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Pore-Space-Partition-Enabled Exceptional Ethane Uptake and Ethane-Selective Ethane−Ethylene Separation

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ABSTRACT: An ideal material for C_2H_6/C_2H_4 separation would simultaneously have the highest C_2H_6 uptake capacity and the highest C_2H_6/C_2H_4 selectivity. But such material is elusive. A benchmark material for ethane-selective C_2H_6/C_2H_4 separation is peroxo-functionalized MOF-74-Fe that exhibits the best known separation performance due to its high C_2H_6/C_2H_4 selectivity (4.4), although its $\rm C_2H_6$ uptake capacity is moderate $(74.3\;cm^3/g).$ Here, we report a family of pore-space-partitioned crystalline porous materials (CPMs) with exceptional C_2H_6 uptake capacity and C_2H_6/C_2H_4 separation potential (i.e., C_2H_4 recovered from the mixture) despite their moderate $\rm C_2H_6/C_2H_4$ selectivity (up to 1.75). The ethane uptake capacity as high as 166.8 cm $^3/g$ at 1 atm and 298 K, more than twice that of peroxo-MOF-74-Fe, has been achieved even though the isosteric heat of adsorption (21.9−30.4 kJ/mol) for these CPMs is as low as about one-third of that for peroxo-MOF-74-Fe (66.8 kJ/mol). While the overall C_2H_6/C_2H_4 separation potentials have not yet surpassed peroxo-MOF-74-Fe, these robust CPMs exhibit outstanding properties including high thermal stability (up to 450 °C) and aqueous stability, low regeneration energy, and a high degree of chemical and geometrical tunability within the same isoreticular framework.

The separation of ethane from ethylene is a vital process in chemical industry and is also energy-intensive. 1 Compared with commonly used cryogenic distillation, adsorptive separation using porous materials would be more energy-efficient.^{[2](#page-3-0),[3](#page-3-0)} Such separation can be performed with either ethane- or ethylene-selective materials,^{[4](#page-3-0)–[8](#page-3-0)} but ethane-selective materials enjoy the simplicity of being able to produce ethylene directly at the outlet.^{[9](#page-3-0) -13 -13} This work seeks to demonstrate a high-performance platform for ethane-selective alkane/alkene separation.

A chemical separation benefits from both uptake capacity and selectivity and can be most efficiently done with a material that excels in both aspects.^{14-[19](#page-4-0)} In practice, however, the interplay between capacity and selectivity makes it hard to maximize both for the same material and a trade-off is often observed.^{[20](#page-4-0)} Regarding the aspect of C_2H_6/C_2H_4 selectivity, a breakthrough was recently reported with the synthesis of a peroxo-functionalized MOF-74-Fe $(Fe₂(O₂)$ dobdc, denoted here peroxo-MOF-74-Fe) with the highest known C_2H_6/C_2H_4 selectivity, leading to the best reported separation performance despite its relatively low ethane uptake $(74.3 \text{ cm}^3/\text{g})$.^{[12](#page-4-0),[21,22](#page-4-0)} In addition to peroxo-MOF-74-Fe, a few other materials have also been reported to have high C_2H_6/C_2H_4 selectivity (>2),^{[10,11](#page-4-0)} but for these materials, both metrics (selectivity and ethane uptake) are well below that of peroxo-MOF-74-Fe, which significantly widens the gap between their separation performance and that of peroxo-MOF-74-Fe. There are also some ethane-selective materials with C_2H_6 uptakes greater than that of peroxo-MOF-74-Fe (e.g., 116.7 $\rm cm^3/g$ for PCN-250), which helps to narrow the gap in separation potential. $23-28$ $23-28$ $23-28$

While the chemical functionalization strategy as shown by peroxo-MOF-74-Fe is effective at tuning host−guest interactions and therefore the C_2H_6/C_2H_4 selectivity, it may at times come with unintended consequences, such as decreased stability. For large-scale gas separation applications, factors such as long-term stability and adsorbent regeneration cost are also important considerations.[29](#page-4-0)[−][37](#page-4-0)

In this work, we report the first application of our pore-space-partition (PSP) strategy,^{[38](#page-4-0)} a geometry-focused method, for the C_2H_6 -selective C_2H_6/C_2H_4 separation. An outstanding feature of this PSP method is that it can dramatically increase the uptake capacity of C_2H_6 to more than 2-fold of the benchmark material peroxo-MOF-74-Fe, and yet the binding strength is just a fraction of that of peroxo-MOF-74-Fe. We show that the C_2H_6/C_2H_4 separation performance can be tuned via any one of three structural modules. Importantly, these materials are highly stable and are capable of excellent ethane-selective C_2H_6/C_2H_4 separation potential due to the ultrahigh C_2H_6 uptake capacity.

The family of materials is built via introduction of a porepartitioning agent into the hexagonal channel of the MIL-88/ MOF-235-type (the acs net) framework,^{[39](#page-4-0)–[41](#page-4-0)} resulting in the partitioned acs net known as the *pacs* net.^{42,[43](#page-4-0)} They have a general framework formula of $[(M1₂M2)(O/OH)L1₃]L2$, where M1 and M2 are the metals in the trimer, L1 is the dicarboxylate ligand for the formation of the acs framework, and L2 is the pore-partitioning agent. All the structural

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modules (i.e., L1, L2, and metal trimers) are tunable with many possibilities. In this work, we have examined the effects of all three modules. The variation in each module is chosen as follows: (Module 1) dicarboxylate ligands - bdc, dmbdc, and ndc; (Module 2) pore-partitioning agent - tpbz, tppy, and tpt; and (Module 3) metal trimers - Co_2V , Co_2Ti , Mg_2V , and Mg_2Ti (Figure 1). While there are 36 permutations by

Figure 1. Three modules of pacs MOFs studied in this work. Three kinds of Ligand Type 1, three kinds of Ligand Type 2 and four kinds of metal trimers are used (BDC = terephthalate; DMBDC = 2,5 dimethylterephthalate; NDC = 1,4-naphthalenedicarboxylate; TPBz = 1,3,5-tri(4-pyridyl)-benzene, TPPy = 2,4,6-tris(4-pyridyl)pyridine, TPT = 2,4,6-tri(4-pyridyl)-1,3,5-triazine).

combining these modules, we are able to establish general trends in ethane-selective C_2H_6/C_2H_4 separation properties with nine combinations reported here.

