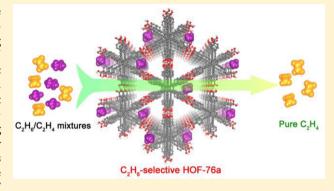


# Selective Ethane/Ethylene Separation in a Robust Microporous Hydrogen-Bonded Organic Framework

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Supporting Information

**ABSTRACT:** The separation of ethane  $(C_2H_6)$  from ethylene (C<sub>2</sub>H<sub>4</sub>) is of prime importance in the production of polymergrade C<sub>2</sub>H<sub>4</sub> for industrial manufacturing. It is very challenging and still remains unexploited to fully realize efficient C<sub>2</sub>H<sub>6</sub>/ C<sub>2</sub>H<sub>4</sub> separation in the emerging hydrogen-bonded organic frameworks (HOFs) due to the weak nature of hydrogen bonds. We herein report the benchmark example of a novel ultrarobust HOF adsorbent (termed as HOF-76a) with a Brunauer-Emmett-Teller surface area exceeding 1100 m<sup>2</sup> g<sup>-1</sup>, exhibiting the preferential binding of C<sub>2</sub>H<sub>6</sub> over C<sub>2</sub>H<sub>4</sub> and thus highly selective separation of C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub>. Theoretical calculations indicate the key role of the nonpolar surface and the suitable triangular channel-like pores in HOF-76a to sterically "match"



better with the nonplanar C<sub>2</sub>H<sub>6</sub> molecule than the planar C<sub>2</sub>H<sub>4</sub>, thus affording overall stronger multipoint van der Waals interactions with C<sub>2</sub>H<sub>6</sub>. The exceptional separation performance of HOF-76a for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation was clearly demonstrated by gas adsorption isotherms, ideal adsorbed solution theory calculations, and simulated and experimental breakthrough curves. Breakthrough experiments on HOF-76a reveal that polymer-grade ethylene gas can be straightforwardly produced from  $50/50 (v/v) C_2H_6/C_2H_4$  mixtures during the first adsorption cycle with a high productivity of 7.2 L/kg at 298 K and 1.01 bar and 18.8 L/kg at 298 K and 5.0 bar, respectively.

#### INTRODUCTION

Ethylene (C2H4) is one of the most important feedstocks in petrochemical industries with a global production capacity of exceeding 170 million tons per year, and is mostly produced by steam cracking of ethane (C2H6) and liquefied petroleum gas.<sup>1,2</sup> Steam crackers inevitably yield many other hydrocarbons (mostly C2H6) that must be removed for polymer production. The purity of ethylene thereby primarily depends on the removal of ethane; this separation process is energyintensive and conventionally dominated by the cryogenic distillation because of the close physicochemical properties of the two components. The total energy used for purification of ethylene and propene even accounts for 0.3% of global energy

consumption, highlighting as one of the most important industrial separation tasks.

Adsorbent-based gas separation has been considered as a viable alternative to replace traditional cryogenic distillation processes. Development of C<sub>2</sub>H<sub>6</sub>-selective adsorbents is more desired for C<sub>2</sub>H<sub>4</sub>/C<sub>2</sub>H<sub>6</sub> separation because pure C<sub>2</sub>H<sub>4</sub> can be directly produced during the adsorption cycle, avoiding an additional C<sub>2</sub>H<sub>4</sub> desorption step and subsequent multiple adsorption-desorption purification cycles of C<sub>2</sub>H<sub>4</sub>-selective materials and thus simplifying the separation process with

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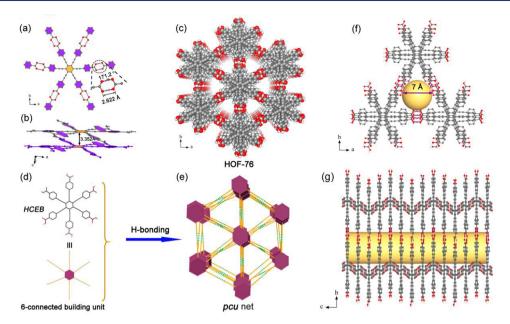


Figure 1. Crystal structure description of HOF-76. (a) View of the connection of adjacent building blocks and the associated hydrogen-bond length and angle. (b) Strong face-to-face  $\pi$ - $\pi$  stacking interactions. (c) Representation of the porous framework of HOF-76. (d) The structure of HCEB as six-connected nodes and (e) pcu-net (green dotted lines represent H-bonds). (g) One-dimensional (1D) triangular channels viewed along c-axis, revealing a moderate size of 7.0 Å in diameter.

remarkable energy saving. However, most of developed adsorbents, such as metal-containing alumina, carbons, zeolites, and metal-organic frameworks (MOFs), commonly take up more amounts of  $C_2H_4$  than  $C_2H_6$ . This is because the C<sub>2</sub>H<sub>4</sub> molecule has a larger quadrupole moment (C<sub>2</sub>H<sub>4</sub>:  $1.50 \times 10^{-26}$  esu cm<sup>2</sup>,  $C_2H_6$ :  $0.65 \times 10^{-26}$  esu cm<sup>2</sup>) and the presence of  $\pi$  electrons, rendering its stronger interactions with metal sites/clusters. Conversely, ethane has a higher polarizability than ethylene  $(44.7 \times 10^{-25} \text{ vs } 42.52 \times 10^{-25}$ cm<sup>3</sup>); so C<sub>2</sub>H<sub>6</sub>-selective adsorbents prefer to possess a pore structure enriched with nonpolar/inert surfaces (e.g., featuring aromatic or aliphatic moieties), wherein dispersion and induction interactions can make major contributions. <sup>10</sup> In this regard, inorganic secondary building units (SBUs) in MOFs are commonly detrimental to building nonpolar/inert surfaces, <sup>11</sup> making most of them absorb more C<sub>2</sub>H<sub>4</sub> over C<sub>2</sub>H<sub>6</sub>. To date, only a handful of C2H6-selective MOFs have been fulfilled by incorporating the well-designed C2H6-affinity sites or inert pore surfaces. 12

Realization of new class of adsorbents that are easy to build nonpolar/inert surfaces is of great importance to target the looked-for C<sub>2</sub>H<sub>6</sub>-selective adsorbents. In this context, hydrogen-bonded organic frameworks (HOFs), as a new type of extended porous materials, come into our sight because they can be self-assembled from organic molecules via intermolecular hydrogen-bonding interactions. 18 The metal-free nature without highly polar groups endows the pores of HOFs natively enriched with the nonpolar/inert surfaces that might lead to some new C<sub>2</sub>H<sub>6</sub>-selective adsorbents. Unlike the wellestablished MOFs and covalent organic frameworks (COFs) that are connected by strong coordinative or covalent bonds, HOFs are constructed by weak hydrogen-bonding interactions that make the framework very difficult to stabilize, and most of them collapse upon removal of solvent molecules by thermal and/or vacuum activation. This drawback led to the first establishment of permanent porosity in HOFs until 2010. 18,19 Evidently, the research of the emerging porous HOFs is still in its comparatively early stage when compared with MOFs and COFs. However, the soft nature of hydrogen bonding endows HOFs with some unique advantages, such as high solution processability and characterization, easy purification, good thermal stability, and facile regeneration and reusage by simple recrystallization. Despite these attractive merits, it is still very challenging to obtain robust HOFs with precise crystal structure, large permanent porosity, and both chemical and thermal stability, which largely hampers its development for gas separation. Until now, no efficient  $C_2H_6$ -selective HOF adsorbents have been fully realized for the important and challenging  $C_2H_6/C_2H_4$  separation.

With the above consideration in mind, we reasoned that constructing the nonpolar/inert pore surfaces within robust HOF materials may lead to the desired C<sub>2</sub>H<sub>6</sub>-selective adsorbents but remains a challenge. We herein designed a C<sub>6</sub>-symmetry organic ligand hexakis(4-carboxyphenylethynyl)benzene (HCEB) with large  $\pi$ -conjugated system and six carboxylate acid groups (Figure 1) and used it as a building unit to construct a novel robust HOF adsorbent (HOF-76). Single-crystal X-ray structure determination revealed that the synergistic effect of high density of strong O-H···O hydrogen bonds and the consecutive strong  $\pi \cdots \pi$  interactions in HOF-76 guarantees the establishment of permanent porosity with a Brunauer-Emmett-Teller (BET) surface area of 1121 m<sup>2</sup> g<sup>-1</sup> and afford its highly thermal and chemical stabilities. As expected, we discovered, for the first time, that the nonpolar/ inert pore surfaces within the activated HOF-76a indeed lead to the preferential adsorption of C2H6 over C2H4, and the separation performance is better than most of MOF materials reported. Simulated and actual breakthrough experiments confirmed that HOF-76a can preferentially capture C<sub>2</sub>H<sub>6</sub> from C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures to directly produce high-purity  $C_2H_4$  gas with a high productivity of 7.2 L/kg at 298 K and 1.01 bar and 18.8 L/kg at 5 bar and 298 K, a value comparable to the record 19.3 L/kg in Fe<sub>2</sub>(O<sub>2</sub>)(dobdc), <sup>13</sup> opening a new

class of porous adsorbents for the challenging  $C_2H_6/C_2H_4$  separation.

