



A novel C₂H₂-selective microporous Cd-MOF for C₂H₂/C₂H₄ and C₂H₂/CO₂ separation

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ABSTRACT

Separation of acetylene (C₂H₂) from ethylene (C₂H₄) or carbon dioxide (CO₂) is a vital and difficult task in petrochemical industry. Owing to the similar physical properties, it is very challenging to synthesize target porous materials with standout separation performance. Herein, we report the utilization of an isophthalic acid linker substituted with pyridine groups to construct a novel microporous Cd-MOF [Cd₂(dpp)₂(DMF)(H₂O)]·DMF·H₂O (1). The Cd-MOF features suitable pore surfaces modified by carboxyl oxygen atoms to contact closely with C₂H₂ molecules, which leads to considerable C₂H₂ adsorption ability of 124.4 cm³ g⁻¹ at 100 kPa and 298 K, significantly higher than C₂H₄ and CO₂. Experimental breakthroughs indicate the Cd-MOF can efficiently separate C₂H₂ from C₂H₂/CO₂, C₂H₂/CH₄, and C₂H₂/C₂H₄ mixtures. The productivities of ≥ 99.996% C₂H₄ purity calculated on the basis of simulated transient breakthroughs for 1/99 C₂H₂/C₂H₄ were determined to be 27.72 and 60.77 L kg⁻¹ at 298 and 273 K, respectively. Molecular simulation demonstrates the pivotal role of the multiple interactions in the framework for C₂H₂.

1. Introduction

The separation and purification of light hydrocarbons is a crucial process for the production of clean energy and high-purity chemicals, but it is also an expensive and energy-intensive technology [1–3]. Among them, acetylene (C₂H₂) is used as important fuel gas in welding/cutting, and is also the basic feedstock to produce several commodity chemicals [4–6]. C₂H₂ mainly originates from the partial combustion of methane (CH₄) or hydrocarbon cracking process, and faces the problem presenting with by-products including carbon dioxide (CO₂) and CH₄ [7–9]. Therefore, separating C₂H₂ from CH₄ or CO₂ mixtures is of industrial importance for obtaining high purity C₂H₂ for the manufacture of value-added products. In another aspect, ethylene (C₂H₄), as the largest feedstock in petrochemical industry, inevitably contains some impurities such as C₂H₂. Trace C₂H₂ would poison the catalyst in the polymerization process of C₂H₄, which needs to be removed to an acceptable level of below about 40 ppm of C₂H₂ [10–12]. Hence, it is needed to purify C₂H₄ from C₂H₂/C₂H₄ mixtures. However, due to

similar physicochemical properties and molecular dimensions [13], the solvent extraction or cryogenic distillation techniques commonly used in industry for C₂H₂/CO₂ and C₂H₂/C₂H₄ separations are energy-intensive processes and also cause severe environmental problems. In contrast, porous material-based physical adsorption separation method provides alternatives with a low-energy and simultaneously achieves a green environmental effect [14–16]. The development of new kinds of porous media is highly desirable to target excellent performance of gas separation.

Metal-organic frameworks (MOFs), self-assembled by organic connectors with metal clusters/ions, are the state-of-the-art porous solid materials because of their powerful predictability as well as tunability on pore size/shape [17–20]. In the past two decades, MOFs have aroused widespread attention in catalysis [21], magnetism [22], drug delivery [23], conductivity [24], and gas separation [25]. Especially for gas separation, MOFs exhibit advantages in accurately controlling the pore size and pore chemical environment through reticular chemistry and crystal engineering strategies [26–30]. Targeted design and synthesis of

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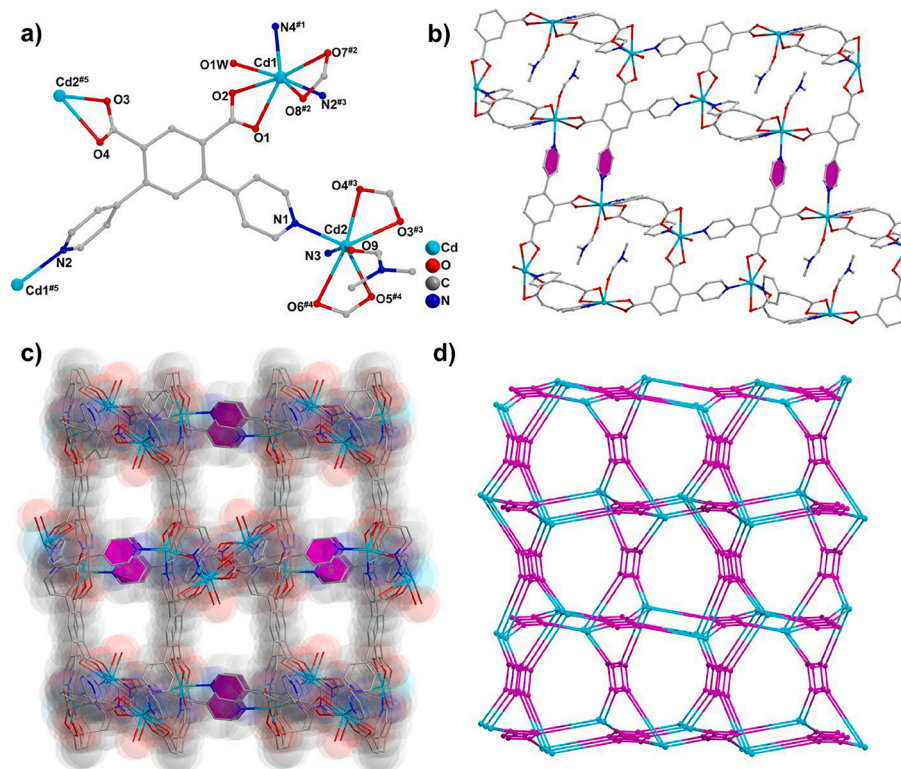


Fig. 1. a) Coordination environment of Cd^{2+} ions; b) 2D corrugated layers on the ac plane; c) 3D porous framework; d) topological net. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

MOF materials according to the differences in molecular size and polarity of the components to be separated is of paramount importance to achieve high gas separation efficiency [31–32]. For example, for gas mixtures with different molecular sizes, the size-exclusion strategy is effective [33,34]. However, for C_2H_2 and CO_2 with similar molecular sizes, additional functional sites (such as Lewis N sites and open metal sites) need to be introduced to enhance the interactions between the framework and one of the gases to achieve separation [35–37]. It is worth noting that the adsorption intensity of active sites should be moderate, and strong strength will result in higher desorption energy, increasing energy consumption in regeneration [38–39]. Therefore, accessible suitable binding sites combining suitable pore sizes in MOFs could significantly enhance the adsorption and separation ability toward C_2H_2 .

It is known that pyridylcarboxylate ligands usually form structurally diverse MOFs, where the carboxylic acid groups have multiple coordination modes, while pyridine group can form stable coordination bonds with M(II) transition metal ions. In this work, we employ a pyridinecarboxylic ligand, 4,6-di(pyridin-4-yl) isophthalic (H_2dpip), for MOF construction. The ligand is that the 4- and 6- positions of the benzene ring in isophthalic acid are replaced by pyridine groups, which can coordinate with the metal ions in four directions that tend to afford open frameworks, which is conducive to the transfer of adsorbates in the process of adsorption and separation. Herein, we developed a novel Cd-MOF, $[\text{Cd}_2(\text{dpip})_2(\text{DMF})(\text{H}_2\text{O})]\cdot\text{DMF}\cdot\text{H}_2\text{O}$ (**1**) via reacting H_2dpip and Cd^{2+} ions under solvothermal conditions. The Cd-MOF has channels modified by carboxyl oxygen atoms, which not only shows higher adsorption capacity for C_2H_2 compared to CO_2 , C_2H_4 , and CH_4 , but also displays dynamic separation for $\text{C}_2\text{H}_2/\text{CO}_2$, $\text{C}_2\text{H}_2/\text{C}_2\text{H}_4$, and $\text{C}_2\text{H}_2/\text{CH}_4$ mixtures.

2. Experimental

2.1. Materials and methods

The details of this section were presented in [Supporting Information](#).

2.2. Synthesis of $[\text{Cd}_2(\text{dpip})_2(\text{DMF})(\text{H}_2\text{O})]\cdot\text{DMF}\cdot\text{H}_2\text{O}$ (**1**)

The mixture of $\text{Cd}(\text{NO}_3)_2\cdot 4\text{H}_2\text{O}$ (0.0308 g, 0.10 mmol) and H_2dpip (0.0154 g, 0.05 mmol) in DMF (4 mL), ethanol (1 mL), and trifluoroacetic acid (0.05 mL) was capped in a 25 mL vessel and heated at $115\text{ }^\circ\text{C}$ for 72 h. After cooling to room temperature, colourless block crystals were obtained (yield: 53%, based on H_2dpip). Anal. Calcd for $\text{C}_{42}\text{H}_{38}\text{Cd}_2\text{N}_6\text{O}_{12}$: C, 48.34; H, 3.67; N, 8.05%. Found: C, 48.15; H, 3.98; N, 8.01%.