The metal combinations such as Co−V trimers in this study are unusual among MOFs. The heterometallic compositions were established by EDS analysis ([Figure S2\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf) and supported by single-crystal X-ray diffraction analysis. It is worth noting that while it is typically difficult to grow large single crystals of vanadium and titanium MOFs for use with a conventional Xray source, crystals around 50 μ m in size can be made readily in this work, $44-47$ $44-47$ $44-47$ and seven structures reported here were analyzed with in-house single-crystal diffraction data.

The materials in this work are highly stable. They are thermally stable up to 450 °C [\(Figures S3](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf)−S4). All of them maintain their structural integrity after repeated gas adsorption−desorption experiments [\(Figure S5\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf). As shown by [Figure](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf) [S6](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf), $Co₂Ti-bdc-tpt$ and $Co₂V-bdc-tpt$ are hydrothermally stable and could retain their high crystallinity even after immersion in water for 24 h.

In addition to partitioning the pore space, pore-partitioning agents deactivate all open-metal sites (OMS) of the parent 6 connected acs framework. While OMSs can be beneficial for ethylene-selective C_2H_4/C_2H_6 C_2H_4/C_2H_6 C_2H_4/C_2H_6 separation,⁴ the total annihilation of OMS in this family of materials could be helpful for ethane-selective C_2H_4/C_2H_6 separation. In fact, all the MOFs in this work show ethane-selective C_2H_4/C_2H_6 separation according to the single-component adsorption isotherms ([Figures S7](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf)−S15). Moreover, the ethane uptakes of these MOFs are remarkably high. Six of nine MOFs have the uptake capacity from 154.2 to 166.8 $\rm cm^3/g$ at 298 K and 1 atm ([Table](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf) $S2$), all of which by far exceed 116.7 cm³/g for PCN-250, a benchmark material for ethane uptake among ethane-selective MOFs. This corresponds to about 6–7 C_2H_6 molecules per formula unit. In particular, the uptake by Mg_2V -bdc-tpt is

166.8 cm³/g (7.45 mmol/g), which is likely the highest among MOFs reported to date (Figure 2 and [Tables S3](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf)−S4).

Figure 2. Experimental C_2H_6 and C_2H_4 adsorption isotherms of Mg_2V -bdc-tpt and Co_2V -bdc-tpt at 298 K.

The high ethane uptakes are accomplished with low adsorption enthalpies ranging from 21.9 to 30.4 kJ/mol at zero coverage [\(Figure S16\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf). In contrast, $Fe₂(O₂)(dobdc)$ has a high ethane adsorption enthalpy (66.8 kJ/mol).^{[12](#page-4-0)} The low Q_{st} of CPMs could be advantageous for adsorbent regeneration due to the reduced energy consumption. The low adsorption enthalpy was further validated by GCMC simulations. The density distribution of C_2H_6 molecules within Co_2V -bdc-tpt was analyzed at different pressures ([Figure S17\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf). No strong adsorption sites were observed on the framework. Also, the density distribution of C_2H_6 molecules was found to be quite dispersive in all the pore space, indicative of the widespread weak adsorption sites that are responsible for the ultrahigh C_2H_6 uptake.

The IAST selectivity has been calculated to evaluate the separation performance [\(Figure S18](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf)). The best selectivity is 1.75 for Co ₂V-bdc-tpt, which is comparable to ZIF-7 (1.5), IRMOF-8 (1.6), PCN-250 (1.9), and MUF-15 (1.96), but significantly lower than $Fe₂(O₂)(dobdc)$ (4.4), Cu(Qc)₂ (3.4), and MAF-49 (2.7) ([Table S3](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf)).

In addition to uptake capacity and selectivity, separation potential, which is a metric incorporating the influence of both factors, is also used to evaluate the separation performance. $48,49$ It represents the maximum amount of pure C_2H_4 that could be recovered from the mixture in a fixed bed adsorber; the separation potentials are calculable from IAST using [eq S3](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf) in the SI. Separation potentials, together with the uptake capacity and selectivity of nine materials, are discussed below when we systematically vary one structural module while keeping two other modules unchanged.

As shown in [Figure 3a](#page-2-0), the BDC ligand was determined to have the best separation potential. We explored the possibility of boosting C_2H_6 uptake by incorporating nonpolar groups which was considered helpful in some structure types studied previously.^{[50](#page-5-0),[51](#page-5-0)} It was observed here that nonpolar functionalization on dicarboxylate ligands does not improve separation performance. An increase in C_2H_6 adsorption enthalpy was indeed observed with the installation of nonpolar groups in the $Co₂V$ compositions from 23.4 for $Co₂V$ -bdc-tpt to 27.9 for $Co₂V$ -dmbdc-tpt and to 30.4 kJ/mol for $Co₂V$ -ndc-tpt [\(Figure](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf) [S15\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf). However, the extra group also led to a significant decrease in surface area (Figure $S19$), which likely resulted in a significant decrease in C_2H_6 uptake capacity. The BET surface

Figure 3. Comparisons of separation potential for C_2H_6/C_2H_4 (50/ 50) mixture: (a) $Co₂V-L1$ -tpt with different dicarboxylate ligands; (b) $Mg_2Ti-bdc-L2$ with different pore-partitioning agents; (c) $M_3-bdc-tpt$ with different metal trimers.

areas decreased from 1328.5 m^2/g for Co₂V-bdc-tpt to 1161.5 m^2/g for Co₂V-dmbdc-tpt and to 472.5 m^2/g for Co₂V-ndctpt. Following the same trend, the C_2H_6 uptake decreased from 159.6 cm³/g for Co₂V-bdc-tpt to 104.6 cm³/g for Co₂Vdmbdc-tpt and to 90.2 cm^3/g for Co₂V-ndc-tpt ([Table S2](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf)). Furthermore, the Co₂V-bdc-tpt also has the highest C_2H_6 / C_2H_4 selectivity among all the phases reported here [\(Figure](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf) [S18\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf). For the Mg_2V system, Mg_2V -bdc-tpt also outperforms Mg2V-dmbdc-tpt in terms of uptake capacity and selectivity ([Figure S21\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf).

