub>6</sub>/C<sub>3</sub>H<sub>6</sub> and framework indicate that **1a** prefers to adsorb C<sub>3</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>6</sub> over C<sub>2</sub>H<sub>4</sub>.

Furthermore, the different binding sites in **1a** for C<sub>2</sub>H<sub>4</sub>, C<sub>3</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>6</sub> at 298 K under 100 kPa were further studied by GCMC simulations. As shown in Fig. 7, the MOF exhibits two adsorption sites for C<sub>2</sub>H<sub>4</sub> and three important sites for C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>6</sub>. C<sub>3</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>6</sub> molecules are involved in more contacts with the accessible sites of  $\pi$  centers from ligands and O/N atoms in framework, forming rich C-H... $\pi$  interactions and C-H...O/N hydrogen bonds (Fig. 7 c,d). Notably, the methyl groups in dmimpym are also useful sites for C<sub>3</sub>H<sub>6</sub> (C<sub>3</sub>H<sub>6</sub>-III) molecules through C-H...C interactions. By contrast, the interactions of the framework with C<sub>2</sub>H<sub>4</sub> are weak due to less contact fashions (Fig. 7a,b), thus forming a priority of adsorption for C<sub>2</sub>H<sub>6</sub>/C<sub>3</sub>H<sub>6</sub> over C<sub>2</sub>H<sub>4</sub>. These simulation results are consistent with the experimental selectivity of **1a** for C<sub>2</sub>H<sub>6</sub>/C<sub>3</sub>H<sub>6</sub> over C<sub>2</sub>H<sub>4</sub>.

### 3.7. Stability test

Considering the real-industrial application, the chemical stability of **1a** toward relative humidity and air environments for a certain period of times was monitored by PXRD and gas adsorption experiments. As given in Fig. 8a, the samples were exposed to the air for 30 days and relative humidity (65%) for 10 days, it remains intact without loss of crystallinity and phase transformation, as indicated by PXRD. Subsequently, C<sub>2</sub>H<sub>6</sub> adsorption isotherms at 298 K obtained after exposing to humidity and air conditions almost unchanged, demonstrating its excellent chemical stability (Fig. 8b).

## 4. Conclusions

In summary, we rationally designed and synthesized a robust MOF with the function of one-step C<sub>2</sub>H<sub>4</sub> purification from C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> and C<sub>3</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures. During design process of MOF, on the one hand, the nonpolar pore surface composed of aromatic rings and methyl groups realized the construction of C<sub>2</sub>H<sub>6</sub>-selective MOF, on the other hand, the introduction of accessible O/N adsorption sites in the ligands further enhanced the interactions between the framework and C<sub>3</sub>H<sub>6</sub> compared with C<sub>2</sub>H<sub>4</sub>. The MOF not only exhibits high C<sub>3</sub>H<sub>6</sub> uptake (53.6 cm<sup>3</sup> g<sup>-1</sup> at 10 kPa and 298 K) but also benchmark C<sub>3</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> selectivity (17.1), surpassing all of the reported porous materials for C<sub>3</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation. The MOF on one hand can efficiently separate the C<sub>2</sub>H<sub>4</sub> from various ratios of C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> and C<sub>3</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures to directly produced high-purity ( $\geq 99.9\%$ ) C<sub>2</sub>H<sub>4</sub> in one-step at 298 and 273 K, on the other hand, greatly increase the purity of C<sub>3</sub>H<sub>6</sub> from C<sub>3</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures. Molecular simulations revealed the nonpolar pore environment and accessible O/N sites synergistically “match” better with C<sub>3</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>6</sub> to provide stronger multiple interactions than C<sub>2</sub>H<sub>4</sub>. This contribution not only reports the benchmark MOF material for C<sub>3</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> and C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation, but also provides some guidance to design nonpolar pore environment with accessible adsorption sites in MOFs for addressing C<sub>2</sub>H<sub>4</sub> purification goal.

### Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

### Data availability

Data will be made available on request.

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**Appendix A. Supplementary data** Additional Figures and selected bond length/angle table. CCDC 2233136 contains the supplementary crystallographic data for this paper. These data can be obtained free of charge from the Cambridge Crystallographic data Center Via [http://www.ccdc.cam.ac.uk/data\\_request/cif](http://www.ccdc.cam.ac.uk/data_request/cif).

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.cej.2023.143056>.

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## *Supporting Information*

### **A Separation MOF with O/N Active Sites in Nonpolar Pore for One-step C<sub>2</sub>H<sub>4</sub> Purification from C<sub>2</sub>H<sub>6</sub> or C<sub>3</sub>H<sub>6</sub> Mixtures**

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## Materials and general methods

All solvents and organic ligand for synthesis were purchased commercially from The Shanghai Tensus Bio-tech Co., Ltd. Elemental analyses of C, H, and N were determined with a Perkin-Elmer 2400C elemental analyzer. Thermalgravimetric analyses (TGA) were carried out in a nitrogen stream using a Netzsch TG209F3 equipment at a heating rate of 10 °C min<sup>-1</sup>. Single crystal diffraction data were collected on a Bruker SMART APEX II CCD single crystal diffractometer. Gas adsorption measurements were performed with an automatic volumetric sorption apparatus (Micrometrics ASAP 2020M). Water sorption was collected by Quantachrome Vstar vapor adsorption equipment. Breakthrough experiments were performed on a Quantachrome dynaSorb BT equipments.