3. Results and discussion

3.1. Structural analysis

The Cd-MOF **1** crystallizes in the monoclinic $P2_1/n$ space group, and includes two Cd(II) ions, two deprotonated dpip ligands, one coordinated water molecule, and one coordinated DMF molecule in the asymmetric unit. The Cd^{2+} center is seven-coordinated in a pentagonal bipyramid geometry formed by four O atoms from two carboxylate linkers ($\text{Cd}-\text{O} = 2.270\text{--}2.603\text{ \AA}$), two pyridyl N atoms ($\text{Cd}-\text{N} = 2.277\text{--}2.378\text{ \AA}$), and one O atom of coordinated solvent molecule (Fig. 1a). Four Cd(II) ions are connected by dpip ligands to form ring-like SBUs, and which are connected by pyridyl rings in ligands to form a two-dimensional (2D) waved layer parallel to the ac plane (Fig. 1b). The adjacent layers are further cross-linked by dpip ligands with Cd(II) ions, forming a porous 3D framework (Fig. 1c). The pore size distribution is estimated to be about $15.1\text{--}16.5\text{ \AA}$ according to the PoreBlazer 4.0 (Figure S1) [40]. The void of framework calculated using a probe of 1.2

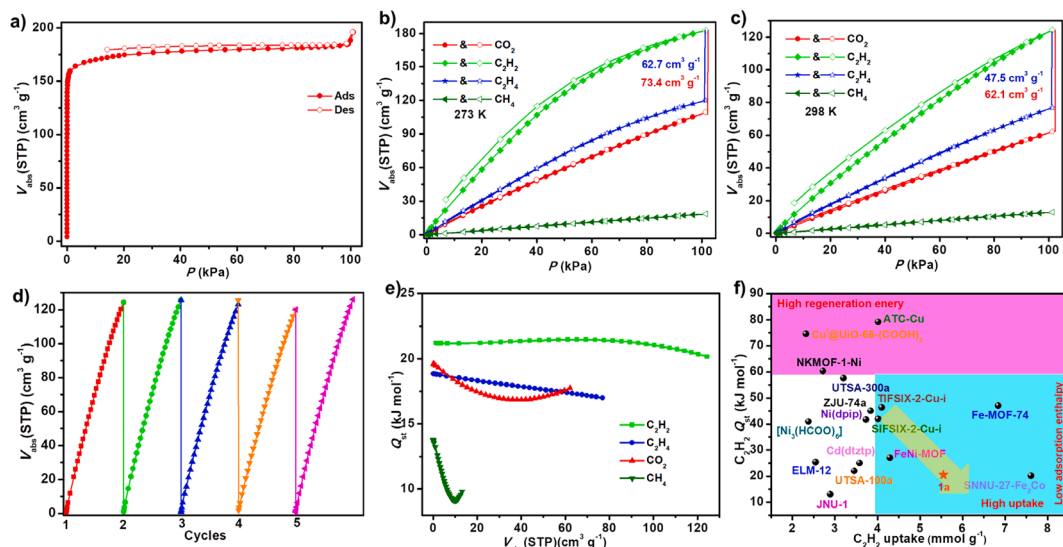


Fig. 2. a) Adsorption isotherm of N_2 at 77 K; b) and c) adsorption isotherms of CO_2 , C_2H_2 , C_2H_4 , and CH_4 for **1a** at 273 and 298 K; d) cycled C_2H_2 adsorption of **1a** at 298 K; e) adsorption heats of **1a**; f) comparison of C_2H_2 uptakes and adsorption heats between **1a** and other MOFs at 298 K.

Å by PLATON program is 55.6% (including coordinated solvent molecules), showing high porosity. The framework possesses a (4,4)-connected ($4^2.8^4$)-*pts* topology, while both Cd(II) ions and ligands are regarded as 4-connection nodes (Fig. 1d).

3.2. PXRD and TGA

The sample purity was verified by the consistency of PXRD between the measured and simulated (Figure S2). TGA shows an 8.8% weight reduction before 120 °C, resulting from the decrease of free water and DMF molecules (calcd. 8.7%) [41,42]. Next, the weight loss of 8.3% from 120 to 250 °C corresponds to the loss of coordinated water and DMF molecules (calcd. 8.7%) (Figure S3). PXRD of **1** shows the thermal stable temperature at about 300 °C as well as the good stability in common organic solvents (acetone, toluene, methanol, acetonitrile, DMF, DMSO), water and different aqueous solutions with pH = 3–10 for 1 day (Figure S4). The desolvated sample **1a** was gained by soaking **1** in CH_2Cl_2 for 48 h, and then heating at 150 °C for 4 h under vacuum, as characterized by the PXRD and TGA analyses.

3.3. Gas adsorption

The permanent porosity of **1a** was investigated by nitrogen (N_2) adsorption experiment at 77 K. As illustrated in Fig. 2a, it exhibits a fully reversible typical type-I adsorption behavior, with the maximum uptake of $195.9 \text{ cm}^3 \text{ g}^{-1}$ at 100 kPa. The fitted Brunauer-Emmett-Teller and

Langmuir surface areas are 698 and $800 \text{ m}^2 \text{ g}^{-1}$ (Figure S5). Moreover, the distribution of pore size localizes in the range of 9–12 Å based on the Horvath-Kawazoe model (Figure S6), matching well with the calculated pore dimensions derived from crystal structure.

The high porosity and bare carboxyl O atoms in the pores of **1a** prompted us to explore C_2H_2 storage and separation behaviors. Single-component equilibrium adsorption isotherms of C_2H_2 , C_2H_4 , CO_2 , and CH_4 on **1a** were recorded at 273, 298 and 313 K (Fig. 2b, 2c, S7 and Table S3), which show a higher uptake for C_2H_2 than other gases. The single-component isotherms revealed that **1a** adsorbs $124.4/182.8 \text{ cm}^3 \text{ g}^{-1}$ of C_2H_2 , $76.8/120.0 \text{ cm}^3 \text{ g}^{-1}$ of C_2H_4 , $62.2/109.4 \text{ cm}^3 \text{ g}^{-1}$ of CO_2 , and $13.1/18.7 \text{ cm}^3 \text{ g}^{-1}$ of CH_4 , and the uptake ratio of C_2H_2/CO_2 , C_2H_2/C_2H_4 , and C_2H_2/CH_4 reached 2/1.67, 1.62/1.52, and 9.5/9.8 at 298 and 273 K under 100 kPa, respectively. Although the C_2H_2 uptake value of **1a** at 298 K is under than some benchmark MOFs like MIL-160 ($191 \text{ cm}^3 \text{ g}^{-1}$) [43], MOF-505 ($148 \text{ cm}^3 \text{ g}^{-1}$) [44], and SNU-27-Fe ($182 \text{ cm}^3 \text{ g}^{-1}$) [16], but is superior or comparable with many famous MOFs, such as JNU-2 ($103 \text{ cm}^3 \text{ g}^{-1}$) [8], NUM-9 ($103 \text{ cm}^3 \text{ g}^{-1}$) [45], UTSA-100a ($95 \text{ cm}^3 \text{ g}^{-1}$) [46], NPU-1 ($114 \text{ cm}^3 \text{ g}^{-1}$) [47], and UPC-60-Cl ($100 \text{ cm}^3 \text{ g}^{-1}$) [48], suggesting promising candidate for C_2H_2 storage. Remarkably, the uptake ratio of C_2H_2/CO_2 at 298 K is superior to some reported MOFs, such as UTSA-74a (1.52 at 298 K) [49], FJU-6-TATB (1.89 at 296 K) [50], $[Ni(tzba)_{0.5}(F)(bpy)]$ (1.45 at 298 K) [51], and FJI-H8-Et (1.75 at 298 K) [35], indicating great separation potential application to address this challenging gas mixtures. Five adsorption–desorption cycle experiments for C_2H_2 reflect great repeatability and regeneration performance

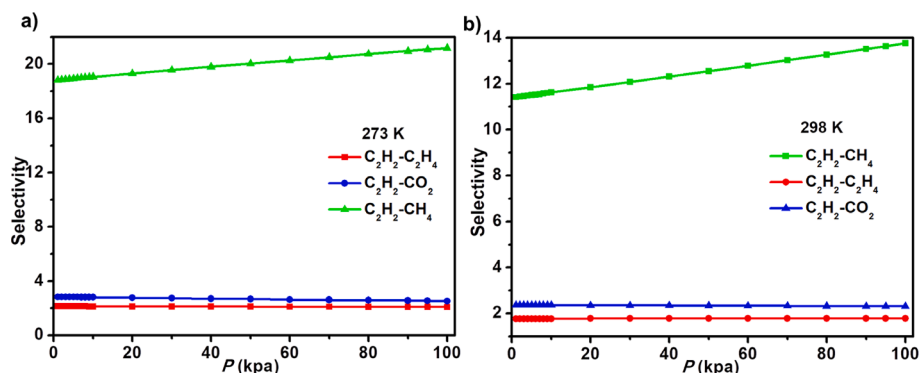


Fig. 3. a) and b) IAST predicted selectivities for equimolar $C_2H_2-CO_2$, $C_2H_2-CH_4$, and $C_2H_2-C_2H_4$ mixtures at 273 and 298 K.