In comparison with the module 1 (dicarboxylates) that exhibits a large impact on C_2H_6/C_2H_4 uptake capacity, the module 2 (pore-partitioning agent) exerts a significant impact on C_2H_6/C_2H_4 selectivity. Overall, the separation potential for

this family of MOFs with a different pore-partitioning agent follows the order of tpt $>$ tpbz $>$ tppy (Figure 3b). These materials ($Mg_2Ti-bdc-tpbz$, $Mg_2Ti-bdc-tppy$, and $Mg_2Ti-bdc$ tpt) have similar surface areas and ethane uptakes, but their selectivity differs obviously. The lowest selectivity for ethane was found in $Mg_2Ti-bdc-tppy$ [\(Figure S18\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf), which led to its lowest separation potential. This may be due to the more basic feature of the pyridine core in the partitioning agent (as compared to the cores in tpt and tpbz), which enables a relatively stronger interaction with ethylene.

With bdc and tpt as the best choices for C_2H_6/C_2H_4 separation, the effects of metal trimers on the separation potential were subsequently evaluated. The order was found to be $Co_2V > Mg_2V > Mg_2Ti > Co_2Ti$ (Figure 3c). The best performance of $Co₂V$ -bdc-tpt among this family is due to the combined effect from uptake capacity and selectivity ([Table](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf) [S2](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf)). Also, vanadium MOFs show higher separation performance than titanium MOFs with the same $M1$ (Co or Mg).

 $Co₂V$ -bdc-tpt and $Mg₂V$ -bdc-tpt could produce 1.88 and 1.73 mmol/g C_2H_4 from the C_2H_6/C_2H_4 (50/50) mixture, respectively. These values are lower than that of $Fe₂O₂(dobdc)$ (1.93), but higher than those of other prominent MOFs (Figure 4a). Other materials reported here such as $Mg_2Ti-bdc$ tpt and $Co₂Ti-bdc-tpt$ also show high separation potentials.

To validate the excellent separation performance, transient breakthrough simulations were also performed using $Co₂V-$

Figure 4. (a) Separation potential (calculated using eq S3 of the SI) versus single-component ethane uptake for the select high-performance ethane-selective materials reported to date (CPM-733: $Co₂V$ bdc-tpt; CPM-233: Mg₂V-bdc-tpt; CPM-223: Mg₂Ti-bdc-tpt; CPM-723: $Co₂Ti-bdc-tpt$). (b) Simulated breakthrough curve for CPM-733. Binary equimolar C_2H_6/C_2H_4 mixtures were used in all calculations and simulations.

bdc-tpt and Mg₂V-bdc-tpt. C_2H_6/C_2H_4 mixtures with two ratios (50/50 and 10/90) were used as feeding gases in the simulation to mimic the industrial separation process. The results clearly show that both materials are capable of separating these two gases. In the simulated breakthrough curves, the C_2H_4 breakthrough occurred first and subsequently reached a plateau, which could thereby produce polymer-grade C_2H_4 before the C_2H_6 breakthrough occurred ([Figure 4](#page-2-0)b). The productivities were also calculated based on the breakthrough curves. For the $50/50 \text{ C}_2H_6/C_2H_4$ mixture, $Co₂V$ -bdc-tpt and Mg₂V-bdc-tpt can produce 0.88 and 0.60 mmol/g C_2H_4 (purity >99.95%), respectively. These values are higher than some benchmark MOFs including MAF-49 (0.52 mmol/g), and $Cu(Qc)$ ₂ (0.49 mmol/g). Similarly, excellent separation performances were also found in $Co₂V$ -bdc-tpt and Mg_2V -bdc-tpt with the 10/90 C_2H_6/C_2H_4 mixture [\(Figure](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf) [S22\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf).

In conclusion, we have made nine heterometallic vanadium and titanium MOFs and systematically investigated the effects of three separate modules on the C_2H_6/C_2H_4 separation performance. Compared with $Fe₂O₂(dobdc)$ with very high selectivity, this family of materials have exceptional high uptake and provide an alternative way to achieve excellent separation performance. Some advantages of this family of materials include high stability, easy adsorbent regeneration, and broad chemical tunability. The correlations between the structure and the C_2H_6/C_2H_4 separation performance revealed in this work could be useful for designing novel high-performance C_2H_6 selective MOFs. Finally, novel metal combinations in this work highlight new possibilities for the construction of trimer-based MOFs.

■ ASSOCIATED CONTENT

³ Supporting Information

The Supporting Information is available free of charge at [https://pubs.acs.org/doi/10.1021/jacs.9b12924.](https://pubs.acs.org/doi/10.1021/jacs.9b12924?goto=supporting-info)

Experimental Procedures and compound characterization data ([PDF\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_001.pdf)

Crystallographic data with CCDC Nos. 1967756− 1967761 [\(CIF](http://pubs.acs.org/doi/suppl/10.1021/jacs.9b12924/suppl_file/ja9b12924_si_002.cif))

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Notes

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Pore-Space-Partition-Enabled Exceptional Ethane **Uptake and Ethane-Selective Ethane-Ethylene** Separation

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EXPERIMENTAL SECTION

Synthesis of [Co2TiO(bdc)3tpt] (Co2Ti-bdc-tpt or CPM-723, CCDC No. 1967757). In a 15 mL glass vial, 52 mg of cobalt chloride (CoCl₂, \sim 0.4 mmol), 50 mg titanocene dichloride (Cp2TiCl2, ~0.2 mmol) 100 mg 1,4-benzenedicarboxylic acid (H2BDC, \sim 0.4 mmol), and 62 mg 2,4,6-tri(4-pyridyl)-1,3,5-triazine (tpt, \sim 0.2 mmol) were dissolved in 3.0 g dimethylformamide (DMF). After being stirred for an hour, the vial was placed in a 130 ℃ oven for 3 days, and the mixture was then cooled to room temperature. Large red polyhedral crystals with small amount of colorless recrystallized organic ligands were obtained. The organic ligands can be removed by being refluxed in dichloromethane for around 12 hours. The phase purity was supported by powder X-ray diffraction.