#### EXPERIMENTAL SECTION

**Materials and Physical Measurements.** All starting chemicals and solvents were purchased from commercial companies and used without further purification.  $^1\mathrm{H}$  and  $^{13}\mathrm{C}$  NMR spectra were recorded on Bruker AVANCE III spectrometers (400 MHz; Figure S1–S4). Thermogravimetric analysis (TGA) was performed on a Netzsch STA 449C thermal analyzer from 30 to 800 °C under nitrogen atmosphere at a heating rate of 5 °C/min. Powder X-ray diffraction (PXRD) patterns were measured by a BRUKER D8 ADVANCE diffractometer employing Cu K radiation operated at 30 kV and 15 mA, scanning over the range 2–45° (2 $\theta$ ) at a rate of 2°/min. Compound 1 and HCEB were prepared according to the literature with slight modification.  $^{23}$ 

Synthesis of Hexakis(4-ethoxycarbonylphenylethynyl)-benzene (1). Under an argon atmosphere, hexabromobenzene (551 mg, 1.0 mmol), ethyl 4-ethynylbenzoate (1.39 g, 8.0 mmol), Pd(PPh<sub>3</sub>)<sub>2</sub>Cl<sub>2</sub> (126 mg, 0.18 mmol), CuI (69 mg, 0.36 mmol), PPh<sub>3</sub> (94 mg, 0.36 mmol), and degassed Et<sub>3</sub>N (50 mL) were combined in a 100 mL two neck round-bottom flask. This mixture was stirred at refluxing temperature for 48 h. After removal of organic solvent, the crude product was purified by column chromatography on silica gel with CH<sub>2</sub>Cl<sub>2</sub>/ethyl acetate (EA; 1:1 v/v) to give the pure product 1 as a yellow solid. Yield: 832 mg, 75%.  $^{1}$ H NMR (CDCl<sub>3</sub>, 400 MHz, ppm):  $\delta$  7.92 (d, J = 7.08 Hz, 12H), 7.51 (d, J = 7.04 Hz, 12H), 4.40 (q, J = 7.12 Hz, 12H), 1.44 (t, J = 7.16 Hz, 18H).  $^{13}$ C NMR (CDCl<sub>3</sub>, ppm):  $\delta$  165.8, 131.7, 130.8, 129.7, 127.8, 127.2, 99.3, 89.5, 61.5, 14.5.

Synthesis of Hexakis(4-carboxyphenylethynyl)benzene ligand (HCEB). A solution of compound 1 (800 mg, 0.72 mmol) in THF (60 mL) was added to a 60 mL water solution of KOH (3.36 g, 60 mmol). The suspension mixture was stirred at 70 °C for 48 h. After removal of organic solvent, the aqueous residue was acidified with 2 M HCl. The resulting precipitate was filtered, washed with water, and dried under vacuum to afford HCEB as a dark yellow solid. Yield: 625 mg, 92%.  $^{1}$ H NMR (DMSO- $^{4}$ G, 400 MHz, ppm):  $\delta$  13.01 (s, 6H), 7.57 (d, J = 7.40 Hz, 12H), 7.20 (d, J = 7.40 Hz, 12H).  $^{13}$ C NMR (DMSO- $^{4}$ G, ppm):  $\delta$  166.3, 131.2, 130.9, 129.1, 126.9, 125.9, 98.8, 89.0.

**Synthesis of HOF-76.** The 50 mg of HCEB ligand was dissolved in a solution of DMSO (15 mL) in a 50 mL beaker, which was put in a 250 mL sealed beaker with 100 mL of acetone. The yellow needle crystals of HOF-76 suitable for single crystal X-ray diffraction analysis were grown by vapor diffusion for several weeks at room temperature. Yield: 45 mg, 90%.

**Single-Crystal X-ray Crystallography.** Single-crystal X-ray diffraction data of HOF-76 was collected on a Bruker D8 VENTURE diffractometer at 100 K using graphite-monochromated Mo Ka ( $\lambda$  = 0.71073 Å) radiation. The structure was solved by direct method and refined on  $F^2$  by full-matrix least-squares methods using SHELXL-97 software package. <sup>24</sup> The solvate molecules of all data were treated as diffuse contribution to the overall scattering without specific atom positions by SQUEEZE/PLATON due to severe disorder of these solvate molecules in the lattices. <sup>25</sup> The crystal data are summarized in Table S3.

**Density Functional Theory Calculations.** Our first-principles density-functional theory (DFT) calculations were performed using the Quantum-Espresso package. A semiempirical addition of dispersive forces to conventional DFT<sup>27</sup> was included in the calculation to account for van der Waals interactions. We used Vanderbilt-type ultrasoft pseudopotentials and the generalized gradient approximation (GGA) with the Perdew–Burke–Ernzerhof (PBE) exchange correlation. We first fully optimized the bare HOF-76a structure, using the primitive unit cell. Then, C<sub>2</sub>H<sub>6</sub> or C<sub>2</sub>H<sub>4</sub> molecules were introduced into the triangular channel-like pores of HOF-76a, and structural relaxations were performed. Various gas adsorption positions and molecular orientations were examined to find the lowest energy configuration. To obtain the gas binding

energies, a single gas molecule placed in a supercell with the same cell dimensions was also relaxed as a reference. The static binding energy (at  $T=0~\rm K$ ) was calculated using  $E_{\rm B}=E_{\rm (HOF-76a)}+E_{\rm (gas)}-E_{\rm (HOF-76a)}$ 

Gas Sorption Measurements. A Micromeritics ASAP 2020 surface area analyzer was used to measure gas adsorption isotherms. To remove all the guest solvents in the framework, the fresh crystal samples were first solvent-exchanged with dry acetone at least 10 times within 2 days, and evacuated at 273 K for 24 h and then 296 K for about 8 h until the outgas rate was 5 mmHg min<sup>-1</sup> to yield the activated samples. The sorption measurement was maintained at 77 K under liquid nitrogen bath. Bath temperatures of 273 and 296 K were precisely controlled with a recirculating control system.

**Column Breakthrough Experiments.** The breakthrough experiments for  $C_2H_6/C_2H_4$  (50/50, v/v),  $C_2H_6/C_2H_4$  (10/90), and  $C_2H_6/C_2H_4/C_4$  (10/92), and  $C_2H_6/C_2H_4/C_4$  (10/87/1/1/1) mixtures were carried out at a flow rate of 1.25 mL/min (298 K, 1.01 or 5.0 bar). Activated HOF-76a powder samples (1.1 g) were packed into  $\Phi$  4 × 150 mm stainless steel column under pure  $N_2$  atmosphere. The sample in each column was compressed as much as possible to obtain the best separation performance, and column voidages are similar for different samples. The horizontal reactor was placed in a temperature controlled environment, maintained at 298 K. The flow rates of all gases mixtures were regulated by mass flow controllers, and the effluent gas stream from the column is monitored by gas chromatography. Prior to breakthrough experiment, the sample was activated by flushing the adsorption bed with helium (He) gas for 2 h at 323 K. The adsorption bed can be regenerated by He flow (100 mL/min) for 1 h at 298 K.

#### ■ RESULTS AND DISCUSSION

The ligand HCEB was readily synthesized on a gram scale using Sonogashira coupling of hexabromobenzene with ethyl 4-ethynylbenzoate, followed by hydrolysis and acidification (Scheme 1). Slow evaporation of an acetone solution into a

Scheme 1. Synthetic Routes to the Organic Building Block of HCEB

DMSO solution of HCEB produced HOF-76 as yellow needle crystals suitable for X-ray analysis. The phase purity of the bulk material was confirmed by matching the experimental and simulated powder X-ray diffraction patterns (PXRD; Figure S5).

Single crystal X-ray diffraction (XRD) analysis revealed that HOF-76 crystallizes in a monoclinic C2/c space group with a hydrogen-bonded three-dimensional (3D) network (Table S3). As shown in Figure 1a, each HCEB molecule in HOF-76 is connected with six neighboring molecules through six pairs of O-H···O hydrogen bonds (H-bonds). The O-H···O distance and angle are 2.622 Å and 171.2°, respectively, which are typical for strong H-bonds with high directionality. Since HCEB molecule has six carboxylate acid units, the density of H-bonds in HOF-76 can reach 6.276 mmol/cc, notably higher than those of robust carboxylic acid-type HOFs reported (Table S4). It is worthy of note that HCEB molecule is not flat but has a twisted conformation, in which two outer carboxyphenyl groups are on the same plane with the central benzene ring, while the other four groups alternately direct up

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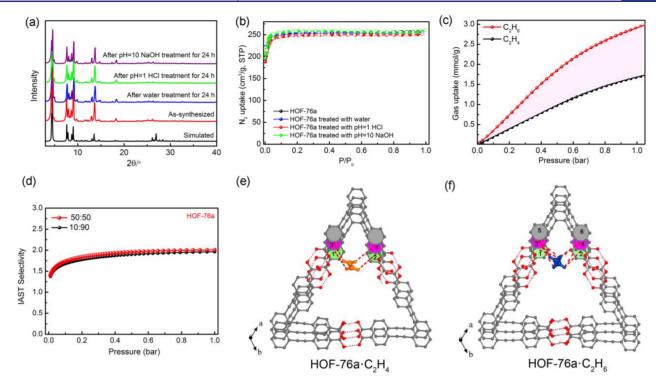