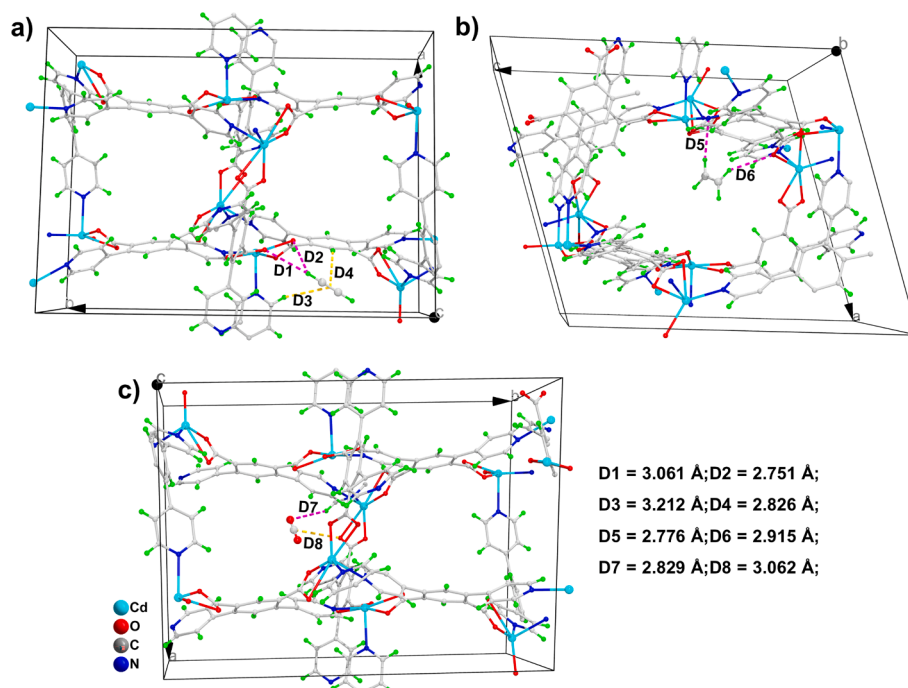


Fig. 4. Adsorption locus in **1a** for C_2H_2 (a), C_2H_4 (b), and CO_2 (c). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

of **1a** (Fig. 2d). The adsorption enthalpy (Q_{st}) was used to evaluate the affinity of **1a** for gas molecules (Figure S8). A notable zero coverage Q_{st} value of **1a** for C_2H_2 (21.2 kJ mol^{-1}) that is higher than those of CH_4 (13.8 kJ mol^{-1}), C_2H_4 (18.8 kJ mol^{-1}), and CO_2 (19.6 kJ mol^{-1}) was indeed observed, illustrating the stronger affinity of **1a** for C_2H_2 than CH_4 , C_2H_4 , and CO_2 (Fig. 2e). Notably, the C_2H_2 Q_{st} value in **1a** is much lower than most MOF materials, such as Fe-MOF-74 [52], ATC-Cu [53], and TIFSIX-2-Cu-I [32], which indicate a less energy consume for C_2H_2

production (Fig. 2f and Table S4).

3.4. Gas selectivity

These differences in absorption uptakes and zero coverage Q_{st} for C_2H_2 and other gases indicate a promising material of **1a** for separating C_2H_2 from C_2H_2 -containing mixtures. To evaluate the separation performance, the ideal adsorbed solution theory (IAST) method was

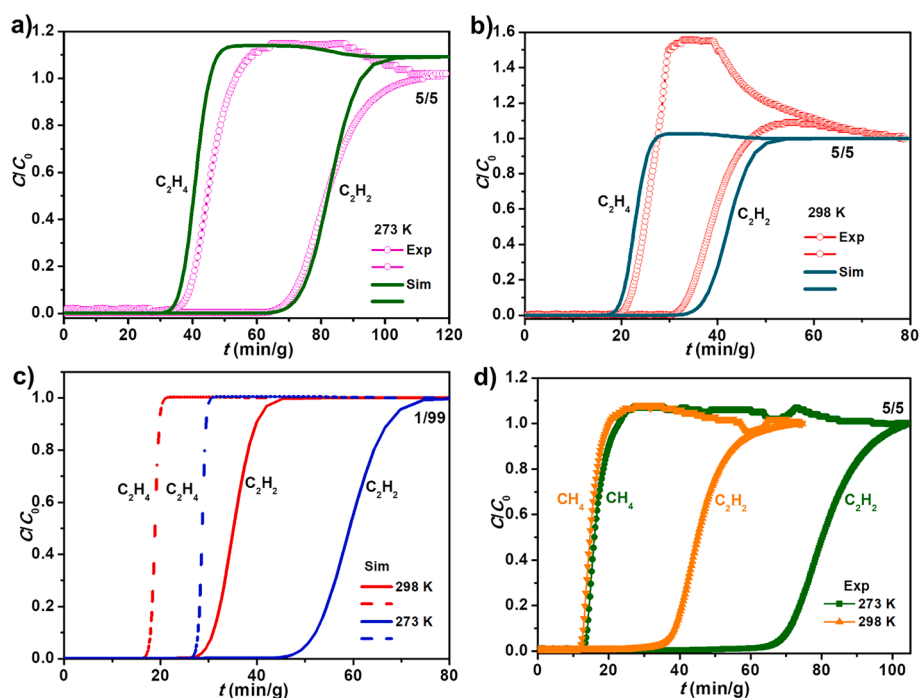


Fig. 5. a) and b) Comparison of the experimental and simulated breakthrough curves for $C_2H_2/C_2H_4/Ar$ (5:5:90, v/v/v) separation at 273 and 298 K; c) simulated breakthrough curves for C_2H_2/C_2H_4 (1:99) separation at 273 and 298 K; d) experimental breakthrough curves for $C_2H_2/CH_4/Ar$ (5:5:90, v/v/v) separation at 273 and 298 K.

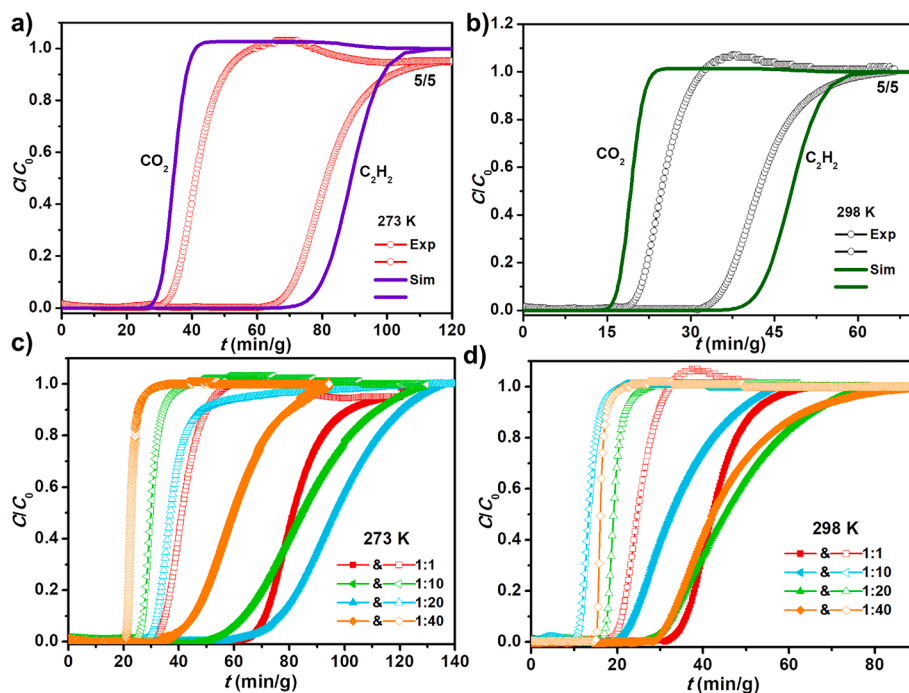


Fig. 6. a) and b) Comparison of experimental and simulated breakthrough curves for $C_2H_2/CO_2/Ar$ (5:5:90, v/v/v) mixtures at 273 and 298 K; c) and d) breakthrough experiments for C_2H_2/CO_2 mixtures at different concentrations at 273 and 298 K.

employed to calculate the C_2H_2 selectivities through fitting sorption isotherms based on 1-site Langmuir model (Figure S9-S10). As depicted in Fig. 3a and 3b, the selectivities of equimolar $C_2H_2-CH_4$ for **1a** are 21.2 and 13.8 under 100 kPa at 273 and 298 K, respectively, which are comparable with [Ni(dpip)] (16.6) [54], SNNU-17 (20.3) [55], and ZJU-16a (17.7) [56]. The selectivities of equimolar $C_2H_2-C_2H_4$ and $C_2H_2-CO_2$ mixtures for **1a** are 2.1 and 1.8, and 2.5 and 2.3 under 100 kPa at 273 and 298 K, respectively. The C_2H_2/CO_2 selectivity value of 2.5 is better than the known materials M'MOF-2a (1.89) [57], Zn-MOF-74a (2) [49], and SNNU-27-Fe (2) [16]. The high C_2H_2 adsorption capacity and separation selectivity endow **1a** with the separation potential for C_2H_2 -containing mixtures.