Synthesis of [Co2V(OH)(bdc)3tpt] (Co2V-bdc-tpt or CPM-733, CCDC No. 1967755). In a 15 mL glass vial, 120 mg of cobalt nitrate hexahydrate (Co(NO3)2·6H2O, ~0.4 mmol), 32 mg vanadium (Ⅲ) chloride (VCl3, ~0.2 mmol), 100 mg 1,4-benzenedicarboxylic acid $(H_2BDC, \sim 0.6 \text{ mmol})$, and 62 mg 2,4,6-tri(4-pyridyl)-1,3,5-triazine (tpt, \sim 0.2 mmol) were dissolved in 3.0 g dimethylformamide (DMF). After being stirred for an hour, the vial was placed in a 130 ℃ oven for 3 days, and the mixture was then cooled to room temperature. Dark-red spindle-shaped crystals were obtained. The phase purity was supported by powder X-ray diffraction.

Synthesis of $[Co_2V(OH)(1,4-ndc)$ *stpt*] (Co_2V -ndc-tpt or CPM-736). In a 15 mL glass vial, 52 mg of cobalt chloride ($CoCl₂$, ~0.4 mmol), 32 mg vanadium (III) chloride (VCl3, ~0.2 mmol), 130 mg 1,4-naphthalenedicarboxylic acid (1,4-H2NDC, ~ 0.6 mmol), and 62 mg 2,4,6-tri(4-pyridyl)-1,3,5-triazine (tpt, ~ 0.2 mmol) were dissolved in 3.0 g dimethylformamide (DMF). After being stirred for an hour, the vial was placed in a 130 ℃ oven for 3 days, and the mixture was then cooled to room temperature. Microcrystalline dark-red spindle-shaped crystals were obtained. The phase purity was supported by powder X-ray diffraction.

Synthesis of $[Co_2V(OH)(dmbdc)$ ₃**tpt**] $(Co_2V\text{-}dmbdc\text{-}tpt$ or CPM-738). In a 15 mL glass vial, cobalt nitrate hexahydrate $(Co(NO₃)₂·6H₂O, ~0.4 mmol)$, 32 mg vanadium (Ⅲ) chloride (VCl3, ~0.2 mmol), 117 mg 2,5-dimethylterephthalic acid (H2DMBDC, ~0.6 mmol), and 62 mg 2,4,6-tri(4-pyridyl)-1,3,5-triazine (tpt, \sim 0.2 mmol) were dissolved in 3.0 g dimethylformamide (DMF). After being stirred for an hour, the vial was placed in a 130 ℃ oven for 3 days, and the mixture was then cooled to room temperature. Microcrystalline dark-red spindle-shaped crystals were obtained. The phase purity was supported by powder X-ray diffraction.

Synthesis of [Mg2TiO(bdc)3tpbz] (Mg2Ti-bdc-tpbz or CPM-223-tpbz, CCDC No. 1967758). In a 15 mL glass vial, 30 mg of magnesium chloride hexahydrate (MgCl₂·6H₂O, ~0.15 mmol), 37 mg titanocene dichloride (Cp₂TiCl₂, ~0.15 mmol), 50 mg 1,4-benzenedicarboxylic acid (H2BDC, ~0.3 mmol), and 31 mg 2,4,6-tri(4-pyridyl)benzene (tpbz, 0.1 mmol) were dissolved in 3.0 g dimethylformamide (DMF). After being stirred for an hour, the vial was placed in a

120 ℃ oven for 4 days, and the mixture was then cooled to room temperature. Large grey truncated hexagonal prism crystals were obtained. The phase purity was supported by powder X-ray diffraction.

Synthesis of [Mg2TiO(bdc)3tppy] (Mg2Ti-bdc-tppy or CPM-223-tppy, CCDC No. 1967759). In a 15 mL glass vial, 30 mg of magnesium chloride hexahydrate (MgCl₂·6H₂O, ~0.15 mmol), 37 mg titanocene dichloride (Cp₂TiCl₂, ~0.15 mmol), 50 mg 1,4-benzenedicarboxylic acid (H2BDC, ~0.3 mmol), and 31 mg 2,4,6-tri(4-pyridyl)pyridine (tpp, 0.1 mmol) were dissolved in 3.0 g dimethylformamide (DMF). After being stirred for an hour, the vial was placed in a 120 ℃ oven for 4 days, and the mixture was then cooled to room temperature. Large grey truncated hexagonal prism crystals were obtained. The phase purity was supported by powder X-ray diffraction.

Synthesis of [Mg2TiO(bdc)3tpt] (Mg2Ti-bdc-tpt or CPM-223, CCDC No. 1967756). The microcrystalline sample of CPM-223 was originally reported in Ref. ^{[S1](#page-57-0)}. Here, with the following modified procedure, we could get large-sized crystals which are suitable for single crystal X-ray diffraction. In a 15 mL glass vial, 30 mg of magnesium chloride hexahydrate (MgCl2·6H2O, ~0.15 mmol), 37 mg titanocene dichloride (Cp₂TiCl₂, ~ 0.15 mmol), 75 mg 1,4-benzenedicarboxylic acid (H₂BDC, ~ 0.3 mmol), and 31 mg 2,4,6-tri(4-pyridyl)-1,3,5-triazine (tpt, ~ 0.1 mmol) were dissolved in 3.0 g dimethylformamide (DMF). After being stirred for an hour, the vial was placed in a 120 ℃ oven for 4 days, and the mixture was then cooled to room

temperature. Large grey truncated hexagonal prism crystals were obtained. The phase purity was supported by powder X-ray diffraction.

Synthesis of [Mg2V(OH)(bdc)3tpt] (Mg2V-bdc-tpt or CPM-233, CCDC No. 1967761). The microcrystalline sample of CPM-233 was originally reported in ref. ^{[S1](#page-57-0)}. Here, with the following modified procedure, we could get large-sized crystals which are suitable for single crystal X-ray diffraction. In a 15 mL glass vial, 40 mg of magnesium chloride hexahydrate (MgCl2·6H2O, ~0.2 mmol), 16 mg vanadium chloride (VCl₃, \sim 0.1 mmol), 50 mg 1,4-benzenedicarboxylic acid (H₂BDC, \sim 0.3 mmol), and 31 mg 2,4,6-tri(4-pyridyl)-1,3,5-triazine (tpt, ~0.1 mmol) were dissolved in 3.0 g dimethylformamide (DMF). 1 drop of concentrated hydrochloric acid (around 25 μL) was subsequently added. After being stirred for an hour, the vial was placed in a 120 ℃ oven for 4 days, and the mixture was then cooled to room temperature. Pink block crystals were obtained. The phase purity was supported by powder X-ray diffraction.