Figure 2. (a) PXRD patterns and (b)  $N_2$  adsorption isotherms (77 K) of HOF-76a and the samples treated with water, HCl (pH 1) and NaOH (pH 10) solution, respectively. (c) Adsorption isotherms of  $C_2H_6$  (red) and  $C_2H_4$  (black) for HOF-76a at 296 K. (d) IAST selectivity of HOF-76a from  $C_2H_6/C_2H_4$  (50/50 and 10/90) gas mixtures. (e and f) Comparison of the preferential  $C_2H_4$  and  $C_2H_6$  adsorption sites and the close vdW contacts within the corner surface of triangular channel-like pores observed by DFT calculations (C, dark gray; O, red; H, white), highlighting the  $C-H\cdots\pi$  interactions in red dashed bonds.

and down with an angle of 20.3-22.6° against the central benzene ring (Figure S6). These structural features enable HCEB to construct a three-dimensionally extended H-bonded network (Figures 1c and S7). The center and outer phenyl rings along with six alkynyls in HCEB form a very large  $\pi$ conjugated system. The distance between two adjacent  $\pi$ conjugated systems is about 3.352 Å (Figure 1b), indicating a very strong face-to-face  $\pi \cdots \pi$  stacking interaction. Topologically, if the HCEB is considered as a six-connected node, HOF-76 possesses the pcu {4<sup>12</sup>6<sup>3</sup>} topology (Figure 1d,e). As shown in Figure 1f,g, HOF-76 exhibits a 1D triangular channellike pore with a moderate size of 7.0 Å in diameter along the c axis. Due to the absence of metal sites/clusters, the channel surfaces are natively enriched with inert aromatic rings that may hold the potential to result in the preferential binding of  $C_2H_6$  over  $C_2H_4$ .

Before evaluating gas sorption properties of HOF-76, we first investigated its thermal and chemical stabilities by PXRD patterns and thermogravimetric analysis (TGA). The foregoing structural studies clearly showed that HOF-76 has extremely high density of H-bonds, short O-H···O distance, and very strong  $\pi \cdots \pi$  interactions (Table S4), which may contribute to high stabilities. As depicted in Figure 2a, HOF-76 indeed exhibits highly thermal and chemical stabilities. The framework can retain its structural integrity without phase change observed after soaking the sample in water, HCl (pH 1) and NaOH (pH 10) solutions for 24 h. There is also no loss of crystallinity when the sample was exposed to air for over 2 months (Figure S8). Along with the stability to water, the thermal stability of HOF-76 is also worthy of being examined. TGA studies revealed that HOF-76a is thermally stable up to 350 °C in an N<sub>2</sub> atmosphere (Figure S9). No obvious phase

changes in the PXRD patterns could be observed even at 300 °C in an air atmosphere, as evidence by various temperature PXRD patterns (Figure S10).

The permanent porosity of the activated HOF-76a was confirmed by nitrogen  $(N_2)$  gas sorption experiments at 77 K. As shown in Figure 2b, HOF-76a takes up 258 cm  $^3$  g  $^{-1}$  N  $_2$  at 77 K and 1 bar, and the N  $_2$  isotherm shows a significant type I sorption behavior without any hysteresis, characteristic of microporous materials. The BET surface area and pore volume of HOF-76a were calculated to be 1121 m  $^2$  g  $^{-1}$  and 0.40 cm  $^3$  g  $^{-1}$  (Figure S12). The pore size distribution determined by N  $_2$  isotherms is shown in Figure S13, and the calculated pore size (8.5 Å) is close to the value (7.0 Å) obtained from the crystal structure. After heating at 300 °C for 2 h or immersing in water, HCl and NaOH solutions for 24 h, the reactivated HOF-76a samples show no obvious decrease on the N  $_2$  uptakes at 77 K compared with the pristine sample, further confirming its excellent chemical and thermal stabilities.

Establishment of the permanent porosity in robust HOF-76a prompted us to examine the single-component adsorption isotherms of  $C_2H_6$  and  $C_2H_4$  at 273 and 296 K up to 1 bar, respectively. As depicted in Figures 2c and S14, HOF-76a shows an obviously preferential adsorption of  $C_2H_6$  over  $C_2H_4$  at both temperatures. The  $C_2H_6$  uptake amount of HOF-76a (2.95 mmol g<sup>-1</sup>) is much higher than that of  $C_2H_4$  (1.67 mmol g<sup>-1</sup>) at 1 bar and 296 K, affording a large  $C_2H_6/C_2H_4$  uptake ratio of 177%. Thus, we successfully realized the "reversed  $C_2H_6/C_2H_4$  adsorption" in HOF-76a. The adsorption heat ( $Q_{st}$ ) of  $C_2H_6$  for HOF-76a was calculated to be ~22.8 kJ/mol at zero coverage, slightly higher than that of  $C_2H_4$  (Figure S16). Due to the absence of metal sites/clusters and polar binding sites, this  $Q_{st}$  value of  $C_2H_6$  is significantly lower than

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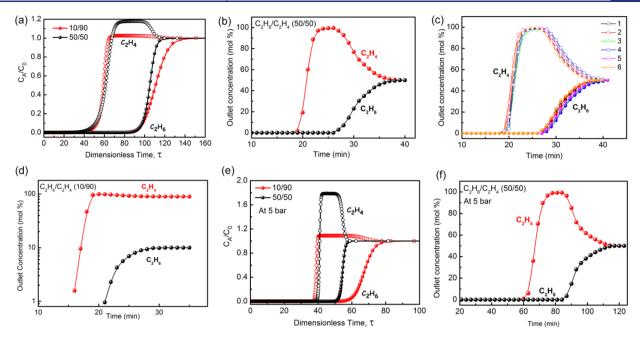


Figure 3. (a) Simulated breakthrough curves for  $C_2H_6/C_2H_4$  (50/50 and 10/90) separation ( $C_A/C_0$ , outlet concentration/feed concentration). (b) Experimental column breakthrough curves for a  $C_2H_6/C_2H_4$  (50/50) mixture in an absorber bed packed with HOF-76a at 298 K and 1.01 bar. (c) The recyclability of HOF-76a under multiple mixed-gas column breakthrough tests. (d) Experimental breakthrough curves for a  $C_2H_6/C_2H_4$  (10/90) mixture. (e) Simulated breakthrough curves for  $C_2H_6/C_2H_4$  mixtures at 5 bar. (f) Experimental breakthrough curves for a  $C_2H_6/C_2H_4$  (50/50) mixture at 298 K and 5 bar.

that in most of the  $C_2H_6$ -selective MOFs, such as MAF-49 (61 kJ/mol)<sup>12</sup> and  $Fe_2(O_2)(dobdc)$  (66.8 kJ/mol).<sup>13</sup>

Ideal adsorbed solution theory (IAST) was used to calculate the adsorption selectivity of HOF-76a for 50/50 and 10/90  $C_2H_6/C_2H_4$  mixtures at room temperature (RT), respectively. As shown in Figure 2d, HOF-76a exhibits a high C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> selectivity of up to 2.0 for both gas mixtures, which is larger than most of the developed MOF materials and HOF-BTB (1.4),  $^{20d}$  and only lower than  $Fe_2(O_2)(dobdc)$ ,  $^{13}$   $Cu(Qc)_2$ ,  $^{15}$  MAF-49,  $^{12}$  and UiO-66–2CF $_3$  (Figure S18 and Table S5). Further, the  $C_2H_6/C_2H_4$  uptake ratio (177%) of HOF-76a at 1 bar outperforms almost all of the reported C<sub>2</sub>H<sub>6</sub>-selective materials except Cu(Qc)<sub>2</sub> (Figure S18), further confirming its exceptional selectivity. The C<sub>2</sub>H<sub>6</sub> uptake of HOF-76a (2.95 mmol/g) at 1 bar and RT is also notably higher than those of benchmark UiO-66–2CF<sub>3</sub> (0.86 mmol/g), <sup>17a</sup> Cu(Qc)<sub>2</sub> (1.85 mmol/g), <sup>15</sup> and MAF-49 (1.73 mmol/g). <sup>12</sup> Overall, these results indicate that HOF-76a is placed among the best performing C<sub>2</sub>H<sub>6</sub>-selective materials reported. It is needed to point out that HOFs are a new kind of porous materials, and the research of porous HOFs for gas separation is still in its early stage. The development of robust HOF materials for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation is much more challenging than the well-established MOFs at this stage. However, HOFs hold some unique advantages that are different from MOFs, such as the ease of building nonpolar pores, solution processability and characterization, easy purification, and straightforward regeneration and reusage by simple recrystallization. These advantages make the pursuit of porous HOFs as separating adsorbents very attractive and become mutually complementary to the existing MOFs for the potential practical applications. In addition, HOF-76 also possesses ultrahigh chemical/thermal stabilities that are commonly absent in most C<sub>2</sub>H<sub>6</sub>-selective MOFs.