3.5. Molecular simulations

To in-depth understand the adsorption mechanism, Grand Canonical Monte Carlo simulations were conducted to explore the interplay among MOF and gas molecules. The calculated preferential adsorption locations for C_2H_2 , C_2H_4 , and CO_2 were shown in Fig. 4a-c. For C_2H_2 molecule, one $-CH$ unit that lies close the accessible carboxyl O atoms of two independent ligands forms two $C-H\cdots O$ hydrogen bonds. Meantime, abundant π electron cloud of the triple bond group in C_2H_2 makes it form $C-H\cdots\pi$ contacts with the pyridine and benzene rings. Differently, C_2H_4 only forms one $C-H\cdots N$ and one $C-H\cdots O$ hydrogen bonds with the pyridine N atom and carboxyl O atom. For CO_2 , its O and C atoms form one $C-H\cdots O$ and $C\cdots O$ interactions ($C\cdots O = 3.062 \text{ \AA}$) with the $-CH$ fragment in the pyridine ring and carboxyl O atom, respectively (Fig. 4c). In addition, the electrostatic potential (ESP) of **1a** mapped onto the Connolly surface results reveal that the negative potential regions of the triple bond group in C_2H_2 molecule is mainly stabilized by the attractive interactions with positive potential region near hydrogen atom of pyridine group (Figure S11). It is worth noting that since the open Cd^{2+} sites do not point into the channel, no $M\cdots\pi$ interaction is formed between the gas molecules and the framework, which is consistent with the low Q_{st} value observed experimentally. The simulations reveal the important dedication of the accessible adsorption locus including pyridine ring and carboxyl groups, which form

multiform of supramolecular interactions between the framework and gas molecules. The more and stronger interactions for C_2H_2 lead to the most loading for C_2H_2 compared to CO_2 and C_2H_4 .

3.6. Breakthrough separation

To validate the separation feasibility of **1a** in the packed column, transient breakthrough simulations for $C_2H_2/C_2H_4/Ar$ (5:5:90, v/v/v) and C_2H_2/C_2H_4 (1/99, v/v) mixtures were done at 273/298 K under 100 kPa using the methodology described by Krishna (see also Supporting Information) [58,59]. The simulated curves in Fig. 5a,b display that the C_2H_2/C_2H_4 mixtures were efficiently separated by **1a**. The experimental breakthrough of gas mixtures (Ar as carrier gas) with a total rate of 5 mL min^{-1} flowed over a fixed-bed of **1a** sample at 298 and 273 K under 100 kPa were then tested. The experimental results were highly close to the simulated results at both temperatures; wherein C_2H_4 were first detected at the outlet while C_2H_2 was retained in bed for with longer times (Fig. 5a,b). And the time intervals are 11.4 and 28.6 min g^{-1} at 298 and 273 K, respectively. In addition, based on the experimental breakthrough curves, the amounts of C_2H_2 adsorbed by **1a** during the dynamic separation process can be calculated as 12.1 and $22.7 \text{ cm}^3 \text{ g}^{-1}$ from C_2H_2/C_2H_4 mixtures at 298 and 273 K. Furthermore, to thoroughly evaluate the desired target purity for C_2H_4 which contains less than about 40 ppm of C_2H_2 can be met with **1a**, we conducted more breakthrough simulations for 1/99 C_2H_2/C_2H_4 mixtures under 100 kPa pressure at 273 and 298 K (Fig. 5c). The productivities of $\geq 99.996\%$ C_2H_4 purity were determined to be 27.72 and 60.77 L kg^{-1} in a single adsorption process, which are lower than the values of ZU-62-Ni (201.07 L kg^{-1}) [60], M'MOF-3a (144.52 L kg^{-1}) [61], UTSA-220 (75.24 L kg^{-1}) [62]. In addition, the experimental breakthrough of $C_2H_2/CH_4/Ar$ (5:5:90, v/v/v) at 273 and 298 K were also conducted (Fig. 5d). The results display a highly efficient separation of C_2H_2 from the equimolar C_2H_2/CH_4 mixture can be realized by **1a** with the amounts of adsorbed C_2H_2 to be 12.5 and $22.2 \text{ cm}^3 \text{ g}^{-1}$ at 298 and 273 K, respectively.

Furthermore, breakthrough experiments were also implemented to validate the separation ability for C_2H_2/CO_2 mixtures. For the ratios of 1:1 mixture, Fig. 6a reveals that CO_2 breakthrough occurred

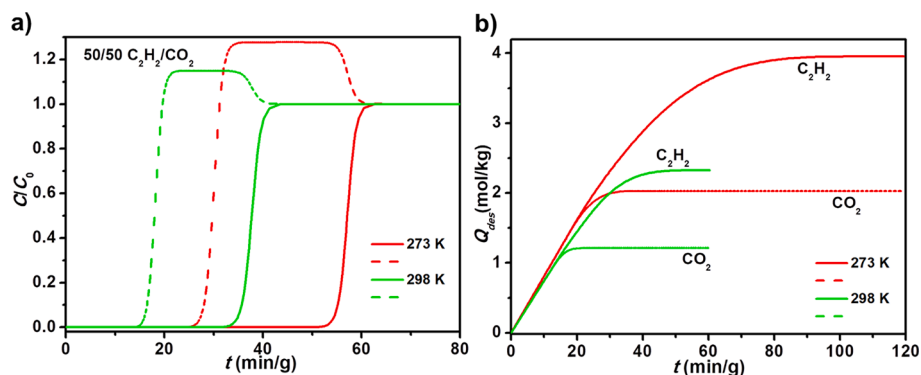


Fig. 7. a) Simulated breakthrough curves for 50/50 C₂H₂/CO₂ mixtures; b) cumulative moles of C₂H₂ and CO₂ recovered during blowdown.

significantly earlier in comparison to C₂H₂ (18.7 versus 32.6 min g⁻¹ for 298 K and 31.5 versus 62.8 min g⁻¹ for 273 K), revealing the stronger interactions between the framework and C₂H₂ molecules. During this process, the adsorbed C₂H₂ amounts were calculated to be 22.4 and 13.2 cm³ g⁻¹ at 273 and 298 K, respectively, however, the co-adsorbed CO₂ amount were significantly low (11.1 and 6.5 cm³ g⁻¹). On the basis of dynamic breakthrough experiments, the separation factor for **1a** reaches up to 2.0 at 273 and 298 K, which are comparable with the reports in best-performing FJU-6-TATB [50], FeNi-M⁺MOF [63], NKMOF-1-Ni [64], and FJU-90 [65]. For other ratios of C₂H₂-CO₂ mixtures (1:10, 1:20, and 1:40), **1a** also displays great separation ability (Fig. 6c and 6d). Cycling breakthrough experiments reveal the excellent reusability and regeneration of **1a**, which shows fully reproducible at least five consecutive adsorption-desorption runs through sweeping by Ar for 30 min at 313 K (Figure S12). In addition, the PXRD pattern and N₂ adsorption of the samples collected before and after the gas adsorption tests and breakthrough experiments are consistent (Figure S2 and Figure S13), verifying the framework stability.