### X-Ray Crystallography

A Bruker Smart Apex II CCD detector was used to collect the single crystal data at 216(2) K using Mo K $\alpha$  radiation ( $\lambda = 0.71073 \text{ \AA}$ ). The structure was solved by direct methods and refined by full-matrix least-squares refinement based on  $F^2$  with the Olex 2 program. The non-hydrogen atoms were refined anisotropically with the hydrogen atoms added at their geometrically ideal positions and refined isotropically. As the disordered solvent molecules in the structure cannot be located, the SQUEEZE routine of Platon program was applied in refining. The formula of complex was got by the single crystal analysis together with elemental microanalyses and TGA data. Relevant crystallographic results are listed in Table S1. Selected bond lengths and angles are provided in Table S2.

### N<sub>2</sub> Sorption Isotherm

Before gas sorption experiments, All the as-synthesized samples were immersed in acetone for 3 days, during which the solvent was decanted and freshly replenished three times a day. All the samples were activated under vacuum at 333 K for 4 hours to obtain activated **1a**. Gas sorption measurements were then conducted using a Micrometrics ASAP 2020M gas adsorption analyzer.

### Transient breakthrough simulations

Transient breakthrough simulations were carried out for the same set of operating conditions as in the experimental data sets, using the methodology described in earlier publications.<sup>1-5</sup> In

these simulations, intra-crystalline diffusion influences are ignored. For **1a**, there is excellent match between the experiments and simulations. From the breakthrough simulations, the productivities of 99.9% pure C<sub>2</sub>H<sub>4</sub> were determined; these are expressed in the units of L per kg of **1a**.

### **Breakthrough Experiments**

The breakthrough experiment was performed on the Quantachrome dynaSorb BT equipments at 298 K and 1 bar with an equal volume of mixed gas (gas A: gas B: Ar = 5% : 5% : 90%, Ar as the carrier gas, flow rate = 5 mL min<sup>-1</sup>). The activated MOF (1 g) was filled into a packed column of  $\phi$  4.2×80 mm, and then the packed column was washed with Ar at a rate of 7 mL min<sup>-1</sup> at 333 K for 35 minutes to further activate the samples. Between two breakthrough experiments, the adsorbent was regenerated by Ar flow of 7 mL min<sup>-1</sup> for 35 min at 333 K to guarantee a complete removal of the adsorbed gases.

### **GCMC Simulation**

Grand canonical Monte Carlo (GCMC) simulations were performed for the gas adsorption in the framework by the Sorption module of Material Studio (Accelrys. Materials Studio Getting Started, release 5.0). The framework was considered to be rigid, and the optimized gas and epoxide molecules were used. The partial charges for atoms of the framework were derived from QEq method and QEq neutral 1.0 parameter. One unit cell was used during the simulations. The interaction energies between the gas molecules and framework were computed through the Coulomb and Lennard-Jones 6-12 (LJ) potentials. All parameters for the atoms were modeled with the universal force field (UFF) embedded in the MS modeling package. A cutoff distance of 12.5 Å was used for LJ interactions, and the Coulombic interactions were calculated by using Ewald summation. For each run, the 5 × 10<sup>6</sup> maximum loading steps, 5 × 10<sup>6</sup> production steps were employed.

### **Fitting of experimental data on pure component isotherms**

The unary isotherm data for C<sub>2</sub>H<sub>4</sub>, C<sub>2</sub>H<sub>6</sub>, and C<sub>3</sub>H<sub>6</sub>, measured at two different temperatures 273 K, and 298 K in **1a** were fitted with good accuracy using the dual-site Langmuir-Freundlich model, where we distinguish two distinct adsorption sites A and B:

$$q = \frac{q_{sat,A} b_A p^{v_A}}{1 + b_A p^{v_A}} + \frac{q_{sat,B} b_B p^{v_B}}{1 + b_B p^{v_B}} \quad (S1)$$

In eq (S1), the Langmuir-Freundlich parameters  $b_A, b_B$  are both temperature dependent

$$b_A = b_{A0} \exp\left(\frac{E_A}{RT}\right); \quad b_b = b_{B0} \exp\left(\frac{E_B}{RT}\right) \quad (S2)$$

In eq (S2),  $E_A, E_B$  are the energy parameters associated with sites A, and B, respectively.

The fit parameters are provided in Table S,

### Isosteric heat of adsorption

The isosteric heat of adsorption,  $Q_{st}$ , is defined as

$$Q_{st} = -RT^2 \left( \frac{\partial \ln p}{\partial T} \right)_q \quad (S3)$$

where the derivative in the right member of eq (S3) is determined at constant adsorbate loading,  $q$ . The derivative was determined by analytic differentiation of the combination of eq (S1), eq (S2), and eq (S3).

