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## Modeling Issues in Zeolite Applications

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### I. INTRODUCTION

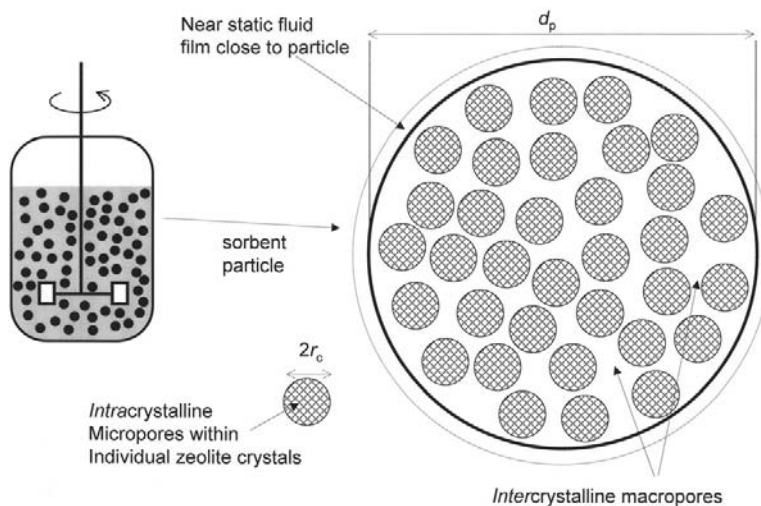
Zeolitic materials are used as sorbents and catalysts in a variety of processes within the chemical, petroleum, petrochemical, and food industries. Zeolite crystals are incorporated into binders (such as amorphous aluminosilicate) and perhaps a diluent (typically a clay mineral), and used in the form of powder (in fluidized beds) or pellets (in fixed beds). Alternatively, zeolite crystals are coated onto a porous membrane support and used in (catalytic) membrane permeation devices.

Zeolite-based processes are carried out either under steady-state, unsteady-state, or cyclic conditions. Fixed-bed adsorbers are typically operated under transient conditions. Zeolite membrane processes typically operate under steady-state conditions. Simulated moving-bed adsorbers operate under cyclic conditions. While many of the transport issues can be understood from the standpoint of classical diffusion and flow, special attention needs to be paid to the proper description of mixture sorption and diffusion in zeolites. The purpose of this chapter is to highlight the special features of zeolite sorption and diffusion by means of several illustrative examples of practical importance.

### II. TRANSIENT UPTAKE OF A SINGLE COMPONENT WITHIN A ZEOLITE

Let us begin by considering the case of a batch adsorber in which zeolite particles are brought into contact with a fluid phase containing a component species  $i$  that diffuses into the particle (of diameter  $d_p$ ) into which the zeolite crystals are embedded; see Fig. 1. There are three steps in the intraparticle diffusion process.

1. Component  $i$  in the bulk fluid phase surrounding the particle has first to diffuse across the stagnant layer, of thickness  $\delta_f$ , surrounding the particle. The stagnant “film” thickness  $\delta_f$  is determined by the fluid–particle hydrodynamics. Higher Reynolds numbers will lead to smaller  $\delta_f$  values and, consequently, lower film diffusion resistance.
2. Next, component  $i$  diffuses into the *macropores*. For a fluid at least four resistances contribute to transport in the macropore. These are *Knudsen diffusion* (a transport process when the fluid is essentially gaseous in nature, where molecular collisions between the diffusing species and the pore walls predominate); *surface diffusion* (diffusive motion of adsorbed species over the



**Fig. 1** Schematic showing a batch adsorber in which the (spherical) zeolite sorbent particles are exposed to a well-mixed fluid environment.

walls of the pores); *bulk diffusion* (a molecular transport process mediated primarily by collisions between the diffusing species themselves); and *viscous flow* (species convected by laminar flow down a pressure gradient inside the pore). With liquids in macropores only bulk and surface diffusion can be expected to make significant contributions to the effective macropore diffusivity.

3. Finally, component  $i$  diffuses inside the zeolite crystallites (assumed to be spherical with radius  $r_c$ ). This diffusion process is termed intracrystalline or *micropore diffusion*, and is also the focus of [Chapter 10](#) in this volume.

We focus attention on the situation where intracrystalline diffusion is the “controlling” resistance. The reader is referred to standard texts of Ruthven (1–4), Yang (5), Do (6), and Chen et al. (7) for more detailed discussions of modeling macropore diffusion and external transport. The transient uptake within the zeolite crystallites is described by the following partial differential equation:

$$\frac{\partial q_i}{\partial t} = -\frac{1}{\rho} \frac{1}{r^2} \frac{\partial}{\partial r} (r^2 N_i) \quad (1)$$

where  $q_i$  is the molar loading of species  $i$ , expressed as moles adsorbed per kg of zeolite;  $N_i$  is the molar flux, expressed as mol/m<sup>2</sup>/s;  $r$  is the radial distance coordinate;  $\rho$  is the zeolite density expressed in kg/m<sup>3</sup>. Alternatively, we may express the loading as the number of molecules per unit cell  $\Theta_i$ :

$$\frac{\partial \Theta_i}{\partial t} = -\frac{1}{\rho} \frac{1}{r^2} \frac{\partial}{\partial r} (r^2 N_i) \quad (2)$$

In this case, the density  $\rho$  is expressed as the number of unit cells per m<sup>3</sup> and the flux  $N_i$  as molecules/m<sup>2</sup>/s. For MFI zeolite, for example a loading of four molecules per unit cell corresponds to 0.6935 mol/kg.

For the solution of Eq. (1) or (2), we need a constitutive equation for the flux  $N_i$ .

## A. Fick, Onsager, and Maxwell-Stefan Approaches

The molar flux is commonly taken to be proportional to the gradient of the component loading:

$$N_i = -\rho D_i \frac{\partial q_i}{\partial r} = -\rho q_{i,sat} D_i \frac{\partial \theta_i}{\partial r} = -\rho \Theta_{i,sat} D_i \frac{\partial \theta_i}{\partial r} \quad (3)$$

where  $D_i$  is the Fick, or transport, diffusivity of species  $i$ ;  $q_{i,sat}$  and  $\Theta_{i,sat}$  are the saturation loadings of species  $i$ ;  $\theta_i$  is the fractional occupancy, which obeys:

$$\theta_i = \frac{q_i}{q_{i,sat}} = \frac{\Theta_i}{\Theta_{i,sat}} \quad (4)$$

An alternative to the Fick formulation is the Onsager approach of irreversible thermodynamics; in this approach, the chemical potential gradients are recognized as the “proper” driving forces for diffusion:

$$N_i = \rho \Theta_{i,sat} L_i \frac{1}{RT} \frac{\partial \mu_i}{\partial r} \quad (5)$$

where  $R$  is the gas constant;  $T$  is the temperature;  $L_i$  is the Onsager coefficient;  $\mu_i$  is the chemical potential of sorbed species  $i$ . Assuming (local) equilibrium between the sorbed species and the bulk fluid phase, we have the following relationship for the chemical potential  $\mu_i$ :

$$\mu_i = \mu_i^0 + RT \ln(f_i) \quad (6)$$

where  $\mu_i^0$  is the chemical potential in the chosen standard state and  $f_i$  is the fugacity. The chemical potential gradients may be expressed in terms of the occupancy gradient:

$$\frac{1}{RT} \frac{\partial \mu_i}{\partial r} = \frac{1}{\theta_i} \Gamma \frac{\partial \theta_i}{\partial r}; \quad \Gamma \equiv \theta_i \frac{\partial \ln f_i}{\partial \theta_i} \quad (7)$$

where  $\Gamma$  is the thermodynamic “correction” factor. If the bulk fluid phase is gaseous and the system pressures are not too high, the component partial pressure,  $p_i$ , can be used in place of the component fugacity,  $f_i$ , *i.e.*  $f_i \approx p_i$ .

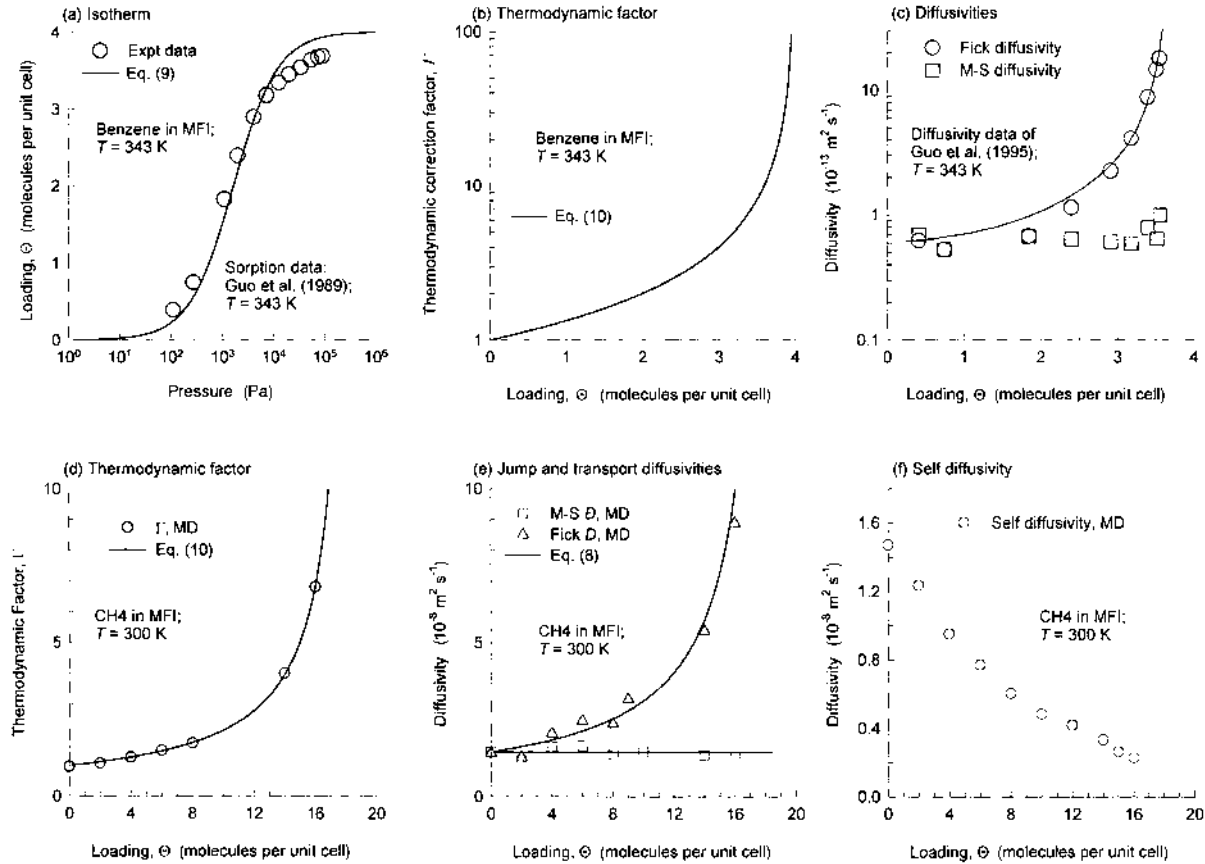
The interrelation between the Fick and Onsager coefficient is

$$\frac{L_i}{\theta_i} = \frac{D_i}{\Gamma} \equiv \mathcal{D}_i \quad (8)$$

Experimental data for several systems show that  $\mathcal{D}_i \equiv D_i/\Gamma$  is practically independent of the loading; this coefficient is referred to as the “corrected” or “jump” diffusivity. The corrected diffusivity  $\mathcal{D}_i$  is also identical to diffusivity that arises from the Maxwell-Stefan description of zeolite diffusion, to be developed below. As discussed in [Chapter 10](#) in this volume, for strong confinement of guest molecules in zeolites, the Maxwell-Stefan diffusivity often decreases with loading, following a  $\mathcal{D}_i(0)(1-\theta_i)$  where  $\mathcal{D}_i(0)$  is the zero-loading diffusivity of the pure component.

Consider the sorption data for benzene in MFI at a temperature  $T = 343$  K (8); cf. [Fig. 2a](#). The experimental data are reasonably well represented by the Langmuir isotherm:

$$\Theta_i = \frac{\Theta_{i,sat} b_i P}{1 + b_i P}; \quad \theta_i = \frac{b_i P}{1 + b_i P} \quad (9)$$



**Fig. 2** (a) Pure component isotherms for sorption of benzene on MFI at a temperature of 343 K. (Experimental data from Ref. 8.) The Langmuir model parameters are  $\Theta_{1,\text{sat}} = 4$ ,  $b_{1,A} = 6 \times 10^{-4} \text{ Pa}^{-1}$ . (b) The thermodynamic correction factor calculated using the Langmuir model. (c) Fick and Maxwell-Stefan diffusivity data for benzene in MFI at 343 K. (Data from Ref. 9). Molecular dynamics simulations of (d) thermodynamic factor, (e) jump and transport, and (f) self-diffusivities of CH<sub>4</sub> in MFI at 300 K. (MD data from Ref. 10.)

where the saturation capacity  $\Theta_{i,\text{sat}}$  is four molecules per unit cell of MFI and the Langmuir constant  $b_i = 6 \times 10^{-4} \text{ Pa}^{-1}$ . The thermodynamic correction factor can be determined from Eq. (7) as follows:

$$\Gamma = \frac{1}{1 - \Theta_i/\Theta_{i,\text{sat}}} = \frac{1}{1 - \theta_i} \quad (10)$$

Figure 2b shows the variation of the thermodynamic factor with molecular loading. Notice the sharp increase in  $\Gamma$  as  $\Theta_i$  approaches the saturation capacity,  $\Theta_{i,\text{sat}}$  (=4). The Fick diffusivity data for benzene in MFI measured by Shah et al. (9) are shown in Fig. 2c and are seen to parallel the behavior of  $\Gamma$ . As seen in Fig. 2c  $D_i$  increases sharply as  $\Theta_i$  approaches the saturation capacity,  $\Theta_{i,\text{sat}}$  (=4). The Maxwell-Stefan diffusivity  $\mathcal{D}_i$  is practically independent of the sorbate loading; see the square symbols in Fig. 2c.

Consider now the Molecular Dynamics (MD) simulation results of Maginn et al. (10) for diffusion of  $\text{CH}_4$  in MFI at 300 K for  $\Gamma$ ,  $\mathcal{D}$ , and  $D$  that have been reproduced in Fig. 2d and e. The simulated  $\Gamma$  follows Langmuirian behavior with a fitted value of  $\Theta_{\text{sat}} = 18.76$  molecules per unit cell; see Fig. 2d. From Fig. 2e, we note that the jump, or Maxwell-Stefan diffusivity, is essentially independent of loading. The transport  $D$ , which is the product,  $\mathcal{D}\Gamma$ , shows the same trend as does  $\Gamma$ ; see Fig. 2b. The MD simulations for the self-diffusivity  $D^*$  show that these decrease with loading; this is due to vacancy correlation effects as discussed by Paschek and Krishna (11). The self-diffusivity cannot be identified with the jump diffusivity except at zero loadings.