Next, the transient breakthrough simulations for 50/50 C₂H₂/CO₂ mixtures for both the adsorption and desorption phase were also performed to validate the feasibility of **1a** for C₂H₂ purification from equimolar C₂H₂/CO₂ mixtures (Fig. 7a,b). The desorption cycle was simulated by applying deep vacuum as described by Wang et al [66]. The results in Fig. 7 show that **1a** can efficiently separate this mixtures with 2.32 and 3.95 mol kg⁻¹ of C₂H₂ productivity at 298 and 273 K, respectively, which are comparable with the benchmark MOFs, such as ZNU-1 (2.9 mol kg⁻¹ at 298 K) and BSF-3 (2.44 mol kg⁻¹ at 298 K), and greatly exceeds the reported top-performance adsorbents, including DICRO-4-Cu-i (1.26 mol kg⁻¹ at 298 K), MUF-17 (1.79 mol kg⁻¹ at 298 K), and BSF-1 (0.96 mol kg⁻¹ at 298 K) [67].

4. Conclusions

In conclusion, a novel Cd-MOF was constructed by solvothermal method toward C₂H₂-related separation. The Cd-MOF features suitable pores surfaces modified by carboxylic acid oxygen atoms, affording ultrahigh C₂H₂ adsorption uptake of 124.4 cm³ g⁻¹, which is significantly higher than C₂H₄ and CO₂ under ambient conditions. Breakthrough experiments confirm the effective separation of C₂H₂ from C₂H₂-C₂H₄, C₂H₂-CH₄, and C₂H₂-CO₂ mixtures by Cd-MOF. The simulated transient breakthroughs for 1/99 C₂H₂/C₂H₄ mixtures demonstrate that Cd-MOF can yield high-purity (>99.996%) C₂H₄ with high productivities of 27.72 at 298 K and 60.77 L kg⁻¹ at 273 K. Furthermore, together with the low heat of adsorption (20.8 kJ mol⁻¹) and cycle tests of sorption isotherms and breakthrough curves, it indicates a low regeneration energy of Cd-MOF for during the process of C₂H₂ separation. This work provides a novel Cd-MOF to achieve excellent C₂H₂-related separation performance to push industrial implementation of MOF materials for some important gas separations.

CRediT authorship contribution statement

Yong-Zhi Li: Writing – original draft, Funding acquisition. **Rajamani Krishna:** Formal analysis, Software. **Fan Xu:** Formal analysis. **Wan-Fang Zhang:** Software. **Yanwei Sui:** Formal analysis, Funding acquisition. **Lei Hou:** Writing – review & editing, Supervision. **Yao-Yu Wang:** Formal analysis, Resources. **Zhonghua Zhu:** Formal analysis.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data availability

Data will be made available on request.

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Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.seppur.2022.122678>.

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Supporting Information

A novel C₂H₂-selective microporous Cd-MOF for C₂H₂/C₂H₄ and C₂H₂/CO₂ separation

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X-Ray Crystallography

A Bruker Smart Apex II CCD detector was used to collect the single crystal data at 189(2) K using Mo K α radiation ($\lambda = 0.71073 \text{ \AA}$). The structure was solved by direct methods and refined by full-matrix least-squares refinement based on F^2 with the SHELXTL program. The non-hydrogen atoms were refined anisotropically with the hydrogen atoms added at their geometrically ideal positions and refined isotropically. As the disordered solvent H₂O molecules in the structure cannot be located, the SQUEEZE routine of Platon program was applied in refining. The formula of complex was got by the single crystal analysis together with elemental microanalyses and TGA data. Relevant crystallographic results are listed in Table S1. Selected bond lengths and angles are provided in Table S2.

N₂ Sorption Isotherm

Before gas sorption experiments, All the as-synthesized samples were immersed in CH₂Cl₂ for 3 days, during which the solvent was decanted and freshly replenished three times a day. All the samples were activated under vacuum at 423 K for 4 hours. Gas sorption measurements were then conducted using a Micrometrics ASAP 2020M gas adsorption analyzer.

Breakthrough Experiments

The breakthrough experiment was performed on the Quantachrome dynaSorb BT equipments at 298 K and 1 bar with an equal volume of mixed gas (gas A: gas B: Ar = 5% : 5% : 90%, Ar as the carrier gas, flow rate = 5 mL min⁻¹). The activated **1a** (0.85 g) was filled into a packed column of 4.2×80 mm, and then the packed column was washed with Ar at a rate of 7 mL min⁻¹ at 353 K for 35 minutes to further activate the samples. Between two breakthrough experiments, the adsorbent was regenerated by Ar flow of 7 mL min⁻¹ for 35 min at 353 K to guarantee a complete removal of the adsorbed gases.

GCMC Simulation

Grand canonical Monte Carlo (GCMC) simulations were performed for the gas adsorption in the framework by the Sorption module of Material Studio (Accelrys. Materials Studio Getting Started, release 5.0). The framework was considered to be rigid, and the optimized gases were used. The atom charges and bond lengths for the gas molecules are as follows:

CO₂, C = 0.576000e, O = -0.288000e, C-O = 1.168 \AA .

C₂H₂, C = -0.238847 e, H = 0.238847 e, C-C = 1.198 Å, C-H = 1.065 Å.

C₂H₄, C = -0.295071 e, H = 0.147536 e, C-C = 1.334 Å, C-H = 1.090 Å.

The partial charges for atoms of the framework were derived from QEq method and QEq neutral 1.0 parameter. One unit cell was used during the simulations. The interaction energies between the gas molecules and framework were computed through the Coulomb and Lennard-Jones 6-12 (LJ) potentials. All parameters for the atoms were modeled with the universal force field (UFF) embedded in the MS modeling package. A cutoff distance of 12.5 Å was used for LJ interactions, and the Coulombic interactions were calculated by using Ewald summation. For each run, the 3×10^6 maximum loading steps, 3×10^6 production steps were employed.

Fitting Adsorption Heat of Pure Component Isotherms

$$\ln P = \ln N + 1/T \sum_{i=0}^m a_i N^i + \sum_{i=0}^n b_i N^i \quad Q_{st} = -R \sum_{i=0}^m a_i N^i$$

The above virial expression was used to fit the combined isotherm data for **1a** at 273.15 and 298 K, where P is the pressure, N is the adsorbed amount, T is the temperature, a_i and b_i are virial coefficients, and m and N are the number of coefficients used to describe the isotherms. Q_{st} is the coverage-dependent enthalpy of adsorption and R is the universal gas constant.

Gas Selectivity Prediction via IAST

The experimental isotherm data for pure C₂H₂, C₂H₄, CO₂ and CH₄ were fitted using a single-site Langmuir equation model:

$$q = a \frac{b * p^{1-c}}{1 + b * p^{1-c}}$$

Where q and p are adsorbed amounts and the pressure of component i, respectively.

The adsorption selectivities for binary mixtures of C₂H₂/C₂H₄, C₂H₂/CO₂ and C₂H₂/CH₄, defined by

$$S_{i/j} = \frac{x_i^* y_j}{x_j^* y_i}$$

were respectively calculated using the Ideal Adsorption Solution Theory (IAST). Where x_i is

the mole fraction of component i in the adsorbed phase and y_i is the mole fraction of component i in the bulk.

Transient breakthrough simulations

Transient breakthrough simulations were carried out for the same set of operating conditions as in the experimental data sets, using the methodology described in earlier publications.[S1-S5] Breakthrough simulations were conducted for the following set of conditions. In these simulations, intra-crystalline diffusion influences are ignored. In all the simulations: length of packed bed, $L = 8$ cm; inner diameter of packed bed, $d = 4.2$ mm; mass of **1a** = 0.41 g; framework density for **1a** = 0.89 g cm⁻³; total volumetric flow rate of gas mixture at inlet = 5.0 mL min⁻¹ (298 K and 273 K, 100 kPa).

Breakthrough simulations were also performed for 1/99 C₂H₂/C₂H₄ mixtures at 100 kPa total pressure (without inert Ar); the productivities of 99.996% pure C₂H₄ were determined; these are expressed in the units of L per kg of MOF.

Breakthrough simulations were also performed for 50/50 C₂H₂/CO₂ mixtures (without inert Ar) for both the adsorption and desorption phase. The desorption cycle was simulated by applying deep vacuum as described by Wang et al.[S6] The desorption cycle is initiated just before C₂H₂ breaks through in the adsorption cycle.