Synthesis of [Mg2V(OH)(dmbdc)3tpt] (Mg2V-dmbdc-tpt or CPM-238, CCDC No. 1967760). In a 15 mL glass vial, 40 mg of magnesium chloride hexahydrate (MgCl₂·6H₂O, ~0.2 mmol), 16 mg vanadium chloride (VCl₃, ~0.1 mmol), 58 mg 2,5-dimethylterephthalic acid $(H_2DMBDC, \sim 0.3$ mmol), and 31 mg 2,4,6-tri(4-pyridyl)-1,3,5-triazine (tpt, \sim 0.1 mmol) were dissolved in 3.0 g dimethylformamide (DMF). 1 drop of concentrated hydrochloric acid (around 25 μL) was subsequently added. After being stirred for an hour, the vial was placed in a 120 ℃ oven for 4 days, and the mixture was then cooled to room temperature. Pink block crystals were obtained. The phase purity was supported by powder X-ray diffraction.

Single-Crystal X-ray Diffraction Characterization. The single-crystal X-ray diffraction measurements were performed on a Bruker diffractometer using graphite-monochromated MoK α (λ = 0.71073 Å) radiation at room temperature. Diffraction data were integrated and scaled by 'multi-scan' method with the Bruker APEX software. The structure was solved by intrinsic phasing which was embedded in 'APEX Ⅲ' software and the refinement against all reflections of the compound was performed using 'APEX Ⅲ'. All non-hydrogen framework atoms were refined anisotropically. All the hydrogen atoms were calculated. CCDC 1967756-1967761 contain the supplementary crystallographic data for this paper. These data are provided free of charge by the Cambridge Crystallographic Data Centre.

Powder X-ray Diffraction (PXRD) Characterization. Powder X-ray diffraction experiments were performed on a PANalytical X'Pert Pro MPD diffractometer, equipped with a linear X'Celerator detector, which was operating at 40 kV and 35 mA (Cu K α radiation, $\lambda = 1.5418$ Å). The data collection was performed at room temperature in the range from 5° to 40° with a step size of $\sim 0.008^{\circ}$. The simulated powder pattern was obtained from the single crystal data.

Thermogravimetric (TG) Measurement. A TA Instruments TGA Q500 thermal analyzer was used to measure the TG curve by heating the sample from 30 ℃ to 800 ℃ with heating rate of 5℃/min under nitrogen flow. The flow rate of the nitrogen gas was controlled at about 60 milliliters per minute.

Water Stability Tests for CPM-723 (Co2Ti-bdc-tpt) and CPM-733 (Co2Ti-bdc-tpt). In one batch**,** around 10 mg of samples were immersed in 10 mL of water at room temperature. After 24 hours, the samples were washed by methanol and dried in 80 ℃ oven for PXRD experiments.

Thermal Stability Tests for CPM-223. Around 10 mg of samples were heated from room temperature to different temperatures with a heating rate of 10℃/min under nitrogen flow using TGA Q500 instrument. The samples were kept at targeted temperature for one hour and then were allowed to be cooled to room temperature naturally. The processed samples were then subject to PXRD experiments.

Energy dispersive spectroscopy (EDS). The semi-quantitative elemental analyses of different heterometallic MOF samples were performed by using a FEI NNS450 field emission scanning electron microscope equipped with 50 mm² X-Max50 SDD energy dispersive spectroscopy (EDS) detector. Data acquisition was performed with an accelerating voltage of 15 kV and 20 s accumulation time.

X-ray Photoelectron Spectroscopy (XPS). XPS characterization was carried out by using a Kratos AXIS ULTRADLD XPS system equipped with an Al Kα monochromated X-ray source and a 165-mm mean radius electron energy hemispherical analyzer.

Gas Sorption Measurement. Gas sorption measurements were carried out on a Micromeritics ASAP 2020 and ASAP 2020 PLUS Physisorption Analyzers. Prior to the measurement, the as-synthesized sample was purified by DMF and immersed in dichloromethane (extra dry) for three days. During each day, the solution was refreshed. After solvent exchange, the upper solvent was decanted. The sample was first dried under air flow gently and was subsequently transferred into the test tube. The degas process was performed under room temperature for 2 hours and further dried at 80 \degree C for another 10 hours. Notably, longer exposure time to air will reduce the gas uptake, especially for magnesium crystals. The BET surface area of all the samples was calculated from the N_2 adsorption isotherm with the pressure range of $P/P_0 < 0.1$ due to the microporous pore.

Isosteric Heat of Adsorption (Q_{st}) **.** The isosteric heats of adsorption for all the gases were calculated using the isotherms at 273 K and 298 K, following the Clausius-Clapeyron equation. It was done with the calculation program embedded in the software of ASAP 2020 plus. High accuracy of the Q_{st} was found in all the calculations as evidenced by the linearity in the isosters.

Selectivity by IAST. To evaluate the C_2H_6/C_2H_4 separation performance, the selectivity was calculated by ideal adsorbed solution theory (IAST). Single-site Langmuir-Freundlich (SLF) model was employed to fit the gas adsorption isotherms over the entire pressure range. SLF model can be written as:

$$
N = \frac{N_{sat}bp^{1/n}}{1 + bp^{1/n}}
$$
 (1)

Where *N* is the quantity adsorbed, *p* is the pressure of bulk gas at equilibrium with adsorbed phase, *Nsat* is the saturation loadings for adsorption site, and *b* are the affinity parameters. 1/n is the index of heterogeneity. The R factors for all the fitting are higher than 99.9%.

The detailed methodology for calculating the amount of A and B adsorption from a mixture by IAST is described elsewhere.⁵² The adsorption selectivity is finally defined as:

$$
selectivity = \frac{q_A}{p_A} \frac{q_B}{p_B} \tag{2}
$$

where q_i ($i = A$ or B) is the uptake quantity in the mixture and p_i is the feeding partial pressure of component i.

