

**Gas Separation**

# Boosting Ethane/Ethylene Separation by MOFs through the Amino-Functionalization of Pores

Gang-Ding Wang, Rajamani Krishna, Yong-Zhi Li, Wen-Juan Shi, Lei Hou,\* Yao-Yu Wang, and Zhonghua Zhu

**Abstract:** Adsorption technology based on ethane-selective materials is a promising alternative to energy-intensive cryogenic distillation for separating ethane ( $C_2H_6$ ) and ethylene ( $C_2H_4$ ). We employed a pore engineering strategy to tune the pore environment of a metal-organic framework (MOF) through organic functional groups and boosted the  $C_2H_6/C_2H_4$  separation of the MOF. Introduction of amino ( $-NH_2$ ) groups into Tb-MOF-76 not only decreased pore sizes but also facilitated multiple guest-host interactions in confined pores. The  $NH_2$ -functionalized Tb-MOF-76( $NH_2$ ) has increased  $C_2H_6$  and  $C_2H_4$  uptakes and  $C_2H_6/C_2H_4$  selectivity. The results of experimental and simulated transient breakthroughs reveal that Tb-MOF-76( $NH_2$ ) has significantly improved one-step separation performance for  $C_2H_6/C_2H_4$  mixtures with a high  $C_2H_4$  (>99.95%) productivity of  $17.66 \text{ L kg}^{-1}$  compared to  $7.53 \text{ L kg}^{-1}$  by Tb-MOF-76, resulting from the suitable pore confinement and accessible  $-NH_2$  groups on pore surfaces.

## Introduction

Ethylene ( $C_2H_4$ ) is the most demanded raw material in the petrochemical industry, with a global production exceeding 210 million tons in 2021.  $C_2H_4$  is mainly produced by the thermal decomposition of ethane ( $C_2H_6$ ) and steam cracking of fossil fuels, and inevitably contains a certain amount of  $C_2H_6$  residue (5%–9%) that must be cut down to guarantee the polymerization utilization of  $C_2H_4$ .<sup>[1]</sup> However, the separation of  $C_2H_6$  and  $C_2H_4$  is extremely challenging

because of similar kinetic diameters and boiling points between them ( $C_2H_6$ :  $4.44 \text{ \AA}$ ,  $184.55 \text{ K}$ ;  $C_2H_4$ :  $4.16 \text{ \AA}$ ,  $169.42 \text{ K}$ ).<sup>[2]</sup> Thus far, the well-developed separation of  $C_2H_4/C_2H_6$  has been realized by energy-intensive distillation operated in a large distillation tower at low temperatures and high pressures.<sup>[3]</sup> Exploiting efficient separation techniques aiming to produce polymer-grade  $C_2H_4$  is highly imperative.<sup>[4]</sup>

Adsorbent-based separation using porous materials has attracted particular attention for low energy consumption and high efficiency.<sup>[5]</sup>  $C_2H_6/C_2H_4$  separation can be implemented by  $C_2H_4$ -selective or  $C_2H_6$ -selective adsorbents.  $C_2H_4$ -selective adsorbents preferentially adsorb  $C_2H_4$  over  $C_2H_6$ , however, the subsequent desorption process is needed to obtain  $C_2H_4$  product, meanwhile the product is generally of low purity due to  $C_2H_6$  co-adsorption in adsorbents.<sup>[6]</sup> By comparison,  $C_2H_6$ -selective adsorbents that preferentially capture  $C_2H_6$  impurity over  $C_2H_4$  attain one-step harvest of  $C_2H_4$  in single breakthrough operation process.<sup>[7]</sup> At present, the  $C_2H_6$ -selective adsorbents have remained relatively underexplored as installing  $C_2H_4$ -binding sites in adsorbents is easier to operate than  $C_2H_6$ -binding sites. So the exploitation of excellent  $C_2H_6$ -selective adsorbents is the desired pursuit for  $C_2H_4$  purity goal.

Metal-organic frameworks (MOFs) are a new type of porous materials with wide variety of application prospects in many fields, especially the unique of tunable pore environment and easy modification enable them to be ideal platforms for designing  $C_2H_6$ -selective adsorbents.<sup>[8]</sup> These adsorbents require specific recognition for  $C_2H_6$  over  $C_2H_4$  and high uptake for  $C_2H_6$ . Hitherto, although some  $C_2H_6$ -selective MOFs have been reported,<sup>[9]</sup> however, the deficiencies, such as low selectivity, unsatisfied capacity, and inferior structural stability are also usually encountered. For instance, creating inert/non-polar pore environment is effective to construct  $C_2H_6$ -selective MOFs, however, the materials exhibit relatively low loading for  $C_2H_6$  due to lacking strong  $C_2H_6$ -interacting sites.<sup>[10]</sup> The functionalization of open metal sites (OMSs) through  $O_2$  reported by Chen et al. greatly enhances the  $C_2H_6$  uptakes, yielding a recorded  $C_2H_6/C_2H_4$  selectivity of 4.4, which shows a pioneering work in creating  $C_2H_6$ -selective MOFs.<sup>[7b]</sup> This strategy needs removing the coordinated solvents followed by introducing  $O_2$  molecules to bind the OMSs, and the material is sensitive to air/moisture and must be handled in a glove box. However, shielding the OMSs by coordinated solvent molecules avoids the metal- $\pi$  interactions between OMSs and  $C_2H_4$ , providing an alternative to creating  $C_2H_6$ -selective

[\*] Dr. G.-D. Wang, Dr. Y.-Z. Li, Dr. W.-J. Shi, Prof. L. Hou, Prof. Y.-Y. Wang  
 Key Laboratory of Synthetic and Natural Functional Molecule of the Ministry of Education, College of Chemistry & Materials Science, Northwest University  
 Xi'an, 710127 (P. R. China)  
 E-mail: lhou2009@nwu.edu.cn  
 Prof. R. Krishna  
 Van 't Hoff Institute for Molecular Sciences, University of Amsterdam  
 1098 XH Amsterdam (The Netherlands)  
 Prof. Z. Zhu  
 School of Chemical Engineering, The University of Queensland  
 Brisbane 4072 (Australia)

MOFs. In addition, there are also other  $C_2H_6$ -selective MOFs that mainly rely on synergy in pore-matching effects and multiple interactions between the framework and  $C_2H_6$  with more C–H bonds.<sup>[11]</sup>

Targeting the development of outstanding separation materials for  $C_2H_6/C_2H_4$  mixtures, isoreticular chemistry allows us through pore engineering strategy to precisely design and regulate the pores of MOFs. The pore environment of  $C_2H_6$ -selective MOFs can be tuned by modifying linkers with organic groups, thus it would improve the  $C_2H_6/C_2H_4$  separation potential of MOFs. However, only very rare examples were reported on the study of  $C_2H_6/C_2H_4$  separation by pore engineering strategy in MOFs.<sup>[3b,8c,9c,10,11]</sup> Amino ( $-NH_2$ ) group effects on the gas adsorption and separation were well demonstrated by other research groups.<sup>[12]</sup> For example, using  $-NH_2$  groups or diamine compounds to modify the organic linkers or metal centers in MOFs, the  $CO_2$  adsorption amounts were greatly increased.<sup>[12a,e,f]</sup> In addition,  $-NH_2$  groups can also serve as accessible sites to form multiple C–H...N hydrogen bonds with light hydrocarbon molecules to enhance the interactions between MOFs and hydrocarbons.<sup>[13]</sup> However, there are very few comparison studies on the  $C_2H_6/C_2H_4$  mixtures separation between amino-functionalized MOFs and parent MOFs.<sup>[8b]</sup> So the comparative in-depth study on  $C_2H_6/C_2H_4$  separation properties in the pair of isoreticular MOFs are urgently needed.

With this in mind, we noticed Tb-MOF-76 based on 1,3,5-benzenetricarboxylate (BTC) as a platform due to its analogues of M–MOF-76 (M = Y, Sm, Eu, and Dy) showed the reversed  $C_2H_6/C_2H_4$  adsorption selectivity.<sup>[14]</sup> According to isoreticular chemistry, isomorphic Tb-MOF-76( $NH_2$ ) was readily obtained by using the amino-functionalized linker 2-amino-1,3,5-benzenetricarboxylate ( $NH_2$ -BTC), which provides an elegant example of controlling pore chemistry for advancing  $C_2H_6/C_2H_4$  separation. It was found that the introduction of  $-NH_2$  groups decreased pore sizes from  $7.9 \times 7.9 \text{ \AA}^2$  in Tb-MOF-76 to  $7.2 \times 7.2 \text{ \AA}^2$  in Tb-MOF-76( $NH_2$ ), facilitating the multiple guest-host interactions in confined pores. Through obstructing the OMSs with the water ligands, both MOFs show the  $C_2H_6$ -selective adsorption behavior. Compared to Tb-MOF-76, Tb-MOF-76( $NH_2$ ) has increased  $C_2H_4$  and  $C_2H_6$  uptakes and  $C_2H_6/C_2H_4$  selectivity (1.7 vs 2.1), superior to most reported  $C_2H_6$ -selective MOFs, and also displays the improved separation performance for 1/1, 1/9, and 1/15 (v/v)  $C_2H_6/C_2H_4$  mixtures, obtaining the pure  $C_2H_4$  by one step. Thus, the installation of  $-NH_2$  groups optimizes the pore environment elaborately, endowing Tb-MOF-76( $NH_2$ ) with great potential as a superior adsorbent for  $C_2H_4$  purification from  $C_2H_6/C_2H_4$  mixtures.

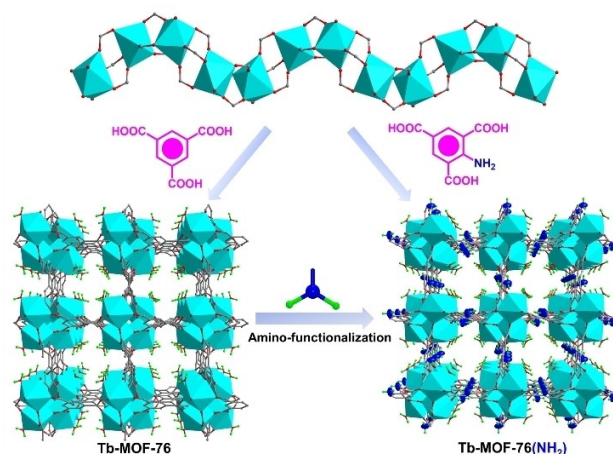
## Results and Discussion

In performing functional modification, it is of paramount importance to synthesize MOFs with high stability. Herein, adjustable and highly stable Tb-MOF-76 was selected as a platform for  $-NH_2$  functionalization. The high quality of single crystals of Tb-MOF-76( $NH_2$ ) can be prepared through

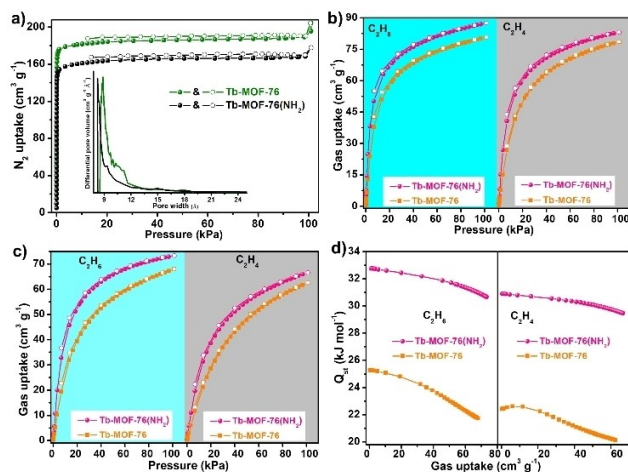
the solvothermal reaction of  $NH_2$ -BTC with  $Tb^{3+}$  ions, which is isostructural to Tb-MOF-76 crystallized in a  $P4_122$  space group (CCDC No. 2192249, as shown in Figure S1–S3, Table S1 and S2).<sup>[15]</sup> The  $Tb^{3+}$  centres have distorted pentagonal-bipyramidal geometries (Figure S4), and are bridged by carboxylates to form a helical rod-shaped secondary building unit (SBU) (Figure 1). The SBUs are connected with organic linkers to construct an open framework possessing square channels. Compared to Tb-MOF-76, Tb-MOF-76( $NH_2$ ) has free  $-NH_2$  groups pointing into the channels, which decreases the pore dimensions to  $7.2 \times 7.2 \text{ \AA}^2$  from  $7.9 \times 7.9 \text{ \AA}^2$  (excluding the van der Waals radii) in Tb-MOF-76 (Figure 1). The pore sizes of Tb-MOF-76( $NH_2$ ) is closer to the kinetic diameter of  $C_2H_4$  (4.16  $\text{\AA}$ ) and  $C_2H_6$  (4.44  $\text{\AA}$ ), which would impose multipoint adsorbing sites for  $C_2H_4$  and  $C_2H_6$  molecules considering size-matching effect.