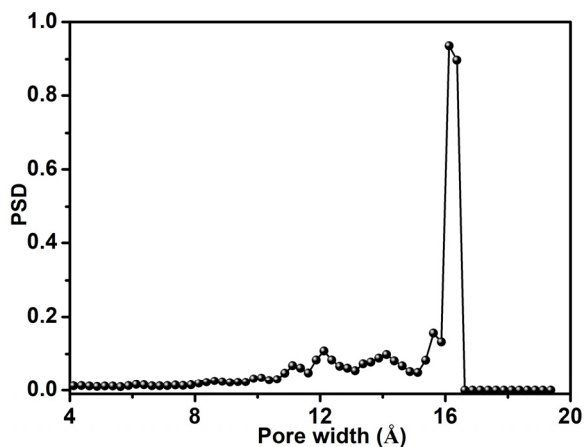


Figure S1. Pore distribution curve calculated from poreblazer software

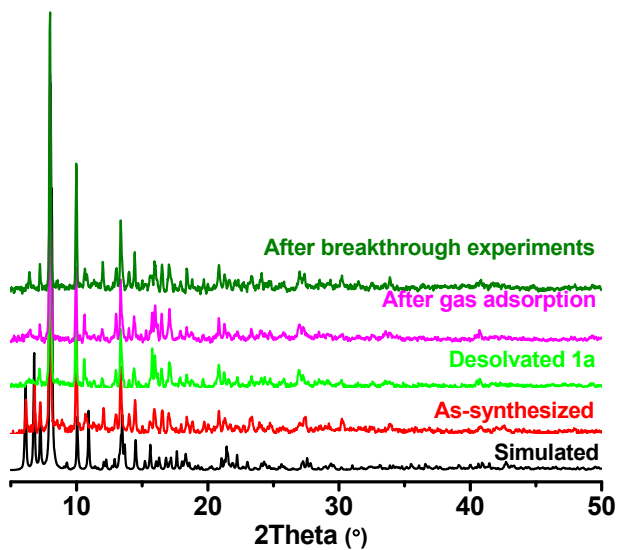


Figure S2. PXRD patterns of complex 1.

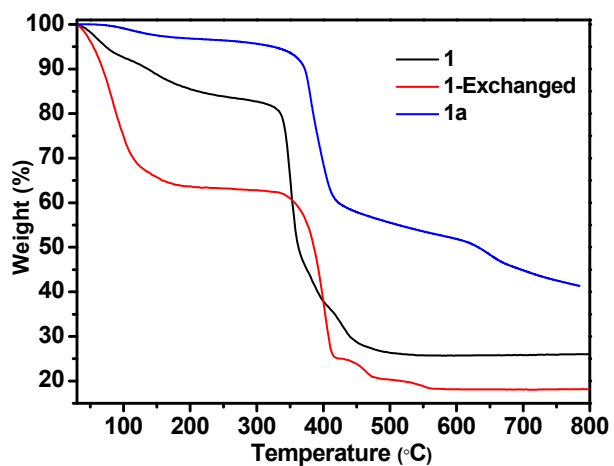


Figure S3. TGA curves of as-synthesized, CH₂Cl₂-exchanged and activated samples of complex 1.

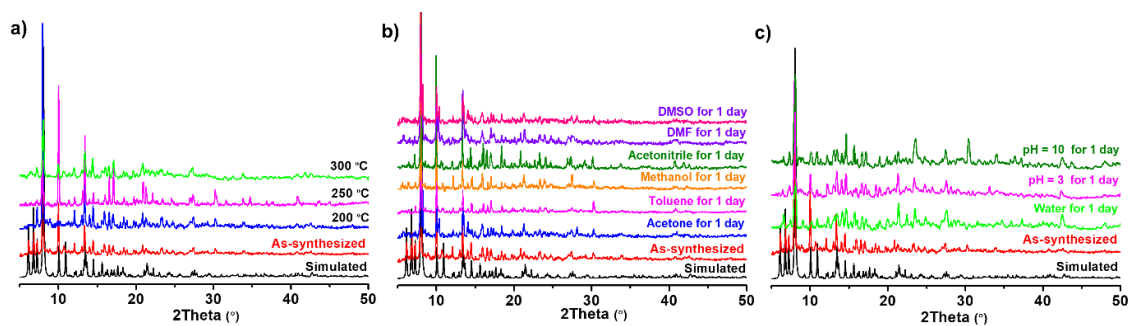
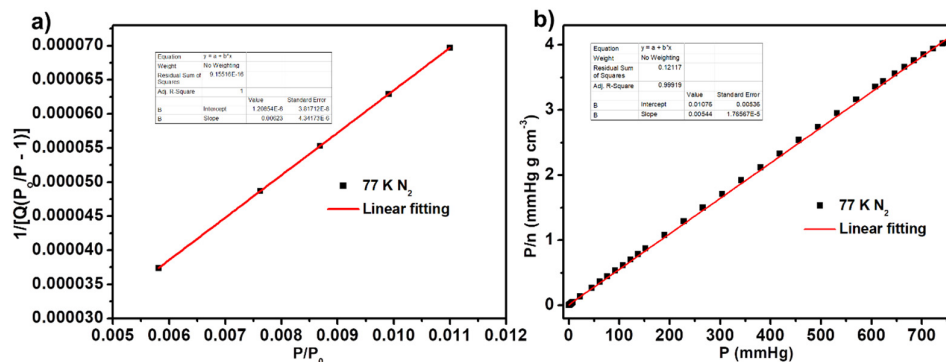


Figure S4. PXRD patterns of 1 after different treatments: a) heated at different temperatures;

b) soaked in organic solvents; c) water tolerability experiments.



$$S_{\text{BET}} = 1/(1.20854 \times 10^{-6} + 0.00623)/22414 \times 6.023 \times 10^{23} \times 0.162 \times 10^{-18} = 698 \text{ m}^2 \text{ g}^{-1}$$

$$S_{\text{Langmuir}} = (1/0.00544)/22414 \times 6.023 \times 10^{23} \times 0.162 \times 10^{-18} = 800 \text{ m}^2 \text{ g}^{-1}$$

Figure S5. a) BET surface area plot and b) Langmuir surface area plot for 1a.

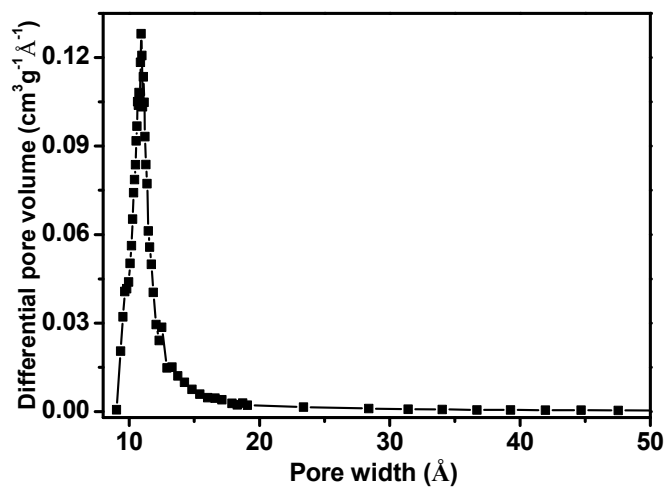


Figure S6. Pore distribution curve calculated from the N₂ adsorption isotherm at 77 K for 1a using the Horvath-Kawazoe model.

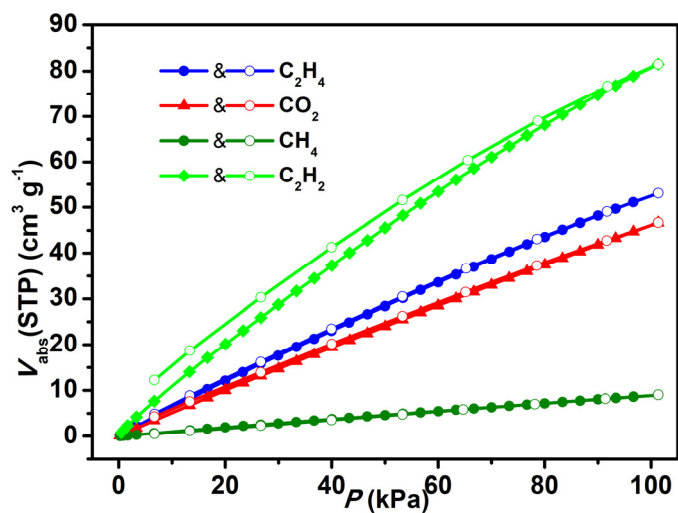
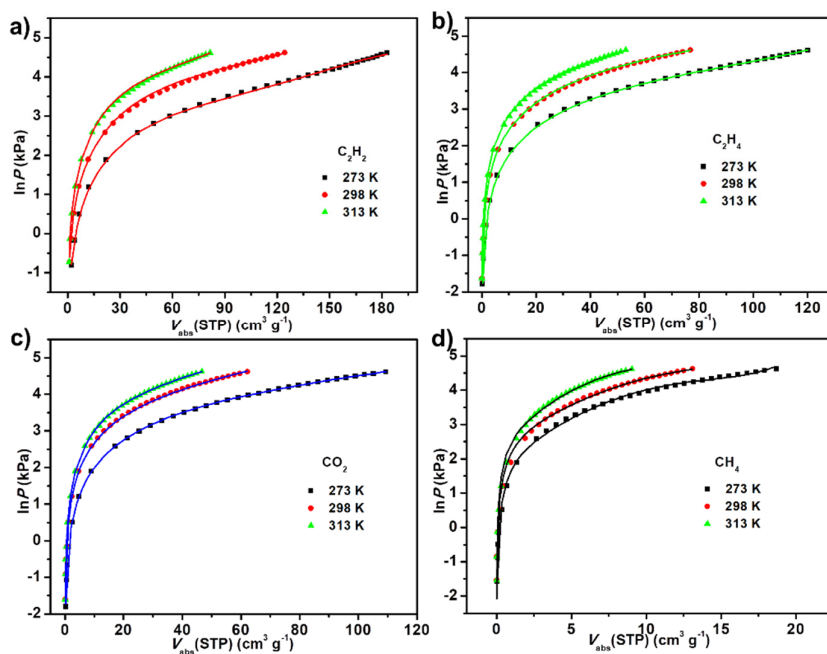