To gain better insight into the role of the nonpolar pore surface on the selective C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> adsorption in HOF-76a, we performed calculations using first-principles dispersioncorrected density functional theory (DFT-D) method. We found that, for both C<sub>2</sub>H<sub>4</sub> and C<sub>2</sub>H<sub>6</sub> molecules, the primary adsorption sites are located at the corners of the triangular channel-like pores. The lowest-energy gas binding configurations are shown in Figure 2e,f. For clarity, we only showed one adsorbed gas molecule at the corner site. Within each unit cell, there exist 24 such "corner sites", which are crystallographically equivalent (Figure S19). The corresponding calculated static binding energies of C<sub>2</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>4</sub> are 33.6 and 29.3 kJ mol<sup>-1</sup>, respectively. Along the channel axis, the binding energy variation is small (within ~5 kJ mol<sup>-1</sup>). As expected, the interaction between the gas molecule and the nonpolar pore surface is of van der Waals type, and the gas binding strength is modest. Nevertheless, the binding of C<sub>2</sub>H<sub>6</sub> is still notably stronger than C<sub>2</sub>H<sub>4</sub>. This is partly because the nonplanar C<sub>2</sub>H<sub>6</sub> molecule sterically "matches" better to the corner surface of the triangular channel-like pore than the planar  $C_2H_4$  molecule. As a result, multiple  $C-H\cdots\pi$ interactions exist between the C2H6 molecule and phenyl rings (Figures 2e and S20), in which all six hydrogens of C<sub>2</sub>H<sub>6</sub> can interact with six adjacent phenyl rings 1–6 and the H $\cdots\pi$ distances are calculated to be 4.116-4.857 Å. In contrast, the C<sub>2</sub>H<sub>4</sub> molecule shows contacts only with four phenyl rings 1–4 with the longer  $H \cdot \cdot \cdot \pi$  distances of 4.372–5.046 Å. Thus, the lower binding energy of C<sub>2</sub>H<sub>4</sub> can be attributed to the lack of strong permanent dipoles on the nonpolar pore surface of HOF framework and the less number of  $C-H\cdots\pi$  interactions. Accordingly, for ethane, the more  $C-H\cdots\pi$  interactions and its higher polarizability indicate the higher binding affinity. This is fully consistent with our experimental observations ( $Q_{st}$  etc.). In addition, we note that a full occupancy of these corner sites would correspond to 3.2 mmol/g gas uptake, which is close to

the experimental  $C_2H_6$  uptake (2.95 mmol/g) at RT and 1 bar, indicating that the corner sites are heavily populated. In contrast, for  $C_2H_4$ , the uptake at RT and 1 bar is much lower (1.67 mmol/g), suggesting that only about half of the corner sites are populated due to its weaker binding affinity and worse sterical "match" with the pores. All of these results can explain the adsorption/separation mechanisms at 1 bar qualitatively.

Transient breakthrough simulations were first performed for HOF-76a in fixed-bed adsorption processes at 1 bar and 298 K to determine the feasibility of C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation. As shown in Figure 3a, efficient separations can be accomplished by HOF-76a for both of  $C_2H_6/C_2H_4$  gas mixtures (50/50 and 10/90), wherein C<sub>2</sub>H<sub>4</sub> breakthrough occurred first and C<sub>2</sub>H<sub>6</sub> passed through the fixed bed after a certain time ( $\tau_{\text{break}}$ ). Next, the experimental breakthrough studies were conducted in a packed column of HOF-76a under actual  $C_2H_6/C_2H_4$  (50/50) mixtures at 298 K. The breakthrough data depicted in Figure 3b clearly show that HOF-76a can effectively separate C<sub>2</sub>H<sub>6</sub>/ C<sub>2</sub>H<sub>4</sub> mixtures. C<sub>2</sub>H<sub>4</sub> gas eluted through the adsorption bed first to yield an outflow of pure gas with an undetectable amount of C<sub>2</sub>H<sub>6</sub> (the detection limit of the instrument is 0.01%), affording a desirable purity of over 99.9%. In contrast, C<sub>2</sub>H<sub>6</sub> broke through the bed after 27 min because C<sub>2</sub>H<sub>6</sub> is more efficiently adsorbed. These experiments are consistent well with simulated breakthrough results. During the breakthrough process of HOF-76a, the pure C<sub>2</sub>H<sub>4</sub> production from the outlet effluent for a given cycle was calculated to be 7.2 L/ kg (Figure 3b), outperforming most of the top-performing MOFs such as MAF-49 (6.2 L/kg), <sup>12</sup> Cu(Qc)<sub>2</sub> (4.4 L/kg), <sup>15</sup> and PCN-250 (3.36 L/kg). <sup>16a</sup> Subsequently, we performed multiple mixed-gas (C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> at 50/50) column breakthrough tests to examine the preservation of separation performance of HOF-76a at ambient conditions. The breakthrough times for both C<sub>2</sub>H<sub>6</sub> and C<sub>2</sub>H<sub>4</sub> remains almost unchanged within six continuous cycles, confirming its good recyclability for C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation (Figures 3c, S21, and 22).

The feed gases in the practical production of high-purity  $C_2H_4$  sometimes have relatively low  $C_2H_6$  concentrations (6–10%) contaminated by low levels of impurities such as  $CH_4$ ,  $H_2$ , and  $C_2H_2$ . We thus conducted a series of breakthrough experiments on HOF-76a for  $C_2H_6/C_2H_4$  (10/90) and  $C_2H_6/C_2H_4/CH_4/H_2/C_2H_2$  (10/87/1/1/1) mixtures. As shown in Figures 3d and S23, highly efficient separation was also realized for both gas mixtures, indicating that HOF-76a can be used to purify  $C_2H_4$  from the mixtures with low  $C_2H_6$  concentrations, even in the presence of  $CH_4$ ,  $H_2$ , and  $C_2H_2$  impurities. Additionally, breakthrough experiments under moisture (from 446 to 2130 ppm) revealed that HOF-76a can maintain its separation performance under more extreme moisture conditions than that would be found in a practical process (Figure S24).

Given that an industrial application is potentially under high pressure (typically 5–10 bar),  $^{17a}$  as required by pressure-swing adsorption (PSA) processes, we thus investigated the separation performance of HOF-76a at 5 bar. The  $C_2H_6$  and  $C_2H_4$  isotherms were measured at 298 K up to 5 bar (Figure S25), wherein the  $C_2H_6$  uptake can increase to 6.3 mmol g $^{-1}$ . Accordingly to IAST calculations for 50/50 and 10/90 mixtures at 5 bar (Figure S26), we found that HOF-76a can maintain its selectivity to around 2. As shown in Figure 3e,f, both the simulated and actual breakthrough curves clearly demonstrated that HOF-76a is capable of removing  $C_2H_6$  from

50/50 and 10/90  $C_2H_6/C_2H_4$  mixtures at 5 bar and 298 K (Figure S28), and the simulations match well with the experiments (Figure S29). At this high-pressure, the production of high-purity  $C_2H_4$  from the outlet effluent for 50/50 mixtures can be enhanced to 18.8 L/kg, a value comparable to the record  $Fe_2(O_2)(dobdc)^{13}$  (19.3 L/kg at ambient conditions). To the best of our knowledge, this is the first example of porous materials whose reversed  $C_2H_6/C_2H_4$  separation at high pressure has been well established by both simulated and experimental breakthrough, enabling HOF-76a to be a potential material for industrial  $C_2H_6/C_2H_4$  separation application.

#### CONCLUSION

In summary, we have realized the best example of an ultrarobust HOF adsorbent (HOF-76a) that exhibits the preferential adsorption of ethane over ethylene, affording the unusual reversed  $C_2H_6/C_2H_4$  adsorption. The foregoing results showed that this material not only possesses extraordinary thermal stability and water stability, but also exhibits exceptional C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> separation performance superior to most of the C<sub>2</sub>H<sub>6</sub>-selective MOFs developed. As revealed by DFT calculation results, the nature of nonpolar/inert surfaces in HOF-76a and the suitable triangular channel-like pores to "match" with the C<sub>2</sub>H<sub>6</sub> molecule play the important roles for the preferential interactions with C<sub>2</sub>H<sub>6</sub> over C<sub>2</sub>H<sub>4</sub>. Simulated and experimental breakthrough curves confirmed that HOF-76a can efficiently separate C<sub>2</sub>H<sub>6</sub> from various C<sub>2</sub>H<sub>6</sub>/C<sub>2</sub>H<sub>4</sub> mixtures to directly produce high-purity C2H4 gas at ambient pressure and high pressure conditions, respectively. These results revealed in this work may shed some light on the development of this new kind of HOF materials for the industrially important  $C_2H_6/C_2H_4$  separation in the future.

#### ASSOCIATED CONTENT

#### S Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/jacs.9b12428.

Full details for experimental procedures, single crystal X-ray crystallography, gas sorption and stability tests, breakthrough experiments, and DFT calculation data (PDF)

Crystallographic data for HOF-76 (CIF)

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#### **Notes**

The authors declare no competing financial interest.