Thermogravimetric analysis (TGA) showed the removal of coordinated water molecules at a high temperature for the two MOFs (Figure S5 and S6). During activation process of MOFs, the coordinated water molecules were held in order to avoid the formation of OMSs that on one hand are unfavorable to binding  $C_2H_6$  over  $C_2H_4$ , on the other hand are easily attacked by water in realistic work environment. For this regard, the MOFs exchanged in methanol for 72 h were activated at 373 K under high vacuum to remove lattice solvents, as shown in TGA curves (Figure S5 and S6). Power X-ray diffraction (PXRD) of two desolvated MOFs remain unchanged, indicating the framework rigidity (Figure S2 and S3). Both MOFs show typically type-I isotherms for  $N_2$  at 77 K (Figure 2a). As expected, the introduction of  $-NH_2$  groups moderately decreases the pore sizes of Tb-MOF-76( $NH_2$ ) (8.2  $\text{\AA}$ ) compared to Tb-MOF-76 (8.8  $\text{\AA}$ ), the smaller pore sizes would be more favorable to form multiple contacts between the framework and gases.

To evaluate the tuning on absorption performance via  $-NH_2$  modification, the sorption isotherms of  $C_2H_6$  and  $C_2H_4$  were measured at 273 and 298 K (Figure 2b and 2c). At 100 kPa, the loadings of Tb-MOF-76 for  $C_2H_6$  and  $C_2H_4$



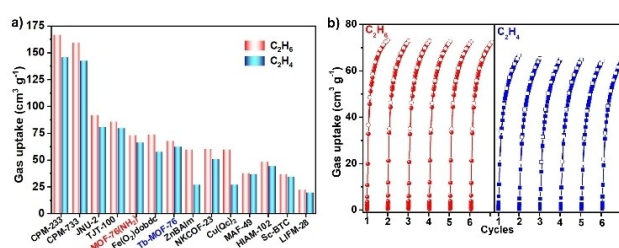
**Figure 1.** Isostructural frameworks of Tb-MOF-76 and Tb-MOF-76( $NH_2$ ) assembled by rod-shaped SBUs and BTC/ $NH_2$ -BTC linkers.



**Figure 2.** a)  $N_2$  sorption isotherms at 77 K, the insert indicates pore size distribution; b)  $C_2H_6$  and  $C_2H_4$  sorption isotherms at 273 K; c)  $C_2H_6$  and  $C_2H_4$  sorption isotherms at 298 K; d)  $Q_{st}$  plots of  $C_2H_6$  and  $C_2H_4$ .

are 68.0 and 62.6  $cm^3 g^{-1}$  at 298 K and 80.6 and 78.6  $cm^3 g^{-1}$  at 273 K, respectively, while the loadings of Tb-MOF-76( $NH_2$ ) for  $C_2H_6$  and  $C_2H_4$  are increased to 73.3 and 66.6  $cm^3 g^{-1}$  at 298 K and 87.5 and 83.1  $cm^3 g^{-1}$  at 273 K, respectively. Compared to Tb-MOF-76, besides higher uptakes, Tb-MOF-76( $NH_2$ ) also exhibited steeper adsorption isotherms of  $C_2H_6$  and  $C_2H_4$ , implying the immobilization of  $-NH_2$  groups strengthens the adsorbate-adsorbent interactions. This finding coincides with the calculated results of isosteric heat of adsorption ( $Q_{st}$ ) by fitting the gas adsorption isotherms at 273, 298, and 313 K to the virial equation (Figure 2d, S7–S12), which display notably higher  $Q_{st}$  values in two MOFs for  $C_2H_6$  (32.8–30.7  $kJ mol^{-1}$  for Tb-MOF-76( $NH_2$ ), 25.2–21.7  $kJ mol^{-1}$  for Tb-MOF-76) compared to  $C_2H_4$  (30.9–29.5  $kJ mol^{-1}$  for Tb-MOF-76( $NH_2$ ), 22.4–20.1  $kJ mol^{-1}$  for Tb-MOF-76) in the measured pressure region. Meanwhile, Tb-MOF-76( $NH_2$ ) shows higher  $Q_{st}$  for the two gases relative to Tb-MOF-76. The decreased pore sizes in Tb-MOF-76( $NH_2$ ) allow closer interactions between the framework and gas molecules, improving adsorption affinity. For two MOFs, besides the higher  $Q_{st}$  values of  $C_2H_6$  than  $C_2H_4$ , the isotherms of  $C_2H_6$  are also increased more sharply than  $C_2H_4$  (Figure S11 and S12). These observations validate the stronger affinity of the framework toward  $C_2H_6$  over  $C_2H_4$ , supporting the  $C_2H_6$ -selective behavior in two MOFs.

Although the  $C_2H_6$  uptake of Tb-MOF-76( $NH_2$ ) at 298 K and 100 kPa is lower than some benchmark  $C_2H_6$ -selective MOFs (Figure 3a), such as CPM-233 (166.8  $cm^3 g^{-1}$ ),<sup>[16]</sup> CPM-733 (159.6  $cm^3 g^{-1}$ ),<sup>[16]</sup> JNU-2 (92  $cm^3 g^{-1}$ ),<sup>[17]</sup> and TJT-100 (86  $cm^3 g^{-1}$ ),<sup>[18]</sup> but exceeds or is comparable with most reported top-performance adsorbents for  $C_2H_6/C_2H_4$  separation, including Cu(Qc)<sub>2</sub> (60  $cm^3 g^{-1}$ ),<sup>[11]</sup> MAF-49 (38  $cm^3 g^{-1}$ ),<sup>[13a]</sup> NKCOF-23 (60.5  $cm^3 g^{-1}$ ),<sup>[7a]</sup> and HIAM-102 (48.3  $cm^3 g^{-1}$ ).<sup>[19]</sup> The cycle tests demonstrate the facile reactivation of Tb-MOF-76( $NH_2$ ), wherein the sorption

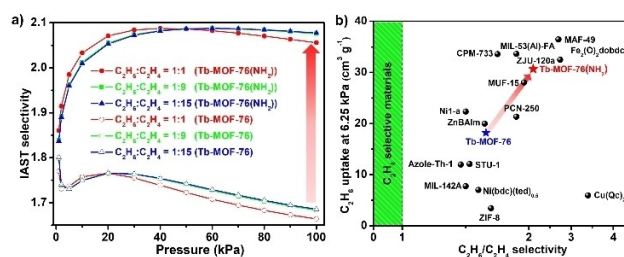


**Figure 3.** a) Comparison of  $C_2H_6$  and  $C_2H_4$  uptakes in some  $C_2H_6$ -selective materials; b)  $C_2H_6$  and  $C_2H_4$  sorption cycles for Tb-MOF-76( $NH_2$ ) at 298 K.

isotherms of  $C_2H_6$  and  $C_2H_4$  are reversible and have no decrease in adsorption capacities (Figure 3b).

Ideal adsorbed solution theory (IAST) was utilized to assess and compare the selectivity of Tb-MOF-76( $NH_2$ ) and Tb-MOF-76 for 1/1, 1/9, and 1/15  $C_2H_6/C_2H_4$  mixtures at 298 K (Figure S13 and S14). Tb-MOF-76 exhibits the  $C_2H_6/C_2H_4$  selectivity of about 1.7 for these mixtures at 100 kPa, while Tb-MOF-76( $NH_2$ ) shows an obviously high selectivity of about 2.1 (Figure 4a). Since the low content of  $C_2H_6$  in actual cracked gas mixtures ( $C_2H_6/C_2H_4$ , 1:15), it is crucial that the material has a high loading for  $C_2H_6$  at low partial pressure. Figure 4b presents a comparison of Tb-MOF-76( $NH_2$ ) with some typical  $C_2H_6$ -selective MOFs when we set  $C_2H_6/C_2H_4$  selectivity at 100 kPa and  $C_2H_6$  uptake at 6.25 kPa (partial pressure of  $C_2H_6$  in cracked gas mixtures) as concurrent objectives. The selectivity and  $C_2H_6$  uptake in Tb-MOF-76( $NH_2$ ) are only lower than benchmark MAF-49 (2.7),<sup>[13a]</sup> Fe<sub>2</sub>(O<sub>2</sub>)(dobdc) (4.4),<sup>[7b]</sup> and ZJU-120a (2.74),<sup>[10]</sup> but higher than some top-performing  $C_2H_6$ -selective MOFs, such as MUF-15 (1.96),<sup>[20]</sup> Azole-Th-1 (1.46),<sup>[21]</sup> CPM-733 (1.75),<sup>[16]</sup> and ZIF-8 (1.7).<sup>[22]</sup> Taken together, previously reported  $C_2H_6$ -selective MOFs commonly display either low selectivity or few  $C_2H_6$  uptake at low partial pressure, Tb-MOF-76( $NH_2$ ) exhibits a well balance in uptake and selectivity, rendering it among the benchmark material for this important separation.

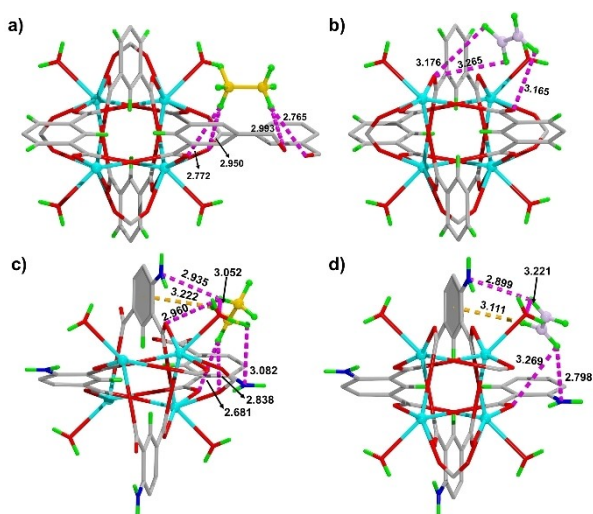
To gain deeper insights into the origin of enhanced gas uptake and selectivity by  $-NH_2$  functionalization, grand canonical Monte Carlo (GCMC) simulations were done to reveal the adsorption details. It shows that both  $C_2H_4$  and  $C_2H_6$  molecules are located at the corners of channels near