Figure S7. C₂H₂, C₂H₄, CO₂ and CH₄ adsorption isotherms of **1a** at 313 K

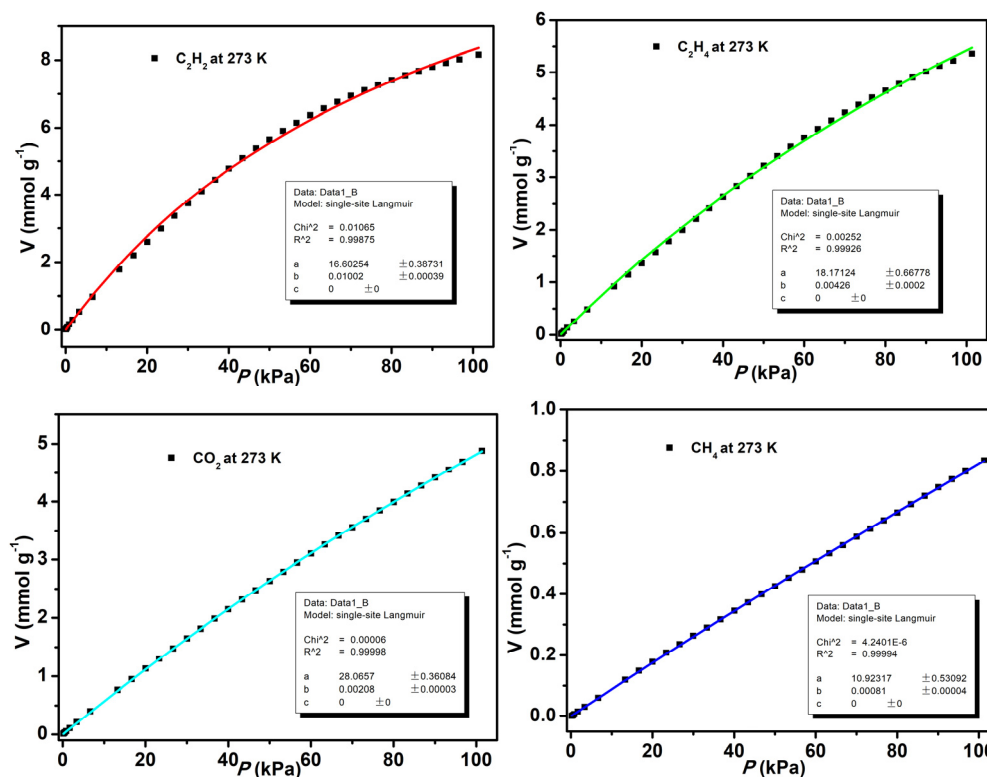


Viral Coefficient	C ₂ H ₂ Value	C ₂ H ₄ Value	CO ₂ Value	CH ₄ Value
a0	-2551.31286	-2267.46136	-2361.92993	-1658.8596
a1	1.05751	3.56382	18.32388	67.39201
a2	-0.06069	-0.01959	-0.27418	5.83396
a3	0.00067	0.00013	0.00062	-0.92708
a4	-1.4824E-6	1.8476E-7	-1.9939E-6	0.02308
b0	7.75517	7.674	8.23916	7.92459
b1	0.01883	-0.00459	-0.0547	-0.45337

b2	-0.00014	-0.00003	0.00076	0.02339
Chi ²	0.00174	0.0008	0.00081	0.0145
R ²	0.99916	0.99975	0.99975	0.99534

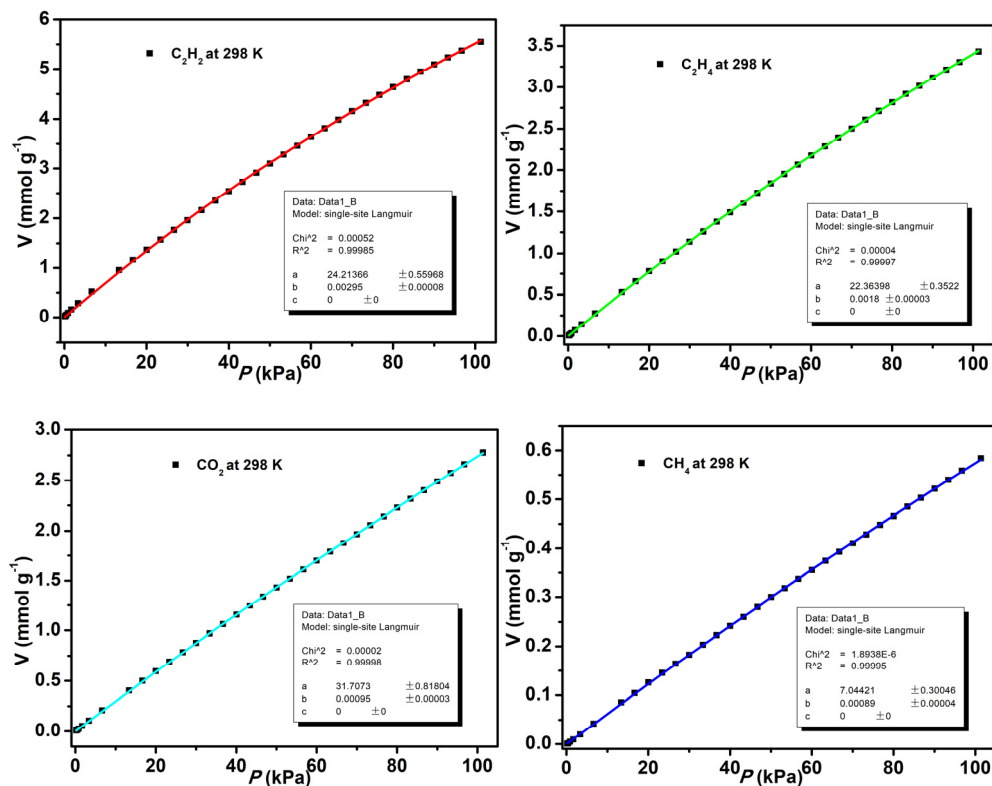
Figure S8. Fitted C₂H₂, C₂H₄, CO₂ and CH₄ isotherms of **1a** measured at 273, 298 and 313

K.



	a	b	Chi ²	R ²
C ₂ H ₂	16.60254	0.01002	0.01065	0.99875
C ₂ H ₄	18.17124	0.00426	0.00252	0.99926
CO ₂	28.0657	0.00208	0.00006	0.99998
CH ₄	10.92317	0.00081	4.2401E-6	0.99994

Figure S9. C₂H₂, C₂H₄, CO₂, CH₄ adsorption isotherms of **1a** with fitted by single-site Langmuir isotherm model at 273 K.



	A1	b1	Chi ²	R ²
C ₂ H ₂	24.21366	0.00295	0.00052	0.99985
C ₂ H ₄	22.36398	0.0018	0.00004	0.99997
CO ₂	31.7073	0.00095	0.00002	0.99998
CH ₄	7.04421	0.00089	1.8938E-6	0.99995

Figure S10. C₂H₂, C₂H₄, CO₂, CH₄ adsorption isotherms of **1a** with fitted by single-site Langmuir isotherm model at 298 K.

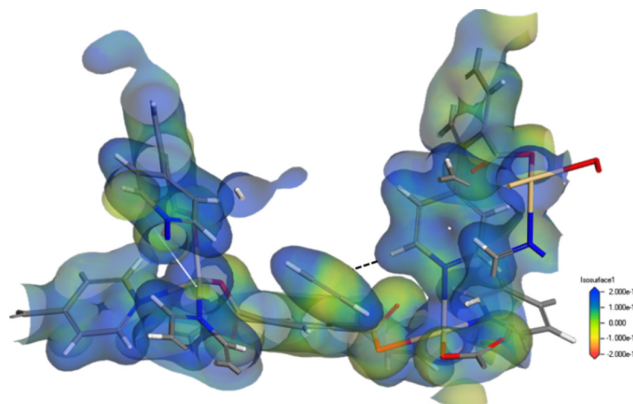


Figure S11. Electrostatic potential of **1a**@C₂H₂ mapped onto the Connolly surface simulated by DMol³ using GGA (PBE) function.