## B. Adsorption vs. Desorption Rates

An important consequence of the nonlinear loading dependence of the Fick diffusivity following  $D_i = \mathcal{D}_i/(1 - \theta_i)$  is that *adsorption* and *desorption* processes do not proceed at the same rate. During the adsorption process the Fick diffusivity increases with time, i.e., with loading. Conversely, during the desorption process the Fick diffusivity decreases with time and therefore proceeds considerably more slowly than adsorption. This asymmetry is illustrated by calculations for diffusion of ethane in 4A zeolite presented in Fig. 3, which were obtained by solving the partial differential Eq. (1) subject to the following conditions:

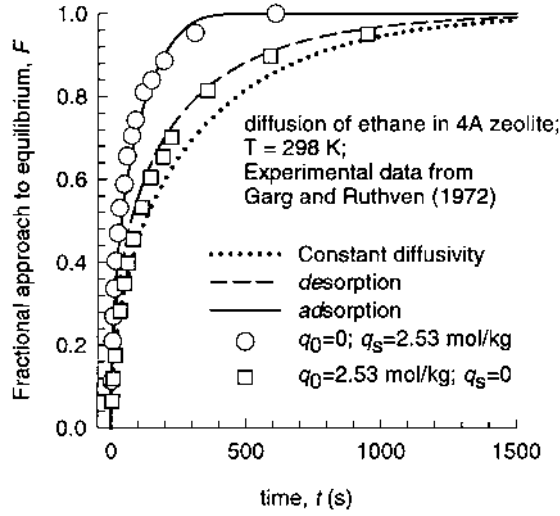
$$\text{Initial condition: } t = 0; \quad 0 < r < r_c: \quad q_i = q_{i,0}; \quad \Theta_i = \Theta_{i,0}; \quad \theta_i = \theta_{i,0} \quad (11)$$

$$\text{Boundary (surface) condition: } t > 0; \quad r = r_c: \quad q_i = q_{i,s}; \quad \Theta_i = \Theta_{i,s}; \quad \theta_i = \theta_{i,s} \quad (12)$$

Here  $q_{i,0}$  is the initial loading and  $q_{i,s}$  is the loading that is in equilibrium with the bulk fluid phase. The Maxwell-Stefan diffusivity of ethane 4A zeolite is taken as  $\mathcal{D}_i/r_c^2 = 2.45 \times 10^{-4} \text{ s}^{-1}$  following Garg and Ruthven (12). The method of lines (13) was used to solve the set of Eqs. (1), (3), and (10). The  $y$  axis in Fig. 2 represents the fractional approach to equilibrium, defined as  $(\bar{q}_i - q_{i,0})/(q_{i,s} - q_{i,0})$  where  $\bar{q}_i$  is the average loading within the particle at any time  $t$ , defined by

$$\bar{q}_i = \frac{3}{r_c^3} \int_0^{r_c} q_i r^2 dr \quad (13)$$

As can be seen in Fig. 3, there is excellent agreement between the experiments of Garg and Ruthven (12) and the simulation results. Also, the results shown in Fig. 3 confirm the asymmetry in adsorption and desorption kinetics.



**Fig. 3** Adsorption and desorption of ethane in 4A zeolite. (Experimental data from Ref. 12.) The simulation results with  $D_i/r_c^2 = 2.45 \times 10^{-4} \text{ s}^{-1}$ .

### C. Constant Fick Diffusivity and LDF Models

If the Fick diffusivity is independent of the loading, Eq. (1) can be solved analytically to obtain

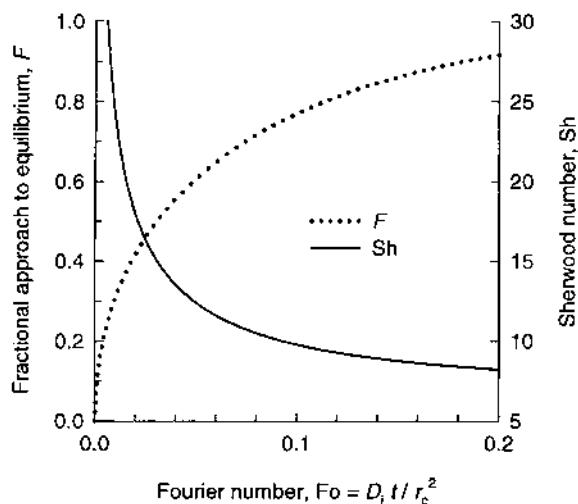
$$\frac{(\bar{q}_i - q_{i,0})}{(q_{i,s} - q_{i,0})} \equiv F = 1 - \frac{6}{\pi^2} \sum_{m=1}^{\infty} \frac{\exp(-m \pi^2 \frac{D_i}{r_c^2} t)}{m^2} \quad (14)$$

For comparison purposes, this constant- $D$  solution is also plotted in Fig. 3 with a dotted line. It is clear that thermodynamic correction factor, given by Eq. (10), has a significant influence on adsorption and desorption rates.

Equation (14) can be used to obtain the time-averaged Sherwood number ( $Sh$ ) within the zeolite crystals:

$$Sh \equiv \frac{k(2r_c)}{D} \equiv - \frac{2}{3 \left( \frac{D_i}{r_c^2} t \right)} \ln(1 - F) \quad (15)$$

where  $F$  is the fractional approach to equilibrium given by Eq. (14), and  $k$  in Eq. (15) is defined by the second equality. The variation of the  $Sh$  and  $F$  with the Fourier number,  $tD/r_c^2$ , is shown in Fig. 4. In the adsorption literature a constant value of  $Sh = 10$ , corresponding to a 75% approach to equilibrium, is usually taken to model uptake in batch adsorbers and breakthrough curves in packed beds. This assumption is usually referred to as the linear driving force (LDF) approximation. In general, the LDF approach is not appropriate for modeling of zeolitic adsorption because of the strong dependence of the Fick diffusivity on the loading. The LDF approach is probably good enough when the fractional loading  $\theta$  in the zeolite is below about 0.2 during the entire process. For such cases, the constant- $D$  assumption may be justified. For all other cases, we have to contend



**Fig. 4** Simulation of intraparticle diffusion using constant diffusivity model.

with solving the nonlinear partial differential equations (1) within the zeolite crystal. We do not discuss the LDF approximation further in this chapter and refer the reader to standard texts (1–6).

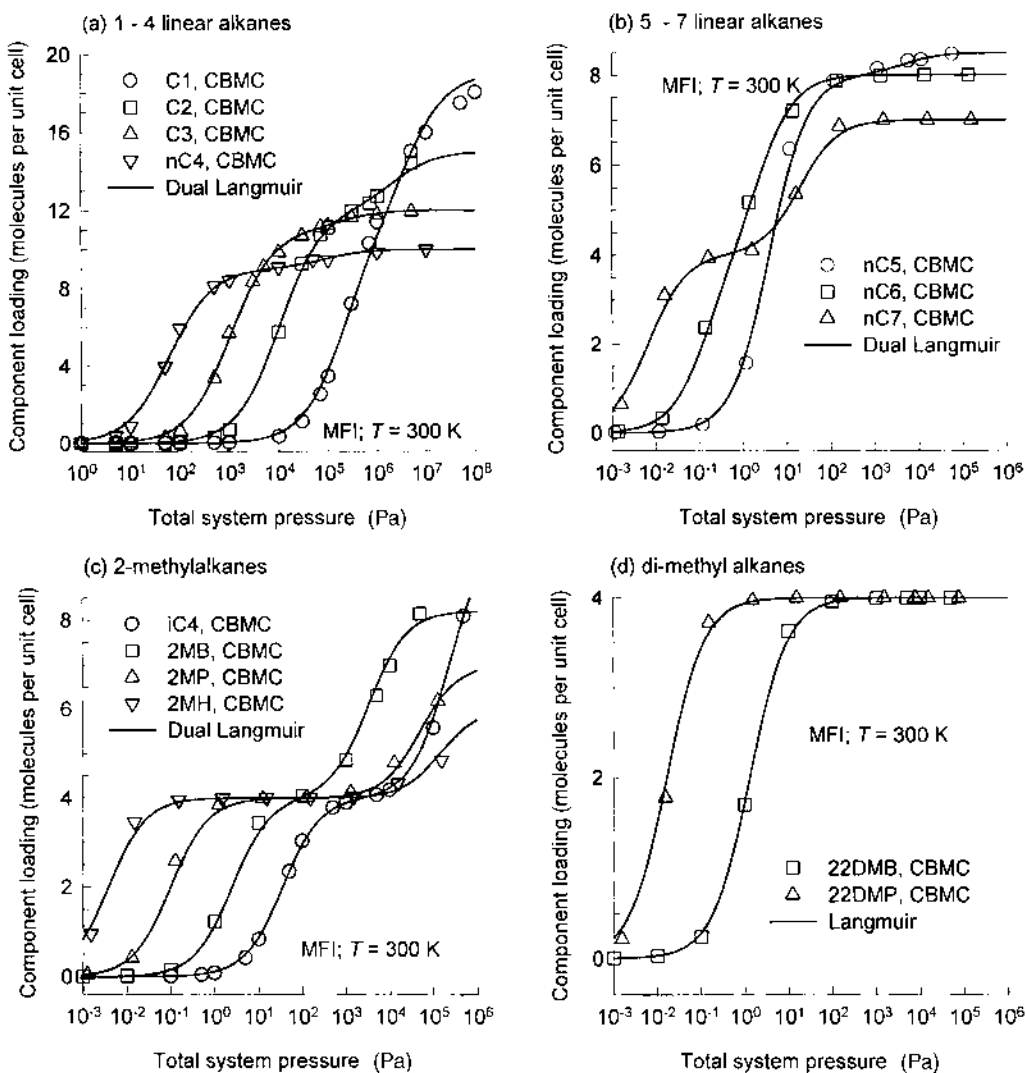
#### D. Isotherm Inflection Influence

In the foregoing discussions we have used the Langmuir isotherm to describe the sorption isotherm; in many cases the isotherms exhibit inflection behavior. Configurational bias Monte Carlo (CBMC) simulations of the isotherms of alkanes in the one- to seven-carbon-atom range at 300 K in MFI zeolite are shown in Fig. 5; the CBMC simulation technique has been described in detail in publications by Smit and coworkers (14–21) (see also Chapter 9 in this volume by the same authors). Normal heptane shows a pronounced inflection at a loading of  $\Theta = 4$ . Normal hexane shows a slight inflection at this loading due to “commensurate freezing” effects (22). All 2-methylalkanes show inflection behavior (see Fig. 5c); this is because these molecules prefer to locate at the intersections between straight and zig-zag channels, which offers more “leg room” (20). At  $\Theta = 4$  all intersections are fully occupied. To locate the 2-methylalkanes within the channel interiors requires an extra push, leading to inflection behavior. The 2,2-dimethylbutane (22DMB) and 2,2-dimethylpentane (22DMP) molecules are too bulky to be located at the channel interiors, and both molecules show a maximal (saturation) loading of 4 (see Fig. 5d).

A simple model accounting for inflection behavior is the dual-site Langmuir (DSL) model. In this model the loading,  $\Theta_i^0(P)$ , expressed in molecules per unit cell, is expressed as a function of the pressure  $P$  as follows:

$$\Theta_i^0 \equiv \Theta_{i,A} + \Theta_{i,B} = \frac{\Theta_{i,sat,A} b_{i,A} P}{1 + b_{i,A} P} + \frac{\Theta_{i,sat,B} b_{i,B} P}{1 + b_{i,B} P} \quad (16)$$

The superscript 0 on  $\Theta_i^0(P)$  is used to emphasize that the relation is for *pure* component loadings. In Eq. (16),  $b_{1,A}$  and  $b_{1,B}$  represent the DSL model parameters expressed in  $\text{Pa}^{-1}$



**Fig. 5** Pure component isotherms for linear, 2-methyl- and dimethylalkanes in MFI at 300 K calculated using CBMC simulations (Refs. 14–21). The continuous lines represent the fits of the isotherms using the dual-site Langmuir model with parameters given in Table 1.

and the subscripts A and B refer to two sorption sites within the MFI structure, with different sorption capacities and sorption strengths. The  $\Theta_{i,\text{sat},A}$  and  $\Theta_{i,\text{sat},B}$  represent the saturation capacities of sites A and B, respectively. The fitted parameters for the pure component isotherms, shown in Fig. 5, are listed in Table 1. It is to be noted that the total saturation loading  $\Theta_{i,\text{sat}} = \Theta_{i,\text{sat},A} + \Theta_{i,\text{sat},B}$  is not a fitted parameter but taken from the final plateau value of the sorption isotherm, estimated from CBMC simulations. In general, the saturation loading decreases with increasing carbon number and with increased degree of branching; see Fig. 6.



**Table 1** Dual-Site Langmuir Parameters for Pure Alkanes in MFI at 300 K<sup>a</sup>

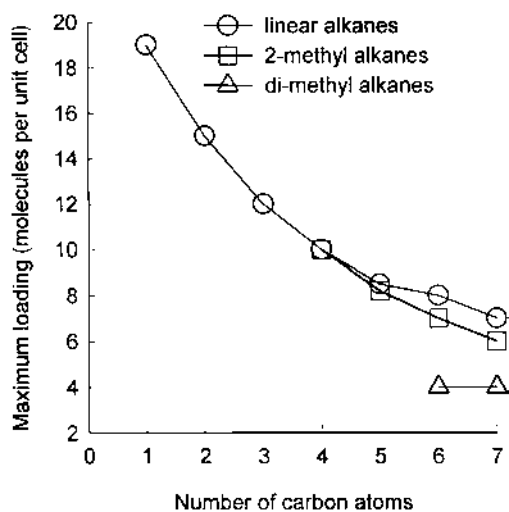
Component	Dual Langmuir parameters			
	Site A		Site B	
	$b_{i,A}$ (Pa <sup>-1</sup> )	$\Theta_{i,sat,A}$ (molecules per unit cell)	$b_{i,B}$ (Pa <sup>-1</sup> )	$\Theta_{i,sat,B}$ (molecules per unit cell)
C1	$4.86 \times 10^{-6}$	11.0	$2.38 \times 10^{-7}$	8.0
C2	$9.73 \times 10^{-5}$	12.0	$4.38 \times 10^{-7}$	3.0
C3	$9.64 \times 10^{-4}$	11.0	$5.06 \times 10^{-6}$	1.0
<i>n</i> C4	$1.63 \times 10^{-2}$	9.0	$1.14 \times 10^{-5}$	1.0
<i>n</i> C5	0.25	8.0	$2 \times 10^{-4}$	0.5
2MB	0.4	4.0	$3 \times 10^{-4}$	4.2
<i>n</i> C6	7.0	4.0	0.4	4.0
2MP	10.0	4.0	$2.0 \times 10^{-5}$	3.0
22DMB	0.76	4.0	—	0
<i>n</i> C7	150	4.0	$5 \times 10^{-2}$	3.0
2MH	260	4.0	$7 \times 10^{-6}$	2.0
22DMP	60	4.0	—	0

<sup>a</sup> Fits correspond to CBMC simulations.