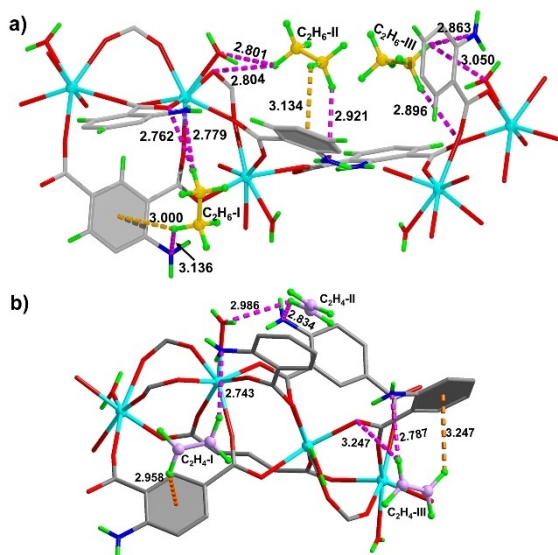
**Figure 4.** a) IAST selectivity of Tb-MOF-76 and Tb-MOF-76( $NH_2$ ) for  $C_2H_6/C_2H_4$  mixtures; b) comparison of  $C_2H_6$  uptakes and  $C_2H_6/C_2H_4$  selectivity in different MOFs.



the rod-shaped SBUs. There only exists C–H...O hydrogen bonds between Tb-MOF-76 and C<sub>2</sub>H<sub>6</sub> or C<sub>2</sub>H<sub>4</sub> molecules, but for Tb-MOF-76(NH<sub>2</sub>) there are not only C–H...O hydrogen bonds but also C–H... $\pi$  and C–H...N interactions from the aromatic rings and –NH<sub>2</sub> groups of ligands. For Tb-MOF-76, the C<sub>2</sub>H<sub>6</sub> molecule is bound to carboxyl O atoms from two BTC through four strong C–H...O interactions (H...O = 2.765–2.993 Å) (Figure 5a), by contrast, three weak C–H...O interactions (H...O distances = 3.165 Å–3.265 Å) formed with the C<sub>2</sub>H<sub>4</sub> molecule are fewer and weaker (Figure 5b), in accord with the selectivity of C<sub>2</sub>H<sub>6</sub> over C<sub>2</sub>H<sub>4</sub>. For Tb-MOF-76(NH<sub>2</sub>), C<sub>2</sub>H<sub>6</sub> interacts with one phenyl ring from the ligand, two amino N atoms and four O atoms from three carboxyl groups and one water ligand to form four C–H...O, two C–H...N, and one C–H... $\pi$  inter-



**Figure 5.** C<sub>2</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>4</sub> preferential adsorption sites in Tb-MOF-76 (a and b) and Tb-MOF-76(NH<sub>2</sub>) (c and d).



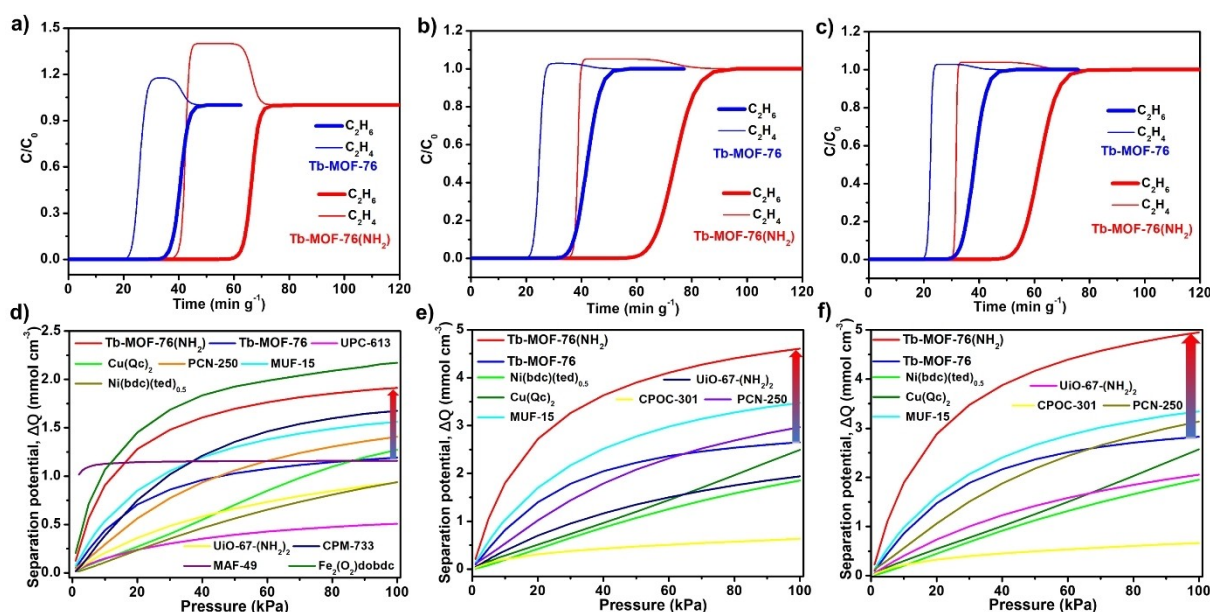
**Figure 6.** Adsorption sites of C<sub>2</sub>H<sub>6</sub> (a) and C<sub>2</sub>H<sub>4</sub> (b) in Tb-MOF-76(NH<sub>2</sub>).

actions with the distances in the range of 2.681–3.082 Å (Figure 5c), but C<sub>2</sub>H<sub>4</sub> forms weaker and fewer C–H...O/N/ $\pi$  interactions with longer distances (2.798–3.221 Å) (Figure 6d). As a result, the smaller pore sizes and more binding sites in Tb-MOF-76(NH<sub>2</sub>) lead to C<sub>2</sub>H<sub>4</sub> and C<sub>2</sub>H<sub>6</sub> molecules in close contact with the pore walls compared to the corresponding gases with Tb-MOF-76. Meanwhile, it also reveals that in two MOFs there exist stronger contacts toward C<sub>2</sub>H<sub>6</sub> than C<sub>2</sub>H<sub>4</sub>, agreeing with their C<sub>2</sub>H<sub>6</sub>-selective features.

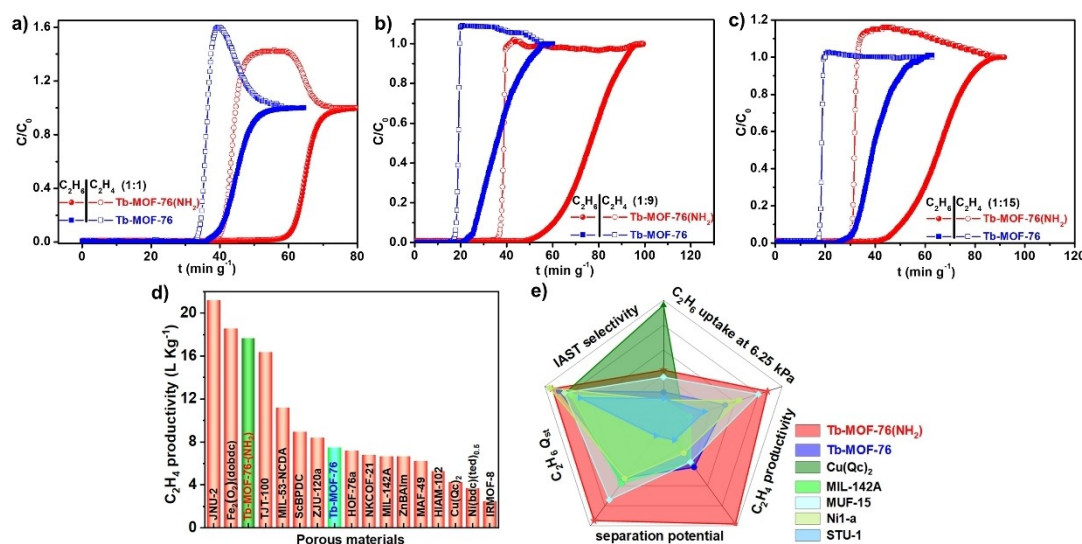
Furthermore, the interactions of C<sub>2</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>4</sub> in Tb-MOF-76(NH<sub>2</sub>) were further studied by simulations at 298 K and 100 kPa. It found three crucial C<sub>2</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>4</sub> molecules interacting with the pore walls, as given in Figure 6. Both C<sub>2</sub>H<sub>6</sub>-I and C<sub>2</sub>H<sub>6</sub>-II form four strong C–H... $\pi$ /N/O interactions with the phenyl groups, amino N atoms, and carboxyl O atoms, while C<sub>2</sub>H<sub>6</sub>-III interacts with carboxyl O and amino N atoms through three strong hydrogen bonds (Figure 6a). Three C<sub>2</sub>H<sub>4</sub> molecules also contact with the framework through C–H... $\pi$ /N/O hydrogen bonds (Figure 6b). In brief, there are more and stronger contacts between the framework and C<sub>2</sub>H<sub>6</sub> compared to C<sub>2</sub>H<sub>4</sub>, thus forming a priority of adsorption for C<sub>2</sub>H<sub>6</sub> over C<sub>2</sub>H<sub>4</sub>.

To validate the positive effect of –NH<sub>2</sub> groups on C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation, transient breakthrough simulations for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures (1/1, 1/9, and 1/15, v/v) on Tb-MOF-76(NH<sub>2</sub>) and Tb-MOF-76 in fixed beds were conducted at 298 K and 100 kPa (see Supporting Information).<sup>[22]</sup> As shown in Figure 7a,b,c, two MOFs can achieve efficient separations for three C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures, wherein C<sub>2</sub>H<sub>4</sub> breakthrough first occurred and subsequently reached a plateau to yield the polymer-grade C<sub>2</sub>H<sub>4</sub>, then C<sub>2</sub>H<sub>6</sub> passed through the fixed bed after long times ( $\tau_{\text{break}}$ ). The separation potential ( $\Delta Q$ ) as a combined selectivity-capacity metric to quantify the mixture separation performance was utilized for further comparison.<sup>[7b,23]</sup> As given in Figure 7d,e,f, the amounts of pure C<sub>2</sub>H<sub>4</sub> can be recovered by Tb-MOF-76(NH<sub>2</sub>) reached up to 1.91, 4.61, and 4.95 mmol cm<sup>−3</sup> for the 1/1, 1/9, and 1/15 mixtures, respectively, greatly outperform the values of 1.19, 2.65, and 2.82 mmol cm<sup>−3</sup> in Tb-MOF-76. The  $\Delta Q$  of Tb-MOF-76(NH<sub>2</sub>) are not good as Fe<sub>2</sub>(O<sub>2</sub>)(dobdc),<sup>[7b]</sup> but are better than other C<sub>2</sub>H<sub>6</sub>-selective materials including MAF-49,<sup>[7b]</sup> UPC-613,<sup>[8d]</sup> PCN-250,<sup>[24]</sup> and UiO-67-(NH<sub>2</sub>)<sub>2</sub>,<sup>[8b]</sup> meaning the most promising prospect for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation (Table S3).

Next, dynamic breakthrough experiments were conducted at 298 K and 1 atm using C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub>/Ar (5/5/90, 1/9/90, and 1/15/84, v/v/v) mixtures with Ar as the carrier gas introduced over the packed columns of Tb-MOF-76(NH<sub>2</sub>) and MOF-76 (flow rate = 7.0 mL min<sup>−1</sup>), respectively. The breakthrough curves depicted in Figure 8a,b,c show the effective separation of C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures, in which Tb-MOF-76(NH<sub>2</sub>) reveals an obviously better separation performance than Tb-MOF-76, in accordance with the single-component sorption, IAST selectivity, and transient breakthrough simulations. Comparing Figure 7a,b,c with Figure 8a,b,c, it is particularly noteworthy that the quantitative agreement between the transient breakthrough simulations with experiments. As predicted, C<sub>2</sub>H<sub>4</sub> first eluted through



**Figure 7.** Simulated breakthrough curves of Tb-MOF-76(NH<sub>2</sub>) and Tb-MOF-76 for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures: a) 1/1, b) 1/9, and c) 1/15; d–f) separation potential of selected MOFs for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures: d) 1/1, e) 1/9, and f) 1/15.