Separation Potential.^{[S3](#page-57-2)} The separation potential (ΔQ) is a combined metric, which considering both uptake capacity and selectivity. It is defined to quantify mixture separations in fixed bed adsorbers. For a C_2H_6/C_2H_4 mixture with mole fractions y_{C2H6} , and $yC2H4=1-yC2H6$, the gravimetric separation potential ΔQ , is calculated from IAST using the formula

$$
\Delta Q = q_{C2H6} \frac{y_{C2H4}}{1 - y_{C2H4}} - q_{C2H4}
$$
\n(3)

where q_{C2H6} and q_{C2H4} are C_2H_6 and C_2H_4 uptake in the mixture, respectively, which are calculated based on IAST theory. For 50/50 mixture, the formula (3) can be simplified as

$$
\Delta Q = q_{C2H6} - q_{C2H4} \tag{4}
$$

The physical significance of ΔQ is that it represents the maximum amount of pure C2H⁴ that can be recovered during the adsorption phase of fixed bed separations.

GCMC Simulation. The GCMC calculations were performed by SORPTION code embedded in the Material Studio (MS) software. Periodic boundary conditions were applied in three dimensions. In addition, the force field of the condensed phase optimized molecular potential for atomistic simulation studies (COMPASS), was used to describe the interatomic interaction. Specifically, the van der Waals interactions with a cutoff of 12 Å were depicted by the Lennard−Jones potential and the electrostatic interactions were described via the Ewald summation method.

DFT calculation. The initial positions of the gas molecules were first estimated by GCMC simulation. The cluster model was cleaved from the unit cell of Co2V-bdc-tpt. The dangling bonds were saturated by H. The DFT (DFT-D2) calculations were performed to describe the interaction between the clusters and gas molecules (ethylene or ethane) using DMol³module that was implemented in Materials Studio.[S4](#page-57-3)[,5](#page-57-4) The Perdew–Burke–Ernzerhof (PBE) exchange-correlation potential combined with the double numerical basis set containing polarization function (DNP) was employed in the calculations. Core electrons were used to set the type of core treatment. The convergence threshold parameters for the optimization were 10⁻⁵ Ha (energy), 2 × 10⁻³ Ha/Å (gradient), and 5×10^{-3} **Å (displacement), respectively.**

The binding energy (BE) between the gas molecules and MOFs was calculated as follows:

$$
= E_{MOFs-gas} - E_{MOFs}
$$

$$
- E_{gas}
$$
 (5)

 ΔE

where *E* **represents the energy of the system after geometry relaxation,** $E_{MOFs-gas}$ is the total energy of the MOFs and gas molecules, E_{MOFs} and E_{gas} **are the energies of the isolated MOFs and gas molecules, respectively.**

Transient Breakthrough Simulation. To determine the productivity of polymer-grade (99.95%) C2H4, we performed transient breakthrough simulations using the simulation methodology described in our previous publications.^{[S3,](#page-57-2)[6](#page-57-5)} For the breakthrough simulations, the following parameter values were used: length of packed bed, $L = 0.3$ m; voidage of packed bed, $\varepsilon = 0.4$; superficial gas velocity at inlet, $u =$ 0.04 m/s. The transient breakthrough simulation results are presented in terms of a dimensionless time, τ , defined by dividing the actual time, t, by the characteristic time, Lεu⁻¹.

Figure S1. Illustration of *pacs-*tpt structure. Metal, green; oxygen, red; carbon, grey;

nitrogen, blue.

Figure S2. EDS analysis of all the MOFs in this study. The atomic ratios of the metals are shown.

Figure S3. Comparisons of TGA traces of activated samples for all the compounds in this study. a: TGA traces of Mg2Ti-bdc MOFs with different ligand 2 ; b: TGA traces of Mg2V-bdc MOFs with different ligand 1; c: TGA traces of Co2V-tpt MOFs with different ligand 1; d: TGA traces of bdc-tpt MOFs with different metal trimers.

Figure S4. PXRD patterns of CPM-223 under different thermal treatments.

Figure S5. Experimental and simulated PXRD patterns for all the compounds in this study. For Co2V-ndc-tpt and Co2V-dmbdc-tpt, the simulated PXRD were generated from Mg2V-ndc-tpt and Mg2V-dmbdc-tpt, respectively, for comparison.

Figure S6. Comparisons of PXRD patterns of as-synthesized samples and the samples after being immersed in water for 24 hours (a: CPM-723; b: CPM-733).

For this family of structures, the occupation of open metal sites by pore-partitioning ligands prevents the water attack on the open metal site, suppresses the breathing effect of MIL-88-type structure, and increases the framework connectivity simultaneously. Such rigidification leads to the robust nature.

Figure S7. Gas adsorption properties for Mg₂Ti-bdc-tpbz. a: C₂H₄ adsorption and C2H⁶ adsorption at 298 K; b: C2H⁴ adsorption and C2H⁶ adsorption at 273 K; c: SLF fitting for C₂H₄ adsorption at 298 K; d: SLF fitting for C₂H₆ adsorption at 298 K.

Figure S8. Gas adsorption properties for Mg₂Ti-bdc-tppy. a: C₂H₄ adsorption and C2H⁶ adsorption at 298 K; b: C2H⁴ adsorption and C2H⁶ adsorption at 273 K; c: SLF fitting for C2H⁴ adsorption at 298 K; d: SLF fitting for C2H⁶ adsorption at 298 K.

Figure S9. Gas adsorption properties for Mg₂Ti-bdc-tpt. a: C₂H₄ adsorption and C₂H₆ adsorption at 298 K; b: C2H⁴ adsorption and C2H⁶ adsorption at 273 K; c: SLF fitting for C2H⁴ adsorption at 298 K; d: SLF fitting for C2H⁶ adsorption at 298 K.

Figure S10. Gas adsorption properties for Mg₂V-bdc-tpt. a: C₂H₄ adsorption and C2H⁶ adsorption at 298 K; b: C2H⁴ adsorption and C2H⁶ adsorption at 273 K; c: SLF fitting for C₂H₄ adsorption at 298 K; d: SLF fitting for C₂H₆ adsorption at 298 K.

Figure S11. Gas adsorption properties for Mg₂V-bdc-tpt. a: C₂H₄ adsorption and C₂H₆ adsorption at 298 K; b: C2H⁴ adsorption and C2H⁶ adsorption at 273 K; c: SLF fitting for C2H⁴ adsorption at 298 K; d: SLF fitting for C2H⁶ adsorption at 298 K.