### Gas Selectivity Prediction via IAST

The experimental isotherm data for pure  $C_2H_4$ ,  $C_2H_6$  and  $C_3H_6$  were fitted using a dual Langmuir-Freundlich (L-F) model:

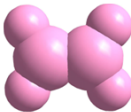
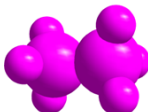
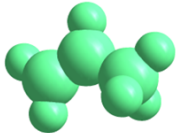
$$q = \frac{a_1 * b_1 * P^{c_1}}{1 + b_1 * P^{c_1}} + \frac{a_2 * b_2 * P^{c_2}}{1 + b_2 * P^{c_2}}$$

Where  $q$  and  $p$  are adsorbed amounts and the pressure of component  $i$ , respectively.

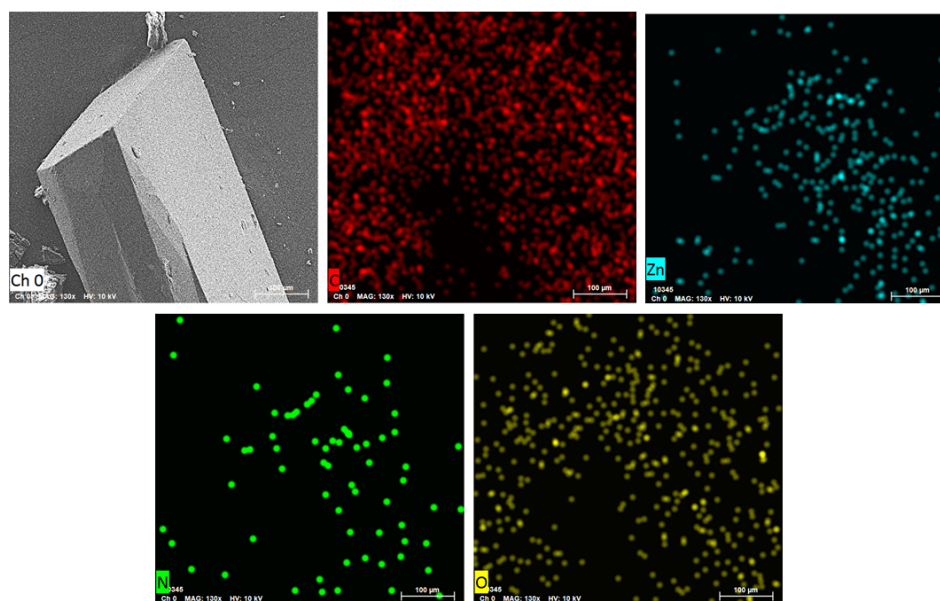
The adsorption selectivities for binary mixtures, defined by

$$S_{i/j} = \frac{x_i^* y_j}{x_j^* y_i}$$

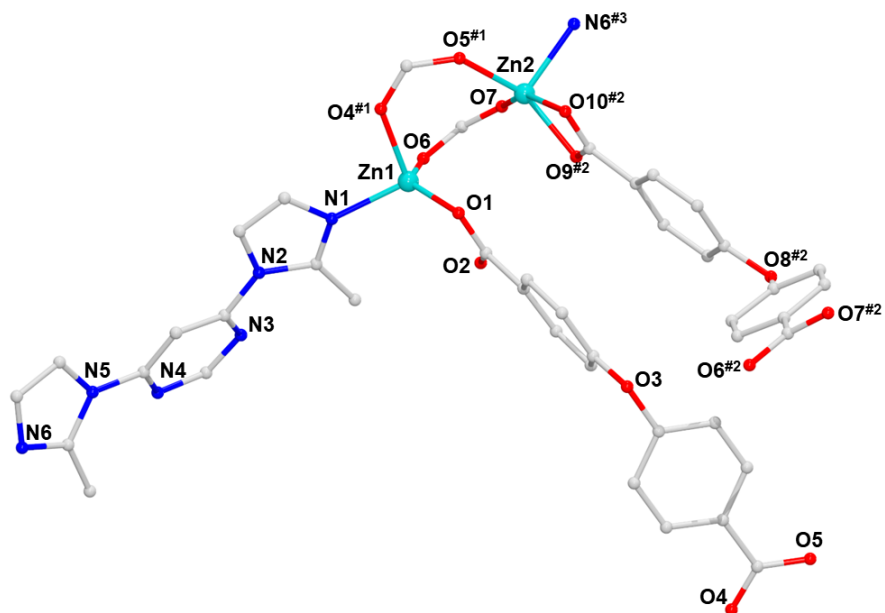
Were respectively calculated using the Ideal Adsorption Solution Theory (IAST). Where  $x_i$  is the mole fraction of component  $i$  in the adsorbed phase and  $y_i$  is the mole fraction of component  $i$  in the bulk.