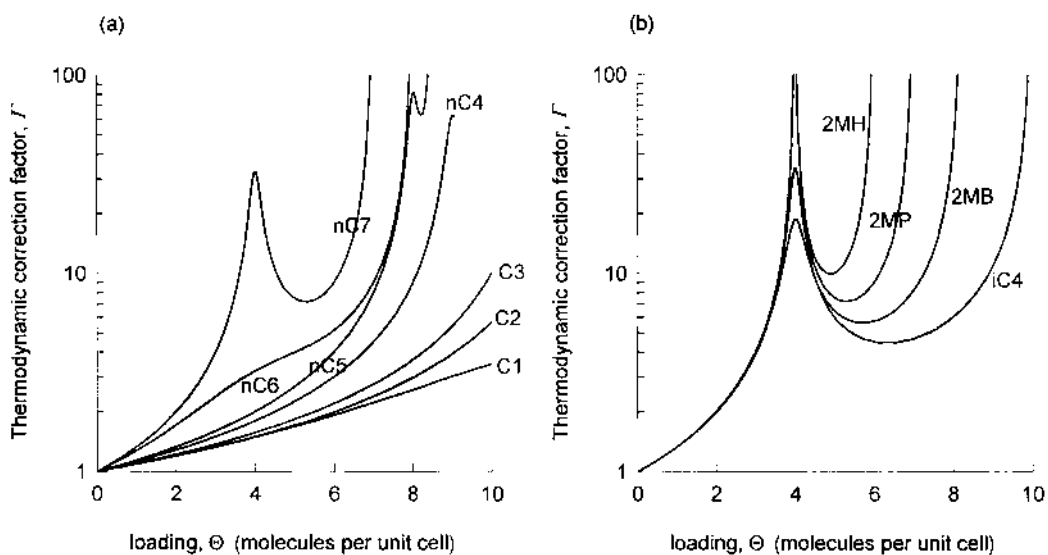
The thermodynamic correction factor for the DSL model is

$$\Gamma = \frac{1}{\frac{\Theta_{i,A}}{\Theta_i} \left(1 - \frac{\Theta_{i,A}}{\Theta_{i,sat,A}}\right) + \frac{\Theta_{i,B}}{\Theta_i} \left(1 - \frac{\Theta_{i,B}}{\Theta_{i,sat,B}}\right)} \quad (17)$$

Calculations of the thermodynamic correction factor for linear and 2-methylalkanes are shown in Fig. 7. The thermodynamic correction factor for *n*C7 and all 2-methylalkanes



**Fig. 6** Maximal saturation loadings of linear, 2-methyl- and dimethylalkanes in MFI at 300 K. CBMC calculations at 300 K using NVT simulations. (From Ref. 16.)



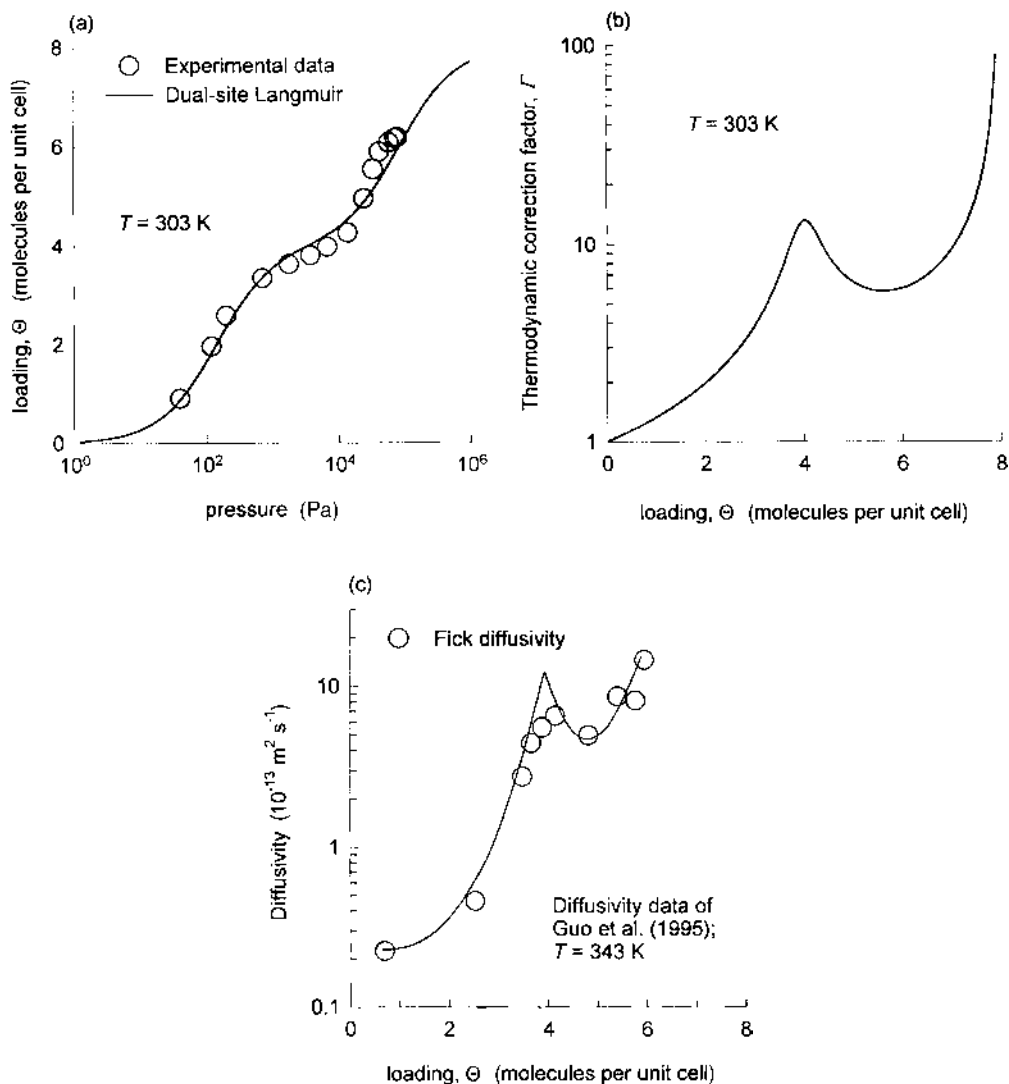
**Fig. 7** Thermodynamic factor for (a) linear alkanes and (b) 2-methylalkanes in MFI at 300 K calculated using the dual-site Langmuir (DSL) model. DSL parameters given in [Table 1](#).

shows two extrema: a maximum at the inflection point  $\Theta = \Theta_{i,sat,A} = 4$  and a minimum at a loading  $\Theta_{i,sat,A} < \Theta < \Theta_{i,sat,A} + \Theta_{i,sat,B}$ .

The isotherm inflection for 2-methylalkanes is due to preferential location of molecules at the intersection. Inflection behavior is also exhibited by benzene in MFI at  $T = 303$  K due to phase transitions ([Fig. 8a](#)). Since the Fick diffusivity is proportional to the thermodynamic factor, it can be expected to also exhibit two extrema if the Maxwell-Stefan diffusivity has a negligible loading dependence. This is verified by the experimental data of Shah et al. (9); see [Fig. 8c](#).

In order to illustrate the influence of isotherm inflection on the uptake characteristics, let us consider desorption of 3-methylpentane (3MP) from MFI zeolite at a temperature of 362 K. The DSL isotherm parameters, obtained from CBMC simulations (14–21), are specified in [Table 2](#). Initially the crystals are equilibrated by exposing to surrounding 3MP vapor at a pressure  $p_{i,0} = 100$  kPa. The equilibrium loading within MFI is uniform and  $\Theta_{i,0} = 5.596$  molecules per unit. At time  $t = 0$ , the surface of the zeolite crystals is exposed to 3MP vapor at a partial pressure  $p_{i,s} = 0.5$  Pa, which gives an equilibrium loading at the outer surface  $\Theta_{i,s} = 0.093$ . Numerical solution of the partial differential equation (2), with the initial and boundary conditions given by Eqs. (11) and (12), yields the  $\Theta_i(t)$  shown in [Fig. 9a](#). We note an inflection in the desorption kinetics, indicated by the arrow. [Figure 9b](#) compares the desorption and adsorption kinetics of 3MP. The desorption kinetics are much slower than the adsorption kinetics; this is evidenced by the fact that at  $Fo \equiv Dt/r_c^2 = 0.04$ , the adsorption process is nearly at equilibrium whereas the desorption process has still a long way to go to equilibration.

A three-site model for sorption of aromatics on ZSM-5 has been proposed by Rudzinski et al. (23) to account for two inflection points observed under certain temperature conditions. The consequences for diffusion can be expected to be interesting but there is no experimental evidence in the literature.



**Fig. 8** (a) Pure component isotherms for sorption of benzene on MFI at a temperature of 303 K. (Experimental data from Ref. 8.) The dual-site Langmuir model parameters are  $\Theta_{\text{sat,A}} = 4$ ,  $\Theta_{\text{sat,B}} = 4$ ,  $b_{1,A} = 7 \times 10^{-3} \text{ Pa}^{-1}$ ,  $b_{1,B} = 1.2 \times 10^{-5} \text{ Pa}^{-1}$ . (b) The thermodynamic correction factor calculated using the DSL model. (c) Fick diffusivity data for benzene in MFI at 303 K. (Data from Ref. 9.)

### III. PERMEATION OF SINGLE COMPONENT ACROSS ZEOLITE MEMBRANE

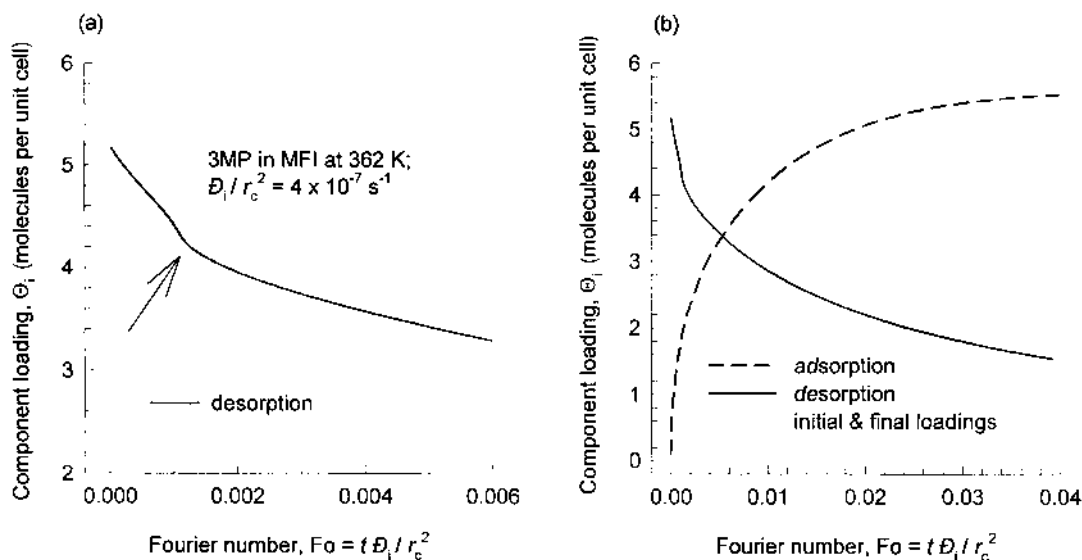
Zeolite membranes are currently being developed for carrying out separations and catalytic reactions. It is important in practice to determine the permeation fluxes across the membrane. Most commonly, zeolite membrane devices are operated under steady-state conditions, though laboratory experiments also monitor the transience prior to achievement of steady state (24–26).

**Table 2** Dual-site Langmuir Parameters for Hexane Isomers in MFI<sup>a</sup>

Component	Temp. (K)	Dual Langmuir parameters			
		Site A		Site B	
		$b_{i,A}$ (Pa <sup>-1</sup> )	$\Theta_{i,sat,A}$ (molecules per unit cell)	$b_{i,B}$ (Pa <sup>-1</sup> )	$\Theta_{i,sat,B}$ (molecules per unit cell)
<i>n</i> C <sub>6</sub>	362	$6.32 \times 10^{-2}$	4.0	$1.7 \times 10^{-3}$	4.0
3MP	362	$4.75 \times 10^{-2}$	4.0	$2.27 \times 10^{-5}$	2.3

<sup>a</sup> Fits correspond to CBMC simulations (from Ref. 30).

We focus here on the permeation of a single component, species *i*, from a well-mixed upstream compartment, across the membrane, to a well-mixed downstream compartment. The permeate gases are often flushed out of the downstream compartment by means of a “sweep” gas (e.g., helium) in order to maintain the partial pressures of the permeants at low values; see Fig. 10. The zeolite crystals are deposited, or grown, onto a support layer consisting, say, of metal wool and/or a macroporous layer of sintered stainless steel particles (25,26). The diffusion through the support layer is akin to transport through the macropores within a sorbent particle and has been considered in detail by van de Graaf (25). The zeolite layer may have nanoscopic defects such as voids and pinholes that can be modeled in the manner described by Nelson et al. (27). In the following we ignore the support resistance and concentrate on the permeation characteristics of a defect-free zeolite membrane layer (of thickness  $\delta$ ). The upstream and downstream faces of the zeolite



**Fig. 9** (a) Kinetics of desorption of 3MP in MFI at 362 K. (b) Comparison of adsorption and desorption kinetics for 3MP.

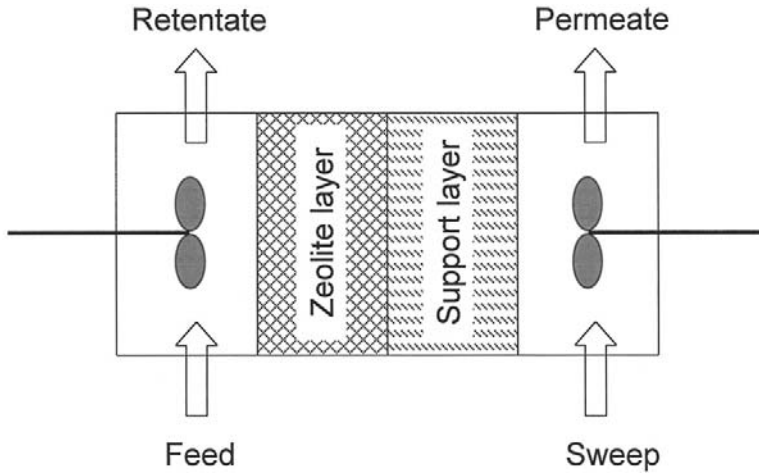


Fig. 10 Schematic of zeolite membrane separation device.

layer are assumed to be in equilibrium with the upstream and downstream partial pressures of the species, respectively:

$$\text{Upstream face: } z = 0; \quad p_i = p_{i,0}; \quad q_i = q_{i,0}; \quad \Theta_i = \Theta_{i,0}; \quad \theta_i = \theta_{i,0} \quad (18)$$

$$\text{Downstream face: } z = \delta; \quad p_i = p_{i,\delta}; \quad q_i = q_{i,\delta}; \quad \Theta_i = \Theta_{i,\delta}; \quad \theta_i = \theta_{i,\delta} \quad (19)$$

The permeation flux is obtained by solving

$$\frac{\partial \theta_i}{\partial t} = - \frac{1}{\rho \Theta_{i,sat}} \frac{\partial N_i}{\partial z} \quad (20)$$

where Eq. (3) is used to describe the single-component permeation flux.