Figure S12. Gas adsorption properties for Co₂Ti-bdc-tpt. a: C₂H₄ adsorption and C₂H₆ adsorption at 298 K; b: C₂H₄ adsorption and C₂H₆ adsorption at 273 K; c: heat of adsorption for C₂H₄ and C₂H₆; d: N₂ adsorption at 77 K; e: SLF fitting for C₂H₄ adsorption at 298 K; f: SLF fitting for C2H⁶ adsorption at 298 K.

Figure S13. Gas adsorption properties for Co₂V-bdc-tpt. a: C₂H₄ adsorption and C₂H₆ adsorption at 298 K; b: C2H⁴ adsorption and C2H⁶ adsorption at 273 K; c: SLF fitting for C2H⁴ adsorption at 298 K; d: SLF fitting for C2H⁶ adsorption at 298 K.

Figure S14. Gas adsorption properties for Co₂V-ndc-tpt. a: C₂H₄ adsorption and C₂H₆ adsorption at 298 K; b: C2H⁴ adsorption and C2H⁶ adsorption at 273 K; c: heat of adsorption for C₂H₄ and C₂H₆; d: N₂ adsorption at 77 K; e: SLF fitting for C₂H₄ adsorption at 298 K; f: SLF fitting for C2H⁶ adsorption at 298 K.

Figure S15. Gas adsorption properties for Co₂V-dmbdc-tpt. a: C₂H₄ adsorption and C_2H_6 adsorption at 298 K; b: C_2H_4 adsorption and C_2H_6 adsorption at 273 K; c: heat of adsorption for C2H⁴ and C2H6; d: N² adsorption at 77 K; e: SLF fitting for C2H⁴ adsorption at 298 K; f: SLF fitting for C2H⁶ adsorption at 298 K.

Figure S16. Comparisons of adsorption enthalpy for C₂H₆ and C₂H₄ adsorption.

Figure S17. Density distribution of C₂H₆ molecules mass center within Co₂V-bdc-tpt under different pressures from GCMC simulations. Pore surface is also shown for comparison.

Figure S18. Comparisons of the IAST Selectivities for C₂H₆/C₂H₄ mixture (50/50). a: Selectivities of Mg2Ti-bdc MOFs with different ligand 2 ; b: Selectivities of Mg₂V-bdc MOFs with different ligand 1; c: Selectivities of Co₂V-tpt MOFs with different ligand 1; d: Selectivities of bdc-tpt MOFs with different metal trimers.

Figure S19. Comparisons of N₂ adsorption at 77 K for all the compounds in this study. a: N_2 adsorption isotherms of Mg₂Ti-bdc MOFs with different ligand 2; b: N₂ adsorption isotherms of Mg₂V-bdc MOFs with different ligand 1; c: N₂ adsorption isotherms of Co₂V-tpt MOFs with different ligand 1; d: N_2 adsorption isotherms of bdc-tpt MOFs with different metal trimers. Note that there is significant adsorption hysteresis in CPM-736 (Co2V-ndc-tpt), which is possibly caused by the defects in the structure due the bulky 1,4-ndc ligand. Similar phenomenon could also be observed in another two 1,4-ndc based *pacs* MOFs (CPM-33d & CPM-236).^{[S1,](#page-57-0)[7](#page-57-6)}

Figure S20. Comparisons of DFT pore size distribution for all the compounds in this study. The pressure range of $P/P_0 < 10^{-3}$ was applied to calculate the pore size due to the micro-porous feature of this family of materials.

Figure S21. Comparison of separation potential of CPM-233 and CPM-238.

Figure S22. Simulated breakthrough curves for CPM-233 and CPM-723 with 50/50

and 10/90 C2H6/C2H⁴ mixture.

Figure 23. Multiple ethane adsorption test of Mg2V-bdc-tpt at 298 K, showing almost no loss of capacity after 5 cycles.

Figure S24. X-ray photoelectron spectroscopy of Co₂Ti-bdc-tpt showing the Co 2p (left) and Ti 2p peaks (right). The satellite peaks shown on Co2p are characteristic of $Co²⁺$. Two peaks located at around 464.4 eV and 458.5 eV, correspond to Ti 2p1/2 and Ti $2p3/2$, respectively, indicating an oxidation state of Ti^{4+} .

Figure S25. The preferential C₂H₄ and C₂H₆ adsorption sites determined by DFT calculation.

We performed DFT calculations to gain insight into the selective adsorption mechanism. The results indicated that the C2H⁶ binding energy at its most preferred binding site is around −35.85 kJ/mol, higher that that for C2H⁴ (−33.76 kJ/mol). As shown in Figure S25, the gas molecules are trapped in the pocket defined by three phenyl rings via van der Waals interactions between the ethane and the neighboring π electron clouds. For C2H6, six hydrogens can interact with the adjacent phenyl rings via C−H···π interaction. In contrast, four hydrogens from C2H⁴ interact with the adjacent phenyl rings. The more interaction points of ethane could account for its higher binding energy.

Table S1. The numbering scheme and names for the MOFs reported in this work

MOFs	BET surface	Pore	C_2H_6	C_2H_4	C_2H_6/C_2H_4	Q^0_{st}	Q^0_{st}	C_2H_6	C_2H_4	CO ₂	CO ₂
	area (m^2/g)	size (\AA)	298K, 1bar $\text{ (cm}^3\text{/g)}$	298K, 1bar $\text{ (cm}^3\text{/g)}$	selectivity	(C ₂ H ₆) (kJ/mol)	(C ₂ H ₄) (kJ/mol)	273 K, 1bar $\text{ (cm}^3\text{/g)}$	273 K, 1bar $\text{ (cm}^3\text{/g)}$	273 K, 1bar $\text{ (cm}^3\text{/g)}$	298 K, 1bar $\text{ (cm}^3\text{/g)}$
CPM-733 $Co2V$ -bdc-tpt	1328.5	7.3	159.6	142.7	1.75	23.4	22.5	186.3	190.0	152.5	75.8
CPM-736 $Co2V$ -ndc-tpt	472.5	5.9	90.2	86.9	1.48	30.4	30.1	108.9	128.9		
CPM-738 $Co2V$ -dmbdc-tpt	1161.5	5.9	104.6	102.1	1.42	27.9	26.5	131.9	140.2		
CPM-723 $Co2Ti-bdc-tpt$	1369.8	6.8	154.7	149.4	1.50	21.7	20.0	187.7	194.9	152.6	76.2
CPM-223 $Mg_2Ti-bdc-tpt$	1460.6	6.8	54.8	142.1	1.57	25.4	25.0	190.8	200.9	167.4	82.5
CPM-223-tppy $Mg2Ti-bdc-tppy$	1599.1	6.8	160.6	164.5	1.28	25.0	22.6	204.0	221.4	182.9	94.8
CPM-223-tpbz $Mg_2Ti-bdc-tpbz$	1661.7	6.8	154.2	140.0	1.51	21.9	23.3	201.4	197.2		
CPM-233 $Mg_2V-bdc-tpt$	1597.9	6.8	166.8	146.1	1.64	27.3	26.7	178.6	189.0	179.3	89.3