Molecular structure						
Molecular formula	Molecular Dimension (Å)			Kinetic diameter (Å)	Polari × 10 <sup>-25</sup> (cm <sup>-3</sup> )	Boiling point(K)
	X	Y	Z			
C <sub>2</sub> H <sub>4</sub>	3.28	4.18	4.84	4.16	42.5	169.4
C <sub>2</sub> H <sub>6</sub>	3.81	4.08	4.82	4.44	44.3-44.7	184.6
C <sub>3</sub> H <sub>6</sub>	4.20	5.30	6.40	4.68	62.6	225.5

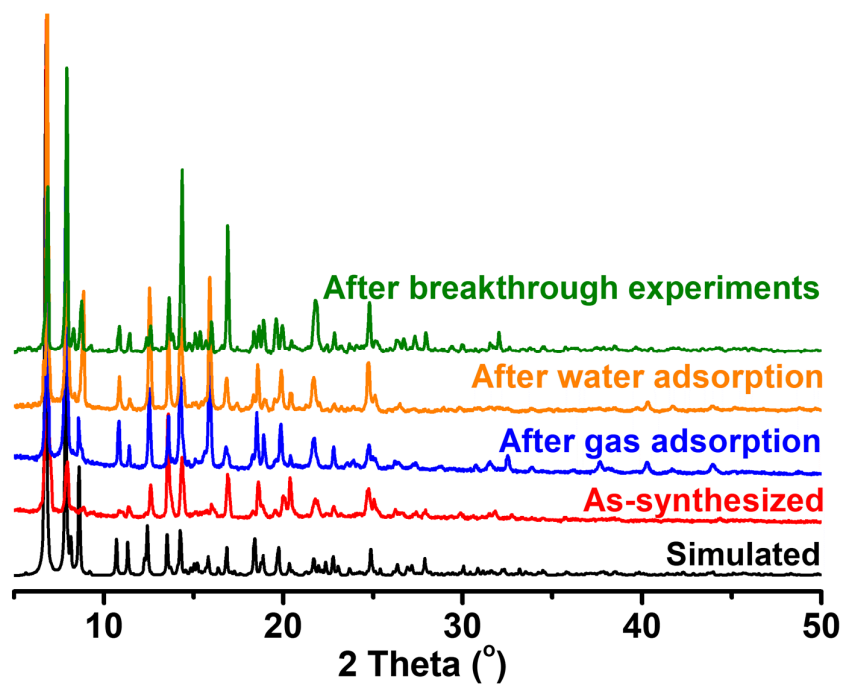
**Scheme S1.** Structures and physical properties of C<sub>2</sub>H<sub>4</sub>, C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>6</sub>.



**Figure S1.** SEM images of as-synthesized MOF, and elemental mapping from SEM-EDX showing uniform distribution of elements (mixed elements, C, Zn, N, O) in the selected area of crystal.



**Figure S2.** Coordination environment of Zn(II) ions.



**Figure S3.** PXRD patterns of **1**.

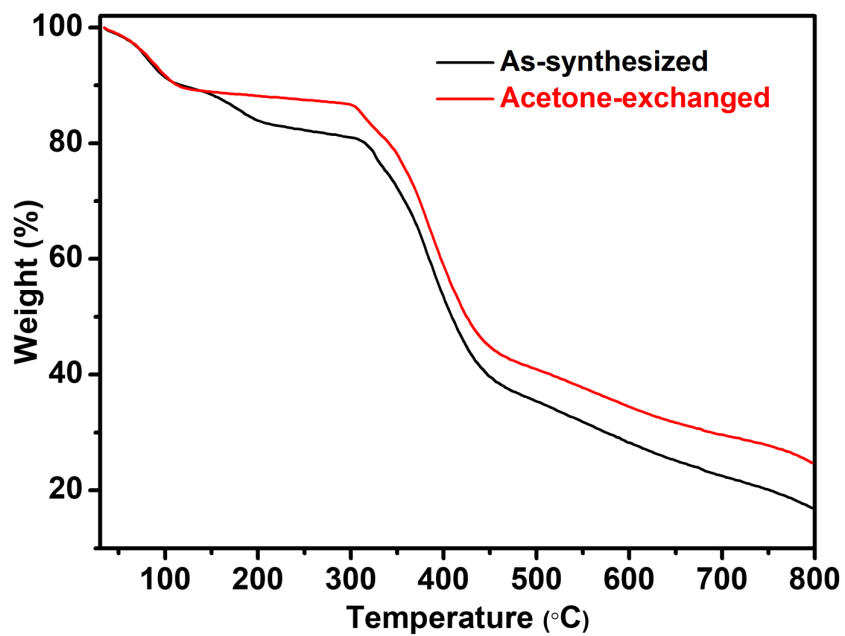


Figure S4. TGA curves of as-synthesized and acetone-exchanged samples of **1**.