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# Selective Ethane/Ethylene Separation in a Robust Microporous Hydrogen-Bonded Organic Framework

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## 1. General materials and procedures

All starting chemicals and solvents were purchased from commercial companies and used without further purification: hexabromobenzene (Alfa Aesar, 97%); ethyl 4-ethynylbenzoate was prepared from 4-Bromobenzoic acid (Alfa Aesar, 98+%); Pd(PPh<sub>3</sub>)<sub>2</sub>Cl<sub>2</sub> (Aladdin, 98+%); CuI (Acros, 98%); PPh<sub>3</sub> (Greagent, 99+%); trimethylsilylacetylene (Macklin, 98+%); D<sub>2</sub>O (Sigma-Alderich, 99.9 atom%); NaOD (Norell, 30 WT% in D<sub>2</sub>O, 99.9 atom%); DC1 (Sigma-Alderich, 35 WT% in D<sub>2</sub>O, 99+ atom%).  $^{1}$ H and  $^{13}$ C NMR spectra were recorded on Bruker AVANCE III spectrometers (400 MHz). Thermogravimetric analyses (TGA) were performed on a Netzsch STA 449C thermal analyzer from 30 to 800 °C under nitrogen atmosphere at a heating rate of 5 °C/minute rate. Powder X–ray diffraction (PXRD) patterns were measured by a BRUKER D8 ADVANCE diffractometer employing Cu-K $_{\alpha}$  radiation operated at 30 kV and 15 mA, scanning over the range 2-45° (20) at a rate of 2°/min.

 $C_2H_4$  (99.99%),  $C_2H_6$  (99.99%), He (99.999%) and mixed gases of  $C_2H_4/C_2H_6=10/90$  (v/v),  $C_2H_4/C_2H_6=50/50$  (v/v), and  $C_2H_6/C_2H_4/C_2H_2/CH_4/H_2$  (10/87/1/1/1) mixtures were purchased from Beijing Special Gas Co. LTD (China).

## 2. Fitting of pure component isotherms

The pure component isotherm data for  $C_2H_4$  and  $C_2H_6$  in HOF-76 were measured at 273 K and 296 K. The data were fitted with the 1-site Langmuir-Freundlich model

$$q = q_{sat} \frac{bp^{v}}{1 + bp^{v}}$$

The parameter b is temperature dependent

$$b = b_0 \exp(E/RT)$$

The fitted parameter values are provided in Table S1-S2 and Figure S27.

### 3. Isosteric heat of adsorption

The binding energy is reflected in the isosteric heat of adsorption,  $Q_{\rm st}$ , defined as

$$Q_{st} = RT^2 \left( \frac{\partial \ln p}{\partial T} \right)_q$$

The  $Q_{\rm st}$  values as function of the molar loadings are provided in Figure S16.

## 4. IAST calculations of adsorption selectivities

The selectivity of preferential adsorption of component 1  $(C_2H_6)$  over component 2  $(C_2H_4)$  can be defined as

$$S_{ads} = \frac{q_1/q_2}{p_1/p_2}$$

Where  $q_1$  and  $q_2$  are the *absolute* component loadings of the adsorbed phase in the mixture, and  $p_1$  and  $p_2$  are the component partial pressures. The component loadings and adsorption selectivity  $S_{\text{ads}}$  for 50/50  $C_2H_6(1)/C_2H_4(2)$  and 10/90  $C_2H_6(1)/C_2H_4(2)$  mixtures in HOF-76a at 296 K were determined using IAST.

### 5. Breakthrough simulations

The performance of industrial fixed bed adsorbers is dictated by a combination of adsorption selectivity and uptake capacity. Transient breakthrough simulations were carried out for 50/50  $C_2H_6(1)/C_2H_4(2)$  and 10/90  $C_2H_6(1)/C_2H_4(2)$  mixtures in HOF-76 operating at 298 K and a total pressure of 1 bar or 5 bar respectively, using the methodology described in earlier publications. For the breakthrough simulations, the following parameter values were used: length of packed bed, L=0.3 m; voidage of packed bed,  $\varepsilon=0.4$ ; superficial gas velocity at inlet, u=0.04 m/s. The transient breakthrough simulation results are presented in terms of a *dimensionless* time,  $\tau=\frac{tu}{L\varepsilon}$ , defined by dividing the actual time, t, by the characteristic time, t.

The breakthrough simulations demonstrate the potential of producing product gas  $C_2H_4$  of required purity during the interval  $\Delta \tau$ .

#### 6. Stability studies

To investigate the chemical stabilities of HOF-76, the as-synthesized samples were soaked in water, HCl (pH = 1) and NaOH (pH = 10) solutions for 24 hours, respectively. After that, each sample was filtered and washed with water and acetone quickly, and then characterized by PXRD measurements in order to determine whether the sample retains its structural integrity. Subsequently, each sample was solvent-exchanged with dry acetone at least 8 times within two days to completely remove the hardly volatile H<sub>2</sub>O molecules in the pores, and then was activated by the activation conditions prior to gas sorption measurements. The 77 K N<sub>2</sub> and 296 K C<sub>2</sub>H<sub>6</sub> sorption isotherms were examined to further determine the chemical stability of HOF-76. The thermal stability of HOF-76 was studied by thermogravimetric analysis and various temperature PXRD patterns. The sample for temperature PXRD patterns was treated with 100°, 150°, 200°, 300° for 2 hours, respectively, and then characterized by PXRD measurements. To further check the thermal stability, the sample treated with 300° was solvent-exchanged by dry acetone and then activated to examine the 77 K N<sub>2</sub> and 296 K C<sub>2</sub>H<sub>6</sub> sorption isotherms.

To further confirm its ultrahigh chemical stability in acid and base solutions, the samples were also soaked into pH = 10 NaOD solution (0.01 mol/L NaOD in  $D_2O$  solution) and pH = 1 HCl solution (0.1 mol/L DCl in  $D_2O$  solution), respectively. After 24 hours, the  $D_2O$  solution was directly used to measure  $^1H$  NMR spectra. As shown in Figure S11, there is no any signal of HCEB ligand observed, strongly demonstrating that the HOF-76 framework remains stable in these acid and base solutions and no ligands are dissociated into the  $D_2O$  solution.

Considering that the  $C_2H_4$  feed gas in industry is often contaminated by trace levels of  $H_2O$  (5-50 ppm), the effect of moisture on breakthrough experiments was carried out on HOF-76a for 50/50  $C_2H_6/C_2H_4$  mixtures containing trace  $H_2O$  (from 446 to 2130 ppm) that are more extreme than that found in a realistic process. Figure S24 showed that the presence of trace  $H_2O$  has a negligible effect on the breakthrough performance of HOF-76a.

## Notation

b Langmuir-Freundlich constant,  $Pa^{-\nu}$ 

q component molar loading of species i, mol kg<sup>-1</sup>

 $q_{\rm sat}$  saturation loading, mol kg<sup>-1</sup>

L length of packed bed adsorber, m

t time, s

T absolute temperature, K

u superficial gas velocity in packed bed, m s<sup>-1</sup>

## **Greek letters**

 $\varepsilon$  voidage of packed bed, dimensionless

v Freundlich exponent, dimensionless

 $\tau$  time, dimensionless

**Table S1.** 1-site Langmuir-Freundlich parameter fits for  $C_2H_4$  and  $C_2H_6$  in HOF-76 at 1 bar and room temperature.

	$q_{ m sat}$	$b_0$ $E$		V	
	mol kg <sup>-1</sup>	Pa <sup>-v</sup>	kJ mol <sup>-1</sup>	dimensionless	
$C_2H_4$	6.3	5.25E-13	35	1.14	
$C_2H_6$	5	2.53E-12	31	1.25	

**Table S2.** 1-site Langmuir-Freundlich parameter fits for  $C_2H_4$  and  $C_2H_6$  in HOF-76 at 5 bar and 298 K.

	$q_{ m sat}$	b	$R^2$	
	mol kg <sup>-1</sup>	Pa <sup>-1</sup>		
$C_2H_4$	11.9	1.73674E-06	0.9953	
$C_2H_6$	9	4.846E-06	0.9857	

Table S3. Crystallographic data and structure refinement results of HOF-76.

	HOF-76			
Formula	$C_{60}H_{30}O_{12}$			
Formula weight	942.84			
Temperature/K	100(2)			
Crystal system	Monoclinic			
Space group	<i>C</i> 2/c			
a, b (Å)	22.539(3), 40.639(5)			
c (Å)	13.7223(17)			
α (°)	90.00			
β (°)	96.088(5)			
γ (°)	90.00			
$V(\mathring{\text{A}}^3)$	12498(3)			
Z	8			
$D_{ m calcd}$ (g cm <sup>-3</sup> )	1.002			
$\mu  (\mathrm{mm}^{-1})$	0.070			
F(000)	3888.0			
Crystal size/mm <sup>3</sup>	$0.0032 \times 0.0028 \times 0.0028$			
GOF	1.005			
$R_{int}$	0.1562			
$R_1$ , $wR_2$ [I>=2 $\sigma$ (I)]	0.0772, 0.2123			
$R_1$ , $wR_2$ [all data]	0.1799, 0.2684			
CCDC number	1907797			

 $<sup>\</sup>frac{1}{16} R1 = \sum |F_0 - F_c| / \sum F_o, \quad wR2 = \sum [w(F_o^2 - F_c^2)^2] / \sum [w(F_o)^2]^{1/2}$ 

**Table S4.** Comparison of the density and features of H-bonds, the distance of  $\pi \cdots \pi$  interactions, and the surface area in HOF-76 and some selected carboxylic acid-type HOFs with established porosities.