Table S2. Summary of gas adsorption properties for MOFs reported in this work

-: not tested

C_2H_6 uptake			C ₂ H ₆ uptake			Q^0_{st}		
			in	Separation		(C ₂ H ₆)		
MOFs	G^a	V^{b}	C_2H_6/C_2H_4	potential	Selectivity	(kJ/mol)	Ref.	
	cm^3/g)	(cm ³ /cm ³)	mixture	(mmol/g)				
			(cm ³ /g)					
Mg2V-bdc-tpt	166.8	139.1	99.2	1.73	1.64	27.3	This	
	160.6	128.6	91.4	0.90	1.28	25.0	work This	
Mg2Ti-bdc-tppy							work	
Co2V-bdc-tpt	159.6	142.0	96.8	1.88	1.75	23.4	This	
							work	
Mg2Ti-bdc-tpt	154.8	126.0	92.1	1.50	1.57	25.4	This work	
Co2Ti-bdc-tpt	154.7	137.5	92.2	1.38	1.50	21.0	This	
							work	
Mg2Ti-bdc-tpbz	154.2	124.4	89.6	1.36	1.51	21.9	This	
							work	
Mg2V-dmbdc-tpt	124.6	103.9	71.3	0.95	1.43	24.7	This work	
PCN-250	116.7	111.7	66.3	1.48	1.9	23.2	${\rm S}8$	
$Ni(BDC)(\text{TED})0.5$	112	114.2	55.6	1.01	1.6	21.5	S9	
$MUF-15$	105.05	131.0	70.1	1.53	1.96c	29.2	${\bf S10}$	
Co2V-dmbdc-tpt	104.6	N.A.	60.5	0.81	1.42	27.9	This	
							work	
IRMOF-8	92	82.4	48.4	1.41	1.6	52.5	S11	
Co2V-ndc-tpt	90.2	N.A.	52.5	0.76	1.48	30.4	This	
							work	

Table S3. Summary of ethane uptake properties in ethane-selective MOFs (~1 bar and \sim 298 K). 50/50 C₂H₆/C₂H₄ mixture was used in all IAST calculations.

^a Gravimetric uptake. ^b Volumetric uptake. ^c The IAST-calculated selectivity of 1.79 was obtained in this work as compared to the value of 1.96 from ref. S8. ^d The calculation of separation potential for MIL-142A, PCN-245, ZIF-8, and ZIF-4, and ZIF-7 are from Table S5 of ref. S8. ^e The value reported in the ref. S11 is 1.2. N.A.: not available.

Table S4. Summary of MOFs with high ethane uptake at ~1 bar and around room temperature. Both ethane-selective MOFs and ethylene selective MOFs are included.

MOFs	Gases	N_{sat} (mmol·g ⁻¹)	b (kPa ^{-1/n})	1/n	R^2
CPM-733	C ₂ H ₄	11.19137	0.01345	0.99567	0.99979
$Co2V$ -bdc-tpt	C ₂ H ₆	8.52615	0.02551	1.14363	0.99993
CPM-736	C_2H_4	5.06259	0.04101	0.9439	0.99991
$Co2V$ -ndc-tpt	C_2H_6	4.92313	0.07117	0.8874	0.99963
CPM-738	C_2H_4	6.31334	0.02611	0.99322	0.99996
$Co2V$ -dmbdc-tpt	C ₂ H ₆	5.49793	0.04235	1.0489	0.99986
CPM-723	C ₂ H ₄	10.74372	0.01265	1.05384	0.99995
$Co2Ti-bdc-tpt$	C ₂ H ₆	8.43098	0.02436	1.12965	0.99994
CPM-223-tpbz	C ₂ H ₄	9.49258	0.01066	1.12633	0.99991
MgTi-bdc-tpb	C ₂ H ₆	8.62664	0.01813	1.16498	0.99993
CPM-223-tppy	C ₂ H ₄	11.62725	0.01191	1.07594	0.99999
$Mg_2Ti-bdc-tpp$	C_2H_6	8.99423	0.02059	1.13631	0.99996
CPM-223	C ₂ H ₄	11.03222	0.01182	1.02759	0.99997
Mg ₂ Ti-bdc-tpt	C_2H_6	8.76043	0.02217	1.10939	0.99996
CPM-233	C ₂ H ₄	11.22692	0.01234	1.02397	0.99995
$Mg2V$ -bdc-tpt	C ₂ H ₆	9.33279	0.02041	1.13942	0.99994
CPM-238	C ₂ H ₄	7.85891	0.02556	0.94192	0.99992
Mg_2V -dmbdc-tpt	C ₂ H ₆	7.00888	0.03889	0.98914	0.99990

Table S5. Summary of the refined parameters with Single-Site Langmuir-Freundlich fitting for the pure adsorption isotherms of C_2H_4 and C_2H_6 .

Table S6. Crystal data and structure refinement for CPM-723 (Co₂Ti-bdc-tpt).

Table S7. Crystal data and structure refinement for CPM-733 (Co2V-bdc-tpt).

Table S8. Crystal data and structure refinement for CPM-223 (Mg₂Ti-bdc-tpt).

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Table S10. Crystal data and structure refinement for CPM-223-tppy (Mg₂Ti-bdc-tppy).

Table S11. Crystal data and structure refinement for CPM-233 (Mg_2V -bdc-tpt).

Table $S12$. Crystal data and structure refinement for CPM-238 (Mg₂V-dmbdc-tpt).

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