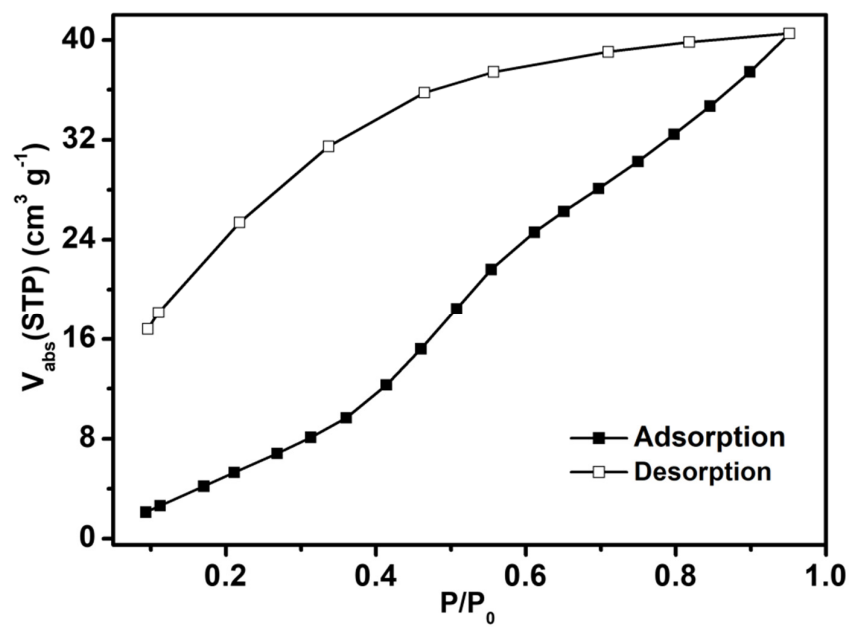
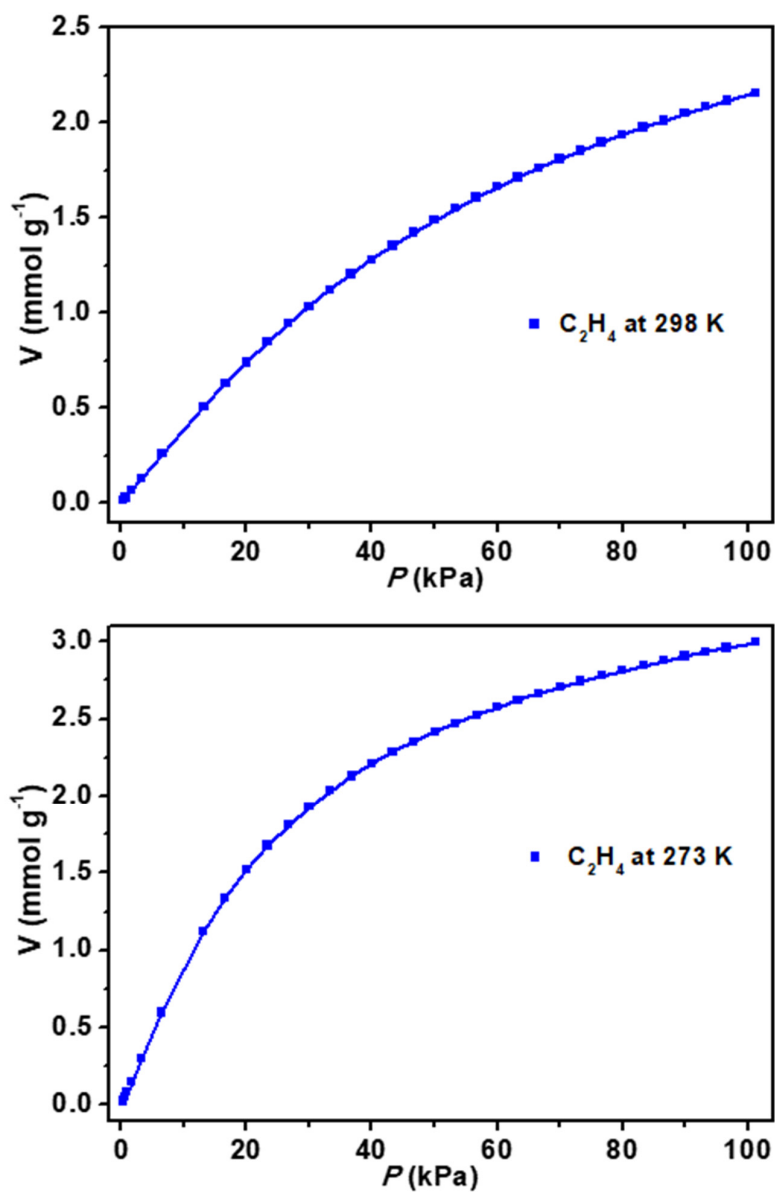
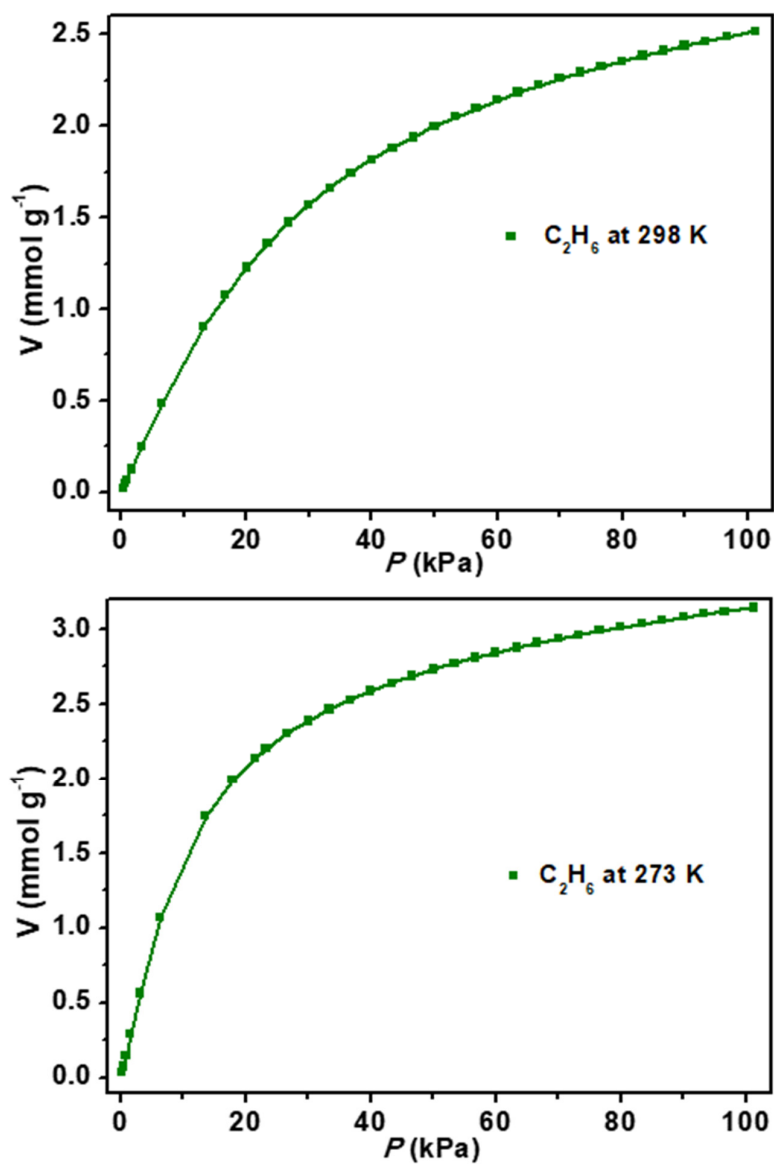