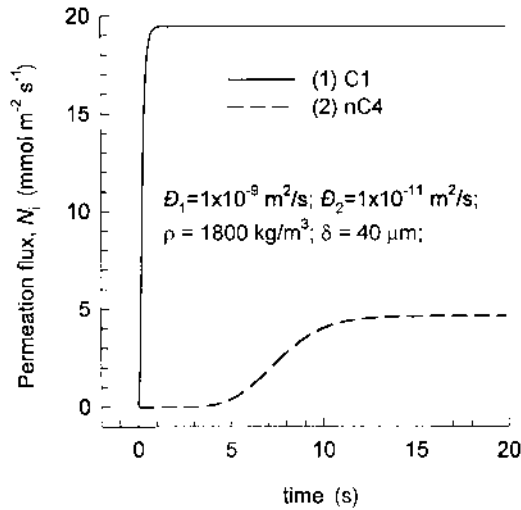
### A. Permeation of Methane and *n*-Butane Across MFI Membrane

For illustration, let us consider transient permeation of methane (C1) across a MFI membrane at 300 K. The upstream partial pressure of C1 is held at  $p_{i,0} = 50$  kPa. The downstream pressure  $p_{i,\delta}$  is maintained at vanishing values by means of a sweep gas. Taking the Maxwell-Stefan diffusivity of methane as  $\mathcal{D}_i = 10^{-9}$  m<sup>2</sup>/s, along with the DSL isotherm parameters reported in Table 1, Eq. (20) can be solved to follow the transience in permeation flux  $N_i$  as steady state is approached; the results are shown in Fig. 11. At steady state, the permeation flux of methane is 19.46 mmol/m<sup>2</sup>/s. For identical upstream and downstream partial pressures, the permeation of *n*-butane (*n*C4), with  $\mathcal{D}_i = 10^{-11}$  m<sup>2</sup>/s, DSL parameters (also given in Table 1) show a much slower approach to the steady-state value of 4.65 mmol/m<sup>2</sup>/s (see Fig. 11) because of its much lower diffusivity value.

For a 50:50 mixture of C1 (1) and *n*C4 (2), each with a 50-kPa upstream partial pressure, we might expect the permeation selectivity,  $S_p$ , defined by

$$S_p = \frac{N_2/N_1}{p_{20}/p_{10}} \quad (21)$$

to be  $4.65/19.46 = 0.239$ . We will see later that this expectation is far removed from reality because of the peculiarities of mixture diffusion in zeolites.



**Fig. 11** Single-component transient permeation of methane (C1) and *n*-butane (*n*C4) across MFI membrane at 300 K.

#### IV. MODELING MIXTURE DIFFUSION WITH THE MAXWELL-STEFAN FORMULATION

For *n*-component diffusion the fluxes  $N_i$  are related to the gradients of the fractional occupancies by the generalization of Fick's law:

$$(\mathbf{N}) = -\rho[\Theta_{sat}][D] \frac{\partial(\theta)}{\partial r} \quad (22)$$

where  $[D]$  is the *n*-dimensional square matrix of Fick diffusivities;  $[\Theta_{sat}]$  is a diagonal matrix with elements  $\Theta_{i,sat}$ , representing the saturation loading of species *i*. The fractional occupancies  $\theta_i$  are defined by Eq. (4). The estimation of the  $n \times n$  elements of  $[D]$  is complicated by the fact that these are influenced not only by the species mobilities (i.e., diffusivities  $\mathcal{D}_i$ ) but also by the sorption thermodynamics. In setting up a proper mixture diffusion theory we need to use chemical potential gradients as the proper driving forces. In the Onsager irreversible thermodynamics (IT) formulation, we generalize Eq. (5) in the following manner:

$$(\mathbf{N}) = -\rho[\Theta_{sat}][L] \frac{1}{RT} \frac{\partial(\mu)}{\partial r} \quad (23)$$

where  $\frac{\partial(\mu)}{\partial r}$  is the column matrix of chemical potential gradients;  $[L]$  is the square matrix of Onsager coefficients having the units  $[\text{m}^2 \text{s}^{-1}]$ . The Onsager matrix  $[L]$  is nondiagonal, in general, and the cross-coefficients portray the coupling between species diffusion. The Onsager reciprocal relations demand that the matrix  $[L]$  be symmetrical, i.e.,

$$L_{ij} = L_{ji}; \quad i, j = 1, 2, \dots, n \quad (i \neq j) \quad (24)$$

The chemical potential gradients in Eq. (23) may be expressed in terms of the gradients of the occupancies by introduction of the matrix of thermodynamic factors  $[\Gamma]$

$$\frac{\theta_i}{RT} \frac{\partial \mu_i}{\partial r} = \sum_{j=1}^n \Gamma_{ij} \frac{\partial \theta_j}{\partial r}; \quad \Gamma_{ij} \equiv \left( \frac{\Theta_{j,sat}}{\Theta_{i,sat}} \right) \frac{\Theta_i}{p_i} \frac{\partial p_i}{\partial \Theta_j}; \quad i, j = 1, 2, \dots, n \quad (25)$$

Knowledge of the sorption isotherm is sufficient to allow estimation of  $[\Gamma]$  and  $\frac{\partial(\mu)}{\partial r}$ . If the  $n$ -component sorption can be described by the multicomponent Langmuir isotherm, the elements of  $[\Gamma]$  are given by

$$\Gamma_{ij} = \delta_{ij} + \frac{\theta_i}{1 - \theta_1 - \theta_2 - \dots - \theta_n}; \quad i, j = 1, 2, \dots, n \quad (26)$$

where  $\delta_{ij}$  is the Kronecker  $\delta$ .

Combining Eqs. (23) and (25), we obtain

$$(N) = -\rho[\Theta_{sat}][L] \begin{bmatrix} 1/\theta_1 & 0 & 0 \\ 0 & \ddots & 0 \\ 0 & 0 & 1/\theta_n \end{bmatrix} [\Gamma] \frac{\partial(\theta)}{\partial r} \quad (27)$$

Comparing Eqs. (22) and (27), we obtain the interrelation:

$$[D] = [L] \begin{bmatrix} 1/\theta_1 & 0 & 0 \\ 0 & \ddots & 0 \\ 0 & 0 & 1/\theta_n \end{bmatrix} [\Gamma] \quad (28)$$

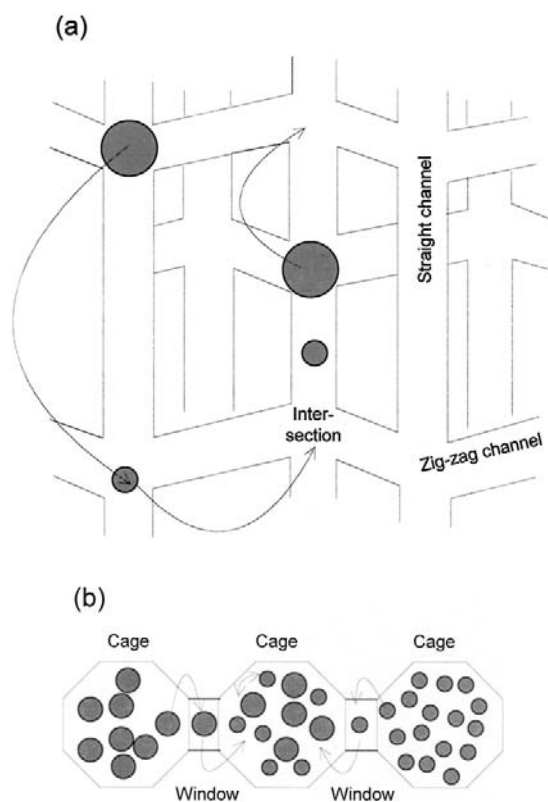
The Fick matrix  $[D]$  can be estimated from knowledge of the Onsager matrix  $[L]$ . In general the thermodynamic correction factor matrix  $[\Gamma]$  is nondiagonal and this makes multicomponent diffusion in zeolites a strongly coupled process.

Unfortunately, the IT theory provides no fundamental guidelines for estimating  $[L]$  from data on pure component transport coefficients. For estimating  $[D]$  it is more convenient to adopt the Maxwell-Stefan formulation, entirely consistent with the theory of IT, in which the chemical potential gradients are written as linear functions of the fluxes (28–31):

$$-\rho \frac{\theta_i}{RT} \nabla \mu_i = \sum_{\substack{j=1 \\ j \neq i}}^n \frac{\Theta_j \mathbf{N}_i - \Theta_i \mathbf{N}_j}{\Theta_{i,sat} \Theta_{j,sat} \mathcal{D}_{ij}} + \frac{\mathbf{N}_i}{\Theta_{i,sat} \mathcal{D}_i}; \quad i = 1, 2, \dots, n \quad (29)$$

We have to reckon in general with two types of Maxwell-Stefan diffusivities:  $\mathcal{D}_i$  and  $\mathcal{D}_{ij}$ . The  $\mathcal{D}_i$  are the diffusivities that reflect interactions between species  $i$  and the zeolite matrix; they are also referred to as jump or “corrected” diffusivities in the literature (1–4) and can be identified with the pure component transport parameters. There are two types of loading dependences which the  $\mathcal{D}_i$  exhibit: (a) for weak confinement of guest molecules,  $\mathcal{D}_i = \mathcal{D}_i(0)$ , the zero-loading pure component diffusivity, and (b) for strong confinement  $\mathcal{D}_i = \mathcal{D}_i(0)(1 - \theta_i)$ . In all of the illustrative examples to be presented below we assume the weak confinement scenario to hold.

Mixture diffusion introduces an additional complication due to guest<sub>*i*</sub>–guest<sub>*j*</sub> interactions. This interaction is embodied in the “interchange” or “exchange” coefficients  $\mathcal{D}_{ij}$ . We can consider this coefficient as representing the facility for counterexchange, i.e., at a sorption site the guest species  $j$  is replaced by the species  $i$ . For intersecting channel structures such as MFI (see Fig. 12a) the interchange process takes place predominantly



**Fig. 12** Pictorial representation of the molecular jumps in (a) MFI structure and (b) cages separated by windows.

at the channel intersections. In structures such as LTA and FAU, molecule–molecule exchange takes place predominantly within the cages (Fig. 12b). The Onsager reciprocal relations require  $\mathcal{D}_{ij} = \mathcal{D}_{ji}$ . The net effect of this counterexchange is a slowing down of a faster moving species due to interactions with a species of lower mobility. Also, a species of lower mobility is accelerated by interactions with another species of higher mobility. An alternative interpretation is to regard  $\mathcal{D}_{ij}$  as quantifying vacancy correlation effects (32–34). For structures such as MFI, consisting of a three-dimensional network of intersecting straight and zig-zag channels, there is a strong correlation between the molecular jumps and the inclusion of the interchange coefficients  $\mathcal{D}_{ij}$  is essential in describing the mixture diffusion process; this has been verified by Paschek and Krishna (34) using kinetic Monte Carlo (KMC) simulations. For estimation of the  $\mathcal{D}_{ij}$  within MFI, they show that the logarithmic interpolation formula, suggested by Krishna and Wesselingh (28):

$$\mathcal{D}_{ij} = [\mathcal{D}_i]^{0_i/(0_i+0_j)} [\mathcal{D}_j]^{0_j/(0_i+0_j)} \quad (30)$$

is of sufficient accuracy.



We now try to interrelate the MS formulation with the Onsager and Fick formulations; to do this we define an  $n$ -dimensional square matrix  $[B]$  with elements

$$B_{ij} = \frac{1}{\mathcal{D}_i} + \sum_{\substack{j=1 \\ j \neq i}}^n \frac{\theta_j}{\mathcal{D}_{ij}}; \quad B_{ij} = \frac{\theta_j}{\mathcal{D}_{ij}}; \quad i, j = 1, 2, \dots, n \quad (31)$$

With this definition of  $[B]$ , Eq. (29) can be cast in  $n$ -dimensional matrix form:

$$(N) = -\rho[\Theta_{sat}][B]^{-1}[\Gamma] \frac{\partial(\theta)}{\partial r} \quad (32)$$

which gives the following expressions for the Onsager and Fick matrices

$$[L] = [B]^{-1} \begin{bmatrix} \theta_1 & 0 & 0 \\ 0 & \ddots & 0 \\ 0 & 0 & \theta_n \end{bmatrix}; \quad [D] = [B]^{-1}[\Gamma] \quad (33)$$

For single-component diffusion, Eq. (33) simplifies to Eq. (8). Equations (31) and (33) show that the interchange coefficients  $\mathcal{D}_{ij}$ , portraying correlated molecular jumps, will influence *all* the elements of  $[L]$  and  $[D]$ . Put another way, the main elements of the Onsager matrix  $L_{ii}$  cannot be identified with pure component diffusion coefficients, as has been erroneously suggested in the literature by Sundaram and Yang (35). In general, the Fick  $[D]$  matrix has large nondiagonal elements and, consequently, the flux of any species  $i$  is strongly coupled to that of all other diffusing species  $j$  in the mixture.

For facile particle–particle exchange, i.e.,  $\mathcal{D}_{ij} \rightarrow \infty$ , vacancy correlation effects tend to get washed out. Facile counterexchange of particles could occur, for example, within the cages of FAU and LTA zeolites when intracage hopping rates are high. We see from Eqs. (31) and (33) that when  $\mathcal{D}_{ij} \rightarrow \infty$ , both  $[B]$  and  $[L]$  matrices reduce to diagonal matrices and the flux relations (29) simplify to give:

$$N_j = -\rho\Theta_{i,sat} \frac{L_{ij}}{RT} \nabla\mu_j \equiv -\rho\Theta_{i,sat} D_i \frac{\theta_i}{RT} \nabla\mu_j; \quad i = 1, 2, \dots, n \quad (34)$$

The off-diagonal elements of the Onsager matrix are also a reflection of (vacancy) correlation effects and the assumption of vanishing off-diagonal elements of  $[L]$  signifies vanishing correlation effects. The set of equations (34), with the multicomponent Langmuir model to estimate mixture isotherms [see Eq. (26)] were first developed by Habgood (36,37) to describe two-component diffusion in zeolite 4A. In zeolite 4A, the intracage hopping is not a limiting factor ( $\mathcal{D}_{ij} \rightarrow \infty$ ). The intercage hopping is governed by guest–host interactions only, as described by Eq. (34). The Habgood model is thus a special limiting case of the Maxwell-Stefan approach.