**Figure 8.** a–c) Experimental breakthrough curves of Tb-MOF-76(NH<sub>2</sub>) and Tb-MOF-76 for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures at 298 K; d) comparison of C<sub>2</sub>H<sub>4</sub> productivity for porous materials; e) comparison of the comprehensive separation performance in different C<sub>2</sub>H<sub>6</sub>-selective MOFs.

the column to yield an outflow of pure C<sub>2</sub>H<sub>4</sub> (> 99.9 %) with an undetectable C<sub>2</sub>H<sub>6</sub> signal, whereas Tb-MOF-76(NH<sub>2</sub>) retained C<sub>2</sub>H<sub>6</sub> until reaching 58.5, 50.7, and 44.1 min g<sup>-1</sup> for 1/1, 1/9, and 1/15 C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures, respectively.

The productivities of ≥ 99.5 % and ≥ 99.95 % C<sub>2</sub>H<sub>4</sub> purity were calculated on the basis of simulating experimental breakthrough curves to compare separation performance (Table S4). For Tb-MOF-76(NH<sub>2</sub>), 6.20, 11.68, and 17.66 L kg<sup>-1</sup> of C<sub>2</sub>H<sub>4</sub> with ≥ 99.95 % purity can be recovered from the 1/1, 1/9, and 1/15 C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures in one cycle; by contrast, the corresponding values of 1.90, 4.48, and 7.53 L kg<sup>-1</sup> for Tb-MOF-76 are significantly lower. Notably,

in industrial practice without containing inert carrier gas, the C<sub>2</sub>H<sub>4</sub> productivity values for Tb-MOF-76(NH<sub>2</sub>) would be higher than the found in our experiments. The value of 17.66 L kg<sup>-1</sup> C<sub>2</sub>H<sub>4</sub> (≥ 99.95 %) for Tb-MOF-76(NH<sub>2</sub>) is nearly 2.4 times for Tb-MOF-76 with 7.53 L kg<sup>-1</sup> (≥ 99.95 %), 2.5 times for HOF-76a with 7.2 L kg<sup>-1</sup> (> 99.9 %),<sup>[25]</sup> 2.8 times for MAF-49 with 6.27 L kg<sup>-1</sup> (> 99.95 %),<sup>[7b]</sup> 4 times for Cu(Qc)<sub>2</sub> with 4.4 L kg<sup>-1</sup> (> 99.9 %),<sup>[11]</sup> and only trails than top-performing JNU-2 with 21.1 L kg<sup>-1</sup> (> 99.99 %)<sup>[17]</sup> and Fe<sub>2</sub>(O<sub>2</sub>)(dobdc) with 18.59 L kg<sup>-1</sup> (> 99.95 %)<sup>[7b]</sup> (Figure 8d). For ≥ 99.5 % C<sub>2</sub>H<sub>4</sub> purity, Tb-MOF-76(NH<sub>2</sub>) also reveals obviously higher

productivities than Tb-MOF-76 (Table S4). The higher separation potential and C<sub>2</sub>H<sub>4</sub> productivity, determined from the breakthrough curves, confirm that the introduction of –NH<sub>2</sub> groups in the framework significantly improves the separation performance of MOFs for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures.

Great recyclability and reusability of adsorbents are essential conditions for practical industrial applications. For Tb-MOF-76(NH<sub>2</sub>), we performed multiple breakthrough experiments of C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures (1/1) under the same conditions, and it showed no any deterioration in separation performance (Figure S15). PXRD patterns also confirmed that the structural stability of Tb-MOF-76(NH<sub>2</sub>) after cycle experiments (Figure S2 and S3). In a nutshell, Tb-MOF-76(NH<sub>2</sub>) displays better one-step separation performance for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures than the reported C<sub>2</sub>H<sub>6</sub>-selective materials in references, in comprehensive consideration of separation potential, C<sub>2</sub>H<sub>4</sub> productivity, C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> selectivity, C<sub>2</sub>H<sub>6</sub> uptake, and C<sub>2</sub>H<sub>6</sub> Q<sub>st</sub> (Figure 8e). These advances enable Tb-MOF-76(NH<sub>2</sub>) to be one of excellent materials for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation.

Considering the real application environments, the stability of Tb-MOF-76(NH<sub>2</sub>) toward air, water humidity, and acid-base environments was monitored by PXRD. As shown in Figure S16, after the samples were exposed in air for 40 days, relative humidity (65%) for 10 days, and different aqueous solutions with pH=3–10 for 1 day, it remains intact with no obvious phase transformation.

## Conclusion

In summary, we have performed an uncommon and crucial systematic comparative example to tune the pore environment by pore engineering strategy using amino-functionalization for improving C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation. The pore environments were successfully engineered by adopting the amino-functionalized linkers to replacing parent linkers in Tb-MOF-76. The obtained material Tb-MOF-76(NH<sub>2</sub>) with the rich –NH<sub>2</sub> groups in pores displays significantly high C<sub>2</sub>H<sub>6</sub> uptakes and C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> adsorption selectivity compared to Tb-MOF-76 (2.1 vs 1.7). Consequently, Tb-MOF-76(NH<sub>2</sub>) greatly improves the separation performance for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures, in which 17.66 Lkg<sup>−1</sup> polymer-grade C<sub>2</sub>H<sub>4</sub> product (≥99.95%) can be directly collected in a single breakthrough process compared to 7.53 Lkg<sup>−1</sup> by Tb-MOF-76. Together with robust framework stability, Tb-MOF-76(NH<sub>2</sub>) would be promising for the application in multiple separation process. The amino-functionalization method presented in this work is efficacious, and will provide an important strategy to facilitate the rational design of MOF adsorbents for efficient challenging industrial C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation.

## Acknowledgements

This work is supported by National Nature Science Foundation of China (21871220 and 21971207).

## Conflict of Interest

The authors declare no conflict of interest.

## Data Availability Statement

The data that support the findings of this study are available in the supplementary material of this article.

**Keywords:** Amino-Functionalization · C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> Separation · Gas Adsorption · Metal–Organic Framework · Porous Engineering

- [1] a) D. S. Sholl, R. P. Lively, *Nature* **2016**, *532*, 435–437; b) L. Zhang, L. Li, E. Hu, L. Yang, K. Shao, L. Yao, K. Jiang, Y. Cui, Y. Yang, B. Li, B. Chen, G. Qian, *Adv. Sci.* **2020**, *7*, 1901918.
- [2] a) J.-R. Li, R. J. Kupplera, H.-C. Zhou, *Chem. Soc. Rev.* **2009**, *38*, 1477–1504; b) M. Kang, S. Yoon, S. Ga, D. W. Kang, S. Han, J. H. Choe, H. Kim, D. W. Kim, Y. G. Chung, C. S. Hong, *Adv. Sci.* **2021**, *8*, 2004940.
- [3] a) H. Wang, D. Luo, E. Velasco, L. Yu, J. Li, *J. Mater. Chem. A* **2021**, *9*, 20874–20896; b) P.-D. Zhang, X. Zhang, X.-Q. Wu, Z.-C. Xu, J.-R. Li, *AIChE J.* **2022**, e17752; c) P. Hu, J. Hu, H. Wang, H. Liu, J. Zhou, Y. Liu, Y. Wang, H. Ji, *ACS Appl. Mater. Interfaces* **2022**, *14*, 15195–15204.
- [4] S. Chu, Y. Cui, N. Liu, *Nat. Mater.* **2017**, *16*, 16–22.
- [5] a) S. Zhou, O. Shekhah, A. Ramírez, P. Lyu, E. Abou-Hamad, J. Jia, J. Li, P. M. Bhatt, Z. Huang, H. Jiang, T. Jin, G. Maurin, J. Gascon, M. Eddaoudi, *Nature* **2022**, *606*, 706–712; b) H. Zeng, M. Xie, T. Wang, R.-J. Wei, X.-J. Xie, Y. Zhao, W. Lu, D. Li, *Nature* **2021**, *595*, 542–548; c) A. Cadiau, K. Adil, P. M. Bhatt, Y. Belmabkhout, M. Eddaoudi, *Science* **2016**, *353*, 137–140; d) P.-Q. Liao, N.-Y. Huang, W.-X. Zhang, J.-P. Zhang, X.-M. Chen, *Science* **2017**, *356*, 1193–1196; e) G.-D. Wang, Y.-Z. Li, W.-J. Shi, L. Hou, Y.-Y. Wang, Z. Zhu, *Angew. Chem. Int. Ed.* **2022**, *61*, e202205427; *Angew. Chem.* **2022**, *134*, e202205427.
- [6] a) S. Mukherjee, D. Sensharma, K.-J. Chen, M. J. Zaworotko, *Chem. Commun.* **2020**, *56*, 10419–10441; b) S. Geng, E. Lin, X. Li, W. Liu, T. Wang, Z. Wang, D. Sensharma, S. Darwish, Y. H. Andaloussi, T. Pham, P. Cheng, M. J. Zaworotko, Y. Chen, Z. Zhang, *J. Am. Chem. Soc.* **2021**, *143*, 8654–8660.
- [7] a) F. Jin, E. Lin, T. Wang, S. Geng, T. Wang, W. Liu, F. Xiong, Z. Wang, Y. Chen, P. Cheng, Z. Zhang, *J. Am. Chem. Soc.* **2022**, *144*, 5643–5652; b) L. Li, R.-B. Lin, R. Krishna, H. Li, S. Xiang, H. Wu, J. Li, W. Zhou, B. Chen, *Science* **2018**, *362*, 443–446.
- [8] a) S.-Q. Yang, F.-Z. Sun, P. Liu, L. Li, R. Krishna, Y.-H. Zhang, Q. Li, L. Zhou, T.-L. Hu, *ACS Appl. Mater. Interfaces* **2021**, *13*, 962–969; b) X.-W. Gu, J.-X. Wang, E. Wu, H. Wu, W. Zhou, G. Qian, B. Chen, B. Li, *J. Am. Chem. Soc.* **2022**, *144*, 2614–2623; c) B. Zhu, J.-W. Cao, S. Mukherjee, T. Pham, T. Zhang, T. Wang, X. Jiang, K. A. Forrest, M. J. Zaworotko, K.-J. Chen, *J. Am. Chem. Soc.* **2021**, *143*, 1485–1492; d) Y. Zhang, C. Hao, W. Fan, M. Fu, X. Wang, Z. Wang, L. Zhu, Y. Li, X. Lu, F. Dai, Z. Kang, R. Wang, W. Guo, S. Hu, D. Sun, *Angew. Chem. Int. Ed.* **2021**, *60*, 11350–11358; *Angew. Chem.* **2021**, *133*, 11451–11459; e) C.-X. Chen, Z.-W. Wei, T. Pham, P. C. Lan, L. Zhang, K. A. Forrest, S. Chen, A. M. Al-Enizi, A. Nafady, C.-Y. Su, S. Ma, *Angew. Chem. Int. Ed.* **2021**, *60*, 9680–9685; *Angew. Chem.* **2021**, *133*, 9766–9771.