Figure S5. Water vapor adsorption and desorption isotherm of **1a** at 298 K.

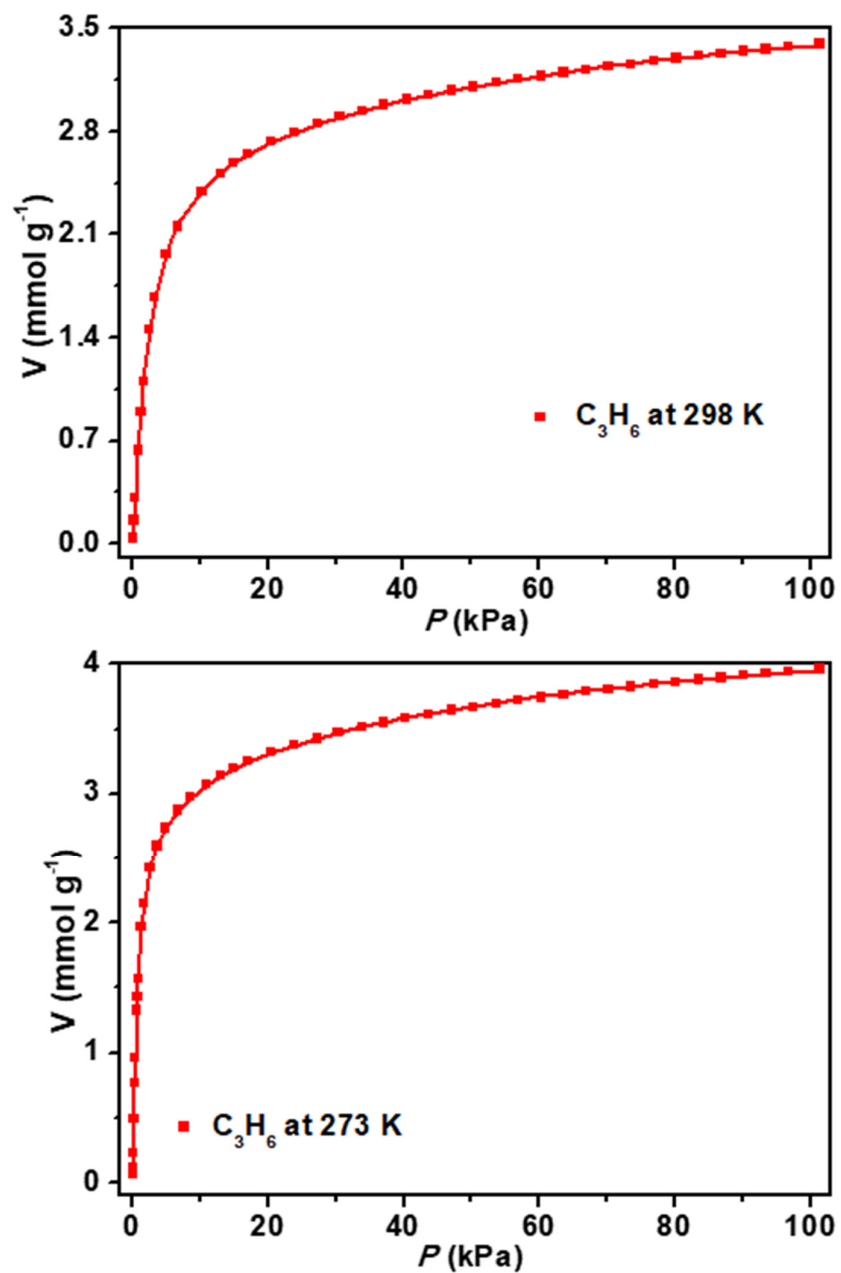


**Figure S6.** C<sub>2</sub>H<sub>4</sub> adsorption isotherms of **1a** with fitted by dual L-F model at 298 K and 273 K, 298 K:  $a_1 = 3.43606$ ,  $b_1 = 0.00921$ ,  $c_1 = 1.12333$ ,  $a_2 = 0.01901$ ,  $b_2 = 0.6084$ ,  $c_2 = 1.18178$ ,  $\text{Chi}^2 = 3.4532\text{E-}7$ ,  $R^2 = 1$ ; 273 K:  $a_1 = 3.52331$ ,  $b_1 = 0.02478$ ,  $c_1 = 1.14267$ ,  $a_2 = 0.17732$ ,  $b_2 = 2.1274\text{E-}9$ ,  $c_2 = 4.25404$ ,  $\text{Chi}^2 = 0.00002$ ,  $R^2 = 0.99999$ .