carboxylic acid-type HOFs	Number of carboxylic acids	Density of H-bonds (mmol/cm <sup>3</sup> )	$d_{\text{O-H}\cdots \text{O}} \atop (\mathring{\text{A}})^a$	Angel of O-H···O bonds <sup>b</sup>	$d_{\pi \cdots \pi \text{ stacking}} $ $(\mathring{A})^c$	$S_{BET}$ $(m^2 g^{-1})^d$	Ref.
<b>HOF-76</b>	6	6.276	2.622	171.2	3.352	1121	This work
HOF-TCBP	4	4.355	2.620	176.3	3.425	2066	[3]
PFC-1	4	4.342	2.597	176.0	3.338	2122	[4]
Ex-1	6	3.120	2.551-2.697	170.4	3.446	_	[5]
T12-1	6	5.765	2.565-2.816	173.8	3.459	557	[5]
PETHOF-1a	6	0.994	2.580	161.8	_	1140	[6]
PETHOF-2a	6	2.553	2.600	160.6	3.30	1690	[6]
HOF-BTB	3	3.254	2.594	166.4	_	955	[7]
IISERP- HOF1	3	4.067	2.592	165.2	_	1025	[8]
TCF-1	4	4.995	2.823	152.6	3.600	_	[9]

<sup>&</sup>lt;sup>a</sup> The distance of O–H···O; <sup>b</sup> the O–H···O bond angel; <sup>c</sup> the distance of  $\pi$ ··· $\pi$  stacking interaction; <sup>d</sup> BET surface areas calculated from 77 K N<sub>2</sub> or 195 K CO<sub>2</sub> isotherms.

**Table S5.** Summary of separation metrics of top-performing ethane-selective materials reported in the literature at 1 bar and room temperature (RT).

C <sub>2</sub> H <sub>6</sub> -selective adsorbents	C <sub>2</sub> H <sub>6</sub> uptake <sup>a</sup> (mmol/g)	C <sub>2</sub> H <sub>4</sub> uptake <sup>a</sup> (mmol/g)	Uptake ratio <sup>b</sup> (%)	Selectivity <sup>c</sup>	$Q_{st, \text{ ethane}}$ $(kJ/mol)^d$	Productivity <sub>exp</sub> (L/kg) <sup>e</sup>	Ref.
HOF-76a	2.95	1.67	177	2.05	22.8	7.2/18.8 <sup>f</sup>	This work
Fe <sub>2</sub> O <sub>2</sub> (dobdc)	3.32	2.53	131	4.4	66.8	19.3	[10]
MUF-15	4.69 <sup>c</sup>	4.15 <sup>c</sup>	113 <sup>c</sup>	1.94 <sup>c</sup>	29.2	14	[11]
MAF-49	1.73	1.69	102	2.7	61	6.2	[12]
$Cu(Qc)_2$	1.85	0.78	235	3.4	29	4.4	[13]
UiO-66-2CF <sub>3</sub>	0.86	0.50	171	2.5	30	-	[14]
PCN-250	5.20	4.22	123	1.9	23.2	3.36	[15]
UiO-66	2.38	1.72	138	1.8	-	-	[14]
UiO-66-NDC	4.3	3.46	124	1.4	-	-	[14]
PCN-245	3.27	2.39	137	1.9	22.8	5.8	[16]
[Ni(bdc)(ted) <sub>0.5</sub>	5.0	3.4	147	1.6	21.5	-	[17]
MIL-53(Al)	2.16 <sup>g</sup>	$1.72^{g}$	126	1.6	22.5	-	[18]
MIL-142(A)	3.8	2.9	131	1.5	-	-	[14]
IR-MOF-8	4.3	3.0	143	1.8	52.5	2.5	[19]
$ZIF-8^h$	2.52	1.51	167	1.7	17.2	0.4	[20]
ZIF-7	1.83	1.80	102	1.5	-	2	[21]
$ZIF-4^h$	2.30	2.20	104	2.15	-	6.6	[22]
UTSA-35	2.43	2.16	112	1.4	-	-	[10]
$HOF$ - $BTB^i$	3.09	2.48	124	1.4	25.4	-	[23]

<sup>&</sup>lt;sup>a</sup> At 1 bar and room temperature;  ${}^bC_2H_6/C_2H_4$  uptake ratio at 1 bar and RT;  ${}^c$  IAST selectivity for 50/50  $C_2H_6/C_2H_4$  gas mixtures;  ${}^d$  The adsorption heat ( $Q_{st}$ ) of  $C_2H_6$  at low surface coverage.  ${}^e$  The pure  $C_2H_4$  productivity calculated from breakthrough experiments at 1 bar and RT on 50/50 gas mixtures.  ${}^f$  The pure  $C_2H_4$  productivity under 5.0 bar and RT.  ${}^g$  At 323 K.  ${}^h$  At 293K.  ${}^i$  The reported literature only described the  $C_2H_4$  and  $C_2H_6$  adsorption isotherms, so the selectivity and uptake ratio were evaluated based on these isotherms.

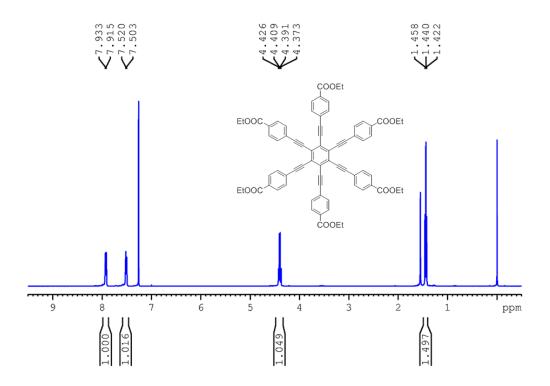


Figure S1. <sup>1</sup>H NMR (CDCl<sub>3</sub>, 400 MHz) spectrum of compound 1.

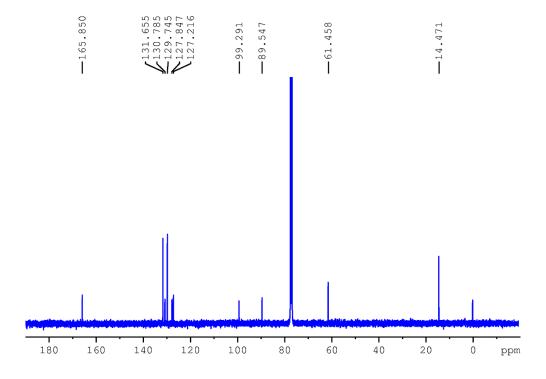


Figure S2. <sup>13</sup>C NMR (CDCl<sub>3</sub>, 100 MHz) spectrum of compound 1.

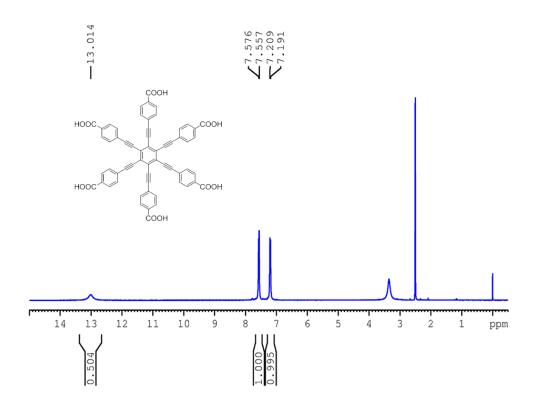


Figure S3. <sup>1</sup>H NMR (DMSO-d<sub>6</sub>, 400 MHz) spectrum of compound HCEB.

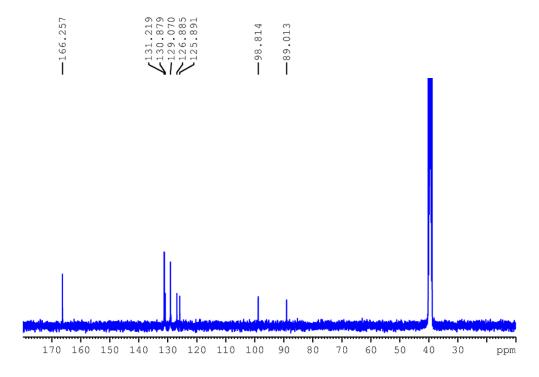
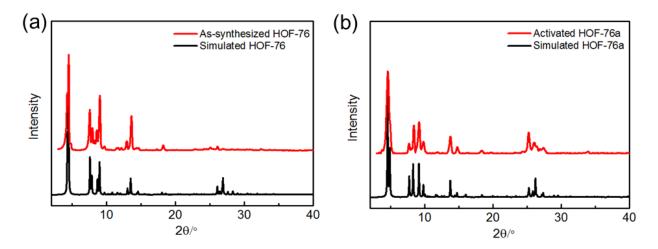
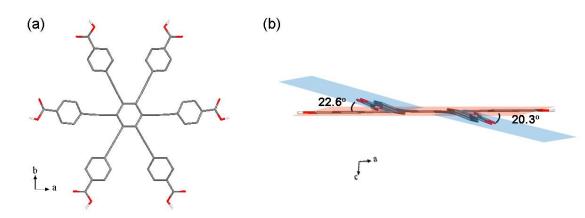


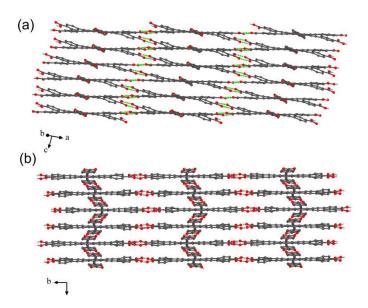
Figure S4. <sup>13</sup>C NMR (DMSO-d<sub>6</sub>, 100 MHz) spectrum of compound HCEB.