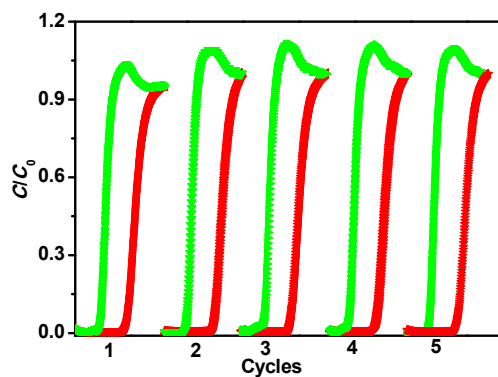


Figure S12. Breakthrough cycling curves for equimolar C_2H_2 - CO_2 mixture at room temperature.

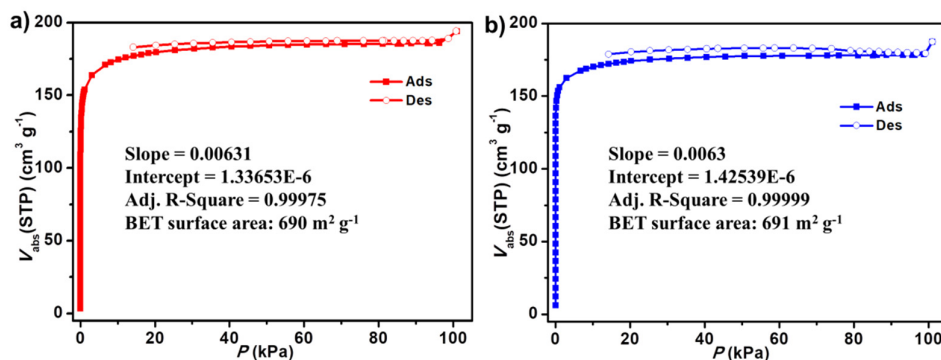


Figure S13. The adsorption isotherm of N_2 at 77K after adsorption experiment (a) and breakthrough experiments (b).

Table S1. Crystal Data and Structure Refinements for **1**.

Chemical formula	$C_{39}H_{29}Cd_2N_5O_{10}$
Formula weight	952.47
T (K)	189(2)
Crystal system, Space group	Monoclinic, $P2(1)/n$
a (Å)	16.9274(13)
b (Å)	22.0209(18)
c (Å)	19.9487(16)
α (°)	90
β (°)	107.005(2)
γ (°)	90
V (Å ³)	7110.9(10)
Z	4

$D_{\text{calcd.}}[\text{g}\cdot\text{cm}^{-3}]$	0.89
μ (mm ⁻¹)	0.638
Reflns collected/unique/ R_{int}	12948/12948/0.0657
Goof	1.031
R_1^a, wR_2^b [$I > 2\sigma$]	$R_1 = 0.1076, wR_2 = 0.3034$
R_1^a, wR_2^b (all data)	$R_1 = 0.1159, wR_2 = 0.3115$

^a $R_1 = \Sigma(|F_o| - |F_c|) / \Sigma|F_o|$. ^b $R_2 = [\Sigma w(F_o^2 - F_c^2)^2 / \Sigma w(F_o^2)^2]^{1/2}$.

Table S2. Selected bond lengths [Å] and angles [°] for **1**.

Cd(1)-N(4)#1	2.283(9)	O(1W)-Cd(1)-O(7)#2	101.9(5)
Cd(1)-O(1)	2.289(8)	N(4)#1-Cd(1)-O(2)	83.1(3)
Cd(1)-O(8)#2	2.301(8)	O(1)-Cd(1)-O(2)	53.2(3)
Cd(1)-N(2)#3	2.325(10)	O(8)#2-Cd(1)-O(2)	136.7(3)
Cd(1)-O(1W)	2.387(11)	N(2)#3-Cd(1)-O(2)	94.2(4)
Cd(1)-O(7)#2	2.485(9)	O(1W)-Cd(1)-O(2)	79.3(4)
Cd(1)-O(2)	2.596(10)	O(7)#2-Cd(1)-O(2)	169.3(3)
Cd(2)-O(5)#4	2.277(7)	O(5)#4-Cd(2)-N(1)	137.6(3)
Cd(2)-N(1)	2.311(8)	O(5)#4-Cd(2)-O(9)	81.7(4)
Cd(2)-O(9)	2.342(10)	N(1)-Cd(2)-O(9)	81.5(4)
Cd(2)-N(3)	2.372(9)	O(5)#4-Cd(2)-N(3)	93.7(3)
Cd(2)-O(3)#3	2.393(11)	N(1)-Cd(2)-N(3)	91.5(3)
Cd(2)-O(4)#3	2.454(11)	O(9)-Cd(2)-N(3)	163.4(4)
Cd(2)-O(6)#4	2.606(7)	O(5)#4-Cd(2)-O(3)#3	87.9(3)
N(4)#1-Cd(1)-O(1)	136.4(3)	N(1)-Cd(2)-O(3)#3	130.3(3)
N(4)#1-Cd(1)-O(8)#2	136.6(3)	O(9)-Cd(2)-O(3)#3	88.9(5)
O(1)-Cd(1)-O(8)#2	85.5(3)	N(3)-Cd(2)-O(3)#3	106.9(4)
N(4)#1-Cd(1)-N(2)#3	94.5(4)	O(5)#4-Cd(2)-O(4)#3	137.1(3)
O(1)-Cd(1)-N(2)#3	88.7(4)	N(1)-Cd(2)-O(4)#3	85.3(3)
O(8)#2-Cd(1)-N(2)#3	97.8(4)	O(9)-Cd(2)-O(4)#3	110.5(5)
N(4)#1-Cd(1)-O(1W)	86.9(4)	N(3)-Cd(2)-O(4)#3	83.7(4)
O(1)-Cd(1)-O(1W)	85.6(4)	O(3)#3-Cd(2)-O(4)#3	52.9(4)
O(8)#2-Cd(1)-O(1W)	85.7(4)	O(5)#4-Cd(2)-O(6)#4	53.7(2)
N(2)#3-Cd(1)-O(1W)	173.2(5)	N(1)-Cd(2)-O(6)#4	85.3(3)
N(4)#1-Cd(1)-O(7)#2	86.3(3)	O(9)-Cd(2)-O(6)#4	80.8(4)
O(1)-Cd(1)-O(7)#2	137.3(3)	N(3)-Cd(2)-O(6)#4	83.6(3)
O(8)#2-Cd(1)-O(7)#2	53.8(3)	O(3)#3-Cd(2)-O(6)#4	141.1(3)
N(2)#3-Cd(1)-O(7)#2	84.9(5)	O(4)#3-Cd(2)-O(6)#4	164.0(3)

Symmetry codes: #1 -x+1, -y+2, -z+1; #2 x-1, y, z; #3 -x+1/2, y-1/2, -z+1/2; #4 -x+1, -y+2, -z; #5 -x+1/2, y+1/2, -z+1/2; #6 x+1, y, z

Table S3. Gas adsorption results.

1a	S_{BET} (m^2/g)	Pore sizes	Adsorption uptake ($\text{cm}^3 \text{g}^{-1}$)				Q_{st} (kJ mol^{-1})			
			C_2H_2	C_2H_4	CH_4	CO_2	C_2H_2	C_2H_4	CH_4	CO_2
273 K	698	10 Å	182.8	120	18.7	109.4				
298 K			124.4	76.8	13.1	62.2	21.2	18.8	13.8	19.6
313 K			81.5	53.1	9.0	46.6				

Table S4. Comparison of Q_{st} and uptake of C_2H_2 in MOFs at 298 K.

	C_2H_2 Uptake (mmol g^{-1})	C_2H_2 Q_{st} (kJ mol^{-1})	Ref.
SNNU-27- Fe_2Co	7.61	20.2	S7
Fe-MOF-74	6.84	47	S8
1a	5.55	20.5	This work
TIFSIX-2-Cu-i	4.1	46.3	S9
FeNi'-MOF	4.29	27	S10
SIFSIX-2-Cu-i	4.02	41.9	S11
PCM-48	3.99	23.6	S12
ZJU-74a	3.84	45	S13
Ni(dpip)	3.73	41.7	S14
Cd(dtztp)	3.58	25	S15
UTSA-100a	3.45	22	S16
UTSA-300a	3.2	57.6	S17
JNU-1	2.89	13	S18
NKMOF-1-Ni	2.72	60.3	S19
ELM-12	2.55	25.4	S20
$[\text{Ni}_3(\text{HCOO})_6]$	2.38	40.9	S21
$\text{CuI}@\text{UiO-66}-(\text{COOH})_2$	2.32	74.5	S22

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