- [9] a) G.-D. Wang, J. Chen, Y.-Z. Li, L. Hou, Y.-Y. Wang, Z. Zhu, *Chem. Eng. J.* **2022**, *433*, 133786; b) Q. Hong, W. Wang, S. Chen, K. Chen, M. Liu, H.-X. Zhang, J. Zhang, *Chem. Mater.* **2022**, *34*, 307–313; c) K. H. Cho, J. W. Yoon, J. H. Lee, J. C. Kim, K. Kim, U.-H. Lee, M. Choi, S. K. Kwak, J.-S. Chang, *J. Mater. Chem. A* **2021**, *9*, 14593–14600; d) S. Jiang, L. Li, L. Guo, C. Song, Q. Yang, Z. Zhang, Y. Yang, Q. Ren, Z. Bao, *Sci. China Chem.* **2021**, *64*, 666–672; e) Y.-P. Li, Y.-N. Zhao, S.-N. Li, D.-Q. Yuan, Y.-C. Jiang, X. Bu, M.-C. Hu, Q.-G. Zhai, *Adv. Sci.* **2021**, *8*, 2003141.
- [10] J. Pei, J.-X. Wang, K. Shao, Y. Yang, Y. Cui, H. Wu, W. Zhou, B. Li, G. Qian, *J. Mater. Chem. A* **2020**, *8*, 3613–3620.
- [11] R.-B. Lin, H. Wu, L. Li, X.-L. Tang, Z. Li, J. Gao, H. Cui, W. Zhou, B. Chen, *J. Am. Chem. Soc.* **2018**, *140*, 12940–12946.
- [12] a) B. Arstad, H. Fjellvåg, K. O. Kongshaug, O. Swang, R. Blom, *Adsorption* **2008**, *14*, 755–762; b) X. Zhang, R. B. Lin, H. Wu, Y. Huang, Y. Ye, J. Duan, W. Zhou, J.-R. Li, B. Chen, *Chem. Eng. J.* **2022**, *431*, 134184; c) S. Couck, J. F. M. Denayer, G. V. Baron, T. Rémy, J. Gascon, F. Kapteijn, *J. Am. Chem. Soc.* **2009**, *131*, 6326–6327; d) Kalmutzki, R. M. Altamimi, F. Fathieh, J. A. Reimer, O. M. Yaghi, *J. Am. Chem. Soc.* **2017**, *139*, 12125–12128; e) B. Dinakar, A. C. Forse, H. Z. H. Jiang, Z. Zhu, J.-H. Lee, E. J. Kim, S. T. Parker, C. J. Pollak, R. L. Siegelman, P. J. Milner, J. A. Reimer, J. R. Long, *J. Am. Chem. Soc.* **2021**, *143*, 15258–15270; f) E. J. Kim, R. L. Siegelman, H. Z. H. Jiang, A. C. Forse, J.-H. Lee, J. D. Martell, P. J. Milner, J. M. Falkowski, J. B. Neaton, J. A. Reimer, S. C. Weston, J. R. Long, *Science* **2020**, *369*, 392–396.
- [13] a) P.-Q. Liao, W.-X. Zhang, J.-P. Zhang, X.-M. Chen, *Nat. Commun.* **2015**, *6*, 8697; b) T.-L. Hu, H. Wang, B. Li, R. Krishna, H. Wu, W. Zhou, Y. Zhao, Y. Han, X. Wang, W. Zhu, Z. Yao, S. Xiang, B. Chen, *Nat. Commun.* **2015**, *6*, 7328.
- [14] C. He, Y. Wang, Y. Chen, X. Wang, J. Yang, L. Li, J. Li, *Ind. Eng. Chem. Res.* **2020**, *59*, 6123–6129.
- [15] Deposition Number 2192249 contains the supplementary crystallographic data for this paper. These data are provided free of charge by the joint Cambridge Crystallographic Data Centre and Fachinformationszentrum Karlsruhe Access Structures service.
- [16] H. Yang, Y. Wang, R. Krishna, X. Jia, Y. Wang, A. N. Hong, C. Dang, H. E. Castillo, X. Bu, P. Feng, *J. Am. Chem. Soc.* **2020**, *142*, 2222–2227.
- [17] H. Zeng, X.-J. Xie, M. Xie, Y.-L. Huang, D. Luo, T. Wang, Y. Zhao, W. Lu, D. Li, *J. Am. Chem. Soc.* **2019**, *141*, 20390–20396.
- [18] H.-G. Hao, Y.-F. Zhao, D.-M. Chen, J.-M. Yu, K. Tan, S. Ma, Y. Chabal, Z.-M. Zhang, J.-M. Dou, Z.-H. Xiao, G. Day, H.-C. Zhou, T.-B. Lu, *Angew. Chem. Int. Ed.* **2018**, *57*, 16067–16071; *Angew. Chem.* **2018**, *130*, 16299–16303.
- [19] J. Liu, J. Miao, S. Ullah, K. Zhou, L. Yu, H. Wang, Y. Wang, T. Thonhauser, J. Li, *ACS Mater. Lett.* **2022**, *4*, 1227–1232.
- [20] O. T. Qazvini, R. Babarao, Z.-L. Shi, Y.-B. Zhang, S. G. Telfer, *J. Am. Chem. Soc.* **2019**, *141*, 5014–5020.
- [21] Z. Xu, X. Xiong, J. Xiong, R. Krishna, L. Li, Y. Fan, F. Luo, B. Chen, *Nat. Commun.* **2020**, *11*, 3163.
- [22] U. Böhme, B. Barth, C. Paula, A. Kuhnt, W. Schwieger, A. Mundstock, J. Caro, M. Hartmann, *Langmuir* **2013**, *29*, 8592–8600.
- [23] a) R. Krishna, *RSC Adv.* **2017**, *7*, 35724–35737; b) R. Krishna, *ACS Omega* **2020**, *5*, 16987–17004.
- [24] Y. Chen, Z. Qiao, H. Wu, D. Lv, R. Shi, Q. Xia, J. Zhou, Z. Li, *Chem. Eng. Sci.* **2018**, *175*, 110–117.
- [25] X. Zhang, L. Li, J.-X. Wang, H.-M. Wen, R. Krishna, H. Wu, W. Zhou, Z.-N. Chen, B. Li, G. Qian, B. Chen, *J. Am. Chem. Soc.* **2020**, *142*, 633–640.

Manuscript received: September 4, 2022

Accepted manuscript online: October 6, 2022

Version of record online: October 26, 2022



## Supporting Information

### **Boosting Ethane/Ethylene Separation by MOFs through the Amino-Functionalization of Pores**

*G.-D. Wang, R. Krishna, Y.-Z. Li, W.-J. Shi, L. Hou\*, Y.-Y. Wang, Z. Zhu*



---

**Table of Contents**

<b>Materials and general methods</b>	S3
<b>Synthesis of MOFs</b>	S4
<b>Powder X-ray diffraction (PXRD)</b>	S5
<b>X-ray crystallography</b>	S6
<b>Thermogravimetric analysis (TGA)</b>	S8
<b>Fitting adsorption heat of pure component isotherms</b>	S9
<b>Gas adsorption isotherm</b>	S11
<b>Gas selectivity prediction via IAST</b>	S12
<b>Molecular simulations</b>	S13
<b>Transient breakthrough simulations</b>	S14
<b>Breakthrough experiment</b>	S15
<b>Stability test</b>	S16
<b>Screening result for some high-performance C<sub>2</sub>H<sub>6</sub> MOF adsorbents</b>	S17
<b>References</b>	S18
<b>Author contributions</b>	S18

---

**Materials and general methods**

The chemicals were purchased commercially from Adamas-beta. Elemental analyses of C, H, and N were determined with a Perkin-Elmer 2400C elemental analyzer. Thermalgravimetric analyses (TGA) were carried out in a nitrogen stream using a Netzsch TG209F3 equipment at a heating rate of 10 °C min<sup>-1</sup>. Single crystal diffraction data were collected on a Bruker SMART APEX II CCD single crystal diffractometer (Supporting Information). Gas adsorption curves were collected with Micromeritics ASAP 2020M and TriStar II 3020 equipments. Breakthrough experiments were performed on a Quantachrome dynaSorb BT equipment.

## Synthesis of MOFs

### Synthesis of Tb-MOF-76.

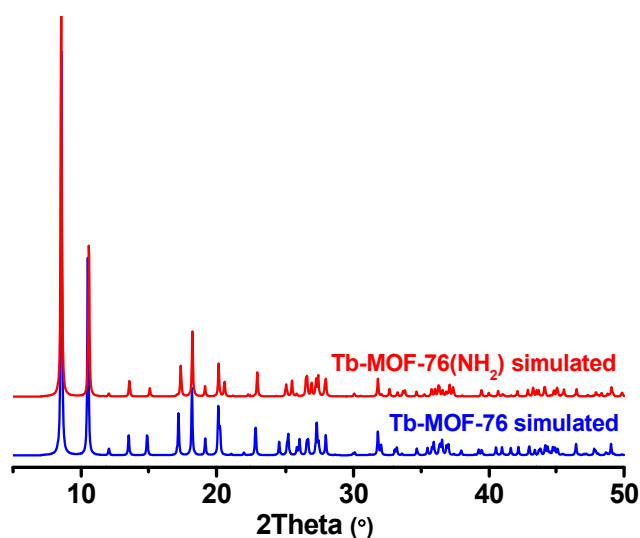
Tb-MOF-76 was synthesized according to reported literature procedure with slight modification.<sup>[1]</sup> In a typical process, Tb(NO<sub>3</sub>)·6H<sub>2</sub>O (0.0456 g), 1,3,5-benzenetricarboxylate (BTC) (0.021 g), N,N-dimethylformamide (DMF) (4 mL), and H<sub>2</sub>O (4 mL) were placed in a 15 mL Teflon-lined reactor. The mixture was heated at 105 °C for 72 h and slowly cooled to room temperature. Needle-like crystals of MOF-76 were collected by filtration.

### Synthesis of Tb-MOF-76(NH<sub>2</sub>).

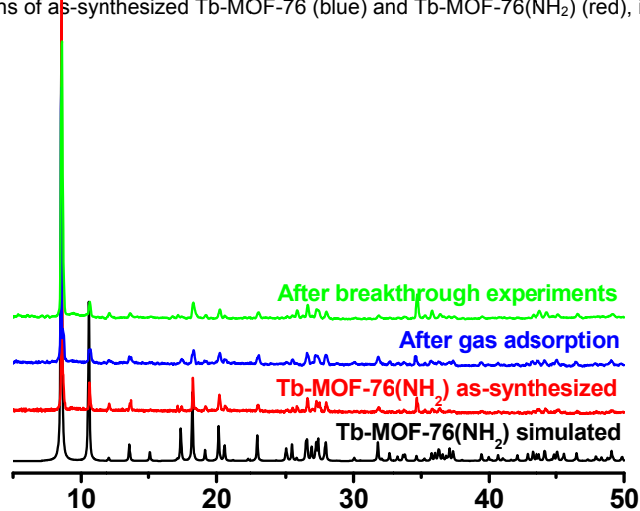
In a typical process, Tb(NO<sub>3</sub>)·6H<sub>2</sub>O (0.0456 g), 1,3,5-benzenetricarboxylate-NH<sub>2</sub> (BTC-NH<sub>2</sub>) (0.022 g), N-Methylformamide (NMF) (3 mL), CH<sub>3</sub>CN (0.5 mL), H<sub>2</sub>O (0.5 mL) and HCl (0.5 mL) were placed in a 15 mL Teflon-lined reactor. The mixture was heated at 130 °C for 72 h and slowly cooled to room temperature. Needle-like crystals of MOF-76(NH<sub>2</sub>) were collected by filtration.



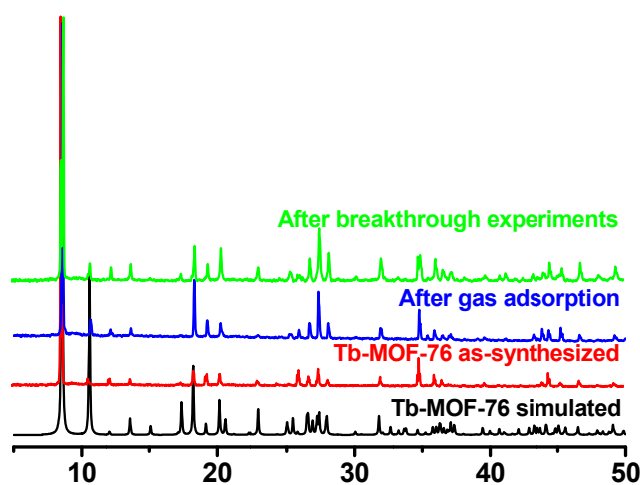
## Powder X-ray diffraction (PXRD)



**Figure S1.** The PXRD patterns of as-synthesized Tb-MOF-76 (blue) and Tb-MOF-76(NH<sub>2</sub>) (red), indicating their same structure.