**Figure S7.** C<sub>2</sub>H<sub>6</sub> adsorption isotherms of **1a** with fitted by dual L-F model at 298 K and 273 K, 298 K:  $a_1 = 3.06171$ ,  $b_1 = 0.02323$ ,  $c_1 = 1.12105$ ,  $a_2 = 0.09758$ ,  $b_2 = 2.3658E-9$ ,  $c_2 = 4.34811$ ,  $\text{Chi}^2 = 8.1286E-6$ ,  $R^2 = 0.99999$ ; 273 K:  $a_1 = 2.97636$ ,  $b_1 = 0.05995$ ,  $c_1 = 1.20031$ ,  $a_2 = 0.8132$ ,  $b_2 = 0.00025$ ,  $c_2 = 1.73607$ ,  $\text{Chi}^2 = 0.00003$ ,  $R^2 = 0.99998$ .



**Figure S8.** C<sub>3</sub>H<sub>6</sub> adsorption isotherms of **1a** with fitted by dual L-F model at 298 K and 273 K, 298 K:  $a_1 = 2.60175$ ,  $b_1 = 0.38696$ ,  $c_1 = 1.20055$ ,  $a_2 = 1.34085$ ,  $b_2 = 0.01105$ ,  $c_2 = 1.06743$ ,  $\text{Chi}^2 = 0.00002$ ,  $R^2 = 0.99998$ ; 273 K:  $a_1 = 2.6841$ ,  $b_1 = 0.10486$ ,  $c_1 = 0.59355$ ,  $a_2 = 2.29332$ ,  $b_2 = 1.83345$ ,  $c_2 = 1.41842$ ,  $\text{Chi}^2 = 0.00004$ ,  $R^2 = 0.99997$ .

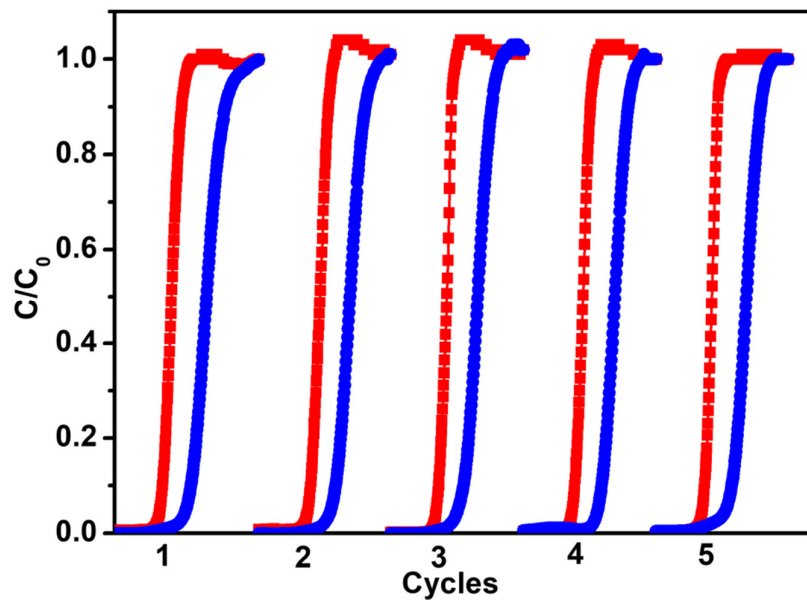


Figure S9. Cycling tests for equimolar  $C_2H_6/C_2H_4$  mixture at 298 K.

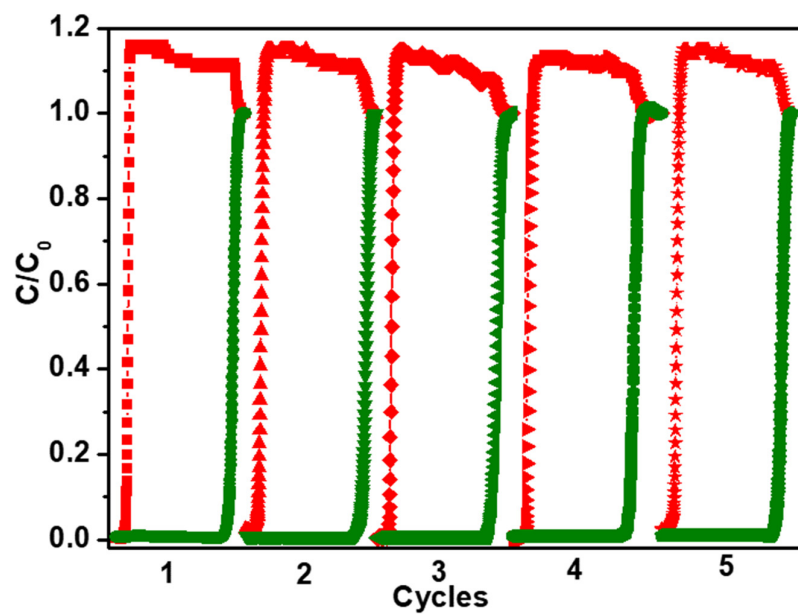


Figure S10. Cycling tests for equimolar  $C_3H_6/C_2H_4$  mixture at 298 K.