The mixture sorption characteristic influences mixture diffusion in two ways: (a) mixture sorption determines the magnitudes of the driving forces  $\partial(\theta)/\partial r$ , and (b) they contribute to *coupling* of the diffusion process due to the presence of the nondiagonal elements in  $[\Gamma]$ ; even when Eq. (34) applies, the species diffusion is still coupled. The proper modeling of mixture sorption is the first essential step toward a proper modeling of mixture diffusion. In particular size and configurational entropy effects during mixture sorption need to be properly addressed. The entropy effects influencing mixture sorption also have a significant influence on mixture diffusion, as will be illustrated below by means of several illustrative examples (see also [Chapter 9](#) in this volume).

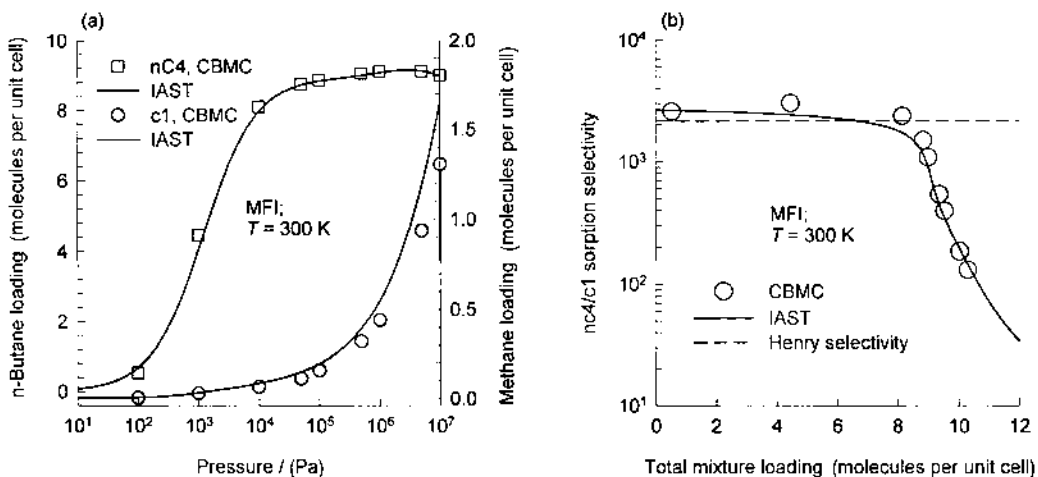
## V. ILLUSTRATIVE EXAMPLES OF BINARY MIXTURE DIFFUSION

### A. Permeation of Methane–*n*-Butane Mixture Across MFI Membrane

Consider permeation of a 95:5 binary mixture of methane (C1) and *n*-butane (*n*C4) across an MFI membrane at 300 K. The first important step in modeling of diffusion is to model the pure component and mixture sorption characteristics; this is required in the determination of the component driving forces  $\partial(\theta)/\partial r$  and the thermodynamic correction factors  $[\Gamma]$ . The pure component isotherms at 300 K in MFI zeolite, obtained from CBMC simulations, are shown in Fig. 5a, along with the DSL fits using the parameters specified in Table 1. For a 95:5 mixture of C1 and *n*C4, the component loadings in the mixture obtained from CBMC simulations are shown in Fig. 13a. The loading of C1 increases monotonically with increasing pressure. On the other hand the loading of *n*C4 reaches a plateau value for pressures in the 1- to 5-MPa range. Increasing the total system pressure beyond 5 MPa leads to a very slight decline in the loading of *n*C4. In Fig. 13b, we plot the sorption selectivity,  $S$ , defined by:

$$S = \frac{\Theta_2/\Theta_1}{p_2/p_1} \quad (35)$$

where  $p_1$  and  $p_2$  are the partial pressures in the bulk gas phase. For mixture loadings,  $\Theta_{\text{mix}} = \Theta_1 + \Theta_2$ , below 8, the sorption selectivity of *n*C4 with respect to C1 is practically constant and equals that calculated from the corresponding Henry coefficients, i.e., 2200. However, as  $\Theta_{\text{mix}}$  increases beyond 8, the sorption selectivity decreases dramatically to values about one to two orders of magnitude lower. Near saturation loadings, the vacant spaces in the zeolite are more easily occupied by the smaller methane molecule. This is a size entropy effect that favors *smaller* molecules at high pressures. It is clear that size entropy effects counter the usual enthalpic effect of chain length; increase in the chain length favors the adsorption enthalpy of the *larger* molecule.



**Fig. 13** (a) Sorption loadings of 95:5 binary mixture of C1 and *n*C4 in MFI at 300 K. (b) *n*C4/C1 sorption selectivity. The continuous lines represent the predictions of the IAST.

From a practical point of view, it is important to be able to predict the mixture isotherms from pure component isotherm data. It is clear that the multicomponent Langmuir isotherm will be totally unsuccessful in this regard because the sorption selectivity predicted by this model will be independent of the mixture loading. Let us try to estimate the mixture loadings from the pure component isotherms using the ideal adsorbed solution theory (IAST) of Myers and Prausnitz (38). Briefly, the basic equation of IAST theory is the analogue of Raoult's law for vapor–liquid equilibrium, i.e.:

$$Py_i = P_i^0(\pi)x_i; \quad i = 1, 2, \dots, n \quad (36)$$

where  $x_i$  is the mole fraction in the adsorbed phase

$$x_i = \frac{\Theta_j}{\Theta_1 + \Theta_2 + \dots + \Theta_n} \quad (37)$$

and  $P_i^0(\pi)$  is the pressure for sorption of every pure component  $i$ , which yields the same spreading pressure,  $\pi$ , as that for the mixture. The spreading pressure is defined by the Gibbs adsorption isotherm

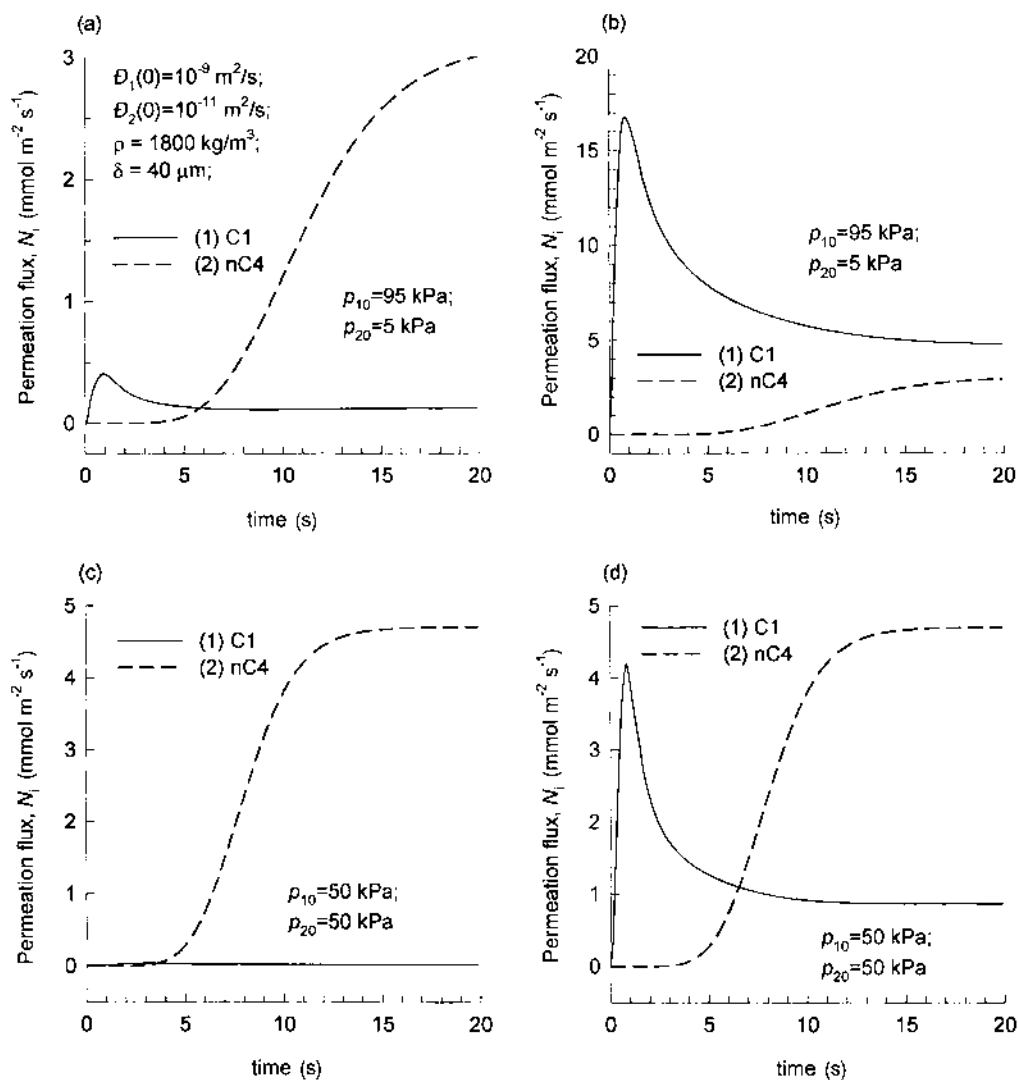
$$\frac{\pi A}{k_B T} = \rho \int_{P=0}^{P=P_i^0} \frac{\Theta_i^0(P)}{P} dP \quad (38)$$

where  $A$  is the adsorbent surface area per  $\text{m}^3$  of adsorbent,  $k_B$  is Boltzmann's constant,  $\rho$  is the density of MFI expressed in terms of the number of unit cells per  $\text{m}^3$ , and  $\Theta_i^0(P)$  is the pure component isotherm given by Eq. (16). The total amount adsorbed is obtained from

$$\Theta_{mix} \cong \Theta_1 + \Theta_2 \dots + \Theta_n = \frac{1}{\frac{x_1}{\Theta_1^0(P_1^0)} + \frac{x_2}{\Theta_2^0(P_2^0)} + \dots + \frac{x_n}{\Theta_n^0(P_n^0)}} \quad (39)$$

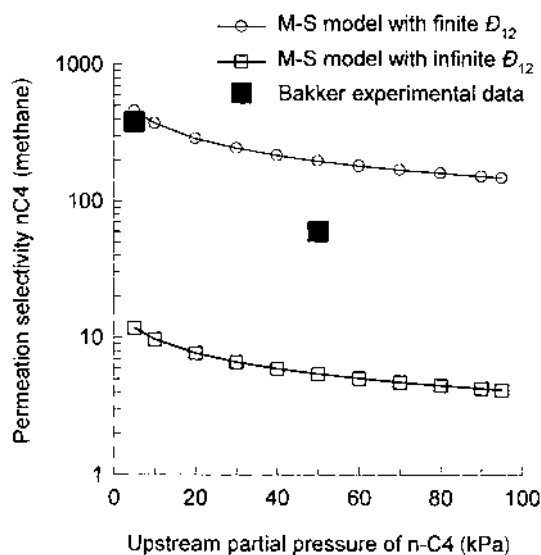
The set of Eqs. (16), (36)–(39) need to be solved numerically to obtain the loadings of the individual components in the mixture. We see in Fig. 13a and b that the IAST predictions are in excellent agreement with the CBMC simulation results. Size entropy effects are properly accounted for in the IAST mixture model.

Now let us consider permeation of the C1 (1) to  $n$ C4 (2) mixture across an MFI membrane for a situation with upstream partial pressures  $p_{1,0} = 95$  kPa,  $p_{2,0} = 5$  kPa. The downstream partial pressures  $p_{i,\delta}$  are assumed to be maintained at vanishing values by means of a sweep gas. The boundary conditions, Eqs. (18) and (19), are thus determined. The set of differential equations (20) are solved together with the (coupled) flux Maxwell-Stefan relations (32) in order to obtain the fluxes  $N_i$  across the membrane. The results are shown in Fig. 14a. The steady-state fluxes of C1 and  $n$ C4 are found, respectively, to be 0.125 and 3  $\text{mmol}/\text{m}^2/\text{s}$  giving an  $n$ C4/C1 permeation selectivity value [see Eq. (21)],  $S_p = 456$ . During the initial transience, the methane flux attains a maximum in the flux (with a value of 0.4  $\text{mmol}/\text{m}^2/\text{s}$ ) at  $t = 1$  s. The reason for this peak is that during the initial period, methane that has a diffusivity value 100 times that of  $n$ C4 diffuses faster through the membrane. However, as time progresses the MFI structure gets increasingly occupied with  $n$ C4 that dislodges the less strongly adsorbed C1. The flux of  $n$ C4 is enhanced with increased  $n$ C4 loading. Concomitantly, the flux of C1 decreases because its loading in MFI decreases. Furthermore, in the Maxwell-Stefan model the interchange coefficient  $\mathcal{D}_{ij}$  serves to slow down methane and speed up  $n$ C4. All these factors leads to a decline in the flux of C1 from its peak value of 0.4 to the steady-state value of 0.125  $\text{mmol}/\text{m}^2/\text{s}$ .



**Fig. 14** (a, b) Transient permeation across MFI membrane of 95:5 mixture of C1 and *n*C4 in MFI at 300 K. (c, d) Transient permeation of 50:50 mixture of C1 and *n*C4 in MFI at 300 K. Two implementations of the Maxwell-Stefan model are compared, with finite and infinite interchange coefficients  $\mathcal{D}_{ij}$ .

The predictions of the fluxes with the M-S model assuming  $\mathcal{D}_{ij} \rightarrow \infty$  and using Eq. (34) are shown in Fig. 14b. There is no slowing down of methane due to interchange and therefore the initial transience shows a peak methane flux of 17 mmol/m<sup>2</sup>/s, which reduces at steady state to 4.8. Assuming  $\mathcal{D}_{ij} \rightarrow \infty$  has a less dramatic influence on the *n*C4 flux, which reaches a steady-state value of 2.96 mmol/m<sup>2</sup>/s. The permeation selectivity is calculated as  $S_p = 0.235$ , significantly lower than the value of 456 using the complete Maxwell-Stefan theory with finite interchange. The experimental value of  $S_p$  for this 95:5 mixture determined by Bakker (26) is 380, quite close to the estimations of the Maxwell-Stefan model including interchange.



**Fig. 15** Dependence of  $nC4/C1$  permeation selectivity on the two implementations of Maxwell-Stefan model, with and without inclusion of the interchange coefficient  $\mathcal{D}_{ij}$ . Also shown are the experimental data of Bakker (Ref. 26).