**Figure S2.** The calculated PXRD pattern from the crystal structure of Tb-MOF-76(NH<sub>2</sub>) (black) and PXRD patterns of as-synthesized Tb-MOF-76(NH<sub>2</sub>) (red), after gas adsorption samples (blue) and after breakthrough experiments (green).



**Figure S3.** The calculated PXRD pattern from the crystal structure of Tb-MOF-76 (black) and PXRD patterns of as-synthesized Tb-MOF-76 (red), after gas adsorption samples (blue) and after breakthrough experiments (green).

## X-ray crystallography

A Bruker Smart Apex II CCD detector was used to collect the single crystal data of Tb-MOF-76(NH<sub>2</sub>) at 180(2) K using Mo K $\alpha$  radiation ( $\lambda = 0.71073$  Å). The structure was solved by direct methods and refined by full-matrix least-squares refinement based on  $F^2$  with the SHELXTL program. The non-hydrogen atoms were refined anisotropically with the hydrogen atoms added at their geometrically ideal positions and refined isotropically. As the disordered solvent molecules in the structure cannot be located, the SQUEEZE routine of Platon program was applied in refining. The formula of complex was get by the single crystal analysis together with elemental microanalyses and TGA data. Relevant crystallographic results are listed in Table S1. Selected bond lengths and angles are provided in Table S2.

**Table S1.** Crystallographic data for Tb-MOF-76(NH<sub>2</sub>) and Tb-MOF-76.<sup>[1]</sup>

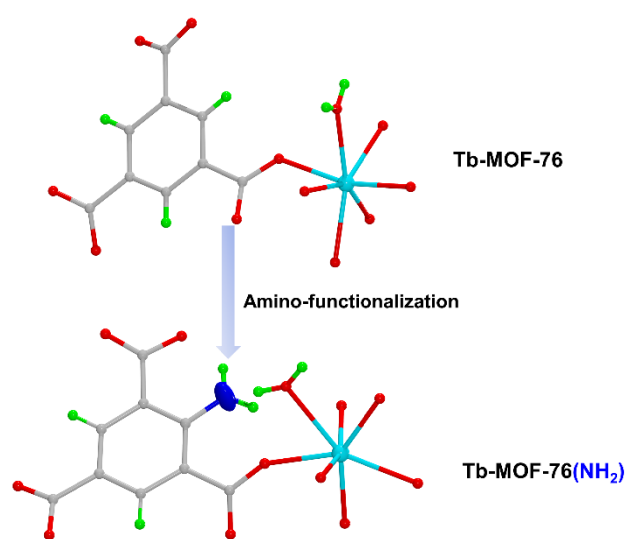
Chemical formula	C <sub>9</sub> H <sub>6</sub> TbNO <sub>7</sub>	C <sub>9</sub> H <sub>5</sub> TbO <sub>7</sub>
Formula weight	399.07	384.05
<i>T</i> (K)	180(2) K	----
Crystal system, Space group	Tetragonal, <i>P</i> 4 <sub>1</sub> 22	Tetragonal, <i>P</i> 4 <sub>3</sub> 22
<i>a</i> (Å), <i>b</i> (Å), <i>c</i> (Å)	10.3464(3)	10.3300(15)
<i>b</i> (Å)	10.3464(3)	10.3300(15)
<i>c</i> (Å)	14.2185(8)	14.510(3)
$\alpha$ (°), $\beta$ (°), $\gamma$ (°)	90,90,90	90,90,90
<i>V</i> (Å <sup>3</sup> )	1522.06(12)	1548.3(4)
<i>Z</i> , <i>D</i> <sub>calcd</sub> [g·cm <sup>-3</sup> ]	4, 1.742	4, 1.639
$\mu$ (mm <sup>-1</sup> ), Goof	4.666, 1.095	4.581, 1.096
Reflns collected/unique/ <i>R</i> <sub>int</sub>	13955/ 1402/ 0.0326	15015/ 1768/ 0.0468
<i>R</i> <sub>1</sub> <sup>a</sup> , <i>wR</i> <sub>2</sub> <sup>b</sup> [ <i>I</i> > 2 $\sigma$ ]	0.0208, 0.0517	0.0188, 0.0453
<i>R</i> <sub>1</sub> <sup>a</sup> , <i>wR</i> <sub>2</sub> <sup>b</sup> (all data)	0.0215, 0.0521	0.0210, 0.0458

$${}^a R_1 = \sum(|F_o| - |F_c|) / \sum |F_o|, \quad {}^b R_2 = [\sum w(F_o^2 - F_c^2)^2 / \sum w(F_o^2)^2]^{1/2}.$$

**Table S2.** Selected bond lengths (Å) and bond angles (deg) for Tb-MOF-76(NH<sub>2</sub>).

Tb(1)-O(1)	2.287(5)	O(1)-Tb(1)-O(2)#4	147.3(2)
Tb(1)-O(1)#1	2.287(5)	O(1)#1-Tb(1)-O(2)#4	75.2(2)
Tb(1)-O(3)#2	2.298(5)	O(3)#2-Tb(1)-O(2)#4	105.3(3)
Tb(1)-O(3)#3	2.298(5)	O(3)#3-Tb(1)-O(2)#4	92.3(3)
Tb(1)-O(2)#4	2.324(6)	O(1)-Tb(1)-O(2)#5	75.2(2)
Tb(1)-O(2)#5	2.324(6)	O(1)#1-Tb(1)-O(2)#5	147.3(2)
Tb(1)-O(1W)	2.501(14)	O(3)#2-Tb(1)-O(2)#5	92.3(3)
O(2)-Tb(1)#7	2.324(6)	O(3)#3-Tb(1)-O(2)#5	105.3(3)
O(3)-Tb(1)#8	2.298(5)	O(2)#4-Tb(1)-O(2)#5	72.1(3)
O(1)-Tb(1)-O(1)#1	137.4(3)	O(1)-Tb(1)-O(1W)	68.72(17)
O(1)-Tb(1)-O(3)#2	77.7(2)	O(1)#1-Tb(1)-O(1W)	68.72(17)
O(1)#1-Tb(1)-O(3)#2	94.4(2)	O(3)#2-Tb(1)-O(1W)	79.2(2)
O(1)-Tb(1)-O(3)#3	94.4(2)	O(3)#3-Tb(1)-O(1W)	79.2(2)
O(1)#1-Tb(1)-O(3)#3	77.7(2)	O(2)#4-Tb(1)-O(1W)	143.93(16)
O(3)#2-Tb(1)-O(3)#3	158.3(4)	O(2)#5-Tb(1)-O(1W)	143.93(16)

Symmetry codes: #1 -y+1,-x+1,-z+1/4; #2 x,-y+2,-z+1/2; #3 y-1,-x+1,z-1/4; #4 x,y-1,z; #5 -y+2,-x+1,-z+1/4.



**Figure S4.** Coordination environments of Tb<sup>3+</sup> ions in Tb-MOF-76 and Tb-MOF-76(NH<sub>2</sub>).



## Thermogravimetric analysis (TGA)

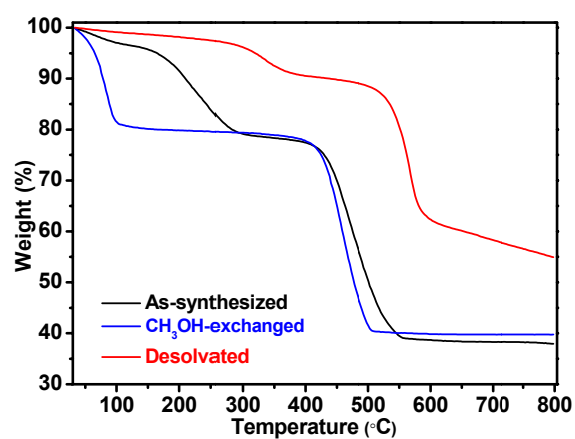


Figure S5. TGA curves of as-synthesized, CH<sub>3</sub>OH-exchanged, and desolvated samples of Tb-MOF-76(NH<sub>2</sub>).

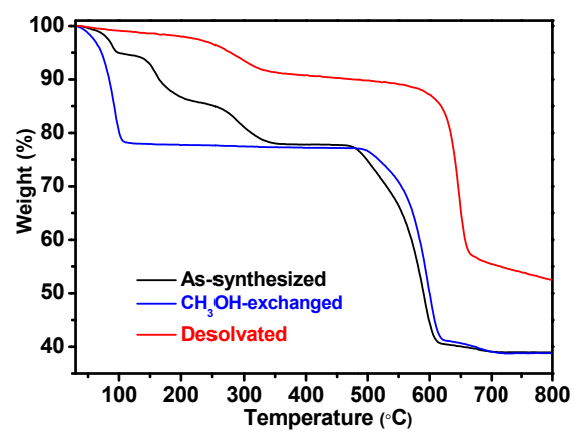
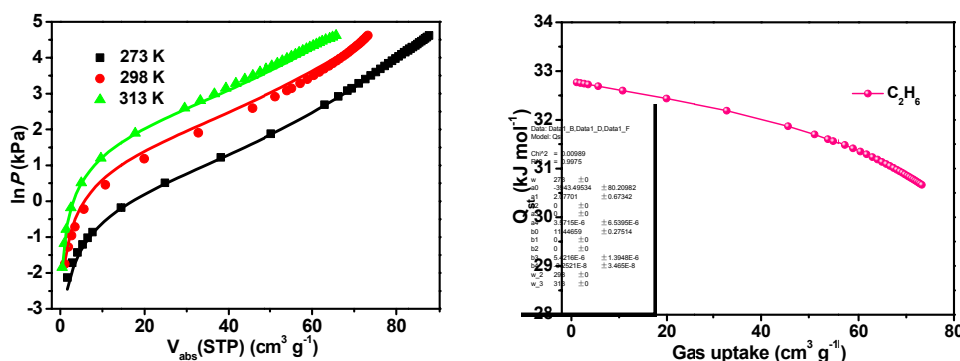


Figure S6. TGA curves of as-synthesized, CH<sub>3</sub>OH-exchanged, and desolvated samples of Tb-MOF-76.

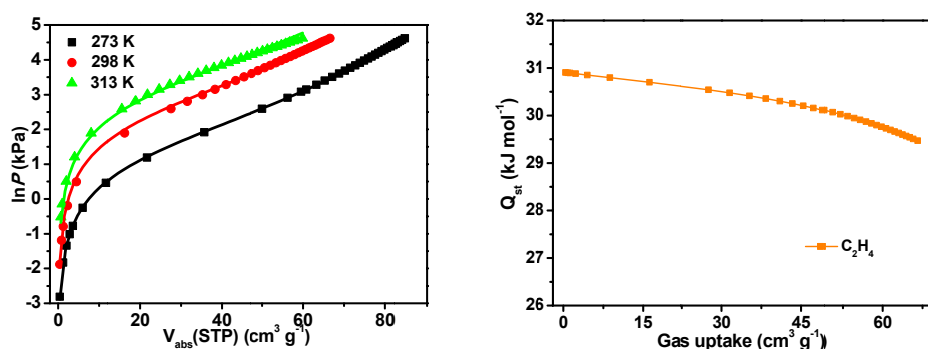
## Fitting adsorption heat of pure component isotherms

$$\ln P = \ln N + 1/T \sum_{i=0}^m a_i N^i + \sum_{i=0}^n b_i N^i \quad Q_{st} = -R \sum_{i=0}^m a_i N^i$$

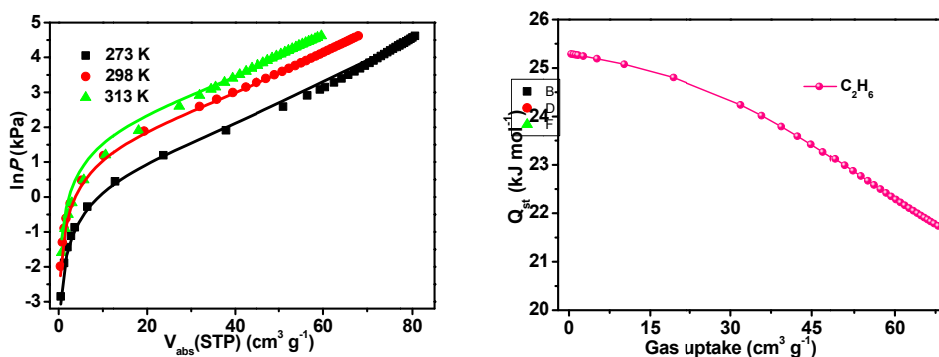
The above virial expression was used to fit the combined isotherm data of the MOFs at 273, 298, and 313 K, where  $P$  is the pressure,  $N$  is the adsorbed amount,  $T$  is the temperature,  $a_i$  and  $b_i$  are virial coefficients, and  $m$  and  $n$  are the number of coefficients used to describe the isotherms.  $Q_{st}$  is the coverage-dependent enthalpy of adsorption and  $R$  is the universal gas constant.