**Table S1.** Crystal Data and Structure Refinements for **1**.

Chemical formula	C <sub>40</sub> H <sub>28</sub> Zn <sub>2</sub> N <sub>6</sub> O <sub>10</sub>
Formula weight	883.42
<i>T</i> (K)	216(2)
Crystal system, Space group	Triclinic, <i>P</i> -1
<i>a</i> (Å)	13.0194(4)
<i>b</i> (Å)	14.4119(4)
<i>c</i> (Å)	16.3911(4)
$\alpha$ (°)	87.5590(10)
$\beta$ (°)	72.9610(10)
$\gamma$ (°)	65.3030(10)
<i>V</i> (Å <sup>3</sup> )	2660.26(13)
<i>Z</i>	2
<i>D</i> <sub>calcd.</sub> [g·cm <sup>-3</sup> ]	1.103
$\mu$ (mm <sup>-1</sup> )	0.950
Reflns collected/unique/ <i>R</i> <sub>int</sub>	42527/9700/0.0369
Goof	1.051
<i>R</i> <sub>1</sub> <sup>a</sup> , <i>wR</i> <sub>2</sub> <sup>b</sup> [ <i>I</i> > 2 $\sigma$ ]	<i>R</i> <sub>1</sub> = 0.0367, <i>wR</i> <sub>2</sub> = 0.1055
<i>R</i> <sub>1</sub> <sup>a</sup> , <i>wR</i> <sub>2</sub> <sup>b</sup> (all data)	<i>R</i> <sub>1</sub> = 0.0482, <i>wR</i> <sub>2</sub> = 0.1118

$${}^aR_1 = \Sigma(|F_o| - |F_c|) / \Sigma|F_o|. \quad {}^bR_2 = [\Sigma w(F_o^2 - F_c^2)^2 / \Sigma w(F_o^2)^2]^{1/2}.$$

**Table S2.** Selected bond lengths [Å] and angles [°] for **1**.

Zn(1)-O(5)	1.9501(18)	O(5)-Zn(1)-N(6)#3	99.62(8)
Zn(1)-O(7)#1	1.9801(19)	O(7)#1-Zn(1)-N(6)#3	98.29(9)
Zn(1)-O(10)#2	1.9576(18)	O(10)#2-Zn(1)-O(7)#1	103.77(8)
Zn(1)-N(6)#3	2.033(2)	O(10)#2-Zn(1)-N(6)#3	106.41(8)
Zn(2)-O(1)	1.9384(19)	O(1)-Zn(2)-O(4)#4	104.17(8)
Zn(2)-O(4)#4	1.9952(19)	O(1)-Zn(2)-O(6)	130.37(9)
Zn(2)-O(6)	1.9404(19)	O(1)-Zn(2)-N(1)	115.05(9)
Zn(2)-N(1)	2.007(2)	O(4)#4-Zn(2)-N(1)	96.21(8)
O(5)-Zn(1)-O(7)#1	106.20(8)	O(6)-Zn(2)-O(4)#4	101.75(9)
O(5)-Zn(1)-O(10)#2	136.55(8)	O(6)-Zn(2)-N(1)	103.19(9)

Symmetry codes: #1 *x*, *y*+1, *z*; #2 *x*-1, *y*+2, *z*; #3 *x*, *y*+1, *z*; #4 *x*, *y*-1, *z*; #5 *x*+1, *y*-2, *z*; #6 *x*, *y*-1, *z*+1.

**Table S3.** Dual-site Langmuir-Freundlich fits for C<sub>2</sub>H<sub>4</sub>, C<sub>2</sub>H<sub>6</sub>, and C<sub>3</sub>H<sub>6</sub>, in **1a**.

	Site A				Site B			
	$\frac{q_{A,sat}}{\text{mol kg}^{-1}}$	$\frac{b_{A,0}}{\text{Pa}^{-\nu_A}}$	$\frac{E_A}{\text{kJ mol}^{-1}}$	$\nu_A$	$\frac{q_{B,sat}}{\text{mol kg}^{-1}}$	$\frac{b_{B,0}}{\text{Pa}^{-\nu_B}}$	$\frac{E_B}{\text{kJ mol}^{-1}}$	$\nu_B$
C <sub>2</sub> H <sub>4</sub>	2.2	1.951E-09	24	0.83	2.1	7.38E-12	30	1.25
C <sub>2</sub> H <sub>6</sub>	2.2	7.684E-09	23.3	0.75	2	6.65E-12	31.5	1.3
C <sub>3</sub> H <sub>6</sub>	2.1	2.855E-13	44.5	1.44	2.6	5.45E-08	24	0.61

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