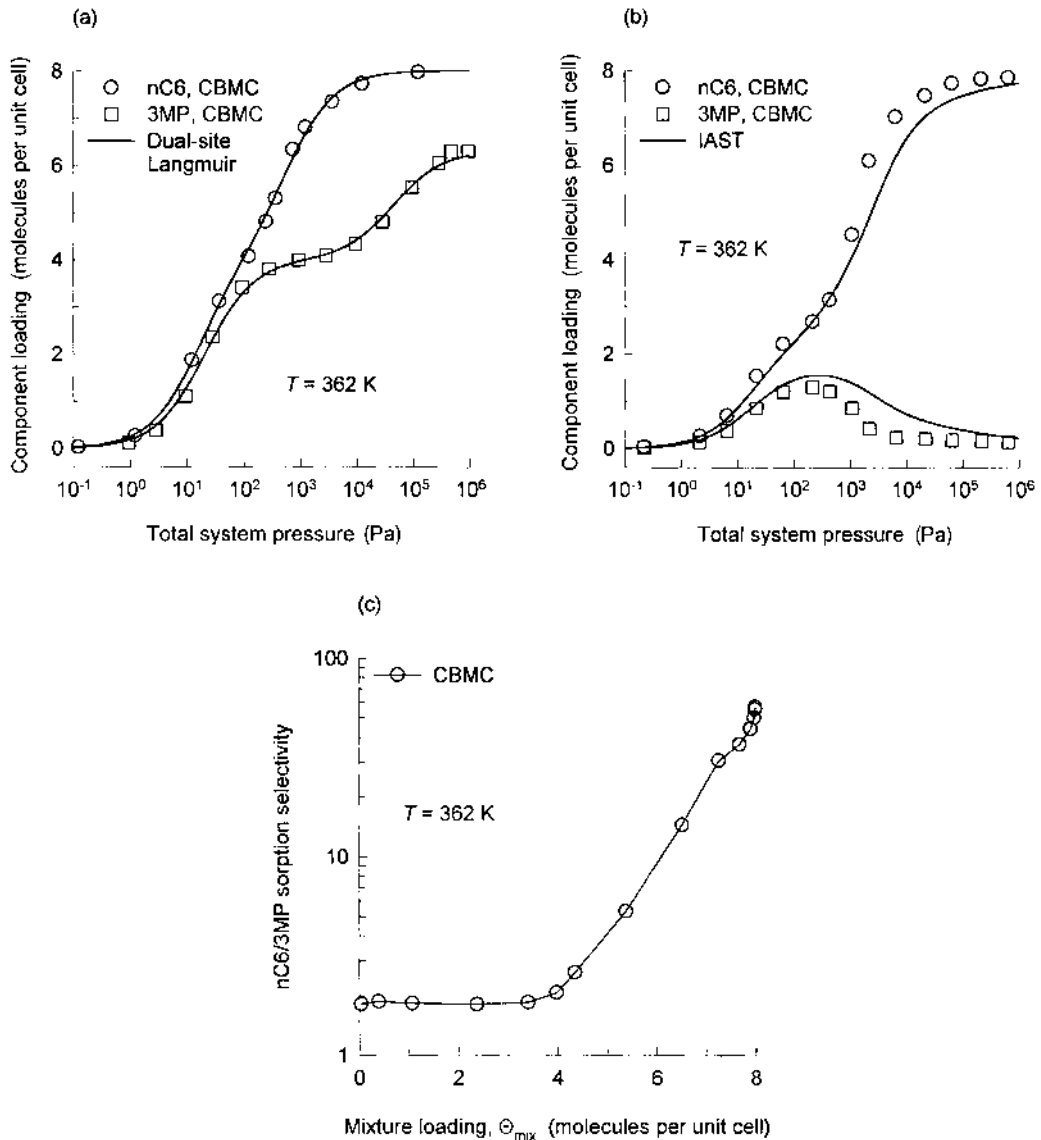
A similar picture emerges for permeation of a 50:50 mixture; see Fig. 14c and d. The respective values of the permeation selectivities are  $S_P = 197$  and 5.4. The experimental value of  $S_P$  for this 50:50 mixture determined by Bakker (26) is 60. Clearly, the interchange process within MFI matrix occurs at a finite rate. In Sec. III.A we have calculated the permeation selectivity on the basis of pure component permeation to be 0.239. Clearly, these mixture permeation selectivities cannot be predicted on the basis of pure component permeation data.

We carried out a series of simulations with varying mixture compositions in the upstream membrane compartment, keeping the total upstream pressure at 100 kPa; the calculations of  $S_P$  with the Maxwell-Stefan model, both with finite interchange, following Eq. (30), and taking  $\mathcal{D}_{ij} \rightarrow \infty$ , are shown in Fig. 15, along with the Bakker experimental data (26). The decrease in the permeation selectivity with increasing partial pressure of  $nC4$  is essentially a size entropy effect that comes into play at high occupancies and favors the smaller methane molecule. If we had used the multicomponent Langmuir isotherm, taking the saturation capacities of  $C1$  and  $nC4$  to be equal to one another, the  $S_P$  would be predicted to be independent of the upstream composition and have a constant value of 800. Kapteijn et al. (29) and Krishna and Paschek (30) have analyzed permeation data for  $C1-C2$  and  $C1-C3$  mixtures across an MFI membrane to stress the need for recognizing the size entropy effects (with the use of the IAST model). Furthermore, the results in Fig. 15 underline the importance of the interchange coefficient  $\mathcal{D}_{ij}$  in the Maxwell-Stefan formulation described by Eq. (29).

## B. Diffusion of $nC_6$ -3MP Mixture in MFI Zeolite

Before analyzing diffusion we need to understand the pure component and mixture sorption behaviors. The pure component isotherms of hexane isomers  $nC_6$  and 3MP in

MFI zeolite at 362 K, obtained from CBMC simulations, are shown in Fig. 16a. The loadings in a 50:50 mixture are shown in Fig. 16b. For  $\Theta_{\text{mix}} < 4$  the isomers have practically the same sorption strength. However, the loading of the monobranched isomer reduces to very low values when  $\Theta_{\text{mix}} > 4$ . The reason for this “exclusion” of the 3MP is because of *configurational* entropy effects which tends to favor the linear isomer. For  $\Theta_{\text{mix}} < 4$ , the 3MP molecules prefer to locate at the intersections between the straight



**Fig. 16** (a) Pure component isotherms for  $nC_6$  and 3MP in MFI at 362 K. CBMC calculations and DSL fits. (b) Component loadings for 50:50 mixture obtained from CBMC simulations compared with IAST predictions. (c)  $nC_6/3MP$  sorption selectivity as a function of the total mixture loading  $\Theta_{\text{mix}}$ .

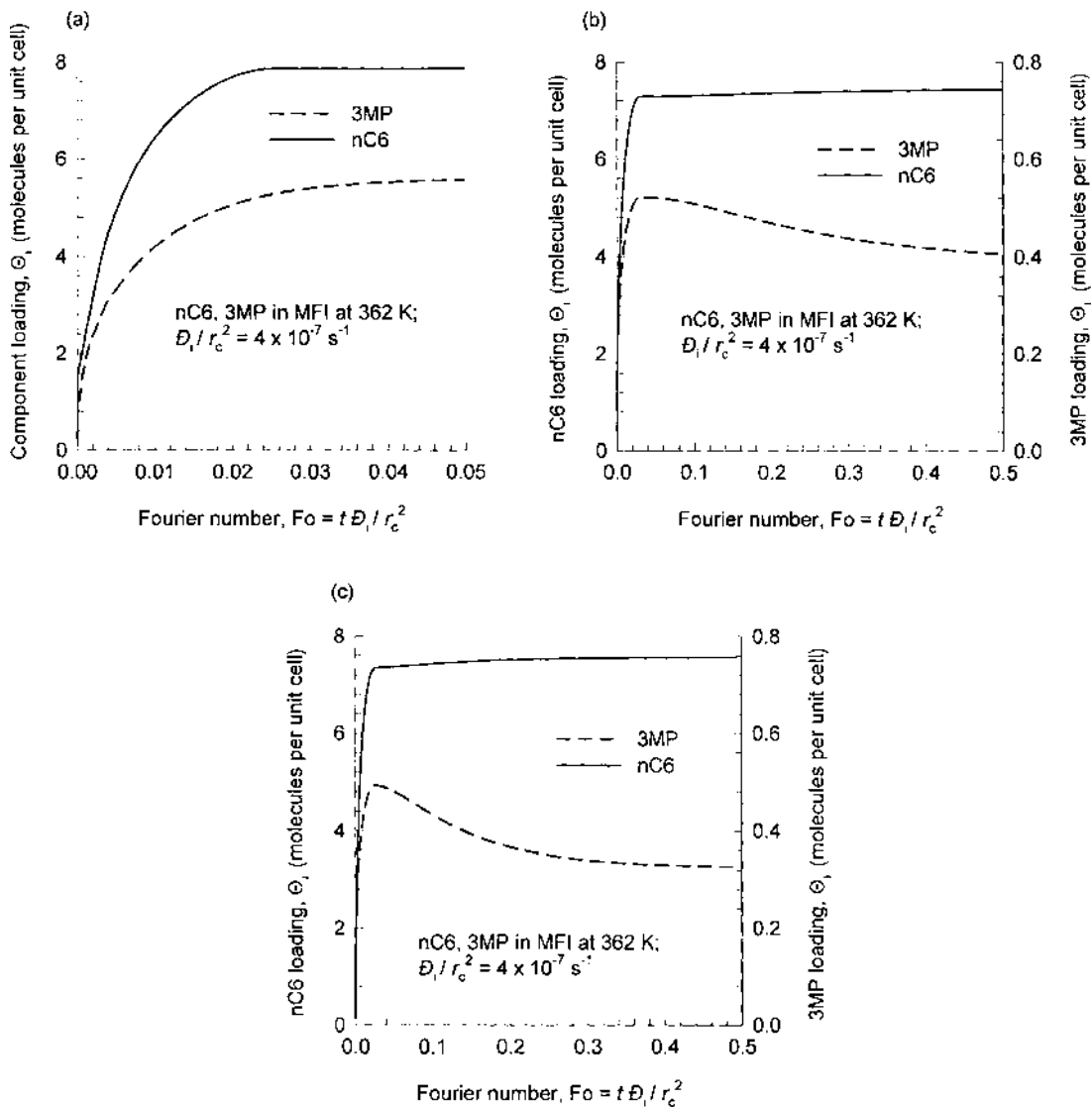
channels and zig-zag channels (16,17,30). The normal alkane can be located anywhere within the MFI matrix. At  $\Theta_{\text{mix}} = 4$ , all of the intersection sites are fully occupied. The 3MP demands an extra “push” to locate within the channel interiors (witness the inflection in the pure component isotherms in Fig. 16a). 3MP suffers a penalty from configurational entropy considerations because these molecules “pack” less efficiently within the MFI matrix; this penalty causes 3MP to be virtually excluded from the MFI matrix near saturation loadings  $\Theta_{\text{mix}} = 8$ . The sorption selectivity,  $S$ , defined by Eq. (35), is plotted in Fig. 16c as a function of the total mixture loading;  $S$  increases significantly above unity values when the mixture loading  $\Theta_{\text{mix}}$  exceeds four molecules per unit cell.

Also shown in Fig. 16b are the IAST calculations of the component loadings with the DSL parameters reported in Table 2. These are in reasonably good agreement with the CBMC simulation results; it appears that the IAST mixture rule properly accounts for configurational entropy effects. Clearly, such effects are not accounted for by the multi-component Langmuir model to predict mixture behavior.

Let us now consider the uptake of pure components  $n\text{C}_6$  (1) and 3MP (2) into a (virgin) spherical MFI crystallite at 362 K. At time  $t = 0$ , the outer surface of the crystal is exposed to a vapor phase with  $p_{1,s} = 50$  kPa;  $p_{2,s} = 50$  kPa; these partial pressures are maintained until equilibrium is achieved. The pure component uptake kinetics, obtained by solution of Eqs. (2), (3), together with (16), are shown in Fig. 17a. In these calculations the pure component Maxwell-Stefan diffusivities are taken to be equal  $\mathcal{D}_i/r_c^2 = 4 \times 10^{-7} \text{ s}^{-1}$  for either isomer. The uptake of a 50:50 mixture of  $n\text{C}_6$  (1) and 3MP (2) with  $p_{1,s} = 50$  kPa;  $p_{2,s} = 50$  kPa, calculated with the Maxwell-Stefan equations (29)–(32) are shown in Fig. 17b. The maximum in the transient loading of 3MP is noteworthy; this maximum is a direct consequence of the maximum in the mixture sorption seen in Fig. 16b. We also note that at equilibrium, the branched isomer is virtually excluded from the MFI matrix. The calculations for the uptake using the Maxwell-Stefan model with  $\mathcal{D}_{ij} \rightarrow \infty$ , Eq. (34), are shown in Fig. 17c. The results are only slightly different from the Maxwell-Stefan model including  $\mathcal{D}_{ij}$ . This result is to be expected because we had assumed the pure component Maxwell-Stefan diffusivities to be equal for either isomer; there is essentially no speeding up or slowing down of either molecule due to mobility differences. The small differences between the complete Maxwell-Stefan model, Eq. (29) and the simplified Eq. (34) are to be ascribed to the differences in the sorption loadings of  $n\text{C}_6$  and 3MP and the influences of these component loadings on the diffusion behavior via the  $[\Gamma]$  matrix.

The mixture results shown in Fig. 17 are of practical importance because they provide a means of separating the hexane isomers, relying on configurational entropy effects. The separation can be achieved by allowing equilibrium to be attained in a batch adsorber. We need to operate with ambient conditions such that mixture loading  $\Theta_{\text{mix}} > 4$ ; at 362 K this corresponds to a total system pressure in excess of 10 kPa. There is some evidence in the patent literature that the entropy concept is being used in commercial practice to separate mixtures of linear and branched alkanes (39).