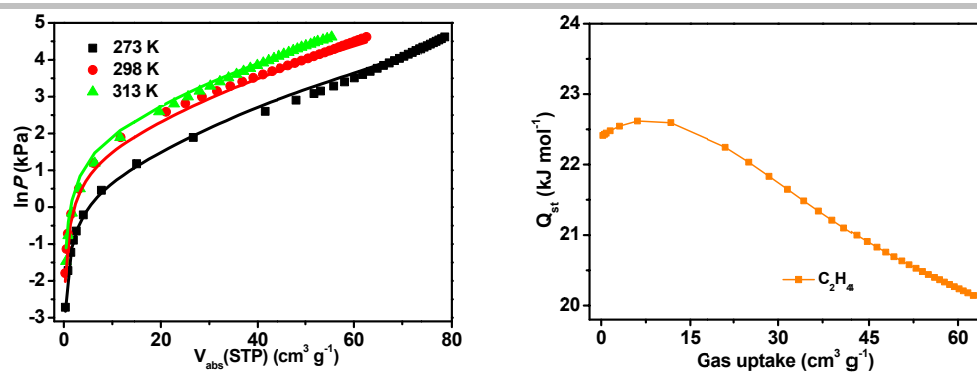
**Figure S7.** Fitted  $C_2H_6$  isotherms of Tb-MOF-76( $NH_2$ ), and the isosteric heats of adsorption ( $Q_{st}$ ). Fitting results,  $a_0 = -3943.49534$ ,  $a_1 = 2.07701$ ,  $a_4 = 3.5715E-6$ ,  $b_0 = 11.44659$ ,  $b_3 = 5.4216E-6$ ,  $b_4 = -3.2521E-8$ ,  $\chi^2 = 0.00989$ ,  $R^2 = 0.9975$ .



**Figure S8.** Fitted  $C_2H_4$  isotherms of Tb-MOF-76( $NH_2$ ), and the isosteric heats of adsorption ( $Q_{st}$ ). Fitting results,  $a_0 = -3717.7822$ ,  $a_1 = 1.52923$ ,  $a_4 = 3.639E-6$ ,  $b_0 = 11.55632$ ,  $b_2 = 0.00016$ ,  $\chi^2 = 0.00369$ ,  $R^2 = 0.99891$ .



**Figure S9.** Fitted  $C_2H_6$  isotherms of Tb-MOF-76, and the isosteric heats of adsorption ( $Q_{st}$ ). Fitting results,  $a_0 = -3042.87311$ ,  $a_1 = 2.41366$ ,  $a_3 = 0.0022$ ,  $a_4 = -0.00002$ ,  $b_0 = 8.8515$ ,  $\chi^2 = 0.02077$ ,  $R^2 = 0.9946$ .



**Figure S10.** Fitted  $\text{C}_2\text{H}_4$  isotherms of Tb-MOF-76, and the isosteric heats of adsorption ( $Q_{\text{st}}$ ). Fitting results,  $a_0 = -2693.8611$ ,  $a_1 = -7.49522$ ,  $a_2 = 0.56767$ ,  $a_3 = -0.00918$ ,  $a_4 = 0.00005$ ,  $b_0 = 8.31135$ ,  $\text{Chi}^2 = 0.01658$ ,  $R^2 = 0.99532$ .



## Gas adsorption isotherm

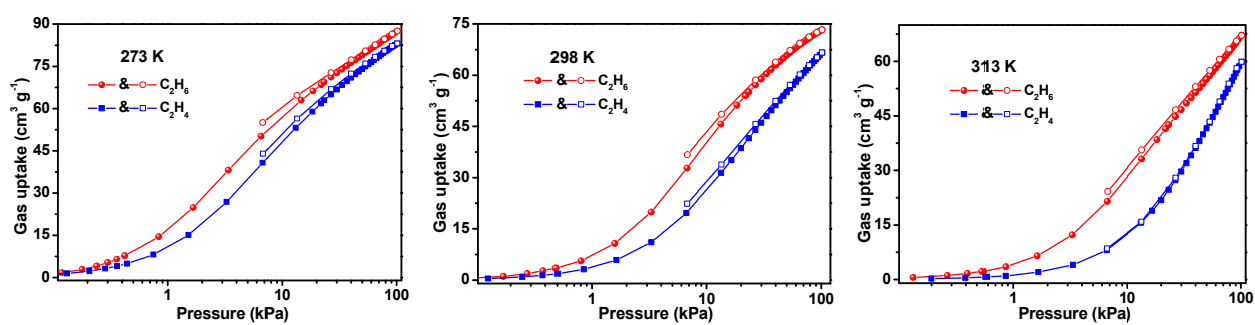


Figure S11. Gas adsorption isotherms of Tb-MOF-76(NH<sub>2</sub>) for C<sub>2</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>4</sub> at 273, 298, and 313 K.

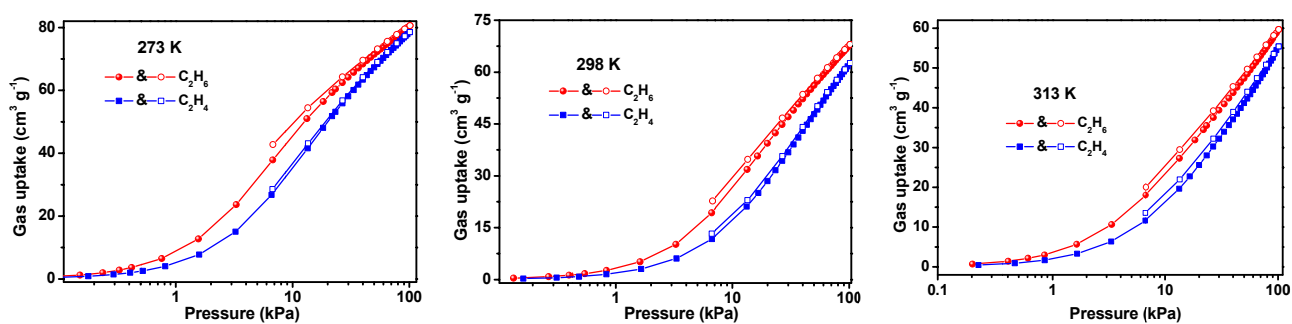


Figure S12. Gas adsorption isotherms of Tb-MOF-76 for C<sub>2</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>4</sub> at 273, 298, and 313 K.

## Gas selectivity prediction via IAST

The experimental isotherm data for pure C<sub>2</sub>H<sub>4</sub> and C<sub>2</sub>H<sub>6</sub> were fitted using a dual Langmuir-Freundlich (L-F) model:

$$q = \frac{a_1 * b_1 * P^{c_1}}{1 + b_1 * P^{c_1}} + \frac{a_2 * b_2 * P^{c_2}}{1 + b_2 * P^{c_2}}$$

Where  $q$  and  $p$  are adsorbed amounts and the pressure of component  $i$ , respectively.

The adsorption selectivities for binary mixtures defined by

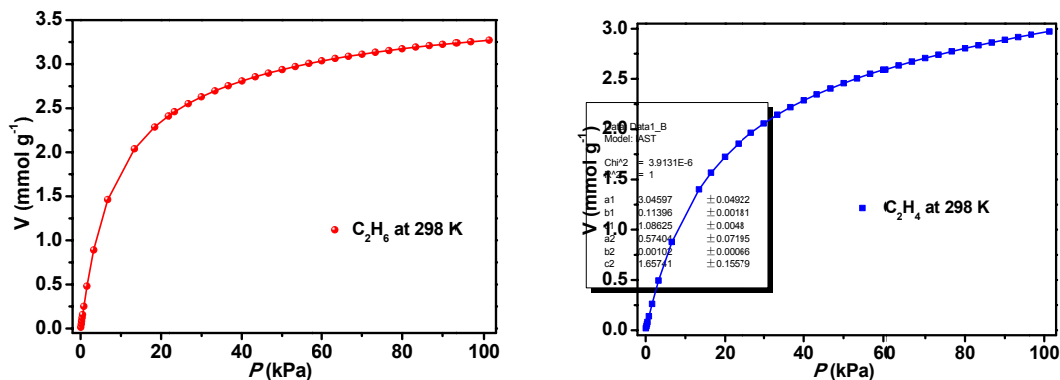
$$S_{i/j} = \frac{x_i * y_j}{x_j * y_i}$$

were respectively calculated using the Ideal Adsorption Solution Theory (IAST). Where  $x_i$  is the mole fraction of component  $i$  in the adsorbed phase and  $y_i$  is the mole fraction of component  $i$  in the bulk.

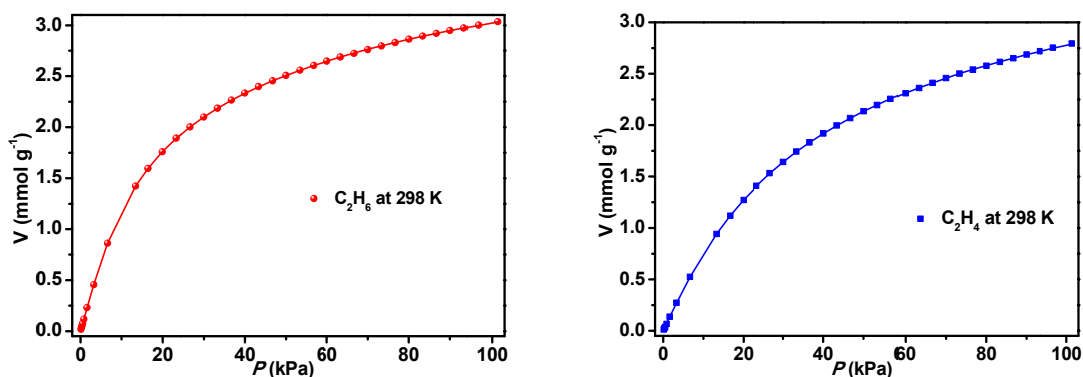
Besides adsorption selectivities, a combined selectivity-capacity metric, called the separation potential,  $\Delta Q$ , was firstly introduced by Rajamani Krishna<sup>[2,3]</sup> to reflect mixture separations capability of the MOF. For a C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixture with mole fractions  $y_{C_2H_6}$ , and  $y_{C_2H_4} = 1 - y_{C_2H_6}$ ,  $\Delta Q$  is calculated from IAST using the formula

$$\Delta Q = (q_{C_2H_6} \frac{y_{C_2H_4}}{y_{C_2H_6}} - q_{C_2H_4}) \rho$$

where  $\rho$  is the crystal framework density of the MOF, expressed say in units of kg m<sup>-3</sup>, or g cm<sup>-3</sup>.