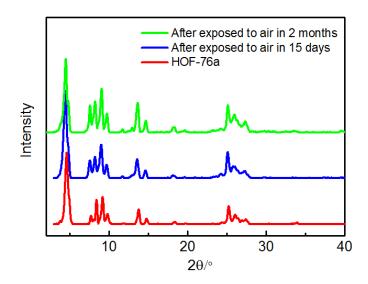
**Figure S5.** (a) Experimental PXRD pattern of as-synthesized HOF-76 (red) and the calculated pattern based on the single-crystal X-ray structure (black). (b) Experimental PXRD pattern of activated HOF-76a (red) and the calculated pattern based on the HOF-76a structural model (black). The agreement between experiment and simulation is very well. In the HOF-76a structural model, the lattice parameters were obtained from indexing the experimental PXRD pattern of activated HOF-76a. The results show that the HOF-76 crystal structure remains essentially the same upon activation (guest removal), in terms of crystal symmetry and framework arrangement. However, the unit cell parameters do change a bit: As-synthesized HOF-76 (from SXRD data): a=22.539; b=40.639; c=13.722; beta=96.088; Activated HOF-76 (from PXRD data): a=23.051; b=36.025; c=14.229; beta=98.095.



**Figure S6.** (a) View of the HCEB ligand along c-axis; (b) along b-axis, indicating that the four outer carboxyphenyl groups alternately direct up and down against the plane of the central benzene ring.



**Figure S7.** Crystal structure description of HOF-76, showing the hydrogen-bonding connections of the HCEB building blocks to form a three-dimensionally extended H-bonded network viewed two perpendicular directions to channels.



**Figure S8.** PXRD patterns of activated HOF-76a samples after exposed to air for more than 2 months.

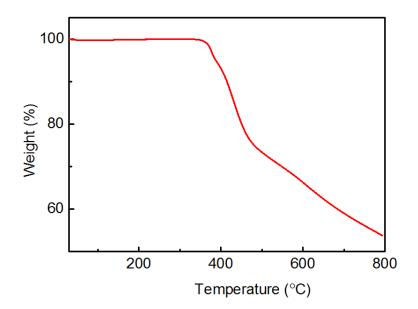
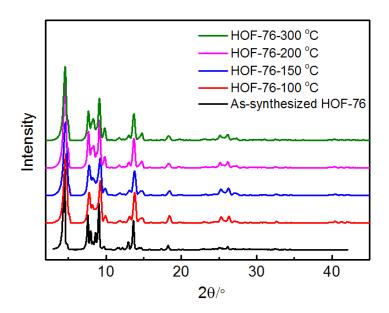
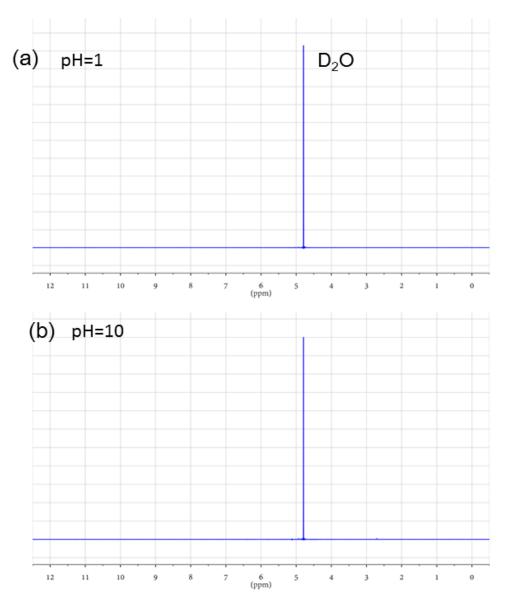


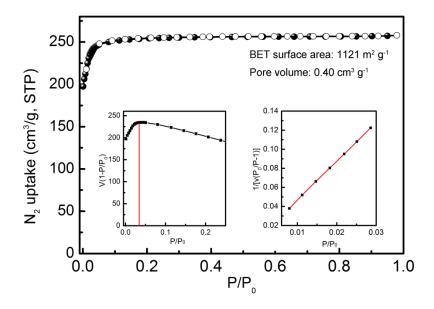
Figure S9. TGA curve of activated HOF-76a.



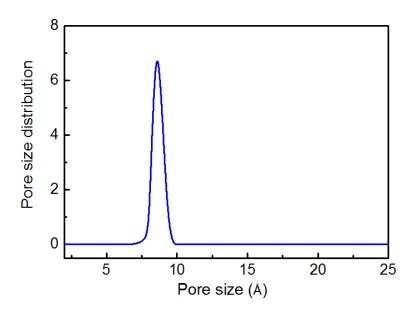
**Figure S10.** PXRD patterns of HOF-76 sample treated with different temperatures for 2 hours, and compared with the XRD patterns of the as-synthesized sample (black).



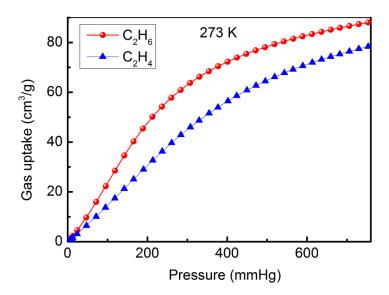
**Figure S11.** <sup>1</sup>H NMR ( $D_2O$ , 400 MHz) spectrum of the  $D_2O$  solution after the HOF-76 sample was soaked into (a) the pH = 1 HCl solution (0.1 mol/L DCl in  $D_2O$  solution) and (b) pH = 10 NaOD solution (0.01 mol/L NaOD in  $D_2O$  solution) for 24 h, and no any ligand signal was detected.



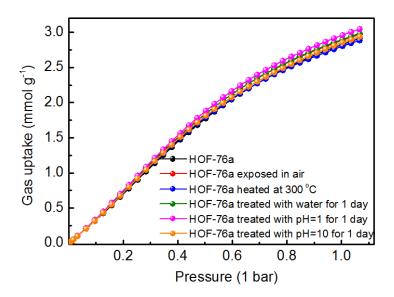
**Figure S12.** Nitrogen isotherm at 77 K with consistency and BET plots for the activated HOF-76a sample.



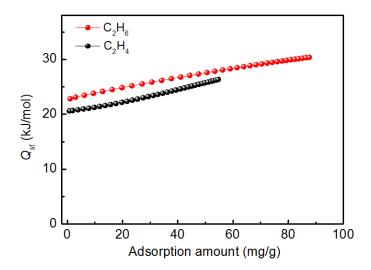
**Figure S13.** The pore size distribution of HOF-76a determined by 77 K  $N_2$  isotherms using Non Local Density Functional Theory (NLDFT) method, revealing a calculated pore size of 8.5 Å.



**Figure S14.** Adsorption isotherms of  $C_2H_6$  (red) and  $C_2H_4$  (blue) for HOF-76a at 273 K up to 1 bar.



**Figure S15.** Gas sorption isotherms of  $C_2H_6$  at 296 K for HOF-76a (black), re-activated HOF-76a after exposure to air for 15 days (red), heated at 300 °C for 2 h, and treated with water-, acid-and base-treated HOF-76a.



**Figure S16.** Adsorption heat of the  $C_2H_6$  (red) and  $C_2H_4$  (black) for HOF-76a.

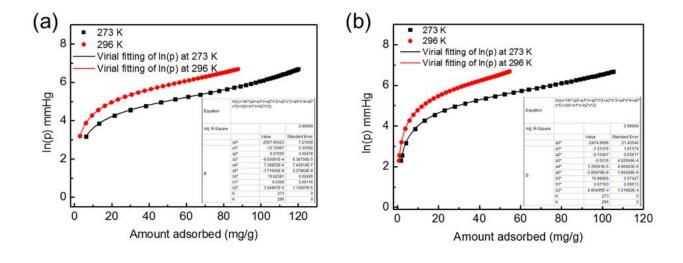
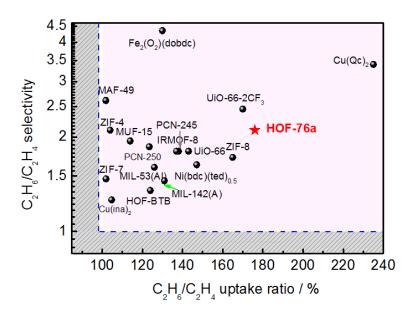
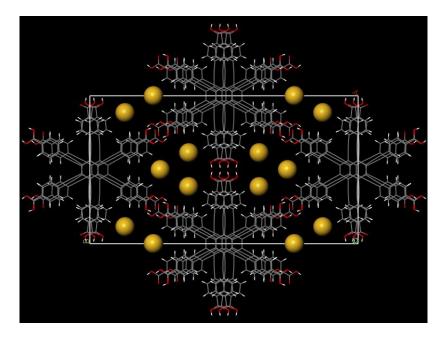


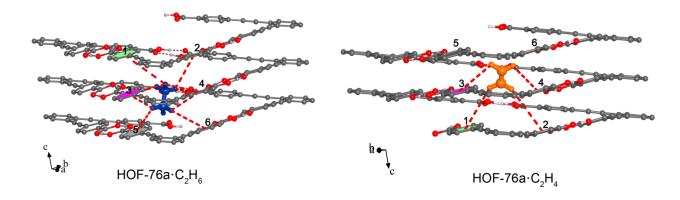
Figure S17. Virial fitting of the  $C_2H_6$  (a) and  $C_2H_4$  (b) adsorption isotherms for HOF-76a.