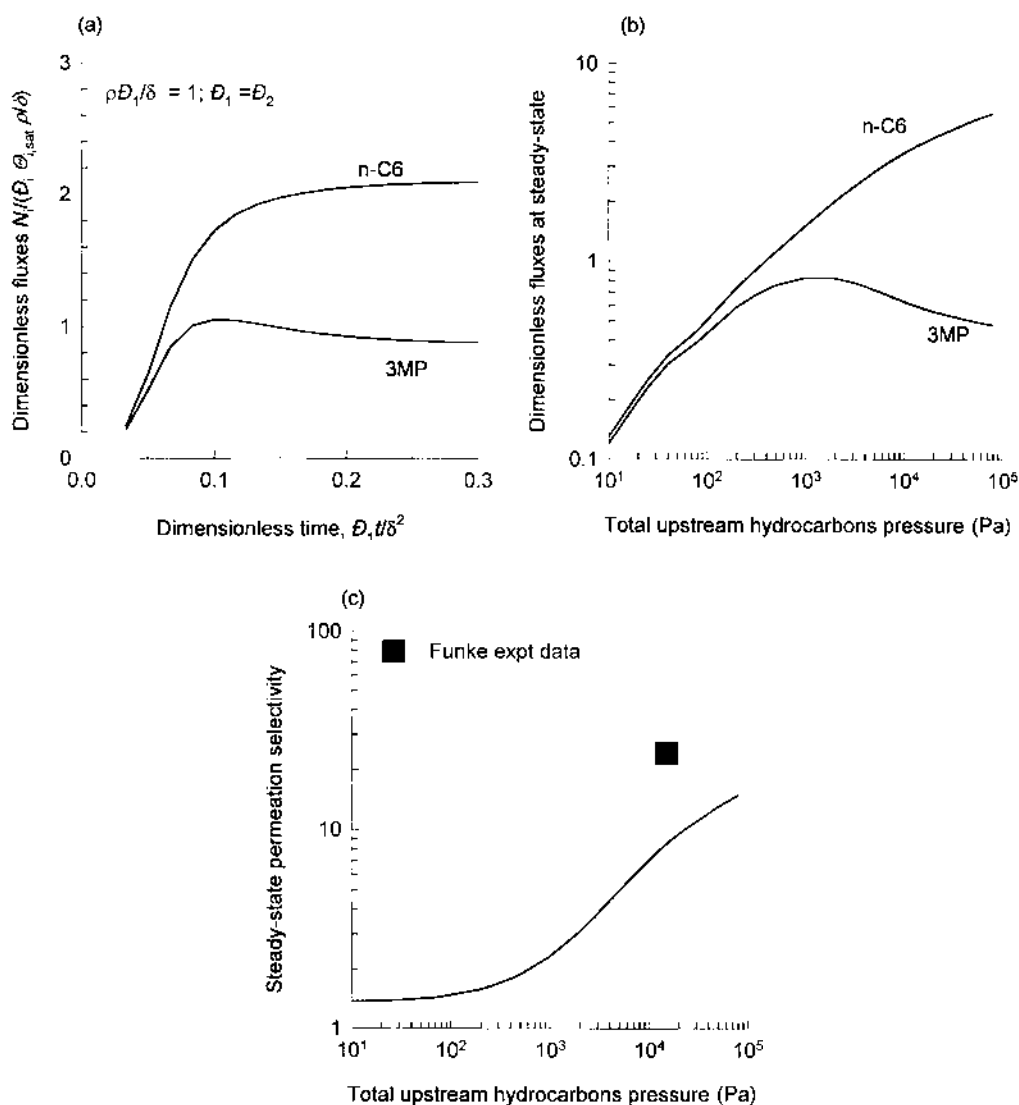
High selectivities for separation of the hexane isomers can also be achieved in a membrane permeation device. Consider permeation of an equimolar mixture of  $n\text{C}_6$  and 3MP across an MFI zeolite membrane, keeping the upstream compartments at a total pressure of 2 kPa. The (normalized) transient permeation fluxes, calculated using the Maxwell-Stefan model are shown in Fig. 18a. The steady-state permeation selectivity  $S_P$  can be calculated to be 2.5. Simulations were carried out for a range of system pressures in the upstream compartment; the results for the fluxes and selectivity are shown in Fig. 18b and c, respectively. We note that values of  $S_P$  in excess of 10 can be obtained when the pressure in the upstream compartment increases beyond 20 kPa; the results in Fig. 18c



**Fig. 17** (a) Transient uptake of pure components  $nC_6$  and 3MP within spherical crystallite of MFI at 362 K. The bulk vapor phase partial pressures are 50 kPa for either component. (b) Transient uptake of 50:50 mixture of  $nC_6$  and 3MP within spherical crystallite of MFI at 362 K. The bulk vapor phase partial pressures are 50 kPa for either component. The calculations are with the Maxwell-Stefan model with finite interchange coefficient  $\mathcal{D}_{ij}$ . (c) Uptake of 50:50 mixture calculated with the Maxwell-Stefan model with  $\mathcal{D}_{ij} \rightarrow \infty$ . The model parameters are given in Table 2.

mirror the sorption selectivity results shown in Fig. 16c. For operation with an upstream pressure of 15 kPa, Funke et al. (40) have experimentally determined a value  $S_P$  of 24; this is in reasonable agreement with our simulation results shown in Fig. 18c when we consider that our simulations were entirely based on CBMC simulations and no experimental data inputs were used. One reason for the higher  $S_P$  found experimentally is probably our assumption that the Maxwell-Stefan diffusivities are equal for either isomer. We would expect  $nC_6$  to have a higher mobility than 3MP. More important is the





**Fig. 18** (a) Transient diffusion fluxes for permeation of 50:50 mixture of  $nC_6$  and 3MP across MFI membrane at 362 K. The upstream partial pressures are  $p_{10} = 1$  kPa,  $p_{20} = 1$  kPa. (b) Steady-state permeation fluxes as a function of upstream hydrocarbons pressure. The Maxwell-Stefan diffusivities of the isomers are taken to be equal, i.e.  $D_1 = D_2$ . (c) Separation selectivities as function of upstream hydrocarbons pressure. (The experimental data point in (c) is from Ref. 40.) The model parameters are given in Table 2. The Maxwell-Stefan model takes account of finite interchange coefficient  $D_{ij}$ , using Eq. (30).

observation of Funke et al. (40) that the permeation selectivity based on pure components was found to be only 1.3. This underlines the fact that subtle configurational entropy effects are at play here.

The same entropy principle can be used to separate  $nC_6$  and 2,2-dimethylbutane (22DMB), as has been demonstrated by Gump et al. (41), and underpinned using CBMC simulations (42).

**Table 3** Pure Component Langmuir Parameters and M-S Diffusivities for N<sub>2</sub> and CH<sub>4</sub> in 4A Zeolite at 193.7 K

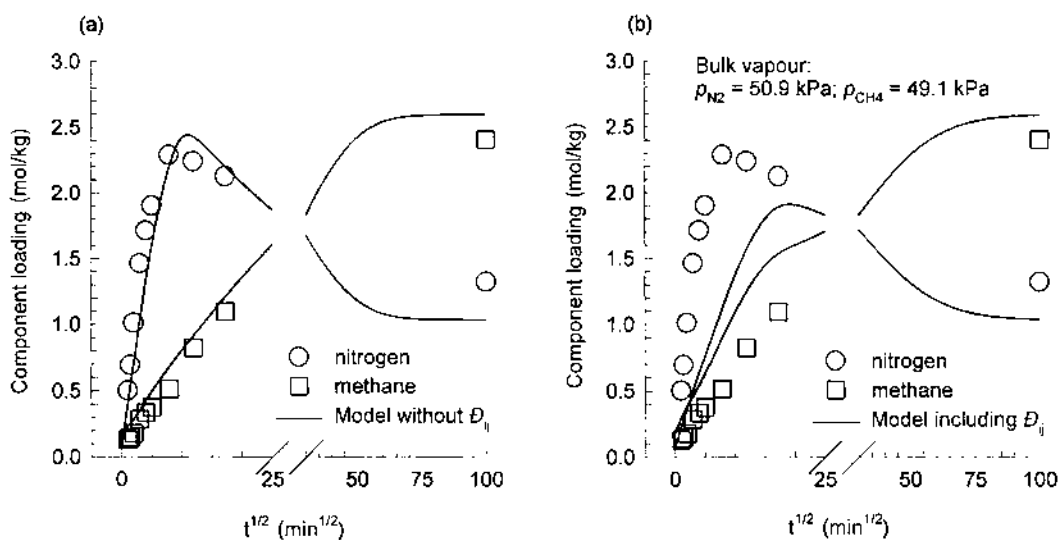
Factor	N <sub>2</sub>	CH <sub>4</sub>
Saturation loading $q_{\text{sat}}$ (mol kg <sup>-1</sup> )	3.75	3.86
Langmuir parameter, $b$ (Pa <sup>-1</sup> )	$1.08 \times 10^{-4}$	$2.56 \times 10^{-4}$
$\mathcal{D}_i$ (m <sup>2</sup> s <sup>-1</sup> )	$3.9 \times 10^{-18}$	$1.8 \times 10^{-19}$
Crystallite radius, $r_c$ (m)	$0.5 \times 10^{-6}$	

Source: Data from Ref. 36.

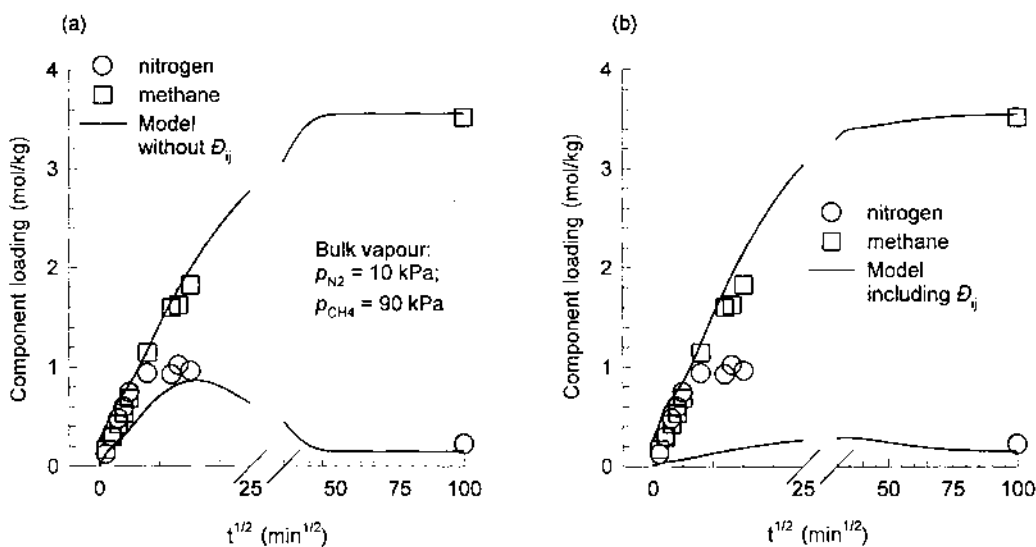
### C. Diffusion of N<sub>2</sub> and CH<sub>4</sub> in Zeolite 4A

Consider uptake of N<sub>2</sub> and CH<sub>4</sub> into a spherical zeolite 4A crystal at 193.7 K. Experimental data for pure component sorption parameters and diffusivities have been published in the classic paper by Habgood (36); these data have been reported in Table 3. The pure component characteristics are interesting because while CH<sub>4</sub> has a higher sorption strength (witness the higher Langmuir  $b$  parameter in Table 3), its diffusivity is considerably lower. The uptake characteristics into 4A zeolite from a bulk vapor mixture maintaining  $p_{\text{N}_2} = 50.9$  kPa and  $p_{\text{CH}_4} = 49.1$  kPa are shown in Fig. 19. The Habgood data (open symbols) show that N<sub>2</sub> exhibits a peak during its transience to equilibrium. The reason for this is clear; starting with (virgin) zeolite, the initial loadings are dominated by the faster diffusing N<sub>2</sub>. As time progresses, the slower diffusing but more strongly adsorbing CH<sub>4</sub> displaces N<sub>2</sub>.

Using the pure component data in Table 3, simulations for the uptake were carried out with the Maxwell-Stefan model, both with finite interchange coefficient  $\mathcal{D}_{ij}$  calculated



**Fig. 19** Transient uptake of 50.9% N<sub>2</sub>, 49.1% CH<sub>4</sub> mixture into spherical crystallite of 4A zeolite at 193.7 K. (Experimental data from Ref. 36.) Two implementations of the Maxwell-Stefan model are compared, with finite and infinite interchange coefficient  $\mathcal{D}_{ij}$ . The model parameters are given in Table 3.



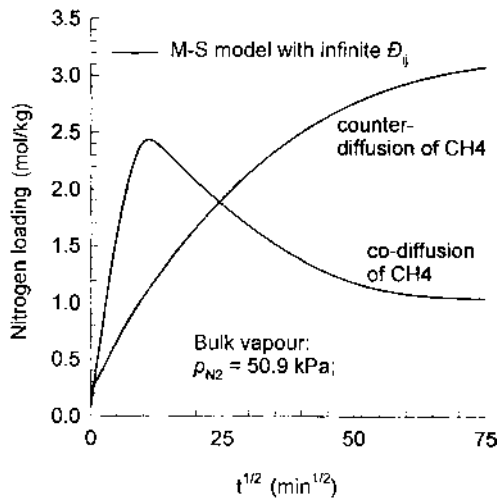
**Fig. 20** Transient uptake of 10%  $N_2$ , 90%  $CH_4$  mixture into spherical crystallite of 4A zeolite at 193.7 K. (Experimental data from Ref. 36.) Two implementations of the Maxwell-Stefan model are compared, with finite and infinite interchange coefficient  $\mathcal{D}_{ij}$ . The model parameters are given in Table 3.

using Eq. (30) (Fig. 19b) and taking  $\mathcal{D}_{ij} \rightarrow \infty$  (Fig. 19a). Both Maxwell-Stefan approaches predict a peak in the  $N_2$  flux, but the model assuming  $\mathcal{D}_{ij} \rightarrow \infty$  does a much better job of quantitatively predicting the uptake profiles, which casts some doubt on the interpolation formula in Eq. (30). Habgood has also published experimental data for uptake into 4A zeolite from a bulk vapour mixture maintaining  $p_{N_2} = 10$  kPa and  $p_{CH_4} = 90$  kPa; these data are shown in Fig. 20, along with the two implementations of the Maxwell-Stefan model. Again we note that that Maxwell-Stefan model, assuming  $\mathcal{D}_{ij} \rightarrow \infty$  does a very good job of predicting the mixture diffusion behavior. It appears that diffusion of  $N_2$  and  $CH_4$  in the zeolite 4A sample used by Habgood proceeds following Eq. (34) and is essentially free from vacancy correlation effects.

#### D. Co- vs. Counterdiffusion of $N_2$ and $CH_4$

In the transient uptake shown in Fig. 19 we had codiffusion of  $N_2$  and  $CH_4$  within 4A zeolite. Let us focus on the transient uptake of  $N_2$  and compare two situations: (a) with codiffusion of  $CH_4$  and (b) with  $CH_4$  diffusing in a direction countercurrent to  $N_2$ . The countercurrent scenario is achieved by preequilibrating the zeolite with  $CH_4$  exposing it to a bulk vapor with  $p_{CH_4} = 49.1$  kPa. This preequilibrated zeolite is then exposed to a bulk vapour with  $p_{N_2} = 50.9$  kPa. Simulations using the Maxwell-Stefan model, taking  $\mathcal{D}_{ij} \rightarrow \infty$ , are compared in Fig. 21. The two uptake characteristics are markedly different. During codiffusion,  $N_2$  and  $CH_4$  compete for sorption sites because they move in the same “direction”;  $N_2$  wins in the early stages, yields the majority of the sorption sites to  $CH_4$  eventually. There is no competition during counterdiffusion but there is cooperation,  $CH_4$  diffuses out and makes way for the incoming  $N_2$ .