**Figure S13.** C<sub>2</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>4</sub> adsorption isotherms of Tb-MOF-76(NH<sub>2</sub>) with fitted by dual L-F model at 298 K, C<sub>2</sub>H<sub>6</sub>: a<sub>1</sub> = 3.04597, b<sub>1</sub> = 0.11396, c<sub>1</sub> = 1.08625, a<sub>2</sub> = 0.57404, b<sub>2</sub> = 0.00102, c<sub>2</sub> = 1.65741, Chi<sup>2</sup> = 3.9131E-6, R<sup>2</sup> = 1; C<sub>2</sub>H<sub>4</sub>: a<sub>1</sub> = 2.96235, b<sub>1</sub> = 0.05795, c<sub>1</sub> = 1.04013, a<sub>2</sub> = 1.1687, b<sub>2</sub> = 0.00053, c<sub>2</sub> = 1.47522, Chi<sup>2</sup> = 7.4485E-7, R<sup>2</sup> = 1.



**Figure S14.** C<sub>2</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>4</sub> adsorption isotherms of Tb-MOF-76 with fitted by dual L-F model at 298 K, C<sub>2</sub>H<sub>6</sub>: a<sub>1</sub> = 3.17612, b<sub>1</sub> = 0.02884, c<sub>1</sub> = 0.81281, a<sub>2</sub> = 1.32366, b<sub>2</sub> = 0.03956, c<sub>2</sub> = 1.42285, Chi<sup>2</sup> = 4.6915E-6, R<sup>2</sup> = 1; C<sub>2</sub>H<sub>4</sub>: a<sub>1</sub> = 3.88323, b<sub>1</sub> = 0.02166, c<sub>1</sub> = 0.98921, a<sub>2</sub> = 0.16476, b<sub>2</sub> = 0.00768, c<sub>2</sub> = 2.00594, Chi<sup>2</sup> = 6.713E-6, R<sup>2</sup> = 0.99999.

### **Molecular simulations**

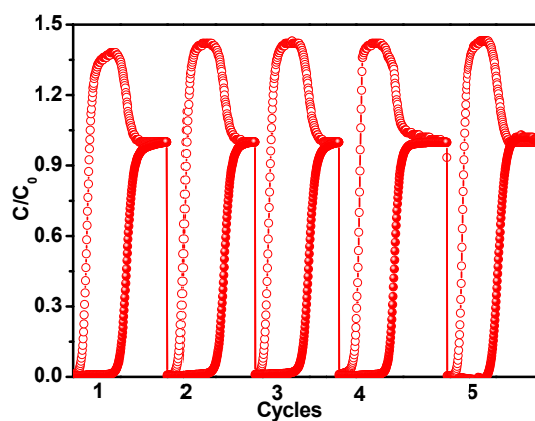
Grand canonical Monte Carlo (GCMC) simulations were performed for the gas adsorption in the framework by the Sorption module of Material Studio (Accelrys. Materials Studio Getting Started). The framework was considered to be rigid, and the optimized gas molecules were used. The partial charges for atoms of the framework were derived from QEq method and QEq neutral 1.0 parameter. One unit cell was used during the simulations. The interaction energies between the gas molecules and framework were computed through the Coulomb and Lennard-Jones 6-12 (LJ) potentials. All parameters for the atoms were modeled with the universal force field (UFF) embedded in the MS modeling package. A cutoff distance of 12.5 Å was used for LJ interactions, and the Coulombic interactions were calculated by using Ewald summation. For each run, the  $3 \times 10^6$  maximum loading steps,  $3 \times 10^6$  production steps were employed.

### Transient breakthrough simulations

Transient breakthrough simulations were carried out for the same set of operating conditions as in the experimental data sets, packed sample weight,  $w = 0.8$  g; length of packed bed,  $L = 8$  cm; diameter of packed bed,  $d = 0.42$  cm; the gas mixtures contain  $C_2H_6/C_2H_4/Ar$  (5/5/90, 1/9/90, and 1/15/84, v/v/v) with Ar as the carrier gas, and a total flow rate of  $7.0 \text{ mL min}^{-1}$  (298 K, 1 atm). using the methodology described in earlier publications.<sup>[S2-6]</sup> In these simulations, intra-crystalline diffusion influences are ignored. For Tb-MOF-76( $NH_2$ ), there is excellent match between the experiments and simulations. From the simulations on the experimental breakthrough curves, the productivities of  $\geq 99.95\%$  and  $\geq 99.5\%$  pure  $C_2H_4$  were determined, which are expressed in the units of L per kg of MOF.

**Breakthrough experiment**

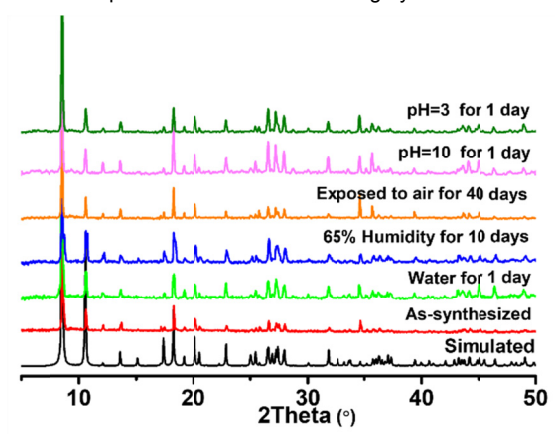
The breakthrough experiment was performed on the Quantachrome dynaSorb BT equipments at 298 K and 100 kPa with different ratios of mixed gas ( $C_2H_6/C_2H_4/Ar = 5/5/90$ ,  $1/9/90$ , and  $1/15/84$ , v/v/v, Ar as the carrier gas, flow rate =  $7 \text{ mL min}^{-1}$ ). The activated Tb-MOF-76( $NH_2$ ) and Tb-MOF-76 (about 0.8 g) was filled into a packed column of  $\phi 4.2 \times 80 \text{ mm}$ , and then the packed column was washed with Ar at a rate of  $7 \text{ mL min}^{-1}$  at 343 K for 30 minutes to further activate the samples. Between two breakthrough experiments, the adsorbent was regenerated by Ar flow of  $7 \text{ mL min}^{-1}$  for 35 min at 343 K to guarantee a complete removal of the adsorbed gas.



**Figure S15.** Five cycles of breakthrough curves of Tb-MOF-76( $NH_2$ ) for equimolar  $C_2H_6/C_2H_4$  mixtures at 298 K.

### Stability test

To investigate the chemical stabilities of Tb-MOF-76(NH<sub>2</sub>), the as-synthesized samples were soaked in water, HCl (pH = 3) and NaOH (pH = 10) solutions for 1 day, exposed to air for 40 days and 65% humidity for 10 days, respectively. And then characterized by PXRD measurements in order to determine whether the sample retains its structural integrity.



**Figure S16.** PXRD patterns of Tb-MOF-76(NH<sub>2</sub>) treated under different conditions.



Screening result for some high-performance C<sub>2</sub>H<sub>6</sub> MOF adsorbents**Table S3.** Comparison of separation potential  $\Delta Q$  for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixture of different adsorbents

Separation potential $\Delta Q$ mmol cm <sup>-3</sup>					
Adsorbents	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/1, v/v)	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/9, v/v)	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/15, v/v)	Crystal density (g cm <sup>-3</sup> )	Ref.
Tb-MOF-76(NH <sub>2</sub> )	1.91	4.61	4.95	1.742	This work
Tb-MOF-76	1.19	2.65	2.82	1.639	This work
Ni(bdc)(ted) <sub>0.5</sub>	0.94	1.85	1.95	0.830	7
Cu(Qc) <sub>2</sub>	1.27	2.49	2.57	1.492	8
PCN-250	1.41	2.96	3.13	0.957	9
MUF-15	1.56	3.47	3.70	1.245	10
Fe <sub>2</sub> (O <sub>2</sub> )dobdc	2.17	--	--	1.255	11
UPC-613	0.51	--	--	1.115	12
UiO-67-(NH <sub>2</sub> ) <sub>2</sub>	0.94	1.94	2.06	0.819	13
MAF-49	1.16	--	--	1.481	14
CPM-733	1.67	--	--	0.890	15

**Table S4.** Comparisons of C<sub>2</sub>H<sub>4</sub> productivities of Tb-MOF-76(NH<sub>2</sub>) and Tb-MOF-76 in experimental breakthrough curves using C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixture as input.

MOFs	Gravimetric productivity (L Kg <sup>-1</sup> ) with different purities of C <sub>2</sub> H <sub>4</sub>		
		≥99.5%	≥99.95%
Tb-MOF-76(NH <sub>2</sub> )	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/1, v/v)	7.53	6.20
	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/9, v/v)	14.99	11.68
	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/15, v/v)	22.66	17.66
Tb-MOF-76	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/1, v/v)	2.88	1.90
	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/9, v/v)	6.53	4.48
	C <sub>2</sub> H <sub>6</sub> /C <sub>2</sub> H <sub>4</sub> (1/15, v/v)	10.77	7.53

## References

- [1] H.-L. Jiang, N. Tsumori and Q. Xu, *Inorg. Chem.* **2010**, 49, 10001-10006.
- [2] R. Krishna, *RSC Adv.* **2017**, 7, 35724-35737.
- [3] R. Krishna, *ACS Omega* **2020**, 5, 16987-17004.
- [4] R. Krishna, *RSC Adv.* **2015**, 5, 52269-52295.
- [5] R. Krishna, *Microporous Mesoporous Mater.* **2014**, 185, 30-50.
- [6] R. Krishna, *Sep. Purif. Technol.* **2018**, 194, 281-300.
- [7] W. Liang, F. Xu, X. Zhou, J. Xiao, Q. Xia, Y. Li, Z. Li, *Chem. Eng. Sci.* **2016**, 148, 275-281.
- [8] R.-B. Lin, H. Wu, L. Li, X.-L. Tang, Z. Li, J. Gao, H. Cui, W. Zhou and B. Chen, *J. Am. Chem. Soc.* **2018**, 140, 12940-12946.
- [9] Y. Chen, Z. Qiao, H. Wu, D. Lv, R. Shi, Q. Xia, J. Zhou, Z. Li, *Chem. Eng. Sci.* **2018**, 175, 110-117.
- [10] O. T. Qazvini, R. Babarao, Z.-L. Shi, Y.-B. Zhang and S. G. Telfer, *J. Am. Chem. Soc.* **2019**, 141, 5014-5020.
- [11] L. Li, R.-B. Lin, R. Krishna, H. Li, S. Xiang, H. Wu, J. Li, W. Zhou, B. Chen, *Science* **2018**, 362, 443-446.
- [12] Y. Wang, C. Hao, W. Fan, M. Fu, X. Wang, Z. Wang, L. Zhu, Y. Li, X. Lu, F. Dai, Z. Kang, R. Wang, W. Guo, S. Hlu and D. Sun, *Angew. Chem. Int. Ed.* **2021**, 60, 11350-11358.
- [13] X.-W. Gu, J.-X. Wang, E. Wu, H. Wu, W. Zhou, G. Qian, B. Chen and B. Li, *J. Am. Chem. Soc.* **2022**, 144, 2614-2623.
- [14] P.-Q. Liao, W.-X. Zhang, J.-P. Zhang, X.-M. Chen, *Nat. Com.* **2015**, 6, 8697.
- [15] H. Yang, Y. Wang, R. Krishna, X. Jia, Y. Wang, A. N. Hong, C. Dang, H. E. Castillo, X. Bu and P. Feng, *J. Am. Chem. Soc.* **2020**, 142, 2222-2227.

## Author contributions

**Gang-Ding Wang:** Experimental, Writing - original draft. **Rajamani Krishna:** Formal analysis, Software. **Yong-Zhi Li:** Formal analysis, Writing - review & editing. **Wen-Juan Shi:** Formal analysis, Software. **Lei Hou:** Software, Writing - review & editing, Supervision, Funding acquisition. **Yao-Yu Wang:** Formal analysis, Resources. **Zhonghua Zhu:** Formal analysis.