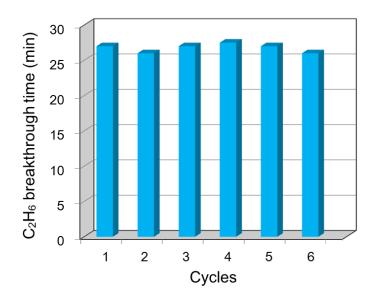
**Figure S18.**  $C_2H_6/C_2H_4$  IAST selectivity/uptake ratio for HOF-76a (2.05 and 177%) verses the indicated  $C_2H_6$ -selective MOF materials at 1 bar and room temperature, indicating that HOF-76a is superior to most of MOFs and the only HOF reported (HOF-BTB: 1.4 and 124%)<sup>23</sup>. HOF-76a is evidently placed among the best-performing materials reported for  $C_2H_6/C_2H_4$  separation.



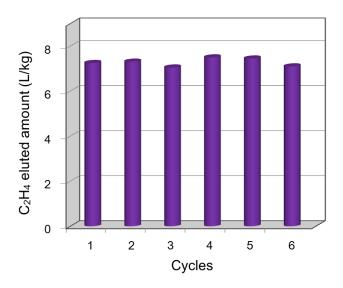
**Figure S19.** A local structure of HOF-76a showing all the corner sites (yellow ball) for gas adsorption, and there exist 24 such "corner sites" within each unit cell.



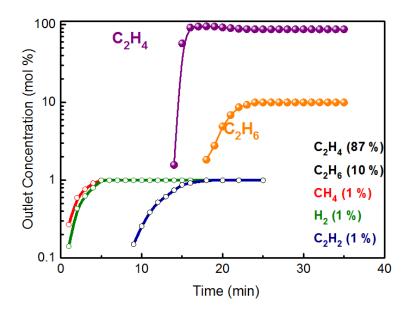
**Figure S20.** Comparison of the preferential  $C_2H_4$  and  $C_2H_6$  adsorption sites and the close vdW contacts within the corner surface of triangular channel-like pores observed by DFT calculations (C, dark gray; O, red; H, white), highlighting the  $C-H\cdots\pi$  interactions in red dashed bonds.



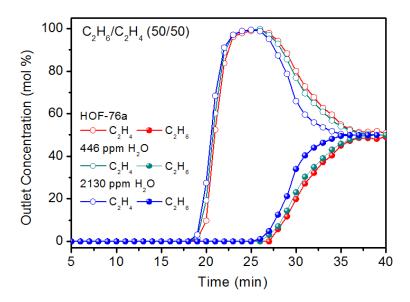
**Figure S21.** Cyclic breakthrough experiments for  $C_2H_6/C_2H_4$  (50/50) separation on HOF-76a, indicating that HOF-76a maintained the  $C_2H_6$  breakthrough time during the separation processes over at least 6 times.



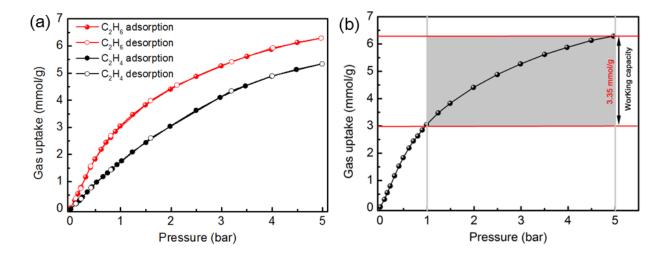
**Figure S22.** Cyclic breakthrough experiments for  $C_2H_6/C_2H_4$  (50/50) separation on HOF-76a, indicating that HOF-76a maintained the  $C_2H_4$  eluted amount during the separation processes over at least 6 times.



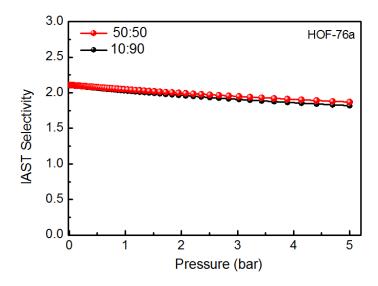
**Figure S23.** Experimental column breakthrough curves for a  $C_2H_6/C_2H_4/C_2H_2/CH_4/H_2$  (10/87/1/1/1) mixture under a total flow of 1.25 mL min<sup>-1</sup> in an absorber bed packed with HOF-76a at 298 K and 1.01 bar.



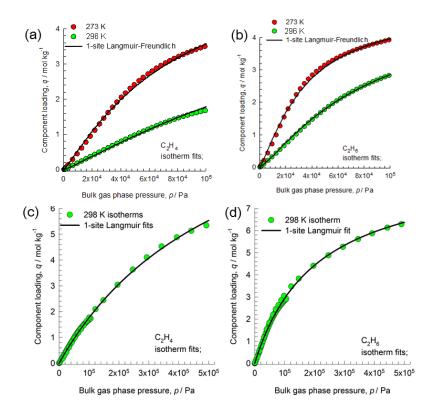
**Figure S24.** Experimental column breakthrough curves for  $C_2H_6/C_2H_4$  separations (50/50, v/v) on HOF-76a at 298 K and 1.01 bar with different amounts of  $H_2O$ , indicating that small amounts of water have a negligible effect on the separation performance.



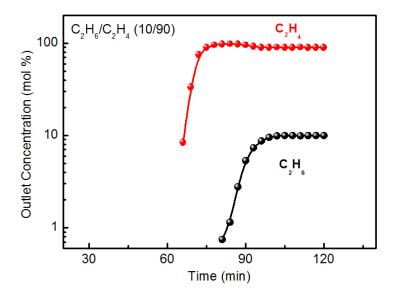
**Figure S25.** (a) Adsorption isotherms of  $C_2H_6$  (red) and  $C_2H_4$  (black) for HOF-76a at 298 K up to 5 bar. (b) Schematic diagram of the determination of the working capacity of HOF-76a at RT, as required by pressure-swing adsorption (PSA) processes. Here the working capacity is defined as the different amounts adsorbed between 1 and 5 bar, if a PSA regeneration step is considered at near-ambient pressure ( $\approx 100 \text{ kPa}$ ). <sup>14</sup>



**Figure S26.** IAST selectivity of HOF-76a from  $C_2H_6/C_2H_4$  (50/50 and 10/90) gas mixtures up to 5 bar, calculated on  $C_2H_6$  and  $C_2H_4$  high-pressure adsorption isotherms at 298 K.



**Figure S27.** (a) Comparison of component loadings for  $C_2H_4$  and (b)  $C_2H_6$  at 1 bar in HOF-76a with the 1-site Langmuir-Freundlich isotherm fits; (c) for  $C_2H_4$  and (d)  $C_2H_6$  at 5 bar and 298 K.



**Figure S28.** Experimental column breakthrough curves for a  $C_2H_6/C_2H_4$  (10/90) mixture in an absorber bed packed with HOF-76a at 298 K and 5.0 bar.

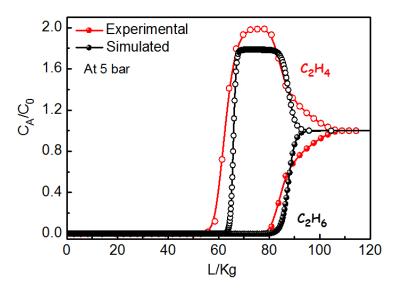
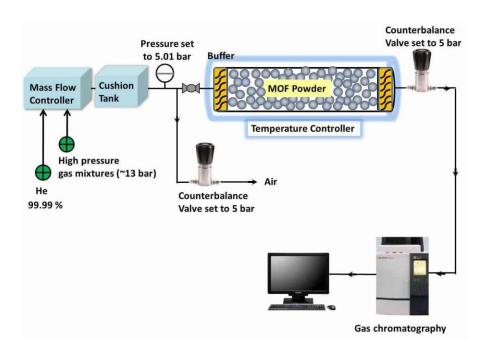


Figure S29. Comparison of the experimental and simulated breakthrough curves for  $C_2H_6/C_2H_4$  separation (50/50, v/v) on HOF-76a under the same separation parameters at 298 K and 5.0 bar, indicating that the simulations are in good agreement with the experimental results. The X-axis represents  $\frac{\text{(time in minutes)} \times \text{(flow rate mL/min STP)}}{\text{(g MOF packed in tube)}} = \text{mL/g}$ .



**Figure S30.** Schematic illustration of the apparatus for the high-pressure breakthrough experiments.

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