The asymmetry in co- and counterdiffusion has also been verified in experimental studies (43).



**Fig. 21** Comparison of transient uptake of  $N_2$  with co- and counterdiffusion of  $CH_4$  mixture into spherical crystallite of 4A zeolite at 193.7 K. The Maxwell-Stefan model calculations assumes  $D_{ij} \rightarrow \infty$ . The model parameters are given in Table 3.

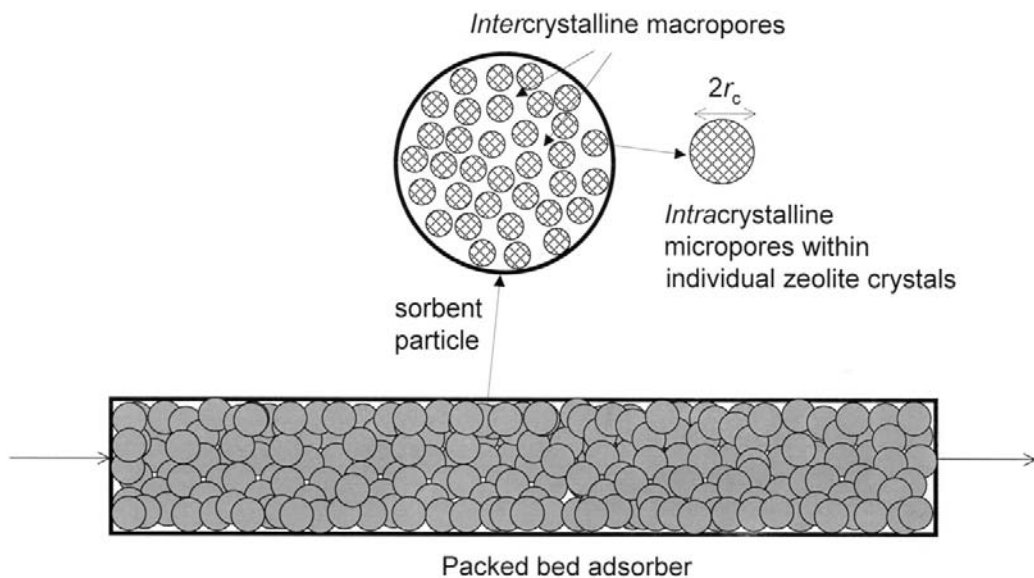
### E. Separation of $O_2$ and $N_2$ Using 4A Zeolite

The separation of  $O_2$  from  $N_2$ , can be achieved by exploiting the differences in their diffusivities in small pore 4A zeolite. The separation process is commonly carried out in a packed bed of sorbent particles (Fig. 22). We now develop a model for breakthrough in this packed bed for the case in which intracrystalline diffusion is the controlling resistance. Assuming plug flow, the concentration at any position and instant of time obtained by solving the following set of partial differential equations (details can be found in Refs. 1–6,44–46).

$$\frac{\partial c_i}{\partial t} = -\frac{\partial(uc_i)}{\partial z} - \left(\frac{1-\varepsilon}{\varepsilon}\right)\rho \frac{\partial q_i}{\partial t} \quad (40)$$

where  $c_i$  is the molar concentration in the gas phase,  $u$  is the fluid phase (absolute) velocity,  $z$  is the axial coordinate distance,  $\varepsilon$  is the bed porosity,  $\rho$  is the density of the zeolite crystals, and  $q_i$  is the average concentration within the spherical particle given by Eq. (13). Usually the LDF approximation is made in order to avoid solving the intraparticle diffusion numerically. Furthermore, published models for breakthrough in packed beds (3,45,46) almost invariably use the multicomponent Langmuir isotherm to describe mixture diffusion. Use of either LDF or the multicomponent Langmuir isotherm is not to be recommended in the general case where subtle entropy effects come into play, affecting sorption and diffusion. For accurate modeling, therefore, there is no avoiding use of the IAST and using a rigorous solution of intracrystalline diffusion.

Consider the breakthrough behavior for air (21%  $O_2$ , 79%  $N_2$ ) at a pressure of 300 kPa in a packed bed of 0.6 m length; other details and parameters are specified in



**Fig. 22** Schematic of adsorbed packed with zeolite particles.

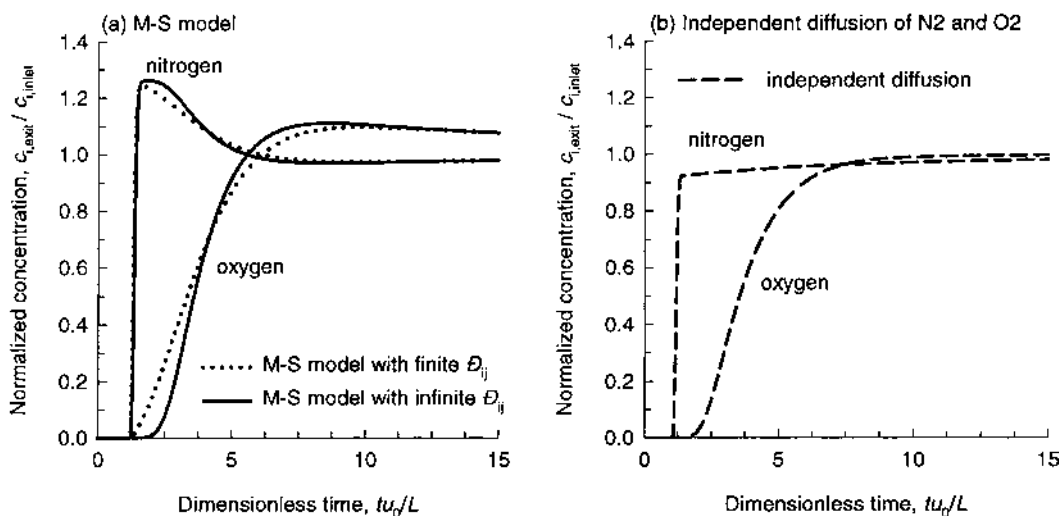
Table 4. As is conventional, we plot the normalized concentrations of the two components at the exit as a function of the dimensionless time  $tu_0/L$  where  $u_0$  is the interstitial gas velocity at the inlet to the packed bed. In Fig. 23a the breakthrough curves with the Maxwell-Stefan model, with finite and infinite interchange coefficient  $\mathcal{D}_{ij}$ , are compared. As expected, there are little differences in the two implementations of the Maxwell-Stefan model when the system approaches steady state. The major differences are during the initial transience. Finite interchange  $\mathcal{D}_{ij}$  tends to bring the breakthrough curves of  $O_2$  and  $N_2$  closer together; this is expected because of the slowing down of the

**Table 4** Pure Component Langmuir Parameters and M-S Diffusivities for  $O_2$  and  $N_2$  in 4A Zeolite at 298 K

Factor	$O_2$	$N_2$
Saturation loading $q_{\text{sat}}$ (mol $\text{kg}^{-1}$ )	9.54	6.68
Langmuir parameter, $b$ ( $\text{Pa}^{-1}$ )	$4.95 \times 10^{-8}$	$1.43 \times 10^{-7}$
$\mathcal{D}_i$ ( $\text{m}^2 \text{s}^{-1}$ )	$1.438 \times 10^{-14}$	$1.519 \times 10^{-16}$
Partial pressures at inlet to packed bed $p$ (kPa)	63.8	240.2
Packed bed voidage $\varepsilon$ (-)		0.4
Length of bed, $L$ (m)		0.6 m
Interstitial gas velocity at inlet to bed, $u_0$ (m/s)		0.0436
Crystallite radius, $r_c$ (m)		$1.3 \times 10^{-6}$

Also given are the parameters of the packed bed.

Source: Data from Refs. 45 and 46.



**Fig. 23** Breakthrough of  $N_2$  and  $O_2$  through packed bed of 4A zeolite particles. (a) Comparison of two implementations of the Maxwell-Stefan model, with finite and infinite interchange coefficient  $\mathcal{D}_{ij}$ . The model parameters are specified in Table 4. (b) Breakthrough curves calculated using an independent diffusion model.

intracrystalline diffusion of  $O_2$  and the concomitant speeding up of  $N_2$ . The slower diffusing  $N_2$  exhibits a “roll-up” in both model implementations. For zeolite 4A, the Maxwell-Stefan implementation with  $\mathcal{D}_{ij} \rightarrow \infty$  is more appropriate. If each of the components were assumed to diffuse independently, oblivious of the other component with respect to both sorption *and* diffusion, the breakthrough behavior is shown in Fig. 23b; in this case, there is no roll-up of  $N_2$ . There is evidence in the literature (47) that a proper model of multicomponent mixture diffusion is essential for the success of this technology for air separation.

## VI. CONCLUSIONS

The proper description of mixture sorption and diffusion is essential in applications of zeolites for separation and reaction. In this chapter we have focused on intracrystalline diffusion process. The following major points and conclusions emerge from the foregoing discussions.

1. Intracrystalline diffusion and sorption processes are intertwined; rigorous models are required to describe both phenomena.
2. Adsorption and desorption of single components proceed at significantly different rates.
3. The sorption isotherm of some types of molecules, e.g., branched alkanes in MFI, show inflection; this inflection behavior has a significant impact on the mixture sorption and diffusion.
4. For mixtures of molecules that differ in their saturation loadings, the proper description of the mixture isotherm requires the use of the IAST; the multicomponent Langmuir isotherm is inadequate in this case. Differences in

the saturation loadings lead to a size entropy effect that favors sorption of the smaller molecule.

5. The size entropy effect can have a significant impact on membrane permeation selectivity.
6. For sorption of mixtures of linear and branched alkanes in MFI, configurational entropy effects come into play; this causes the branched alkanes to be excluded from the zeolite. Configurational entropy effects can be exploited for achieving separations of alkane isomers.
7. The Maxwell-Stefan approach allows the prediction of mixture diffusion for some systems, on the basis of information on the pure component Maxwell-Stefan diffusivities,  $\mathcal{D}_i$ , along with the mixture isotherms (estimated using say the IAST). The interchange coefficient  $\mathcal{D}_{ij}$  accounts for the slowing down of the molecule with the higher mobility, along with the speeding up of the molecule with the lower mobility.
8. In MFI zeolite, the complete Maxwell-Stefan model, with a finite interchange coefficient  $\mathcal{D}_{ij}$  calculated from Eq. (30) is required for proper description of mixture diffusion. This has been verified both from KMC simulations (34) and by comparison with experimental data on membrane permeation (24,26).
9. Published experimental data of Habgood (36) for uptake of  $N_2$  and  $CH_4$  in zeolite 4A is better simulated by the Maxwell-Stefan model in which the interchange occurs at an infinite rate, i.e.,  $\mathcal{D}_{ij} \rightarrow \infty$ .
10. There is a need for more experimental data on mixture diffusion in various zeolite structures in order to gain a better understanding of the influence of guest topology on the mixture diffusion characteristics.
11. Co- and counterdiffusion of binary mixtures are asymmetrical phenomena.
12. Accurate prediction of breakthrough behavior in a packed bed requires proper modeling of intracrystalline mixture diffusion.

## NOMENCLATURE

$A$	surface area of adsorbent, $m^2 kg^{-1}$
$b_I$	parameter in the Langmuir adsorption isotherm, $Pa^{-1}$
$[B]$	square matrix of inverse Maxwell-Stefan coefficients, $m^{-2} s$
$c_i$	molar concentration of species $i$ , $mol m^{-3}$
$D_i$	Fick coefficient of pure component $i$ , $m^2 s^{-1}$
$D^*$	self-diffusivity, $m^2 s^{-1}$
$[D]$	matrix of Fick diffusivities, $m^2/s$
$\mathcal{D}_I$	Maxwell-Stefan diffusivity of species $i$ in zeolite, $m^2/s$
$\mathcal{D}_{ij}$	Maxwell-Stefan diffusivity describing interchange between $i$ and $j$ , $m^2/s$
$f_i$	fugacity of species $i$ ; $f_i = p_i$ for ideal gases, Pa
$F$	fractional approach to equilibrium, dimensionless
$Fo$	Fourier number, $tD/r_c^2$ , dimensionless
$k_B$	Boltzmann constant, $1.38 \times 10^{-23} J molecule^{-1} K^{-1}$
$L$	length of packed bed, m
$L_i$	Onsager coefficient of pure component $i$ , $m^2 s^{-1}$
$[L]$	matrix of Onsager coefficients, $m^2 s^{-1}$
$m$	summation parameter defined in Eq. (14), dimensionless
$N_i$	molar or molecular flux of species $i$ , $mol m^{-2} s^{-1}$ or $molecules m^{-2} s^{-1}$

$P$	system pressure, Pa
$P_i^0$	vapor pressure analog in Eq. (36), Pa
$p_i$	partial pressure of species $i$ , Pa
$q_i$	molar loading of component $i$ , mol kg <sup>-1</sup>
$q_{i,\text{sat}}$	saturation loading of component $i$ , mol kg <sup>-1</sup>
$q_j$	average loading of component $i$ , mol kg <sup>-1</sup>
$r$	radial distance coordinate, m
$r_c$	radius of spherical crystal, m
$R$	gas constant, 8.314 J mol <sup>-1</sup> K <sup>-1</sup>
$Sh$	Sherwood number, dimensionless
$S$	sorption selectivity defined by Eq. (35), dimensionless
$S_P$	permeation selectivity defined by Eq. (21), dimensionless
$t$	time, s
$T$	absolute temperature, K
$u$	abolute fluid velocity in packed bed, m s <sup>-1</sup>
$u_0$	superficial fluid velocity in packed bed, m s <sup>-1</sup>
$x_i$	mole fraction of component $i$ in the adsorbed phase, dimensionless
$y_i$	mole fraction of component $i$ in bulk vapour phase, dimensionless
$z$	distance coordinate along membrane, m

### Greek Letters

$\delta$	thickness of membrane, m
$\varepsilon$	porosity of packed bed, dimensionless
$\Gamma$	thermodynamic correction factor, dimensionless
$[\Gamma]$	matrix of thermodynamic factors, dimensionless
$\theta_i$	fractional surface occupancy of component $i$
$\Theta_i$	molecular loading, molecules per unit cell or per cage
$\Theta_{i,\text{sat}}$	saturation loading, molecules per unit cell or per cage
$\mu_i$	molar chemical potential, J mol <sup>-1</sup>
$\pi$	spreading pressure, Pa m
$\rho$	density, number of unit cells per m <sup>3</sup> or kg m <sup>-3</sup>

### Subscripts

A	referring to site A
B	referring to site B
1	component 1 in binary mixture
2	component 2 in binary mixture
mix	referring to mixture loading
sat	referring to saturation conditions
$i,j$	components in mixture
p	derivative at constant pressure
s	referring to surface of particle

### Superscripts

0	pure component parameter
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## Vector and Matrix Notation

- ( ) component vector  
[ ] square matrix

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